

Molecular lifting, twisting, and curling during metal-assisted polycyclic hydrocarbon dehydrogenation

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Abstract

The atomistic understanding of the dissociation mechanisms for large molecules adsorbed on surfaces is still a challenge in heterogeneous catalysis. This is especially true for polycyclic aromatic hydrocarbons, which represent an important class of organic compounds used to produce novel graphene-based architectures. Here, we show that coronene molecules adsorbed on Ir(111) undergo major conformational changes during dissociation. They firstly tilt upwards with respect to the surface, still keeping their planar configuration, and subsequently experience a rotation, which changes the molecular axis orientation. Upon lifting, the internal C-C strain is initially relieved; as the dehydrogenation proceeds, the molecules experience a progressive increase in the average interatomic distance, and gradually settle to form dome shaped nano-graphene flakes. Our results provide important insight into the complex mechanism of molecular break-up, which could have implications in the synthesis of new carbon-based nanostructured materials

Introduction

The process of surface-assisted cyclodehydrogenation of polycyclic aromatic hydrocarbons (PAHs) has been adopted in the recent years as one of the most effective, versatile, and flexible strategies for the bottom-up synthesis of fullerenes,¹ small graphene flakes (nanographene), and nanoribbons.²⁻⁴ The large number of available precursors is the key to tailor the structural properties of low-dimensional carbon networks via polymerization reactions and to control their chemical and transport properties, which exhibit promising properties for applications in opto- and molecular-electronics. In particular, the band gap problem in graphene can be overcome by exploiting quantum confinement effects, which allow the modification of the band gap by adjusting the width and the type of edge terminations.⁵⁻¹⁰ However, the pathway of molecular dissociation becomes more complex with increasing molecular extension, number of sequential reaction steps, degrees of freedom for molecular motion, and competi-

tion with desorption processes. The initial-state PAH adsorption configuration, which can be influenced by inter-molecular or substrate-mediated interactions, can also strongly affect the dissociation mechanism. For example, the reported thermally induced dehydrogenation of polyphenylene molecules on Cu(111)¹¹ proceeds via six intermediate steps of aryl-aryl coupling reactions. For polyphenylene molecules, the partial dehydrogenation of the large hydrocarbons leads to a H terminated nanographene, although in this case, the interaction with the substrate only plays a marginal role, and the partial reaction barriers are similar for the non-catalyzed dehydrogenation. It is clear that detailed knowledge of the chemical, structural, and electronic transformations involved in each cyclodehydrogenation step is a prerogative for finding more efficient routes to fabricate novel nanographene/nanoribbon structures with tailored properties. In this study we show that thermally assisted cyclodehydrogenation of coronene (C₂₄H₁₂) on Ir(111) takes place through sequential steps that include dramatic changes of the molecule's pristine configuration. For a comprehensive characterization of the reaction process we adopted a combined theoretical and experimental approach where density functional theory (DFT) calculations accompanied the experiments performed combining several techniques, namely fast and high-energy resolution X-ray photoelectron spectroscopy (HR-XPS), near-edge X-ray absorption fine structure (NEXAFS), ultraviolet photoelectron spectroscopy (UPS), angle resolved photoemission spectroscopy (ARPES), temperature programmed desorption (TPD), and low energy electron diffraction (LEED).

Experimental

The Ir(111) single crystal used in this experiment was cleaned in ultra high vacuum (UHV) conditions by repeated cycles of Ar⁺ sputtering at 2 keV and annealing to 1400 K, followed by annealing cycles in oxygen pressure to remove any carbon contaminants. A final annealing in hydrogen atmosphere was performed to remove residual oxygen from the surface. This procedure was optimized to ensure a high quality surface, as judged by LEED and XPS. The

measurements reported in this work were performed in UHV condition, with a base pressure of the order of 1×10^{-10} mbar.

Commercially available (Sigma-Aldrich) coronene powder (99%, purified by sublimation) was used throughout the experiment. Coronene molecules were evaporated in vacuum using a custom evaporator consisting in a Boron Nitride (BN) crucible wrapped with a 1 mm diameter Ta wire used to heat it by resistive dissipation. The crucible was cleaned in vacuum before the experiment by means of a long annealing to increasing temperatures up to 670 K, in order to clean the evaporation cell from impurities. The temperature inside the crucible was measured through a type *K* thermocouple in thermal contact with the coronene powder. Coronene evaporation was achieved by keeping the crucible at a fixed temperature, varying from 400 to 440 K. Before evaporation, coronene was further purified in vacuum by means of repeated short annealing cycles up to 450 K, while monitoring the chamber residual gas composition with a Residual Gas Analyzer system. Calibration of the coronene evaporation rate was achieved by comparing the photoelectron intensity to a reference signal acquired, with the same acquisition parameters, after graphene growth on Ir(111). On the basis of the C(10 × 10)/Ir(9 × 9) moiré unit cell, observed for single layer graphene on Ir(111), the reference coverage was assumed to be 2.47 ML.

High resolution (HR) and temperature programmed (TP) XPS¹² measurements were carried out at the SuperESCA beamline of the Elettra synchrotron radiation facility (Trieste, Italy). A state-of-the-art SPECS PHOIBOS 150 hemispherical analyzer was employed, equipped with a delay line detector developed at Elettra. C 1s spectra were acquired in normal emission geometry with a photon energy of 400 eV, along with a measurement of the Fermi Level on the sample, which was used to align the binding energy scale. The X-ray beam on the sample, at normal emission, measures 300 μm in the horizontal direction and 10 μm in the vertical one. In order to check for possible X-ray induced molecular break-up we performed several scans along the surface to search for C 1s spectral modifications, without finding appreciable differences.

NEXAFS measurements were performed at the SuperESCA beamline of Elettra. Carbon K-edge absorption spectra were acquired in the Auger Electron Yield configuration, with a photoelectron kinetic energy of 260 eV and a photon energy range between 280 eV and 315 eV. Two different angular configurations were used, corresponding to an incidence angle of 0° and 70° with respect to the surface.

Valence band photoemission spectra were acquired at the BaDElPh beamline of Elettra using a state-of-the-art SPECS PHOIBOS 150 hemispherical analyzer combined with a 2D detector consisting of a phosphor screen coupled with a 2D-CCD commercial detector. The UPS spectra were measured with photon energy equal to 34 eV in a $\pm 13^\circ$ wide angular region centered in the K point of the reciprocal space. The ARPES maps were also acquired with 34 eV photon energy.

The LEED experiments were carried out at the Surface Science Laboratory at Elettra-Sincrotrone Trieste using a VG rear-view electron optical system.

The TPD curves were acquired at the Surface Science Laboratory at Elettra using an SRS 200 residual gas analyzer. During the experiment the sample was annealed at a fixed rate of 2.5 K/s by means of three hot W filaments placed a few millimeters behind the sample.

The C 1s spectra were analyzed using a sum of 24 Doniach-Šunjić functions¹³ convoluted with Gaussians, including a Shirley type background. The relative BE shifts between the components were constrained to the ones predicted by DFT, while a rigid shift was imposed to align the experimental BE scale to the theoretical one. The Lorentzian and Gaussian widths, as well as the asymmetry parameters, were constrained to be the same for all peaks. This implicates the following assumptions: the core electronic states corresponding to the different C atoms in the molecule have the same finite lifetime (resulting in an equal Lorentzian broadening); all the core level components are affected to the same extent by thermal phonon broadening, experimental, and inhomogeneous broadening (which are reflected in their Gaussian FWHM); the chance of an electron-hole pair excitation event is the same for each core level (resulting in an equal asymmetry parameter). The intensities were

also constrained to be equal. All experimental data and fitting results have been displayed after subtracting the background found by the fitting operation.

Density functional theory calculations have been carried out using the VASP code. We used the Projector Augmented Wave (PAW) method to account for the core electrons, with the $6s$ and $5d$ electrons of Ir and the $2s$ and $2p$ electrons of C explicitly included in the valence. Single particle orbitals were expanded in plane-waves using a kinetic energy cut-off of 400eV. Surfaces were modeled with the usual slab geometry, using a (9 x 9) supercell with 4 layers, of which the bottom two were kept frozen at the bulk interatomic distances. The vacuum was $\sim 17.5 \text{ \AA}$ for the clean surface and $\sim 15 \text{ \AA}$ for the system with the coronene molecule adsorbed on the surface. We used the Gamma point only to sample the Brillouin zone. Energy barrier were estimated with the climbing-image Nudged Elastic Band method,¹⁴ using up to 11 images. Core level binding energies have been estimated within the final state approximation, where the screening from valence electrons is included, thus providing an accurate estimate of its effects on the core level binding energies. According to previous theoretical studies,¹⁵ the method used in our work calculates the core level shifts with an accuracy better than 50 meV.

Results and Discussion

Coronene was adsorbed on the surface at room temperature, $T = 300 \text{ K}$. First, we performed NEXAFS measurements to address the issue of intact adsorbed molecule orientation with respect to the surface plane. Figure 1(a) reports the C K-edge absorption spectra measured at two different incident angles for sub-monolayer coverage. The most noticeable result is the strongly reduced intensity of the π^* resonance components at 286 eV when the electric field vector is almost parallel to the surface (normal incidence, NI), originating from the C $1s \rightarrow \text{LUMO}$ electronic transitions,¹⁶ and the increased spectral weight at about 292 eV due to the transition to the σ^* orbitals, thus indicating a nearly flat adsorption geometry.

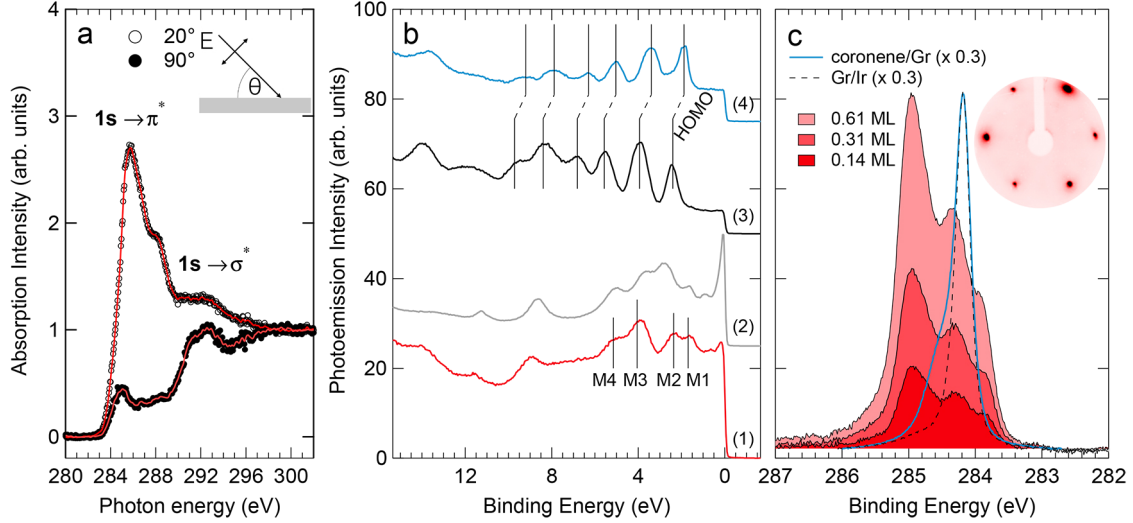


Figure 1: (a) C K-edge NEXAFS spectra for coronene/Ir(111) acquired with linearly polarized radiation, with the electric field vector either parallel (filled circles) or almost normal (open circles) to the crystal surface. (b) Valence band spectra measured at $h\nu = 34\text{ eV}$ of (1-red curve) monolayer coronene on Ir(111), (2-gray curve) clean Ir(111), (3-black) multilayer coronene on Ir(111) and (4-blue) monolayer coronene on graphene/Ir(111). Identified molecular orbitals are, right to left (black bars on spectra (3), (4)): E_{2u} first peak; B_{1g} , B_{2g} , E_{1g} second peak; E_{2u} , A_{2u} , E_{2g} third peak. Molecular orbitals M1, M2, M3, and M4 on spectrum (1), associated to non-dispersing features in the ARPES maps (see Supporting Information Figure S1), have been marked with vertical lines. (c) C 1s spectra measured at $h\nu = 400\text{ eV}$ for three different coverages of coronene on Ir(111), compared with monolayer coronene on graphene/Ir(111) (blue curve) and pristine graphene on Ir(111) (dotted curve), and (inset) LEED pattern acquired at 86 eV electron energy after about 0.14 ML coronene deposition.

The presence of a shoulder extending up to about 288 eV can be explained, beside the C $1s \rightarrow \text{LUMO}+1, +2, +3$ and $+4 \pi^*$ transitions, as due to the interaction of the molecules with the substrate. The strong modification in the electronic structure of coronene upon adsorption is confirmed by the UPS results reported in Fig. 1(b). These valence band (VB) spectra have been measured at normal emission ($h\nu = 34\text{ eV}$) after different surface preparations. The spectrum acquired after deposition of a monolayer of coronene on clean Ir(111) (1-red), shows several peaks attributed to molecular signals, since they appear at different binding energies with respect to those present in the spectrum of clean Ir(111) (2-gray), which is dominated by the large d -band density of states. These molecular states can be associated to non-dispersing features in the ARPES maps (see Supporting Information,

Figure S1). In particular the M1 peak, at 1.8 eV, together with the next state (M2) appearing at 2.4 eV can be attributed to hybridization of the coronene molecular orbitals with the Ir 5d bands. The Ir(111) surface’s ability in creating strong π bonds to PAHs is confirmed by considering the spectral components appearing for other coronene-based interfaces with weaker interactions, *i.e.* coronene multilayers grown on Ir(111) (3-black) and for coronene on graphene/Ir(111) (4-blue). In these cases, besides the rigid shift induced by different charge transfer, the VB spectra are a clear fingerprint of the coronene molecular orbitals (E_{2u} , B_{1g} , B_{2g} , E_{1g} , E_{2u} , A_{2u} , E_{2g}), whose binding energy distribution is in excellent agreement with previous experimental findings and theoretical calculations.^{17–19} Further insight into the modification of the electronic structure due to adsorption was gained by high-resolution XPS C 1s core level spectra, acquired at different molecular coverages (see Fig. 1(c)). The three-peak spectral shape and binding energies are very different from those measured for coronene on graphene (blue curve), and for coronene in gas phase, that displays a two-peak line shape.¹⁶ In addition, the C 1s spectra measured at different coverages show the same line shape regardless of the coronene amount, suggesting that the adsorption configuration and site are not modified by intermolecular interactions in denser molecular layers or by the presence of surface defects, such as steps. It should be noted that the presence of three components cannot be simply justified as due to the geometrically non equivalent carbon atoms in the molecule, *i.e.* those forming the inner hexagon, the middle one, and the outer ring, because the spectral weight does not match the 1:1:2 ratio corresponding to the population of such non equivalent C atoms. The LEED pattern (see inset Fig. 1(c)) does not show any additional diffraction spots, besides those arising from the hexagonal symmetry of the Ir substrate, indicating that the coronene molecules do not pack forming a long-range ordered layer, but are rather randomly distributed among the free adsorption sites.

In order to further shed light on the coronene adsorption geometry, and on the origin of the different core level components, we performed DFT calculations. Several configurations were probed (see Supporting Information, Figure S2), including molecules with symmetry

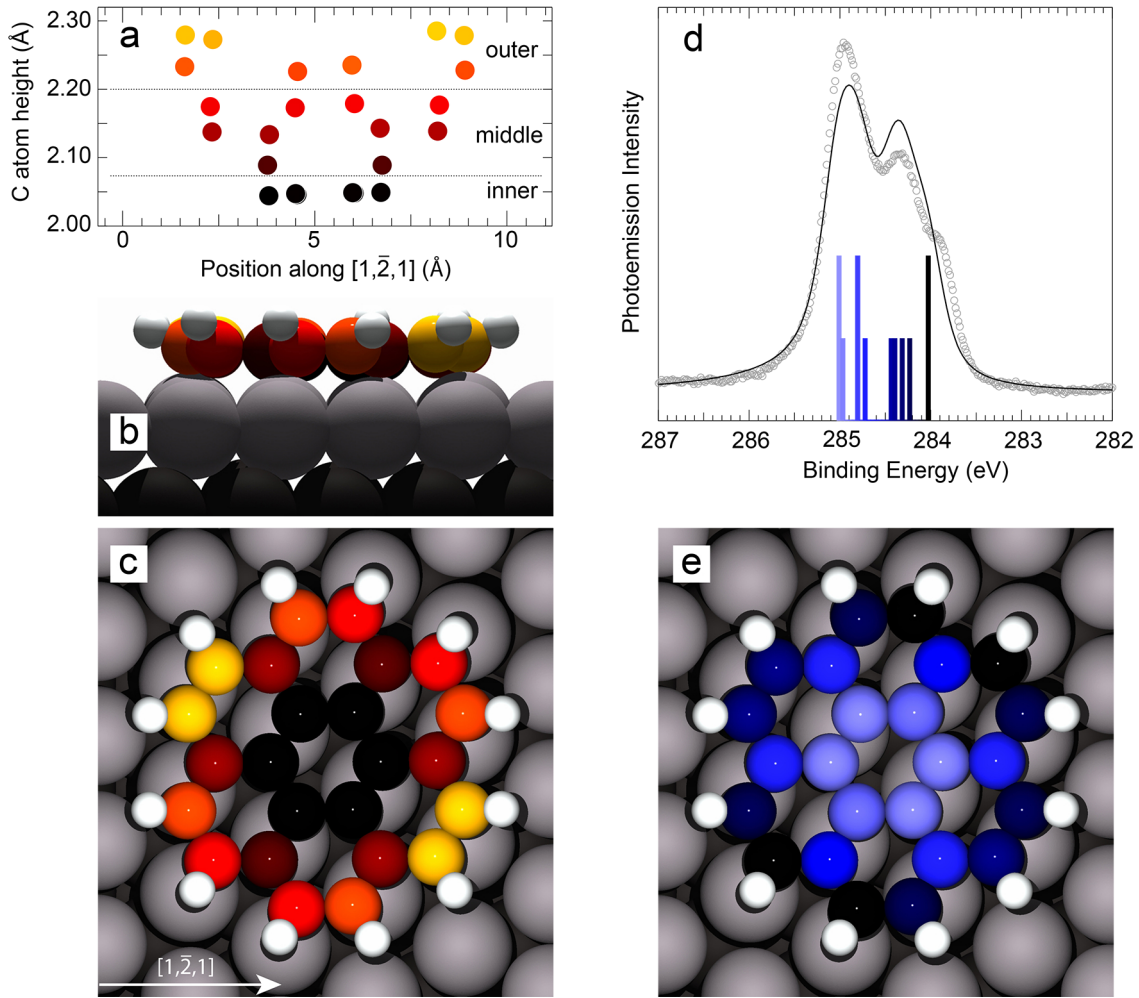


Figure 2: (a) Carbon-Ir distance for the non-equivalent C atoms of the coronene molecule. (b-c) Illustration of the side and top views of the $C_{24}H_{12}$ molecular orientation as found in DFT calculations. The color scale reflects the separation between the C atoms and the substrate beneath. Small differences in the C-Ir distance between apparently equivalent C atoms can be accounted for by considering the small influence of the second Ir layer, which breaks the 6-fold symmetry of the first atomic layer. (d) C 1s spectrum measured at normal emission and $h\nu = 400$ eV (empty circles) together with the fit result (solid line) and the calculated spectral distribution originated from the 24 C atoms. (e) Color scale in the model reflects the BEs of the different C 1s calculated components.

axes oriented along different directions, with different matches with respect to the high-symmetry adsorption sites of the Ir(111) surface, and also in non-planar configurations. The minimum energy configuration corresponds to coronene adsorbed with the inner C hexagon in bridge-site and the molecular axis aligned parallel to the $[101]$ direction of Ir(111) (see Fig. 2(c)). The molecules assume a bowl-like shape (side view shown in Fig. 2(b)), usually

associated to π -conjugated compounds known as buckybowls or π -bowls.^{20,21} These species have been found to commonly form stable adlayers on metal surfaces, as in the case of corannulene ($C_{20}H_{10}$) on Cu(110)^{22,23} and of sumanene ($C_{21}H_{12}$) on Ag(111).²⁴ In our case, the adsorbed molecule develops its bowl opening upwards: C atoms in the inner ring are about 0.23 Å closer to the surface with respect to C atoms in the outer ring (see Fig. 2(a)). The bowl-shaped geometry is enhanced by the hydrogen atoms displaying an average height of 2.74 Å, 0.7 Å farther from the surface than the inner C atoms, and a C–H bond angle ranging from 22° to 39° with respect to the Ir surface plane. In order to test the consistency of the minimum energy adsorption structure we compared in Fig. 2(d) the experimental C 1s core level spectra (empty circles), for which the molecule-substrate interaction is expected to have a strong influence, and fit results (solid line). We computed the C 1s core level binding energy (BE) for each of the 24 C atoms of the coronene molecules, including also final state effects due to core-hole screening. The results, illustrated in Fig. 2(d), show that the computed BEs for the carbon atoms in the coronene molecule are not only linked to some extent to the C-metal distance (C atoms in the central ring showing the highest BEs), but are strongly dependent on the positions with respect to the first-layer Ir atoms. This is especially true for those C atoms sitting in the outer ring and bonded to H atoms, whose C 1s BE depends on the degree of interaction with the substrate Ir atoms, besides the bond with H instead of another C. The good agreement between experimental data and fit results strongly supports the DFT calculated molecular adsorption geometry.

In order to explore the mechanism of coronene dissociation we initially employed TPD, in the temperature range 150–950 K (see Fig. 3(a)), with the aim of finding the threshold of hydrogen desorption (mass to charge ratio $m/z = 2$), which can be used as a fingerprint for C–H break-up. Desorption spectra corresponding to different initial coverages of coronene are reported in red colored scale, indicating that H_2 in gas phase can be detected only above 500 K, with a maximum desorption rate at temperatures slightly higher than 650 K. To verify that on coronene the process of C–H bond dissociation does not take place

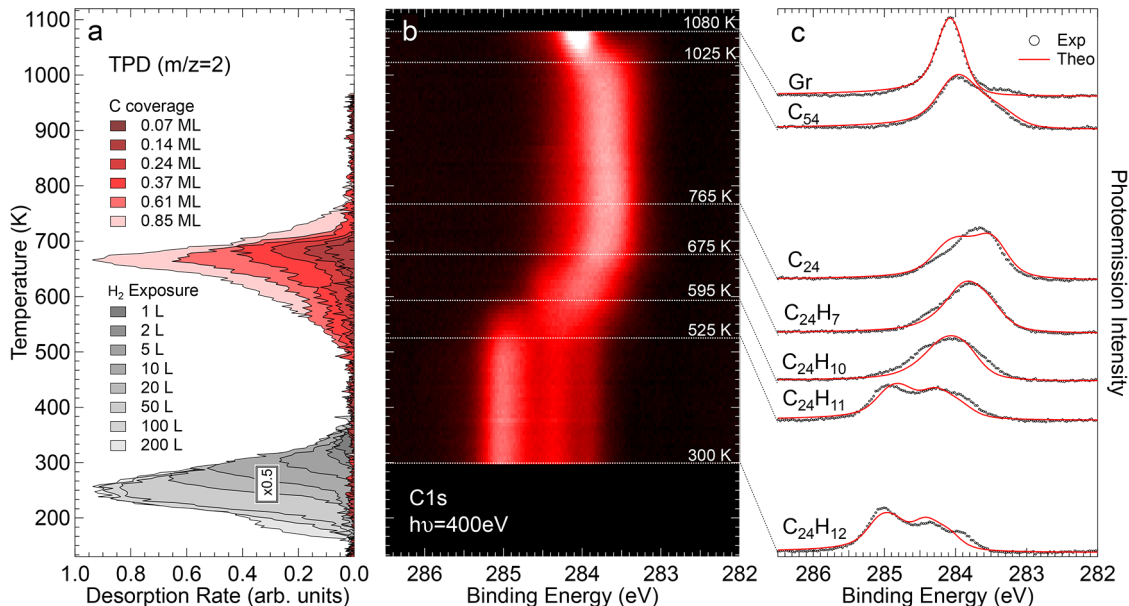


Figure 3: (a) Temperature programmed desorption spectra (red scale) of $m/z = 2$ after coronene deposition at different coverage and $T = 300$ K. For comparison desorption spectra (grey scale) corresponding to $m/z = 2$ after molecular hydrogen adsorption at $T = 100$ K are reported (intensity rescaled by a factor 0.5). (b) Temperature dependent C 1s core level spectra (about 100 spectra) shown as a two-dimensional intensity plot. (c) Comparison between selected experimental C 1s core-level spectra and fit results (red curves).

at lower temperature, we compared the TPD data with a similar H₂ desorption experiment starting from hydrogen adsorption at $T = 100$ K. It is known that H₂ adsorbs dissociatively on Ir(111)^{25,26} and that the onset of desorption is indicative of the recombination of two H atoms after surface diffusion. The large difference in H₂ formation temperature, and the lack of any overlap on the temperature scale of the desorption spectra from H₂ and C₂₄H₁₂, is a clear proof that coronene molecules undergo the first C–H breakup only above 500 K. Further information about the coronene dissociation process was obtained following *in-situ* the temperature evolution of the C 1s core level spectrum from RT to 1120 K. The combination of a high-efficiency electron energy analyzer with a home-made delay-line detection system allowed us to measure each photoemission spectrum in about 30 s (still keeping the overall energy resolution to 40 meV), providing real-time spectroscopic visualization of the intermediate reaction stages. The image plot in Fig. 3 (b) shows the evolution of the C 1s spectrum at increasing temperature. As expected, the spectrum does not change during the

initial stage of the temperature ramp, and only above 525 K, the onset temperature for H₂ desorption in the TPD spectra, clear changes are observed. The spectral intensity shifts to lower binding energy, reaching a rather constant value above 700 K, when the TPD results show that all hydrogen atoms have desorbed. Only above 1000 K the C 1s signal starts shifting back to higher BE. The BE value of 284.10 eV, reached above 1050 K, is a clear fingerprint of high-temperature graphene formation.^{27–30} Notably, during the dissociation process, there is no loss in the overall C 1s spectral intensity, indicating that the energy barrier for C–H bond cleavage is lower than that for desorption of the intact molecule or C containing species. Unambiguous chemical assignment of the experimentally observed C 1s core-level line shape is, however, not easy, due to the high number of possible configurations of the dehydrogenated transient products. In order to interpret the C 1s spectral sequence, and to reach an atomistic understanding of the transient states involved in the thermal breakup of coronene, we resorted to DFT by performing extensive calculations of the energy barriers between the computed intermediate species, using the nudged elastic band (NEB) method.¹⁴ In Fig. 4 we show the twelve molecule-substrate configurations corresponding to the energetically most favored reaction pathway for breaking each C–H bond in n sequential steps (with n from 1 to 12), leaving the dehydrogenated molecules and an additional single H atom on the surface. The reaction-coordinate energy diagrams reported under the geometrical configurations are given with respect to the energy of the $n - 1$ configuration, being $n = 0$ the intact adsorbed coronene molecule. It is important to stress that, despite the higher final state energy of each dehydrogenation step, the overall reaction can easily proceed at the temperature of our experiments, as dissociated H atoms can diffuse very rapidly on the surface forming H₂, which desorbs and is not available for the reverse process. For this reason the final state of structure n does not exactly correspond to the initial state of structure $n + 1$, where a H atom has been removed. What is intriguing in our findings is the complex evolution of the molecular geometry during the dehydrogenation steps. Close inspection of the images shown in Fig. 4 (see also movie in the Supporting Information) indi-

cates that the stepwise dehydrogenation mechanism appears to be driven by the proximity of the C outer ring atom to the final adsorption position. At the beginning, the intact adsorbed molecule has significant internal C–C strain with respect to the gas phase configuration (see Supporting Information, Figure S3). The peripheral C closest to the most favorable position is the one which will lose an H atom in the next reaction step; the dissociation begins (see Fig. 4, $n = 1$) from one of the four equivalent peripheral C atoms closest to the top position, which, after losing its H atom, binds strongly to the surface and approaches its preferred adsorption site, much closer to the surface. The next key step of the dissociation, during the second dehydrogenation ($n = 2$), corresponds to the abrupt molecular lifting from the surface, resulting in an almost complete internal strain relief (see Supporting Information, Figure S3). In fact, three peripheral C atoms are positioned in equivalently favorable sites, giving therefore three candidate C–H bonds for the next dehydrogenation. But the lifting of the molecule has been found to give an energy gain of 0.67 eV with respect to the other configurations, thereby losing one of the two symmetry axes in the molecule, and causing the candidate C–H bond closest to the previously dehydrogenated one to be broken. At this point, one symmetry axis of the molecule with respect to the substrate still remains. Always keeping in mind that the driving mechanism for dehydrogenation is the proximity of the peripheral C atom to a top site, the two C atoms, adjacent to the freshly dehydrogenated carbons, are found to be equivalent. The evolution of the system towards the $n = 3$ state results in a clear twist of the molecule in one direction, in order to better accommodate the newly dehydrogenated C atom in its favored bond site on the substrate. This causes the loss of the second symmetry axis, and the remaining evolution of the dissociation reaction is fully determined. It is significant that during the dehydrogenation process also the atoms of the Ir substrate are slightly rearranged, in particular becoming more elevated with respect to the average surface plane, which eventually contributes to the formation of a stronger molecule-Ir bond. From this step on, the molecule evolves by gradually completing the rotation to the final adsorption geometry, and by curling towards a dome shape configuration, always losing

H atoms next to dehydrogenated bonds. This gradually changing shape increases the internal strain of the part attached to the substrate, whilst always keeping one end lifted until the very end, allowing for some strain to be relieved in the lifted portion of the molecule during the intermediate steps. The topology during the dissociation process is also affected by the contributions of the σ and π orbital overlap with the substrate d-bands, the former tending to cause the C rings to stand perpendicular to the surface, the latter trying to push the C ring parallel to it. The final state configuration is very similar to what has been found for the initial stages of growth of graphene using ethylene: dome-shaped C nanoislands whose interactions with the Ir substrate take place only at the cluster edges.²⁸ With increasing temperature, the mobility of the nanodomains becomes large enough to permit the formation of larger clusters and eventually graphene.³¹

In order to confirm that the reaction intermediates, including the molecular lifting, twisting and curling, are more favorable with respect to the flat behavior, we simulated the C 1s core level spectra for all the carbon atoms in the most important configurations in Fig. 4 (1, 2, 5, 11, and 12), as well as for coronene, a larger nanodome formed by 54 C atoms - C_{54} , and graphene. The fit results, reported in Fig. 3(c) along with the experimental data, show an excellent agreement: the three-component line shape, associated to the $C_{24}H_{12}$ and $C_{24}H_{11}$ molecules, at 525 K changes and shifts to lower BE, and only when larger nanodomains are formed, because of coalescence and nucleation (see fit result corresponding to C_{54}), the spectral intensity distribution gets narrower. The single peak at 284.1 eV, fingerprint of graphene formation on Ir(111), appears only at high temperature (1080 K). We want to stress that the sequential dehydrogenation is a statistical process, and during the annealing there could be several different $C_{24}H_n$ species present on the surface. The amount of each species depends, besides from the energy barrier, also from pre-exponential factors and diffusion constants among other factors, which are difficult to calculate. In Fig. 3 (c) we report the C 1s core level spectra, along with the single molecular species fit result, for the most important intermediate molecules. Beside the initial and final configurations, corresponding

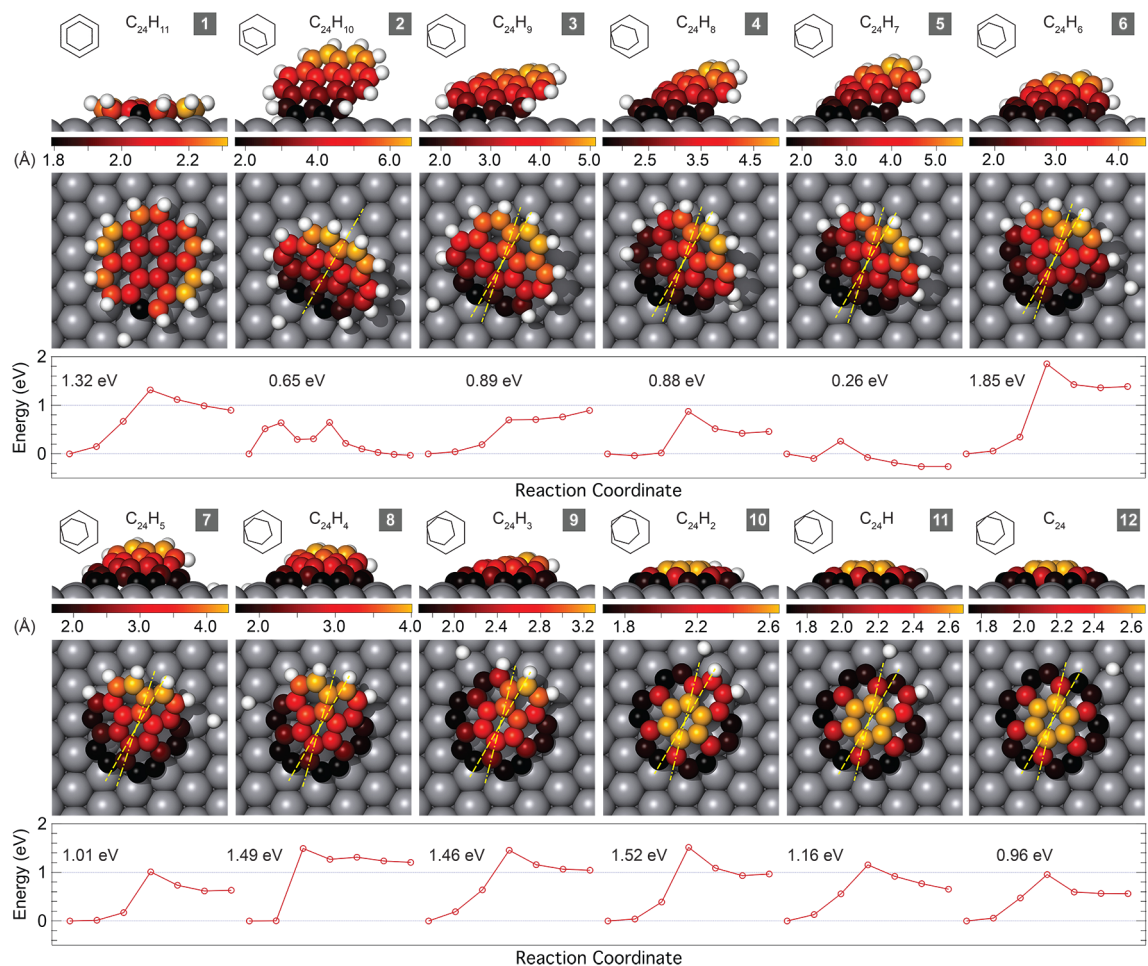


Figure 4: (a) Illustration of final states side and top views of $C_{24}H_n$ through the 12 sequential C–H bond cleavages that bring coronene molecules to the nanodome configuration. Different colors correspond to different C-metal substrate heights. The colors have been rescaled in each step to emphasize the geometrical differences in each configuration (the height scale in Å is also reported for each step of the reaction). Above each illustration, a schematic representation of the coronene to Ir relative orientation is reported. The yellow dash-dotted line in step 2 represents the symmetry axis shared by the adsorbed coronene molecule and the Ir substrate. In steps $n = 3 - 12$, the dashed line represents the original coronene orientation, while the dash-dotted line represents the current one. Below each illustration, a graph representing the energy evolution during the NEB simulations is shown with the total energy barrier for the single dehydrogenation step.

to coronene and the C_{24} nanodome respectively, we report the spectrum associated to $C_{24}H_7$, which has the highest dehydrogenation barrier and is therefore expected to be stable for a larger temperature window on the surface. We also show the fit result associated to the tilted molecule, $C_{24}H_{10}$, since it represents a crucial step in the dissociation pathway, and to

$C_{24}H_{11}$, which represents the first dissociation product. Even though the process illustrated in Fig. 4 is the most energetically favorable, we also tested many other reaction paths that revealed striking differences in terms of geometry, and do not include the process of molecular lifting and curling. For example we found that an alternative possibility is that, after the first dehydrogenation, the molecules break the next C–H bond but still remain flat on the surface. However the final state is 0.67 eV higher in energy. Moreover, the evolution of the reaction following this new sequence would end up with a C_{24} carbon cluster in a geometrical configuration that is by far not energetically favorable. In fact, it is important to stress that the un-rotated C cluster, *i.e.* the one presenting the same azimuthal orientation as the original coronene molecule, has a 2.27 eV higher energy. Notably, the reaction path involving the flat dissociation behavior results in a calculated C 1s core level spectrum with an additional component growing at lower BE (at 282.9 eV), which is not experimentally observed. In fact, during the flat dissociation pathway, some Ir atoms that bind to peripheral C atoms of the partially dissociated molecule, are also bound to another C atom. This produces the theoretically predicted, but not experimentally observed, low BE component in the C 1s spectrum. As a final step of our investigation, we have compared the experimental $C_{24}H_{12}$ coverage evolution with temperature, as extracted from the C 1s photoemission data (see Supporting Information, Figure S4), to a theoretical prediction based on a microkinetic model, assuming that the reaction rate depends on the activation energy of the first C–H bond breaking reaction step through a Boltzmann factor, with an energy barrier deduced from the NEB calculations, allowing only the pre-exponential order of magnitude as a free parameter. The temperature behavior is in good agreement with the experimentally observed decrease of the $C_{24}H_{12}$ population when the pre-exponential factor is equal to 10^{10} . This value, that at first instance seems quite low if compared with the typical value of 10^{13} , is however not surprising. As recently reported by Campbell *et al.* by discussing extensive experimental results,³² ν_{diss} prefactors for dissociation are found typically 1/10 to 1/10⁵ of the prefactors for desorption ν_{des} of the same molecule, with an average ratio (ν_{diss}/ν_{des}) of

about 1/1000, because of entropy loss at the transition state. It is interesting to compare our experimental finding with those obtained for the simplest aromatic molecule, *i.e.* benzene (C_6H_6).³³ Theoretical calculations of benzene adsorption on Cu(100) show that, after the first hydrogen scission starting from its flat adsorption configuration, also the phenyl radical C_6H_5 and the ortho-benzyne species C_6H_4 assume upright configurations. It is interesting to observe that instead, the dissociation of benzene on Pt(111), which is a transition metal quite similar to Ir, results in a tilted configuration with the C atoms adsorbed through C at the top site.³⁴ Our results therefore suggest that the process of thermally activated molecular lifting can be a more general behavior in the dissociation process of similar molecules, also for larger PAHs. We suppose that this is the case for other transition metal surfaces, where the presence of unsaturated C bonds could lead to the development of strong interactions between the partially dehydrogenated molecules and the metal substrates. The main driving force resulting in the tilt and rotation of the molecule before curling, *i.e.* the creation of new bonds with the substrate which breaks the molecular symmetries and allows for some internal strain relief, somehow mimics what happens in the case of heterocyclic aromatic adsorbates such as pyridine, pyrrole and thiophene for which, besides flat-lying configurations, also tilted geometries have been reported.³⁵

Conclusions

We have shown, using a multi-method experimental approach and extensive theoretical calculations, that coronene molecules adsorbed on Ir(111) undergo major conformational changes during the dissociation process, which bring the molecules from a flat, slightly upwards-pointing bowl shape, to graphene, through a series of exotic configurations. Upon the second C–H bond cleavage, the molecules tilt upward with respect to the surface, and then rotate to accommodate the reactive C terminations to the most favorable bond site with the substrate. During the lifting, the C–C strain is initially relieved, while as the dehydrogenation

proceeds, the molecules experience a progressive increase in the average interatomic distance, and gradually settle to form peculiar dome shaped nanographene flakes. By exploiting this reaction mechanism, we envisage the possibility of creating new nanostructures with different functionalities by encapsulating single adatoms below the carbon dome, through diffusion of the new species underneath the carbon disk just before the formation of large carbon clusters, in a similar way as already demonstrated for larger Sn nanoparticles for applications in lithium batteries.³⁶ It is clear that this concept would require an accurate control of the status of the molecular structure, which is a challenging task, but the lifting of the molecule could be an important step to follow this strategy. In fact, the process of intercalation of atomic species below already formed nanographenes, which could have interesting application in spintronics³⁷ and energy storage,³⁸ is strongly hindered by the strong C-metal bonds formed at the edges which prevent the atomic penetration below the C flake. Besides the confinement of magnetic species below nanographene flakes, also the presence of chemically inert species below the nanographenes is expected to modify the electronic edge states giving rise to interesting magnetic phenomena.³⁹ The use of nanostructured surfaces, such as those formed by a high density of monoatomic steps or vacancies, could be a possibility to reduce the geometrical degree of freedom forced by the hexagonal surface of Ir(111), with C_{3v} symmetry, and achieve an easier control of the reaction parameters. Finally, the control of molecular tilt in PAHs could tailor their chemical reactivity, by activating or inhibiting specific reactions that are usually catalyzed by the surface, as reported for example in the case of different carbonyl compounds, more specifically aldehydes, on transition metal substrates.^{40,41}

Supporting Information Available

Angle Resolved Photoemission data, additional DFT calculated adsorption configurations, C-C bond length histograms, microkinetic reaction model.

This material is available free of charge via the Internet at <http://pubs.acs.org/>.

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Notes

The authors declare no competing financial interest.

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References

- (1) Otero, G.; Biddau, G.; Sanchez-Sanchez, C.; Caillard, R.; Lopez, M. F.; Rogero, C.; Palomares, F. J.; Cabello, N.; Basanta, M. A.; Ortega, J.; Mendez, J.; Echavarren, A. M.; Perez, R.; Gomez-Lor, B.; Martin-Gago, J. A. *Nature* **2008**, *454*, 865–868.

- (2) Cai, J.; Ruffieux, P.; Jaafar, R.; Bieri, M.; Braun, T.; Blankenburg, S.; Muoth, M.; Seitsonen, A. P.; Saleh, M.; Feng, X.; Müllen, K.; Fasel, R. *Nature* **2010**, *466*, 470–473.
- (3) Müllen, K.; Rabe, J. P. *Acc. Chem. Res.* **2008**, *41*, 511–520.
- (4) Narita, A.; Wang, X.-Y.; Feng, X.; Müllen, K. *Chem. Soc. Rev.* **2015**, *44*, 6616–6643.
- (5) Son, Y.-W.; Cohen, M. L.; Louie, S. G. *Nature* **2006**, *444*, 347–349.
- (6) Yang, L.; Park, C.-H.; Son, Y.-W.; Cohen, M. L.; Louie, S. G. *Phys. Rev. Lett.* **2007**, *99*, 186801.
- (7) Chen, Y.-C.; de Oteyza, D. G.; Pedramrazi, Z.; Chen, C.; Fischer, F. R.; Crommie, M. F. *ACS Nano* **2013**, *7*, 6123–6128.
- (8) Ruffieux, P.; Cai, J.; Plumb, N. C.; Patthey, L.; Prezzi, D.; Ferretti, A.; Molinari, E.; Feng, X.; Müllen, K.; Pignedoli, C. A.; Fasel, R. *ACS Nano* **2012**, *6*, 6930–6935.
- (9) Fujii, S.; Enoki, T. *Acc. Chem. Res.* **2013**, *46*, 2202–2210.
- (10) Narita, A.; Feng, X.; Müllen, K. *Chem. Rec.* **2015**, *15*, 295–309.
- (11) Treier, M.; Pignedoli, C. A.; Laino, T.; Rieger, R.; Müllen, K.; Passerone, D.; Fasel, R. *Nat. Chem.* **2011**, *3*, 61–67.
- (12) Baraldi, A.; Comelli, G.; Lizzit, S.; Cocco, D.; Paolucci, G.; Rosei, R. *Surf. Sci.* **1996**, *367*, L67–L72.
- (13) Doniach, S.; Šunjić, M. *J. Phys. C: Solid State Phys.* **1970**, *3*, 285.
- (14) Henkelman, G.; Uberuaga, B. P.; Jónsson, H. *J. Chem. Phys.* **2000**, *113*, 9901–9904.
- (15) Köhler, L.; Kresse, G. *Phys. Rev. B* **2004**, *70*, 165405 (1–9).
- (16) Fronzoni, G.; Baseggio, O.; Stener, M.; Hua, W.; Tian, G.; Luo, Y.; Apicella, B.; Alfé, M.; de Simone, M.; Kivimäki, A.; Coreno, M. *J. Chem. Phys.* **2014**, *141*.

- (17) Boschi, R.; Clar, E.; Schmidt, W. *J. Chem. Phys.* **1974**, *60*, 4406–4418.
- (18) Schroeder, P. G.; France, C. B.; Parkinson, B. A.; Schlaf, R. *J. Appl. Phys.* **2002**, *91*, 9095–9107.
- (19) Medjanik, K.; Kutnyakhov, D.; Nepijko, S. A.; Schonhense, G.; Naghavi, S.; Alijani, V.; Felser, C.; Koch, N.; Rieger, R.; Baumgarten, M.; Müllen, K. *Phys. Chem. Chem. Phys.* **2010**, *12*, 7184–7193.
- (20) Wu, Y.-T.; Siegel, J. S. *Chem. Rev.* **2006**, *106*, 4843–4867.
- (21) Tsefrikas, V. M.; Scott, L. T. *Chem. Rev.* **2006**, *106*, 4868–4884.
- (22) Parschau, M.; Fasel, R.; Ernst, K.-H.; Gröning, O.; Brandenberger, L.; Schillinger, R.; Greber, T.; Seitsonen, A.; Wu, Y.-T.; Siegel, J. *Angew. Chem., Int. Ed.* **2007**, *46*, 8258–8261.
- (23) Xiao, W.; Passerone, D.; Ruffieux, P.; Aït-Mansour, K.; Gröning, O.; Tosatti, E.; Siegel, J. S.; Fasel, R. *J. Am. Chem. Soc.* **2008**, *130*, 4767–4771.
- (24) Jaafar, R.; Pignedoli, C. A.; Bussi, G.; Aït-Mansour, K.; Groening, O.; Amaya, T.; Hirao, T.; Fasel, R.; Ruffieux, P. *J. Am. Chem. Soc.* **2014**, *136*, 13666–13671.
- (25) Hagedorn, C. J.; Weiss, M. J.; Weinberg, W. H. *Phys. Rev. B* **1999**, *60*, R14016–R14018.
- (26) Moritani, K.; Okada, M.; Nakamura, M.; Kasai, T.; Murata, Y. *J. Chem. Phys.* **2001**, *115*, 9947–9959.
- (27) Preobrajenski, A. B.; Ng, M. L.; Vinogradov, A. S.; Mårtensson, N. *Phys. Rev. B* **2008**, *78*, 073401.
- (28) Lacovig, P.; Pozzo, M.; Alfè, D.; Vilmercati, P.; Baraldi, A.; Lizzit, S. *Phys. Rev. Lett.* **2009**, *103*, 166101.

- (29) Lizzit, S.; Baraldi, A. *Catal. Today* **2010**, *154*, 68–74.
- (30) Presel, F.; Jabeen, N.; Pozzo, M.; Curcio, D.; Omiciuolo, L.; Lacovig, P.; Lizzit, S.; Alfè, D.; Baraldi, A. *Carbon* **2015**, *93*, 187 – 198.
- (31) Coraux, J.; N’Diaye, A. T.; Engler, M.; Busse, C.; Wall, D.; Buckanie, N.; zu Heringdorf, F.-J. M.; van Gastel, R.; Poelsema, B.; Michely, T. *New J. Phys.* **2009**, *11*, 023006.
- (32) Campbell, C. T.; Árnadóttir, L.; Sellers, J. R. *Z. Phys. Chem.* **2013**, *227*, 1435–1454.
- (33) Lesnard, H.; Bocquet, M.-L.; Lorente, N. *J. Am. Chem. Soc.* **2007**, *129*, 4298–4305.
- (34) Gao, W.; Zheng, W. T.; Jiang, Q. *J. Chem. Phys.* **2008**, *129*.
- (35) Jenkins, S. J. *Proc. R. Soc. London, Ser. A* **2009**, *465*, 2949–2976.
- (36) Cui, G.; Hu, Y.-S.; Zhi, L.; Wu, D.; Lieberwirth, I.; Maier, J.; Müllen, K. *Small* **2007**, *3*, 2066–2069.
- (37) Sicot, M.; Leicht, P.; Zusan, A.; Bouvron, S.; Zander, O.; Weser, M.; Dedkov, Y. S.; Horn, K.; Fonin, M. *ACS Nano* **2012**, *6*, 151–158.
- (38) Xu, F.; Tang, Z.; Huang, S.; Chen, L.; Liang, Y.; Mai, W.; Zhong, H.; Fu, R.; Wu, D. *Nat. Commun.* **2015**, *6*.
- (39) Fujii, S.; Ziatdinov, M.; Ohtsuka, M.; Kusakabe, K.; Kiguchi, M.; Enoki, T. *Faraday Discuss.* **2014**, *173*, 173–199.
- (40) Chiu, M. E.; Watson, D. J.; Kyriakou, G.; Tikhov, M. S.; Lambert, R. M. *Angew. Chem.* **2006**, *118*, 7692–7696.
- (41) Brandt, K.; Chiu, M. E.; Watson, D. J.; Tikhov, M. S.; Lambert, R. M. *J. Am. Chem. Soc.* **2009**, *131*, 17286–17290.

Graphical TOC Entry

