

1 **Electrospun pH-sensitive core-shell polymer nanocomposites**  
2 **fabricated using a tri-axial process**

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41 **Abstract:**

42 A modified tri-axial electrospinning process was developed for the generation of a  
43 new type of pH-sensitive polymer/lipid nanocomposite. The systems produced are  
44 able to promote both dissolution and permeation of a model poorly water-soluble drug.  
45 First, we show that it is possible to run a tri-axial process with only one of the three  
46 fluids being electrospinnable. Using an electrospinnable middle fluid of Eudragit  
47 S100 (ES100) with pure ethanol as the outer solvent and an unspinnable  
48 lecithin-diclofenac sodium (PL-DS) core solution, nanofibers with linear morphology  
49 and clear core/shell structures can be fabricated continuously and smoothly. X-ray  
50 diffraction proved that these nanofibers are structural nanocomposites with the drug  
51 present in an amorphous state. *In vitro* dissolution tests demonstrated that the  
52 formulations could preclude release in acidic conditions, and that the drug was  
53 released from the fibers in two successive steps at neutral pH. The first step is the  
54 dissolution of the shell ES100 and the conversion of the core PL-DS into sub-micron  
55 sized particles. This frees some DS into solution, and later the remaining DS is  
56 gradually released from the PL-DS particles through diffusion. *Ex vivo* permeation  
57 results showed that the composite nanofibers give a more than two-fold uplift in the  
58 amount of DS passing through the colonic membrane as compared to pure DS; 74%  
59 of the transmitted drug was in the form of PL-DS particles. The new tri-axial  
60 electrospinning process developed in this work provides a platform to fabricate  
61 structural nanomaterials, and the core-shell polymer-PL nanocomposites we have  
62 produced have significant potential applications for oral colon-targeted drug delivery.

63 **Keywords:** Tri-axial electrospinning; core-sheath fibers; polymer-lipid  
64 nanocomposites; colon-targeted drug delivery; electrospinnability

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## 70 **1. Introduction**

71 The fabrication of advanced drug delivery systems (DDSs) is increasingly  
72 dependent on the creation of complex architectures and understanding  
73 structure-activity relationships at the nanoscale [1-3]. To this end, core-shell  
74 nanostructures have been very widely studied in the production of functional  
75 nanomaterials, including those for biomedical applications [4-6]. For drug delivery  
76 and controlled release, both the core and shell can be loaded with an active  
77 pharmaceutical ingredient (API) and/or with different types of pharmaceutical  
78 excipients. Applications of such systems include improving the solubility of poorly  
79 water-soluble drugs, controlled release of multiple APIs from a single dosage form, or  
80 tunable multiple phase release [7-9].

81 Over recent years, polymers and lipids have been the most widely used  
82 pharmaceutical excipients, and these materials have acted as the basis for a broad  
83 gamut of novel DDSs, being exploited to alter the biopharmaceutical and  
84 pharmacokinetic properties of the drug molecule for favorable clinical outcomes  
85 [3,10,11]. Numerous core-shell polymeric nanoparticles (NPs) and lipid-based DDS  
86 (such as solid lipid dispersions and liposomes) have been investigated for drug  
87 delivery through varied administration routes [12-15]. Novel strategies derived from  
88 the combined usage of polymers and phospholipids (PLs) have been reported for  
89 some biomedical applications (including controlled release) and are presently of  
90 intense interest in the pharmaceuticals field. However, virtually all the reported  
91 polymer-lipid composites are in the form of microparticles or NPs [4,8,16-18].

92 Core-shell nanofiber-based DDS have received relatively little attention, and to the  
93 best of our knowledge there are no reports of drug-loaded polymer-lipid nanofibers  
94 being used in drug delivery.

95 Electrospun nanofibers, comprising an API loaded into a filament-forming  
96 polymer, have been the focus of much research. They are prepared from a  
97 co-dissolving solution of a drug and polymer; this is ejected from a syringe with  
98 electrical energy used to rapidly evaporate the solvent and yield one-dimensional  
99 fibers with diameters frequently on the nanoscale. This technique is scalable, and  
100 several recent reports address large scale fabrication and the potential for commercial  
101 products [19-22]. The intense research effort invested in these materials thus appears  
102 to be about to yield products which can make a major difference to patients' lives.  
103 Electrospinning is a facile, one-step procedure, and the products form as a visible and  
104 flexible mat which can easily be recovered from the collector without significant loss  
105 of material or damage. The nanofibers produced can further be used as templates to  
106 manipulate molecular self-assembly to create drug-loaded NPs or liposomes; the  
107 electrospinning technique thus provides not only a bridge between fiber-based and  
108 NP-based DDSs, but also between solid and liquid dosage forms [23-26].

109 The most simple, single-fluid, electrospinning process has been explored for  
110 approaching two decades, and the applications of the resultant monolithic nanofibers  
111 have been probed in a wide range of fields. Current developments in electrospinning  
112 are focused in two key areas. The first is the manufacture of electrospun nanofibers on  
113 an industrial scale [27-29]. The second line of research involves developing advanced

114 electrospinning techniques to yield nanofibers with sophisticated structural  
115 characteristics (such as multiple-compartment nanofibers, core-shell nanofibers, or  
116 structured fibers with varied distributions of the API), which in turn impart tunable  
117 and multiple functionalities [30-32]. Because of the popularity of core-shell  
118 nanostructures and the relative ease of the process, coaxial electrospinning (in which  
119 two needles, one nested inside another, are used to handle two working fluids) has  
120 been the focus of much research. Other advanced approaches such as side-by-side  
121 electrospinning (to yield Janus fibers), tri-axial electrospinning (giving three-layer  
122 composites), and other types of multiple-fluid electrospinning have been neglected in  
123 comparison [6,9,33].

124       Compared with single-fluid electrospinning, the standard coaxial experiment has  
125 greatly expanded the range of fibers which can be produced. These include not only  
126 core-shell fibers [34,35], but also fibers prepared from materials without  
127 filament-forming properties [36] and used as templates for creating nanotubes (from  
128 the fiber as a whole) or the “bottom-up” generation of NPs (self-assembled from the  
129 components loaded in the fibers) [26, 37]. For biomedical applications, core-shell  
130 nanofibers proffer a series of new possibilities; for instance, it is possible to protect a  
131 fragile active ingredient such as a protein from the stresses of the electrospinning  
132 processes by confining it to the core, or to vary the APIs concentration in the core and  
133 shell to achieve complex drug release profiles [38-41]. In the traditional coaxial  
134 process the sheath working fluid must be electrospinnable, but a modified process in  
135 which one can utilize unspinnable liquids as the sheath fluid is also possible. The

136 number of polymers which can be directly electrospun is rather limited, but there are  
137 numerous unspinnable liquids, and the modified coaxial process should hence further  
138 expand the range of functional nanofibers which can be produced [38,42,43].

139 The above discussion is focused on the simultaneous processing of two fluids;  
140 working with three or even four fluids simultaneously is also possible, however  
141 [44-49]. For example, Han and Steckl reported tri-layer nanofibers for biphasic  
142 controlled release, using dyes as model active ingredients [49]. In very recent work,  
143 we successfully developed a tri-axial electrospinning process to generate nanofibers  
144 with a gradient distribution of the API, allowing us to achieve zero-order drug release  
145 profiles [31]. However, in all the tri-axial electrospinning processes reported to date,  
146 the three working fluids are all electrospinnable. This limits the applications of the  
147 process. If unspinnable liquids can be processed in combination with spinnable  
148 working solutions, a much broader selection of functional products could be designed  
149 and generated.

150 Building on our previous work developing modified coaxial [38,42,43] and  
151 standard tri-axial electrospinning [50], here we report the first modified tri-axial  
152 electrospinning process. We have used this process to create core-shell fibers  
153 comprising a lipid-drug core and a pH sensitive shell, thereby allowing us to  
154 demonstrate that only an electrospinnable central fluid is required to achieve a  
155 successful tri-axial process. The polymer-lipid nanocomposites produced showed  
156 desirable functional performance in altering the release behavior of the model drug  
157 diclofenac sodium and improving its permeation through the colonic membrane.

## 158 **2 Experimental**

### 159 **2.1. Materials**

160 Eudragit S100 (ES100,  $M_w=135,000$ ), a methacrylic acid/methyl methacrylate  
161 copolymer which only dissolves at  $\text{pH} > 7.0$ , was obtained from Röhm GmbH  
162 (Darmstadt, Germany). Diclofenac sodium (DS, a non-steroidal anti-inflammatory  
163 drug with potent anti-inflammatory, analgesic and antipyretic properties) was  
164 purchased from the Hubei Biocause Pharmaceutical Co., Ltd. (Hubei, China).  
165 Lecithin (PL, extracted from egg yolk, and containing lysophosphatidylcholine,  
166 sphingomyelin, and neutral lipids in minor quantities), N,N-dimethylacetamide  
167 (DMAc), anhydrous ethanol, methylene blue and basic fuchsin were purchased from  
168 the Sinopharm Chemical Reagent Co., Ltd. (Shanghai, China). All other chemicals  
169 used were analytical grade, and water was doubly distilled before use.

### 170 **2.2. Electrospinning**

171 The tri-layer concentric spinneret was homemade. Three syringe pumps  
172 (KDS100, Cole-Parmer, Vernon Hills, IL, USA) and a high-voltage power supply  
173 (ZGF 60kV/2 mA, Shanghai Sute Corp., Shanghai, China) were used for  
174 electrospinning. The collector comprised a flat piece of cardboard wrapped with  
175 aluminum foil. All electrospinning processes were carried out under ambient  
176 conditions ( $21 \pm 5$  °C with a relative humidity of  $47 \pm 5$  %). Experiments were  
177 recorded using a digital camera (PowerShot A490, Canon, Tokyo, Japan). The  
178 spinneret to collector distance was fixed at 15 cm for all experiments.

179 The outer fluid was pure anhydrous ethanol. The middle fluid consisted of 14.0 g

180 ES100 in 100 mL of a mixture of ethanol / DMAc (90:10 v/v). The inner fluid was  
 181 prepared from 3 g PL and 0.6 g DS in 10 mL ethanol. After initial optimization  
 182 experiments, the applied voltage was fixed at 15 kV. To facilitate observation of the  
 183 electrospinning processes, 10 mg/L methylene blue was added to the inner fluid and 5  
 184 mg/L basic fuchsin to the middle fluid. Four different sets of fibers were prepared  
 185 with varied flow rates, as detailed in Table 1.

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187 **Table 1.** Key details of the electrospinning processes and resultant fibers

No.	Process	F <sub>0</sub> <sup>a</sup> (mL/h)	F <sub>M</sub> <sup>a</sup> (mL/h)	F <sub>I</sub> <sup>a</sup> (mL/h)	Morphology <sup>b</sup>	Diameter ( $\mu$ m)
F1	Single	0	3.0	0	Linear	1.27 $\pm$ 0.19
F2		0.5	2.0	0.5	Linear	0.55 $\pm$ 0.06
F3	Tri-axial	0.5	1.6	0.9	Linear, with some beads	0.47 $\pm$ 0.05
F4		1.0	1.6	0.4	Spindles-on-a-string	---

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## 197 2.3. Characterization

### 198 2.3.1. Morphology

199 The morphology of the fibers was determined using a Quanta FEG450 field  
 200 emission scanning electron microscope (FESEM; FEI Corporation, Hillsboro, OR,  
 201 USA). Prior to examination, samples were gold sputter-coated under a nitrogen  
 202 atmosphere to render them electrically conductive. Images were recorded at an  
 203 excitation voltage of 20 kV. The average fiber size was determined by measuring their  
 204 diameters at more than 100 places in FESEM images, using the NIH Image J software  
 205 (National Institutes of Health, MD, USA). To view the cross-sections of sample F2, a  
 206 section of the fiber mat was placed into liquid nitrogen and manually broken before



207 gold coating.

208 Transmission electron microscope (TEM) images of the samples were recorded  
209 on a JEM 2100F field emission TEM (JEOL, Tokyo, Japan). Samples were collected  
210 by fixing a lacey carbon-coated copper grid on the collector and electrospinning  
211 directly onto it for several minutes.

### 212 2.3.2. *Physical form and compatibility*

213 X-ray diffraction (XRD) was conducted using a D/Max-BR diffractometer  
214 (Rigaku, Tokyo, Japan) with Cu K $\alpha$  radiation over the  $2\theta$  range 5 to 60° at 40 kV and  
215 30 mA. Attenuated total reflectance-Fourier transform infrared (ATR-FTIR)  
216 spectroscopy was carried out on a Nicolet-Nexus 670 FTIR spectrometer (Nicolet  
217 Instrument Corporation, Madison, USA) from 500 cm<sup>-1</sup> to 4000 cm<sup>-1</sup> at a resolution  
218 of 2 cm<sup>-1</sup>.

### 219 2.3.3. *In vitro dissolution tests*

220 To determine drug loading efficiency (LE), 0.100 g of the fibers was added into  
221 10 mL of a 10% v/v ethanol solution in water, in order to extract all the loaded DS.  
222 The resultant solutions were diluted using phosphate buffered saline (PBS, pH7.0,  
223 0.1M) to a suitable concentration for UV measurement. The LE was calculated using  
224 the following equation:

$$225 \quad \text{LE(\%)} = (\text{DS mass measured})/(\text{theoretical DS mass in the formulation}) \times 100\%$$

226 *In vitro* dissolution tests were carried out according to the Chinese  
227 Pharmacopoeia (2015 Ed.). Method II, which is a paddle method, was undertaken  
228 using a RCZ-8A dissolution apparatus (Tianjin University Radio Factory, Tianjin,

229 China). 280 mg of fibers F2 or 20 mg of the DS raw material (particle size  $<30\ \mu\text{m}$ )  
230 were first placed in 600 mL of 0.1 M HCl. Two hours later, 2.4 g NaOH was added to  
231 neutralize the dissolution media. The temperature of the dissolution medium was  $37 \pm$   
232  $1\ ^\circ\text{C}$  and the instrument was stirred at 50 rpm. Sink conditions were maintained, with  
233  $C < 0.2C_s$ . At predetermined time points, 5.0 mL aliquots were withdrawn from the  
234 dissolution medium and replaced with distilled water to maintain a constant volume.  
235 After filtration through a  $0.22\ \mu\text{m}$  membrane (Millipore, Billerica, MA, USA) and  
236 appropriate dilution with PBS, samples were analyzed at  $\lambda_{\text{max}} = 276\ \text{nm}$  using a  
237 UV-vis spectrophotometer (UV-2102PL, Unico Instrument Co. Ltd., Shanghai,  
238 China). The cumulative amount of DS released at each time point was back-calculated  
239 from the data obtained against a predetermined calibration curve. Experiments were  
240 performed seven times, and the average results from six of these replicates are  
241 reported as mean  $\pm$  S.D.

242 During the *in vitro* dissolution process, dissolution media from the seventh  
243 replicate was withdrawn and the transmittance at  $\lambda=500\ \text{nm}$  measured using the  
244 UV-vis spectrophotometer. The average hydrodynamic diameter and size distribution  
245 of the particles in the final dissolution medium from these experiments were  
246 determined using a BI-200SM static and dynamic light scattering (SDLC) instrument  
247 (Brookhaven Instruments Corporation, Austin, TX, USA).

#### 248 2.3.4. *Ex vivo* permeation tests

249 *Ex vivo* permeation studies were performed using a RYJ-6A diffusion test  
250 apparatus (Shanghai Huanghai Drug Control Instrument Co., Ltd., Shanghai, China),

251 in which materials were mounted in six Keshary-Chien glass diffusion cells and a  
252 water bath system maintained a constant temperature of  $37 \pm 0.2$  °C. Each cell had a  
253 diffusion area of  $2.60 \text{ cm}^2$ , and the receptor compartment had a capacity of 7.2 mL  
254 PBS (pH7.0, 0.1M). Each donor compartment was filled with 1.0 mL PBS and the  
255 hydrodynamics in the receptor compartment were maintained by stirring at 50 rpm  
256 with a Teflon coated magnetic bead. Large intestines were obtained from pigs after  
257 slaughtering (Baoshan Jiangwan slaughterhouse, Shanghai, China). The intestine was  
258 washed carefully with physiological saline solution (NaCl 0.9% w/v) to remove  
259 non-digested food. The colonic membranes were peeled away from the intestines and  
260 fixed on diffusion cells with the mucosal walls upward. They were equilibrated at  
261  $35$  °C for 30 min before permeation tests.

262 The F2 fibers (140 mg) were placed on the mucosal surface in the chambers.  
263 Samples (1 mL) were withdrawn from the receptor compartment at timed intervals  
264 and 1 mL fresh PBS was added to maintain the volume of fluid here at a constant  
265 level. The aliquots were filtered through a  $0.22 \mu\text{m}$  membrane (Millipore, Billerica,  
266 MA, USA). The absorption of the filtrate was measured at 276 nm to determine the  
267 amount of DS present in the aqueous phase. The semi-solid residue was dissolved  
268 using 10 mL of a 10% v/v ethanol solution in water and diluted with PBS before  
269 measuring absorbance, in order to determine its DS content of. All measurements  
270 were carried out in triplicate. Permeation experiments with 10 mg of pure DS (particle  
271 size  $<30 \mu\text{m}$ ) as a control were also carried out.

#### 272 2.4. Statistical analysis

273 The experimental data are presented as mean  $\pm$  SD. The results from the *in vitro*

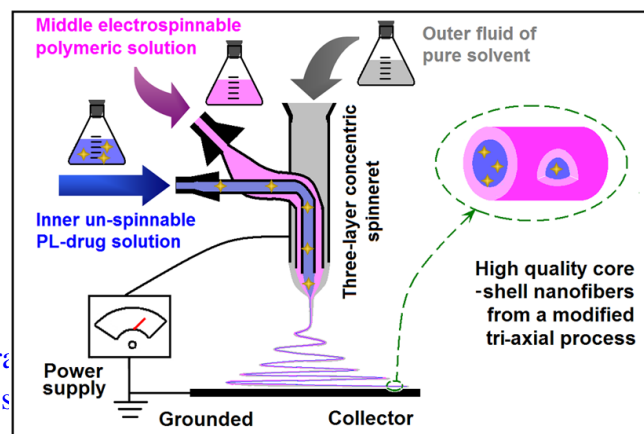
274 dissolution tests and *ex vivo* permeation tests were analyzed using one-way ANOVA.  
275 The threshold significance level was set at 0.05. Thus, *p* (probability) values lower  
276 than 0.05 were considered statistically significant.

### 277 3. Results and discussion

#### 278 3.1. Implementation of modified tri-axial electrospinning

279 A diagram illustrating the modified tri-axial electrospinning process is shown in  
280 Fig. 1. The system consists of four components: three syringe pumps to drive the  
281 working fluids, a power supply, a fiber collector, and a three layer concentric  
282 spinneret. In modified coaxial electrospinning, the use of a spinnable core solution  
283 can ensure a successful process regardless of the electrospinnability of the sheath fluid  
284 [43]. Here, the central solution is electrospinnable, and this is utilized to achieve  
285 tri-axial electrospinning even though the outer fluid is pure solvent and the inner fluid  
286 is unspinnable.

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298 **Fig. 1** A diagram illustrating the modified tri-axial electrospinning process and its use for  
299 preparing core-shell nanofibers.

300 The homemade tri-concentric spinneret and its connection with the power supply  
301 and three working fluids are shown in Fig. 2a. An alligator clip was used to connect  
302 the power supply to the spinneret, which was directly fixed to the syringe holding the

303 outer fluid. The middle and inner fluids were connected to the spinneret through  
304 high-elastic silicon tubing.

305 The design of the spinneret is of critical importance in ensuring a robust and  
306 reproducible electrospinning process [42,48]. A well-designed spinneret must provide  
307 a suitable template for producing the desired nanofiber architectures, and must be  
308 developed bearing in mind the behavior of the working fluids under an electrical field.  
309 The spinneret used in this work is exhibited in the top-right and bottom-right insets of  
310 Fig. 2a. It consists of three concentric capillaries composed of austenitic stainless steel  
311 ( $O_6Cr_{19}Ni_{10}$ , GB24511 in China). The inner, middle and outer capillaries have outer  
312 diameters of 0.4, 1.6, 2.8 mm and inner diameters ( $D_i$ ) of 0.20, 1.3 and 2.2 mm,  
313 respectively. The end of the inner capillary projects 0.2 mm out of the central one,  
314 which similarly projects 0.2 mm from the outer capillary. This design helps to ensure  
315 the encapsulation of the inner fluid by the middle fluid, and in turn the middle by the  
316 outer fluid. This structure should also help to prevent mixing of the working fluids  
317 when they are ejected from the spinneret.

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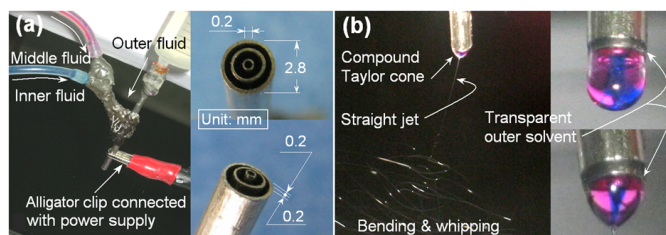
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326 **Fig. 2.** The implementation of modified tri-axial electrospinning: (a) the connection of  
327 the spinneret with the power supply and the working fluids (left), and images of the  
328 tri-concentric spinneret (insets); (b) a digital photograph of the tri-axial process (left),  
329 the tri-layer droplet before a voltage of 15 kV was applied (top-right) and the  
330 compound Taylor cone (bottom-right).

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332 Under optimised conditions (see Section 2.2), successful electrospinning could  
333 be achieved as shown in Fig. 2b. The process involves three steps including Taylor  
334 cone formation, the emission of a straight fluid jet and then an unstable region with  
335 gradually enlarged bending and whipping loops. The top-right inset of Fig. 2b  
336 displays the droplets ejected from the spinneret with no voltage applied. Both the blue  
337 inner fluid and pink middle fluids were observed to diffuse into the outer fluid to  
338 some extent, as demonstrated by their gradually increased sizes and decreased size of  
339 the outer (colourless) solvent moving away from the spinneret. However, the three  
340 working fluids form a clear three-layer compound Taylor cone when a voltage of 15  
341 kV was applied, as shown in the bottom-right inset of Fig. 2b.

342 The modified tri-axial electrospinning process can be run continuously and  
343 smoothly, without any clogging or other adverse phenomena arising. These are  
344 frequently encountered in traditional single-fluid and coaxial electrospinning [50], but  
345 spinning with a pure solvent as the exterior fluid has been shown to reduce incidents  
346 of clogging as well as to improve the uniformity of the fibers produced in the latter

347 process [42]. The use of pure solvent as the outer layer will: 1) lubricate the spinneret  
348 to retard clinging; 2) prevent the formation of semi-solid substance on the surface of  
349 the fluid jets; 3) protect the inner fluid from any environmental fluctuations; and, 4)  
350 lead to a longer drawing period under the electrical field, and thus to narrower fibers.

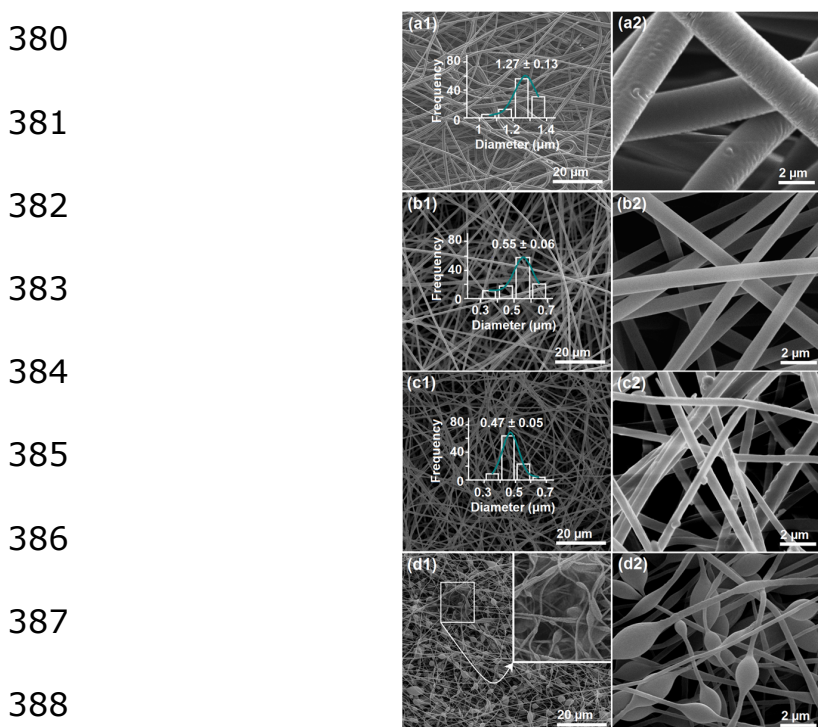
### 351 *3.2. Morphology and core-shell structures of the created nanofibers*

352 FESEM images of fibers F1 to F4 are shown in Fig. 3. When the inner and  
353 outer fluids were turned off, a traditional single-fluid electrospinning of the middle  
354 ES100 solution could be achieved. Although manual intervention was needed  
355 periodically to remove semi-solid substances which collected on the spinneret, the  
356 resultant ES100 fibers were linear without any beads-on-a-string or  
357 spindles-on-a-string morphology (Fig. 3a1 and 3a2). These fibers have an average  
358 diameter of  $1.27 \pm 0.13 \mu\text{m}$ , with an uneven and wrinkled surface (Table 1, Fig. 3a2).  
359 This is a result of barometric pressure, when residual solvent which was not  
360 evaporated during electrospinning escaped from the fibers. Single-fluid  
361 electrospinning easily traps solvent in the fibers because of the formation of a solid  
362 “skin” on the fluid jet during the solidification process.

363 The F2 fibers are linear with an average diameter of  $0.55 \pm 0.06 \mu\text{m}$  and smooth  
364 surfaces (Fig. 3b1 and 3b2, Table 1). This can be attributed to the surrounding outer  
365 solvent and appropriate selection of the flow rates of the three working fluids (0.5, 2.0  
366 and 0.5 mL/h for outer, middle and inner fluids, respectively). If the flow rate of the  
367 outer solvent is kept constant and those of the middle and inner fluid altered to 1.6  
368 and 0.9 mL/h respectively, the resultant F3 material has many beads clinging to the

369 fibers, although the latter are still linear with an average size of  $0.47 \pm 0.05 \mu\text{m}$  (Fig.  
370 3b1 and 3b2, Table 1). It is thought that this high flow rate of the inner fluid causes it  
371 to penetrate the middle and outer fluids to form round PL-DS beads on the fiber  
372 surfaces.

373 If the flow rate of the outer solvent is doubled to 1.0 mL/h, the fibers generated  
374 exhibit a typical spindles-on-a-string morphology (Fig. 3d1 and 3d2). Some  
375 unexpected clumps are also formed within the fiber mat, as shown in the inset of Fig.  
376 3d1. These are ascribed to PL escaping from the inner fluids. A further increase of the  
377 outer solvent flow rate was found to result in an electrospaying process. These results  
378 demonstrate that the selection of flow rates is a key parameter which must be  
379 controlled to ensure the formation of a core-shell nanostructure.

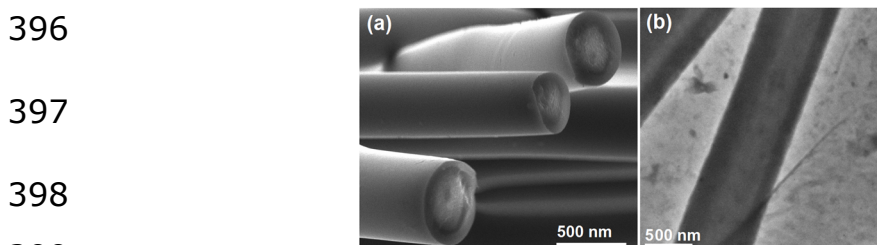


389 **Fig. 3.** FESEM images of the core-shell nanofibers and their size distributions; (a1  
390 and a2) F1; (b1 and b2) F2; (c1 and c2) F3; (d1 and d2) F4, the inset shows a clump  
391 of PL-DS.

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393 FESEM images of cross-sections of F2 (Fig. 4a) and TEM images (Fig. 4b)  
394 demonstrate that the fibers have clear core/shell structures. Both the FESEM and  
395 TEM images suggest that the PL-DS core has a diameter of approximately 300 nm.



400 **Fig. 4.** (a) A FESEM and (b) TEM image of the cross-sections of F2.

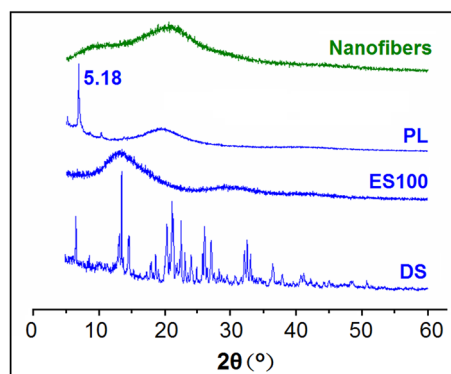
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### 402 3.3. Physical form and component compatibility

403 XRD data are depicted in Fig. 5; these clearly demonstrate that DS is crystalline,  
404 with many sharp Bragg reflections visible in its pattern. ES100 exhibits only a broad  
405 hump, characteristic of an amorphous material. PL exists as a paste at an ambient  
406 temperature of 21 °C, yet shows a sharp reflection at  $2\theta=5.18^\circ$ . This suggests that  
407 there are liquid crystals present in the PL paste, with an ordered lamellar structure as  
408 reported in the early literature [51]. All reflections from PL and DS are absent in the  
409 patterns of the core-shell F2 fibers, suggesting the formation of an amorphous PL-DS  
410 complex.

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**Fig. 5.** XRD patterns of the raw materials (PL, EL100 and DS) and F2.

The potential secondary interactions between the fiber components were investigated using ATR-FTIR, and the results are shown in Fig. 6. DS has three characteristic peaks at 1574, 1553 and 1507  $\text{cm}^{-1}$  arising from its benzene rings. In the spectrum of PL, the  $\text{CH}_2$  symmetric and asymmetric vibrations at 2854  $\text{cm}^{-1}$  and 2923  $\text{cm}^{-1}$  and the antisymmetric stretch of  $\text{N}^+(\text{CH}_3)_3$  at 968  $\text{cm}^{-1}$  comprise the most prominent features. These peaks similarly appear in the spectrum of the fibers, confirming the presence of PL with ES100. However, the characteristic peaks from the benzene rings of DS cannot be seen in the F2 spectrum. This can be attributed to secondary interactions between PL and DS. Hydrophobic interactions, in addition to possible hydrogen bonding and electrostatic interactions, can arise between all three components in F2, as is clear from a consideration of the molecular structures in Fig. 6. These secondary interactions should ensure that the drug and excipients are highly compatible, favorable for the stability of the core-shell nanocomposites.

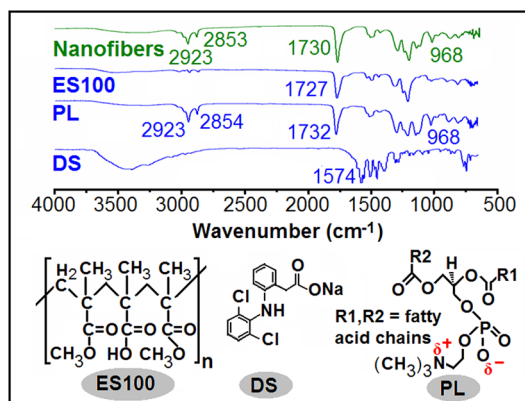
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439 **Fig. 6.** ATR-FTIR spectra and the molecular formulae of the fiber components.

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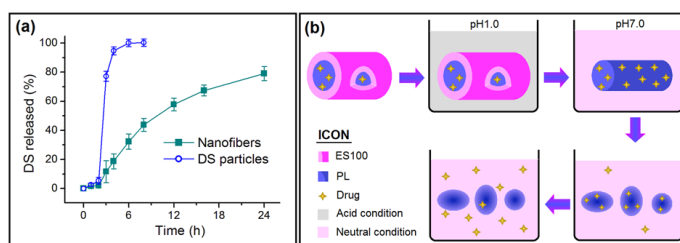
#### 441 3.4. Functional performance

442 DS has a maximum absorbance at 276 nm, which was used to construct a  
443 calibration curve:  $A=0.0085+0.0279 C$  ( $R=0.9997$ ) within a linear range from 2 to 50  
444  $\mu\text{g/mL}$ . The drug content in F2 was first assayed, and found to be  $7.26 \pm 0.31\%$  ( $n =$   
445 6), almost identical to the calculated value of 7.14%.

446 The *in vitro* release profiles of F2 and the DS starting material are shown in Fig.  
447 7a. DS is virtually insoluble in acidic conditions, with a small increase in solubility  
448 when the pH is raised to neutral. After 2 h in acid, 2.8% of DS from the raw material  
449 was freed into the dissolution media. When the pH was raised to neutral, the DS  
450 particles gradually dissolved over *ca.* 3 hours. For F2, 2.1% of the loaded DS was  
451 released during the first 2 h. In the neutral dissolution media, the nanofibers released a  
452 total of 79.1% of the incorporated DS over 22 h.

453 ES100 is a pH-sensitive polymer, and is insoluble at pHs below 7.0; it can thus  
454 be used to target DS to the colon region. DS is a popular API for oral administration  
455 and is frequently used in the treatment of pain and peri-operatively. However, it can  
456 easily result in an anaphylactic reaction, and to an allergic reaction in the digestive

457 tract [52,53]. With traditional electrospun nanofibers, the drug is released by diffusion  
 458 through an insoluble polymeric matrix, or by an erosion mechanism from a  
 459 water-soluble carrier (or a combination of both processes) [39,43]. Here the drug  
 460 release from the core-shell composites is expected to include two successive steps  
 461 (Fig. 7b). First, dissolution of the pH-sensitive ES100 shell will occur, with some  
 462 diffusion of DS from the insoluble core PL. After dissolution of the shell ES100, the  
 463 core PL-DS is not thought to be able to endure the shear forces of stirring applied  
 464 during the experiment and thus we propose that the core is broken up into PL-DS  
 465 particles. The DS is then gradually released from the resultant DS-PL aggregates.  
 466 Thus in the dissolution tests, the released drug (%) corresponds to the DS molecules  
 467 which are in solution (the DS-PL aggregates in suspension are removed by filtration).



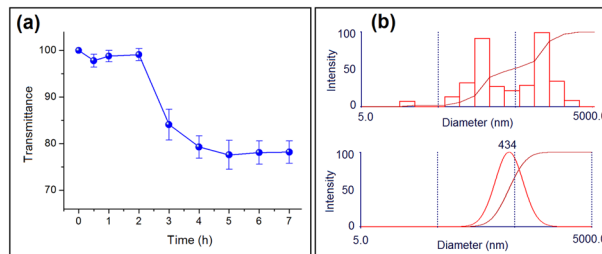
476 **Fig. 7.** *In vitro* dissolution of DS and D2 (a) and the proposed drug release  
 477 mechanism (b).

478

479 To further investigate the drug release mechanism and validate this hypothesis,  
 480 the transmittance of the dissolution media and light scattering studies were performed.  
 481 The changes in transmittance at  $\lambda=500$  nm are shown in Fig. 8a. DS has no  
 482 absorbance above 320 nm, and thus any turbidity of the dissolution media recorded at  
 483 this wavelength must result from the formation of a PL-DS suspension. In the first 2 h,

484 the transmittance remains virtually constant. After the pH is raised to neutral, the  
485 transmittance values decreased for 3 h, after which they level out at around 77%. This  
486 is consistent with the dissolution of the shell ES100 occurring over this period and  
487 resulting in PL-DS nanoparticles.

488 The SDLC results obtained on the final dissolution medium are given in Fig. 8b.  
489 The PL-DS particles formed have an average diameter of 434 nm with a  
490 polydispersity index (PDI) of 0.187.



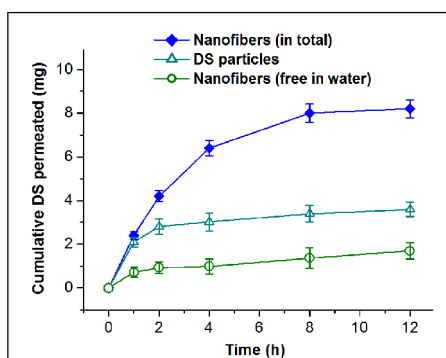
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**Fig. 8.** (a) The transmittance of the dissolution medium measured at 500 nm as a function of time and (b) the sizes of the PL-DS particles measured by SDLC at the end of the dissolution experiment.

497 The results of permeation tests are presented in Fig. 9. DS is a Biopharmaceutical  
498 class II drug, meaning it is poorly water-soluble but is able to effectively cross fatty  
499 membranes [54]. After 12 h only 3.7 mg of the pure drug was transmitted into the  
500 receptor cells. The dissolution of DS is very slow because there is only a very limited  
501 amount of aqueous medium in the donor cell (*cf.* the dissolution experiments, which  
502 are performed under sink conditions) [55]. For the F2 fibers, both dissolved DS and  
503 the PL-DS particles penetrate the bio-membranes into the receptor cells [56].  
504 Although the core-shell nanofibers provided sustained release of DS in dissolution  
505 studies (much slower than the release from pure DS), after 12 h  $8.1 \pm 0.46$  mg DS

506 from F2 had entered the receptor chamber. Of this amount,  $1.7 \pm 0.23$  mg was present  
507 in the aqueous phase (or in particles below 220 nm in size, which could pass through  
508 the filters used). This suggests that  $(8.1-1.7)/8.1 \times 100 = 79\%$  of the DS penetrated  
509 through the mucosal membrane in the format of PL-DS particles. For oral  
510 administration applications, this drug delivery route should alleviate any potential  
511 allergic reactions with the digestive tracts.

512 Many commercial tablets are essentially a physical mixture of drug powders and  
513 polymeric carrier, the latter being added to modulate the drug release behavior. The  
514 combined use of polymer and lipid in the fibers prepared in the work is able to both  
515 protect the API from release in the stomach and provide sustained release in the  
516 colonic region, and also ensure improved trans-membrane permeability, leading to  
517 more effective absorbance. This strategy is particularly useful for oral delivery of  
518 Class IV drugs (which are both poorly water-soluble and have poor permeation  
519 properties). Drug-loaded electrospun fibers can easily be converted into routine oral  
520 solid dosage forms such as tablets and capsules using traditional pharmaceutical  
521 protocols [57-59].



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524

**Fig. 9.** *Ex vivo* permeation profiles of the F2 fibers and pure DS (n=6).

525 *3.5. Perspectives*

526 Coaxial electrospinning is often regarded as a major breakthrough in this field  
527 [60,61]. The fact that only one of the working fluids needs to be electrospinnable for a  
528 successful coaxial process significantly widens the range of materials which can be  
529 processed, and a very broad family of core-shell nanostructures can be produced.  
530 There are only about 100 polymers which can be directly electrospun into fibers, and  
531 often these can only be processed within a narrow window of conditions  
532 (concentration, voltage, etc). The introduction of unspinnable fluids in the modified  
533 coaxial processes greatly expands the capability of this simple technology to produce  
534 nanoscale products from a large range of raw materials. Furthermore, modified  
535 coaxial electrospinning permits all types of liquid phase (including solvents, small  
536 molecule solutions, dilute polymer solutions, suspensions and also emulsions) to be  
537 processed.

538 In this work, we report the first example of modified tri-axial electrospinning.  
539 Similar to modified co-axial spinning, this moves technology beyond the traditional  
540 tri-axial process in which all three working fluids are required to be individually  
541 electrospinnable. In our work, two of the three fluids were unspinnable alone: an  
542 electrospinnable middle layer fluid is sufficient to ensure a successful tri-axial process.  
543 This proof-of-concept work indicates that there are many possibilities in developing  
544 functional nanofibers through the introduction of unspinnable outer-layer and  
545 inner-layer working fluids into tri-axial processes.

546 The feasibility of the different tri-axial electrospinning processes which can be

547 conceived are summarized in Fig. 10. A process with three spinnable working fluids  
 548 (Process I) has been reported in several publications [31,44,45]. Processes II, III and  
 549 IV have two of the three fluids being electrospinnable, and these are feasible provided  
 550 the working fluids are compatible. This report is an example of Process V, with a  
 551 spinnable middle layer fluid used to support unspinnable outer and inner fluids. For  
 552 processes VI and VII, the two unspinnable fluids are adjacent to each other. This may  
 553 result in diffusion of the solutes and formation of a mixture of the two unspinnable  
 554 liquids, and thus it is anticipated that such to tri-axial electrospinning processes will  
 555 result in failure.

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Process	Working fluid			Spinnable fluids	Feasibility	Fiber products	
	Inner	Middle	Outer				
Triaxial electrospinning	I				3	✓	
	II				2	✓	
	III				2	✓	
	IV				2	✓	
	V				1	✓	
	VI				1	✗	✗
	VII				1	✗	✗
Icon	Electro-spinnable	Un-spinnable	Feasible	None or infeasible	Tri-layer	Core-sheath	

567 **Fig. 10.** The feasibility of different tri-axial electrospinning processes.

568

#### 569 4. Conclusions

570 A modified tri-axial electrospinning process was successfully implemented to create  
 571 core-shell nanofibers, in which a spinnable Eudragit S100 (ES100) solution was used  
 572 as the middle fluid to support the outer solvent and an unspinnable phosphatidyl  
 573 choline (PL)/diclofenac sodium (DS) inner solution. This resulted in a continuous and  
 574 trouble-free nanofabrication process. The resultant core-shell nanofibers have a linear



575 morphology with an obvious core-shell structure. XRD demonstrated that the  
576 nanofibers are structural nanocomposites with both the drug DS and also the lipid  
577 carrier PL losing their original crystalline physical forms and being transferred into an  
578 amorphous state. These core (PL-DS)-shell (ES100) nanostructures can protect the  
579 drug from release in acidic conditions to give colon-targeted release. They release the  
580 drug through two successive steps at neutral pH: first, dissolution of the shell ES100  
581 occurs, which is believed to generate PL-DS sub-micron sized particles. Subsequently,  
582 release of DS from the particles occurs. The composite nanofibers lead to more than  
583 twice as much drug permeation through the colonic bio-membrane when compared  
584 with pure DS. The tri-axial electrospinning process developed in this work should  
585 provide a new platform to fabricate structural nanomaterials, and polymer-PL  
586 nanocomposites such as those prepared here can be utilized for effective oral drug  
587 delivery.

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594 Electrical Power Materials, Shanghai Municipal Commission of Education.

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## 770 **Table and Figures Legend**

771

772 **Table 1.** Key details of the electrospinning processes and resultant fibers

773 **Fig. 1.** A diagram of the modified tri-axial electrospinning process and its use for  
774 preparing core-shell drug-loaded nanofibers.

775 **Fig. 2.** The implementation of modified tri-axial electrospinning: (a) the connection of  
776 the spinneret with the power supply and the working fluids (left), and images of the  
777 spinneret (insets); (b) a digital photograph of the tri-axial process (left), the droplet  
778 before a voltage of 15 kV was applied (top-right) and the compound Taylor cone  
779 (bottom-right).

780 **Fig. 3.** FESEM images of the core-shell nanofibers and their size distributions; (a1  
781 and a2) F1; (b1 and b2) F2; (c1 and c2) F3; (d1 and d2) F4. The inset to (d1) shows a  
782 clump of PL-DS.

783 **Fig. 4.** (a) A FESEM image of the cross-sections of F2 and (b) a TEM image showing  
784 the same.

785 **Fig. 5.** XRD patterns of the raw materials (PL, EL100 and DS) and F2.

786 **Fig. 6.** ATR-FTIR spectra and the molecular formula of the fiber components.

787 **Fig. 7.** *In vitro* dissolution of DS and D2 (a) and the proposed drug release  
788 mechanism (b).

789 **Fig. 8.** (a) The transmittance of the dissolution medium measured at 500 nm as a  
790 function of time and (b) the sizes of the PL-DS particles measured by SDLC at the  
791 end of the dissolution experiment.

792 **Fig. 9.** *Ex vivo* permeation profiles of the F2 fibers and pure DS (n=6). It should be  
793 noted that it is not possible in the permeation experiment to distinguish between drug  
794 in solution and in very small particles (< 220 nm) which could pass through the  
795 filtration membrane used. Thus, some portion of the DS which had permeated in the  
796 “free in water” experiment could in fact be in very small nanoparticles.

797 **Fig. 10.** The feasibility of different tri-axial electrospinning processes.

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