

Single-Atom Catalysis

Subsurface Single-Atom Catalyst Enabled by Mechanochemical Synthesis for Oxidation Chemistry

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Abstract: Single-atom catalysts have garnered significant attention due to their exceptional atom utilization and unique properties. However, the practical application of these catalysts is often impeded by challenges such as sintering-induced instability and poisoning of isolated atoms due to strong gas adsorption. In this study, we employed the mechanochemical method to insert single Cu atoms into the subsurface of Fe₂O₃ support. By manipulating the location of single atoms at the surface or subsurface, catalysts with distinct adsorption properties and reaction mechanisms can be achieved. It was observed that the subsurface Cu single atoms in Fe₂O₃ remained isolated under both oxidation and reduction environments, whereas surface Cu single atoms on Fe₂O₃ experienced sintering under reduction conditions. The unique properties of these subsurface single-atom catalysts call for innovations and new understandings in catalyst design.

Introduction

Single-atom catalysts^[1] (SACs) have received significant attention due to their high atom utilization and unique properties.^[1–2] SACs show great potential in various applications, including CO oxidation,^[3] hydrogenation,^[4] organic reactions,^[5] as well as electrocatalytic^[6,7] and photocatalytic reactions.^[8] The catalytic performance of isolated metal atoms is determined by their locations on supports and the associated coordination environment,^[9] with current efforts mainly focused on developing surface single-atom catalysts. However, their application is often limited by poor stability resulting from sintering^[10] and the poisoning of isolated atoms by strong adsorption of impurity gases.^[11] In this regard, tuning the local atomic configuration is a promising method for achieving stable and enhanced catalytic performance.

Ammonia (NH₃) plays a crucial role in the selective catalytic reduction (NH₃-SCR) process, which is used to reduce harmful nitrogen oxide (NO_x) emissions from diesel vehicle exhaust and power plants.^[12] However, in SCR systems, excess NH₃ is needed, where NH₃ is not fully utilized and is released into the environment. Selective catalytic oxidation of ammonia (NH₃-SCO) is one of the most promising methods to remove NH₃ from the exhaust

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gas, which will play a crucial role in the upcoming EU7 standard.^[13] The diesel exhaust generally has a temperature of 500 to 700 °C at 100 % load and normally discharges at a temperature of around 420 °C.^[14] In the downstream NH₃-SCO process following NH₃-SCR, high-performance catalysts are essential to achieve complete conversion of NH₃ to N₂ and water (H₂O), particularly at low temperatures (T < 400 °C).

Noble metals supported on metal oxides have been widely recognized for their high catalytic activity in NH₃ oxidation at low temperatures.^[15] However, a major challenge associated with these catalysts is the tendency for over-oxidation, leading to the formation of nitrogen oxides.^[16] To replace noble metals, the development of transition metal catalysts with excellent low-temperature activity and good stability has become a central topic in NH₃-SCO. However, even the more selective options among these catalysts, such as Cu, have not exhibited sufficient activity to be practically viable.^[17] In recent years, iron oxide (Fe₂O₃) has been proposed as a promising material for deNO_x applications, due to its superior performance in medium- and high-temperature ranges (> 300 °C) and good SO₂ resistance in the NH₃-SCR reaction.^[18] Kang et al.^[19] achieved selective activation of oxygen by employing a cerium (Ce) cation framework to reduce the energy of 3d orbitals of isolated copper (Cu) sites. In our previous study,^[20] it was observed that the presence of atomic Cu–O–Fe enhances NH₃ conversion on Fe₂O₃ surfaces. Remarkably, with the lowest 1s→3d transition energy, atomic Cu–O–Fe achieves a twofold increase in NH₃ conversion in comparison to CuO clusters at 573 K. Furthermore, it maintains high N₂ selectivity at 673 K under conditions involving an inlet NH₃ concentration of 5000 ppm. Cu single atoms over Fe₂O₃ can achieve high NH₃ conversion and N₂ selectivity simultaneously at high temperatures, while the low-temperature activity was much poorer than that of noble-metal catalysts.

Here, we inserted single Cu atoms into the subsurface of Fe₂O₃ support via the mechanochemical method.^[21] The subsurface single atom reveals remarkably different dynamics and adsorption behaviours under reaction conditions. Subsurface Cu sites in the lattice created O vacancies, facilitating the activation of oxygen and redox of surface Fe₂O₃. Compared with surface Cu single atoms, subsurface Cu single atoms exhibited high resistance against sintering and resilience to NO poisoning. Additionally, the subsurface Cu atoms remained dispersed under both oxidizing and reducing environments at high temperatures, yielding 100 % NH₃ conversion at 513 K, which is 3.7 times higher than that of surface Cu atoms catalysts. These results provide a new strategy for the rational design of active and stable single-atom catalysts for practical applications. The ability to control the location of single atoms via simple mechanochemical methods highlights the potential of this subsurface single-atom catalyst.

Results and Discussion

The fresh and surface CuO–Fe₂O₃ catalyst was prepared by the coprecipitation method with 1 wt % Cu. Subsequently, the Cu single atoms were incorporated into the subsurface via ball milling, as depicted in Figure 1a. Microwave plasma-atomic absorption spectroscopy (MP-AES) confirmed that the Cu ratio of the ball-milled catalyst was similar to that in the fresh catalyst (Figure S1). Due to the similar atomic numbers of Cu and Fe, it is not possible to observe Cu atoms on the Fe₂O₃ support in STEM. Energy-dispersive X-ray spectroscopy (EDX) mappings showed that Cu species were homogeneously dispersed in both fresh and ball-milled samples (Figure 1b, Figures S2 and S3). Extended X-ray absorption fine structure (EXAFS) spectra further demonstrated atomically dispersed Cu, with no evidence of a Cu–Cu scattering path in the spectra of both fresh and ball-milled samples (Figure 1c, fitting results in Figure S4 and Table S1). To distinguish Cu–Fe from Cu–Cu, we applied continuous Cauchy wavelet analysis (CCWT) to the k²-weighted EXAFS spectra (Figure S5 and Supporting Note 1), since CCWT has proven to be a very useful tool for obtaining structural information from complex systems.^[22] An increase in the coordination number of Cu–Fe (from 2.2 to 2.7, Table S1) implied that Cu single atoms penetrated the interior of the Fe₂O₃.

Surface-sensitive X-ray photoelectron spectroscopy (XPS; Figures S6 and S7) revealed the presence of Cu in the fresh catalyst, but could not detect a clear Cu signal in the ball-milled catalyst, indicating that the isolated Cu atoms were not located on the surface of Fe₂O₃. The inelastic mean free path (IMFP) of Cu 2p_{3/2} electrons in a matrix of Fe₂O₃ copper is 1.3 nm as calculated using the TPP-2M model.^[23] Near edge X-ray absorption fine structures (NEXAFS) spectra for Cu L-edge were measured using both total electron yield (TEY) and Auger electron yield (AEY). TEY provides information about chemical environments with depth sensitivities of 2–5 nm, while AEY is more surface-sensitive and provides information within 1 nm.^[24] The AEY (Figure 1d) could not effectively detect the Cu signal for the ball-milled catalyst, while the TEY (Figure 1e) showed the presence of Cu in the deeper region. The difference between AEY and TEY suggests that Cu was not present on the surface of Fe₂O₃ but was located in the subsurface after ball milling, consistent with the XPS and EXAFS results. These findings confirm that the surface Cu single atoms can be inserted into the subsurface of Fe₂O₃ support through ball milling. To further confirm the position of subsurface Cu in Fe₂O₃, the formation energies were calculated to determine the doping sites of the Cu atom in Fe₂O₃, and the replacement of Fe atoms is a stable structure (Figure S8 and Supporting Note 2).

The physical adsorption showed that the surface area of the catalyst decreased after ball milling (Figure S9), which could be attributed to pore collapse. X-ray diffraction (XRD; Figure S10) patterns of both fresh CuO–Fe₂O₃ catalyst and ball-milled catalyst showed an ordered α -Fe₂O₃ phase. The half-peak width of the ball-milled CuO–Fe₂O₃ increased, indicating a smaller crystalline domain after ball-

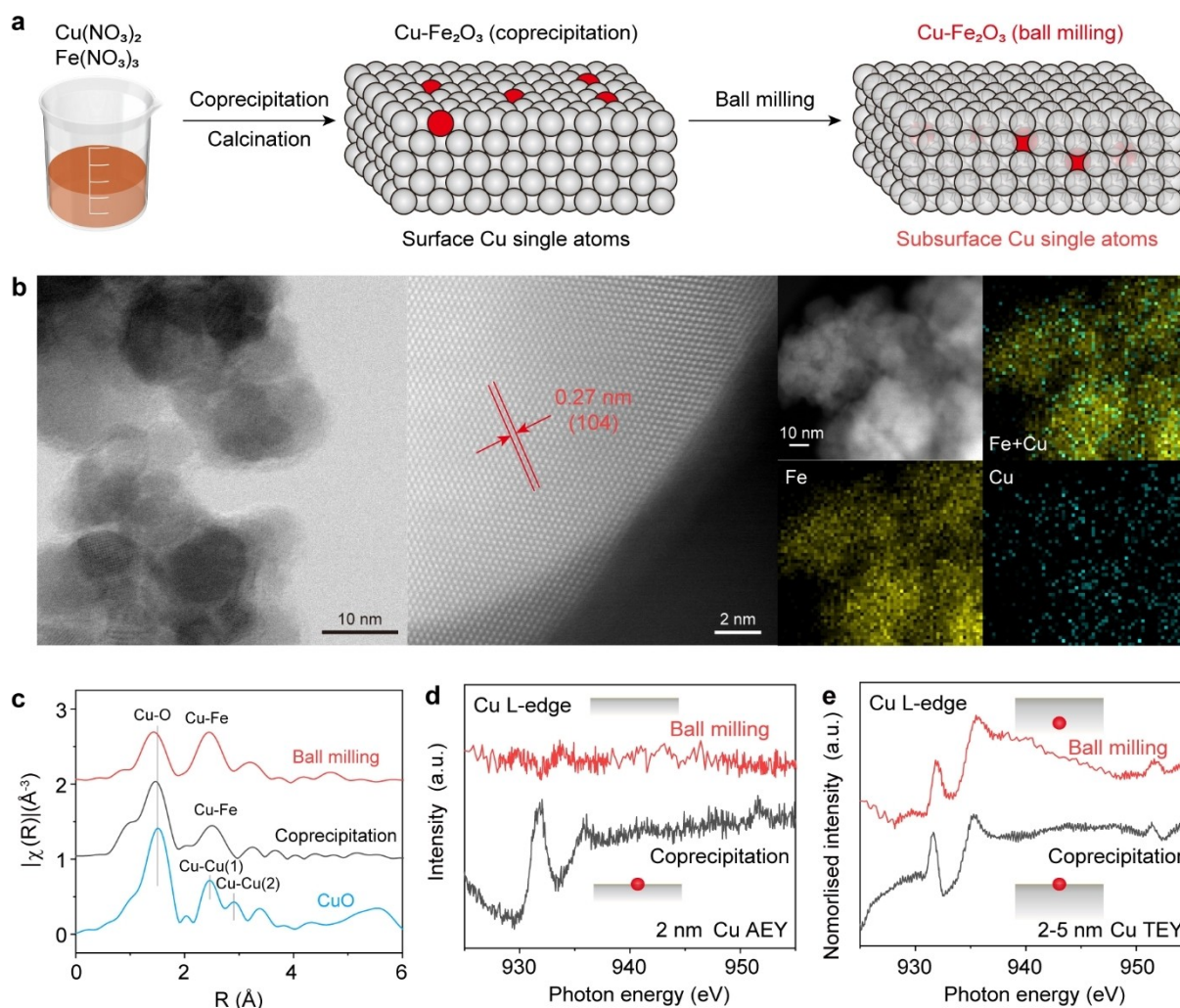


Figure 1. Synthetic Scheme and ex situ characterizations of catalysts. **a**, Schematic illustration of immersing Cu into Fe₂O₃ via ball milling. **b**, Scanning transmission electron microscopy image and the corresponding energy-dispersive X-ray spectroscopy elemental mapping images of Fe Kα₁, Cu Kα₁ and the reconstructed Cu + Fe composition. **c**, Fourier transform of *k*²-weighted EXAFS spectrum of the as-synthesized 1 wt% CuO–Fe₂O₃ (coprecipitation), 1 wt% CuO–Fe₂O₃ (ball milling) catalysts and CuO standard. **d**, Cu L₃ edge NEXAFS AEY spectra. **e**, Cu L₃ edge NEXAFS TEY spectra.

milling. High-angle annular dark-field imaging scanning transmission electron microscopy (HAADF-STEM) confirmed that the particles became smaller after ball milling for both Fe₂O₃ and CuO–Fe₂O₃ catalysts (Figures S11–S14). Overall, the results demonstrate that Cu single sites were homogeneously dispersed over the Fe₂O₃ support, and that ball milling was effective in incorporating these single atoms into the subsurface.

Surface oxygen vacancies are a common structural feature resulting from the reductive reconstruction of iron oxide catalysts. Additionally, hydroxyl groups serve as important intermediates in reactions involving H₂O, H₂, and hydrocarbons that are catalysed by iron oxide. The ball-milled catalyst has the highest content of adsorbed O species, even higher than the H₂-activated catalyst (Figure S7b), which is due to the strong adsorption of H₂O on the surface of defected oxides.^[25] These results suggest that the

surface Fe₂O₃ undergoes reconstruction during the ball milling process.

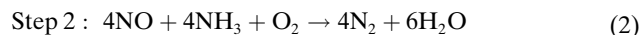
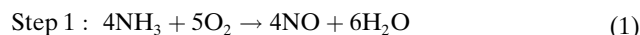
The structure of subsurface single-atom catalysts (SACs) differs from that of doped metal oxides. For doped metal oxides, the doping elements are evenly distributed in the catalyst, including on the surface. In comparison, for subsurface single-atom catalysts, the single atoms are located beneath the surface. As a result, the active sites for doped metal oxides are usually in the form of M_A–O–M_B, while such interfaces are not exposed on the surface for subsurface single-atom. Based on our NEXAFS results, the depth of subsurface Cu species is around 1–5 nm, and their unique position influences their interaction with the surface. This positioning allows them to contribute significantly to the catalytic activity due to their accessibility and modification of surfaces.

The NH₃ oxidation activity of the ball-milled CuO–Fe₂O₃ catalyst was compared with other samples.

Fresh catalyst (surface Cu SAC) and ball-milled catalyst (subsurface Cu SAC) were investigated under realistic NH_3 slip conditions (1000 ppm NH_3 , weight hourly space velocity (WHSV) of $120 \text{ ml}_{\text{NH}_3} \cdot \text{h}^{-1} \cdot \text{g}^{-1}$). The subsurface Cu SAC shifts the conversion profile 80 K towards the low-temperature region and NH_3 is completely converted at 513 K (Figure 2a). After evaluating NH_3 -SCO performance at a higher NH_3 concentration (5000 ppm NH_3 , WHSV of $600 \text{ ml}_{\text{NH}_3} \cdot \text{h}^{-1} \cdot \text{g}^{-1}$), the subsurface Cu SAC showed similar N_2 yield with $\text{Pt-Al}_2\text{O}_3$ catalyst from 473 K to 673 K (Figures S15 and S16). At 573 K, the subsurface Cu SAC exhibited the highest N_2 yield (Figure 2b), with 5.6 times and 2.4 times improvement relative to the Fe_2O_3 and surface Cu SAC, respectively. Compared to other catalyst systems in the literature, the 1 wt% $\text{CuO-Fe}_2\text{O}_3$ (ball milling) catalyst achieved the highest N_2 productivity of Cu-based catalysts (Figure S17). Additionally, the subsurface Cu SAC even outperformed most noble metal catalysts regarding N_2 productivity, N_2 selectivity, WHSV, the temperature of full conversion and stability (Figure S18). The subsurface Cu SAC was also found to be stable under a WHSV of $1200 \text{ ml}_{\text{NH}_3} \cdot \text{h}^{-1} \cdot \text{g}^{-1}$ conditions at 573 K, with good stability over a 100-hour testing period (Figure S19).

The internal selective catalytic reduction (i-SCR) mechanism is widely accepted for transition metal catalysts.^[15,17a] This mechanism includes two steps. First, NH_3 is oxidized to NO (Eq. 1). Then, the unreacted NH_3 reacts with as-formed NO to produce N_2 (Eq. 2). Compared with noble metals,

transition metals generally show lower catalytic activity due to the slow oxidation rates in step 1. The key aspect in this reaction is the activation of O_2 .



The NH_3 oxidation kinetics exhibited a lower apparent activation energy for the subsurface Cu SAC ($\sim 45 \text{ kJ mol}^{-1}$, Figure 2c) in comparison with that of the surface Cu SAC ($\sim 76 \text{ kJ mol}^{-1}$). Moreover, the subsurface Cu SAC exhibited a lower reaction order in O_2 (+0.01 versus +0.62) and a higher reaction order in NH_3 (+0.51 versus -0.26) (Figure 2d,e). The negative order in NH_3 and positive order in O_2 of surface Cu SAC catalyst are consistent with the classical Langmuir–Hinshelwood (L–H) oxidation kinetics,^[10c,26] in which NH_3 or intermediate NO poisoning inhibits the adsorption and activation of O_2 . In contrast to the fresh catalyst, the O_2 order of ~ 0 for the subsurface Cu SAC suggests more facile oxygen activation.^[10c,27] The positive NH_3 order indicates that subsurface Cu single atoms are not poisoned by NH_3 and NO during NH_3 oxidation. These results imply a different mechanism for the subsurface Cu SAC, suggesting that variations in the location of single atoms significantly influence the surface structure and adsorption behaviours.

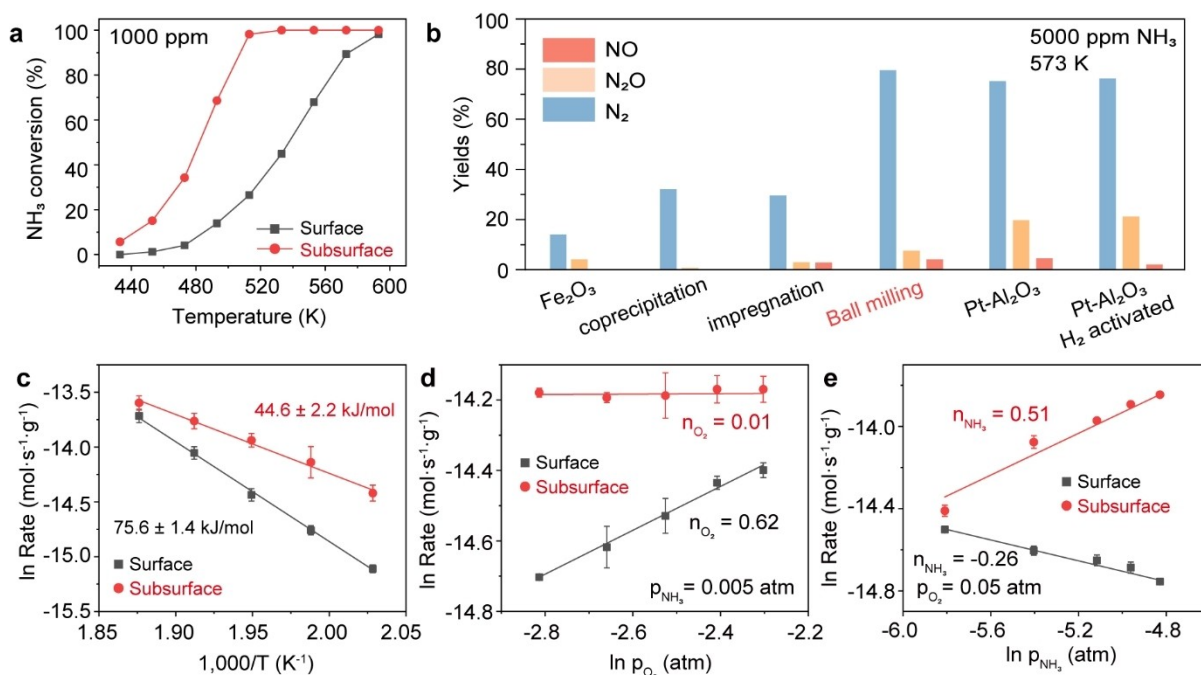


Figure 2. Catalytic reactivity of Cu with different coordination environments. **a**, Steady-state NH_3 conversion plots (1000 ppm NH_3 , 5% O_2 , He balance, and WHSV of $120 \text{ ml}_{\text{NH}_3} \cdot \text{h}^{-1} \cdot \text{g}^{-1}$). **b**, Yields of N_2 , N_2O and NO for various catalysts at 573 K (5000 ppm NH_3 , 5% O_2 , He balance, and WHSV of $600 \text{ ml}_{\text{NH}_3} \cdot \text{h}^{-1} \cdot \text{g}^{-1}$). **c**, Arrhenius-type plots for NH_3 oxidation over the fresh sample (surface Cu SAC) and the ball-milled sample (subsurface Cu SAC), where a difference in the $1000/T$ scale should be noted. **d**, Reaction rate as a function of the O_2 partial pressure measured at 493 K. **e**, Reaction rate as a function of the NH_3 partial pressure measured at 493 K. Reaction orders are estimated as slopes of the linear fit. The data points and error bars represent the average and standard deviation based on multiple measurements on the same catalyst at different times during the experiment.

To confirm the importance of subsurface Cu atoms, we loaded Cu species on the surface of ball-milled Fe_2O_3 by impregnation. After the addition of 1 wt % Cu on the surface, the ball-milled Fe_2O_3 shows an increase in the activity (Figure S20). In our previous study, it is found that the bridging O^{2-} of Cu–O–Fe has a lower antibonding orbital energy and thus weaker Cu–O/Fe–O strength. In selective NH_3 oxidation, the weak Cu–O/Fe–O bond enables fast Cu redox for NH_3 conversion. However, the NH_3 conversion is still lower than the ball-milled Cu– Fe_2O_3 catalysts. These results further support the hypothesis that the weak adsorption of NO for subsurface Cu atoms and oxygen vacancies are crucial for enhancing the catalytic activity.

The zero-order dependence on O_2 partial pressure in the subsurface Cu SAC suggests that O_2 dissociation is not the rate-limiting step. The positive ammonia reaction order for the subsurface Cu SAC points to the Mars–van Krevelen (MvK) mechanism, in which O_2 activation proceeds at the O vacancies of the catalyst^[28] (Figure 3a). Reducibility is a key feature of oxide catalysts that follow the MvK mechanism in oxidation reactions. A typical description of oxide reducibility is the cost of oxygen vacancy formation, which measures the tendency of the oxide to lose oxygen or donate it to the adsorbed species.^[29]

Using near-ambient pressure (NAP)-NEXAFS, we measured the reducibility of the catalyst surface. The spectra of the O K-edge demonstrate transitions from the O-1s state to unoccupied states. The hybridized O 2p–Fe 3d levels exhibit predominantly 3d character and are split into t_{2g} and

e_g states at 530.7 eV and 532.1 eV (Figure 3b), respectively. With the increasing temperature in the NH_3 atmosphere, the t_{2g} and e_g double-peak splitting of ball-milled 1 wt % CuO– Fe_2O_3 (subsurface Cu SAC) becomes less pronounced, while fresh 1 wt % CuO– Fe_2O_3 (surface Cu SAC) remains essentially unchanged. The line shape observed in the present ball-milled sample at 573 K suggests the presence of Fe_3O_4 crystal symmetry.^[30] This can be further confirmed by the Fe L_3 -edge NAP-NEXAFS, as shown in Figure 3c. The increased peak of the L_3 peak at lower energy can be assigned to the formation of Fe^{2+} in the ball-milled catalyst,^[31] while the fresh catalyst keeps unchanged from 373 K to 573 K under 1 mbar NH_3 (Figure S21). These results suggest the enhanced redox behaviour of surface Fe after ball milling.

To verify the role of subsurface Cu single atoms, we also measured the O K-edge of ball-milled Fe_2O_3 . The ball-milled Fe_2O_3 shows similar reducibility to the fresh CuO– Fe_2O_3 catalysts, showing no significant reduction under NH_3 at 573 K (Figure S22 and Supporting Note 3). These results confirm that the surface Fe_2O_3 with Cu in the subsurface is prone to reduction, while the surface Fe_2O_3 with Cu on the surface keeps stable under NH_3 at 573 K. From this perspective, the subsurface Cu SAC has different active sites from the surface Cu SAC, suggesting variations in the local structure of single-atom catalysts influence the reaction mechanism.

The X-ray absorption near-edge structure (XANES) spectra showed that Cu species in the ball-milled sample are

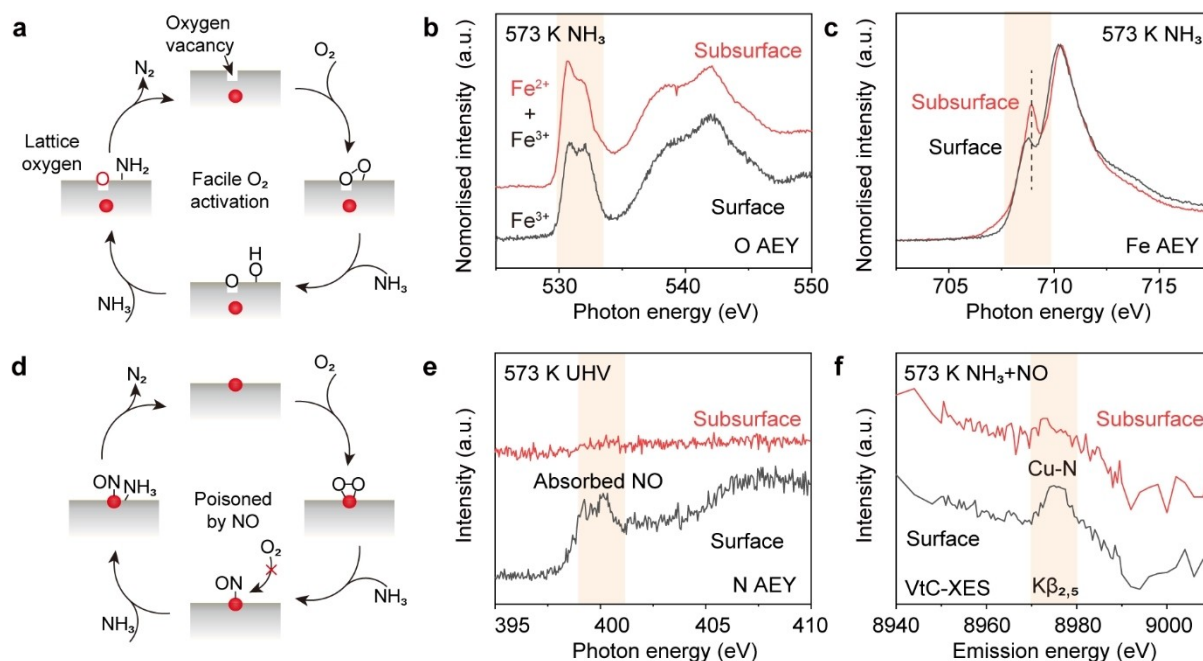


Figure 3. Structure and adsorption behaviours of subsurface Cu single-atom catalyst. **a**, MvK type reaction paths for the ball-milled catalyst with subsurface Cu single atoms. **b**, O–K edge NAP-NEXAFS of the fresh catalyst (surface Cu SAC) and the ball-milled catalyst (subsurface Cu SAC) under 1 mbar NH_3 at 573 K. **c**, Fe– L_3 edge NAP-NEXAFS of the fresh catalyst (surface Cu SAC) and the ball-milled catalyst (subsurface Cu SAC) under 1 mbar NH_3 at 573 K. **d**, L–H type reaction paths for the fresh catalyst with surface Cu single atoms. **e**, N K-edge (AEY mode) under 1 mbar NH_3 at 573 K. **f**, VtC-XES spectra of ball-milled catalyst (top) and fresh catalyst (bottom) under $\text{NH}_3 + \text{NO}$ at 573 K. Due to the adsorption of Fe and low Cu concentration of the catalysts, the obtained spectra were much noisier.

in the oxidation state of +1 and +2, while Cu in the fresh sample is in the form of Cu^{2+} (Figure S23a). Both catalysts have similar $1s \rightarrow 4p$ transition energy for Cu^{2+} , which is higher than that of the CuO standard (Figure S23b). This can be explained by the electron-withdrawing effect of Fe.^[19–20] Compared with the surface Cu SAC, the subsurface Cu SAC has decreased Cu–O coordination (from 3.2 to 2.6, Table S1), indicating the formation of O vacancy in the catalyst.

Adsorbed NO_x species have been reported to block the active sites for O_2 activation in NH_3 oxidation, leading to loss of activity.^[32] This is consistent with the kinetics of the fresh $\text{CuO-Fe}_2\text{O}_3$ catalyst (Figure 3d). Competitive adsorption experiments of NH_3 , O_2 and NO over the fresh catalyst and the ball-milled catalyst were conducted at room temperature (Figure S24). For the surface Cu SAC, the adsorbed NO (Figure S25) is not desorbed under ultra-high vacuum (UHV) and is not replaced by NH_3 (Figure S26) and O_2 (Figure S27). However, the adsorbed NH_3 is desorbed under UHV. For the subsurface Cu SAC, both the adsorbed NO and NH_3 were desorbed under UHV. Subsequent desorption experiments were conducted from room temperature (RT) to 573 K. NO was adsorbed on the catalyst at RT and the desorption of surface species was monitored with the raising of temperature from RT to 573 K (Figure S28). The adsorbed N species disappeared quickly over the subsurface Cu SAC catalyst, while the peaks for adsorbed N species were stable even under 573 K over the surface Cu SAC (Figure 3e). These results indicate that the surface Cu species play an important role in NO adsorption, which agrees with DFT calculations (Figure S29 and Table S2).

To further prove the different adsorption behaviours of surface Cu and subsurface Cu, valence-to-core ($\text{K}\beta_{2,5}$) X-ray emission spectroscopy (VIC-XES) was measured under different conditions. Recognizable changes were still observed under different atmospheres. The most significant difference was the intensity of the peak of the two catalysts (Figure 3f). The Cu in fresh catalyst has higher $\text{K}\beta_{2,5}$ intensity under NH_3 and $\text{NH}_3 + \text{NO}$ conditions, while the spectra of Cu in the subsurface are similar in different environments (Figure S30). These results further indicate that the surface Cu can coordinate with N, while the subsurface Cu can hardly coordinate with N during the reaction. The in situ XES results are consistent with the NAP-NEXAFS results. The strong adsorption of NO on catalysts may cause deactivation because of the loss of active sites for NH_3 and O_2 adsorption. High resilience to NO poisoning of ball-milled catalyst is attributed to durable single Cu atoms in the Fe_2O_3 subsurface.

In situ diffuse reflectance Fourier transform spectrometry (DRIFTS) was used to explore the reaction intermediates of different catalysts. The co-introduction of NH_3 and O_2 was recorded for surface and subsurface Cu SACs from 473 K to 573 K, as shown in Figure 4.

The co-introduction of NH_3 and O_2 was recorded for surface and subsurface Cu SACs from 473 K to 573 K, as shown in Figure 4. The bands at 1140 and 1404 cm^{-1} on the surface Cu SAC can be assigned to ammonia species bound to Lewis acid sites^[33] and Brønsted acid sites,^[34] respectively.

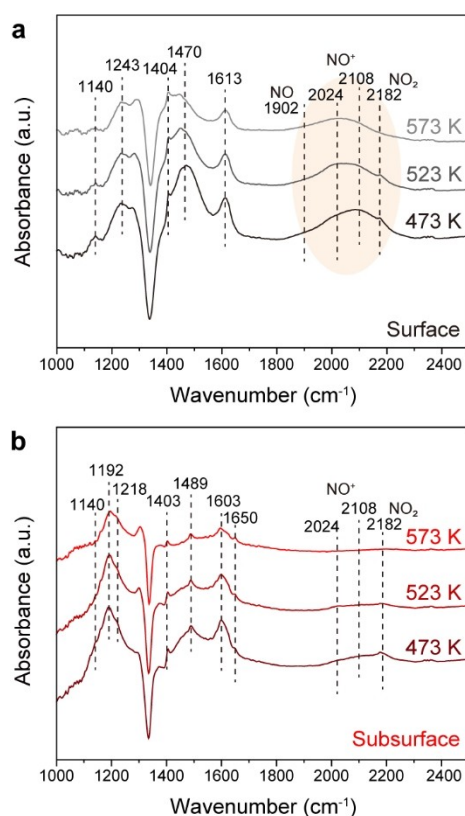


Figure 4. In situ DRIFTS study. In situ DRIFT spectra of NH_3 oxidation over (a) 1 wt% $\text{CuO-Fe}_2\text{O}_3$ (fresh, surface Cu SAC) and (b) 1 wt% $\text{CuO-Fe}_2\text{O}_3$ (ball milling, subsurface Cu SAC) at 473 K, 523 K and 573 K. Feed: 5000 ppm NH_3 , 5% O_2 , He as balance.

The band at 1902 cm^{-1} could be ascribed to the adsorbed NO species,^[35] while such a band cannot be detected on the surface of the subsurface Cu SAC. The monodentate nitrate^[36] (1243 and 1470 cm^{-1}) and bridging nitrate^[37] (1613 cm^{-1}) species were observed on the surface $\text{Cu-Fe}_2\text{O}_3$ catalyst, whereas no such bands emerged on the subsurface $\text{Cu-Fe}_2\text{O}_3$ catalyst. In the i-SCR mechanism, nitrate species are often regarded as representative reaction intermediates.^[38] In this mechanism, NH_3 undergoes oxidation, leading to the formation of various types of nitrates, which subsequently engage in SCR reaction pathways by reacting with NH_3 . The absence of nitrate and nitrite species indicates that the subsurface Cu SAC did not follow the conventional i-SCR mechanism. For subsurface Cu SAC, the bands at 1403, 1489 and 1650 cm^{-1} were assigned to adsorbed NH_4^+ on Brønsted acid sites,^[39] while the bands at 1140, 1192, 1218 and 1603 cm^{-1} were ascribed to NH_3 adsorbed on the Lewis acid sites.^[40]

The bands observed at 2024 and 2108 cm^{-1} on both catalysts were attributed to NO^+ species adsorbed on various cationic positions.^[41] The bands at 2182 cm^{-1} could be assigned to adsorbed NO_2 ,^[42] which was consumed at elevated temperatures for both catalysts. Such consumable $-\text{NO}_2$ indicates the occurrence of the fast i-SCR mechanism,^[43] wherein NO_2 further reacts with ammonia to form N_2 and H_2O , as demonstrated below:



Recently, Lan et al.^[44] reported that the limited selective oxidation of NH_3 into N_2 over $\text{CuO}/\text{Al}_2\text{O}_3$ was attributed to the poor adsorption and activation of reactants, which is due to the strong adsorption of intermediates on conventional CuO sites. The accumulations of NO^+ intermediates were observed on the surface Cu SAC , while the adsorbed $-\text{NO}_2$ was consumed at 573 K. This observation indicates the limited O_2 activation of the surface Cu SAC , which could only oxidize the adsorbed NH_3 into $-\text{NO}_x$ intermediates rather than facilitate the subsequent fast i-SCR reaction. In contrast, NO^+ species are not observed for the subsurface Cu SAC at 573 K (Figure 4b), indicating the fast transformation from NO^+ species to other species. The poison effect of NO over the surface Cu SAC agrees well with the NAP-NEXAFS and VtC-XES results.

Liu et al.^[45] found that the rapid depletion and replenishment of lattice oxygen facilitated the transformation of NO^+ species and the formation of N_2 . Combining in situ experiments, the subsurface Cu SAC showed better redox performance of Fe_2O_3 and weak adsorption of intermediates, leading to higher activity in $\text{NH}_3\text{-SCO}$. As for the surface Cu SAC , the formation of monodentate nitrate and the consumption of $-\text{NO}_2$ adsorbed species under high temperatures indicated that the $\text{NH}_3\text{-SCO}$ reaction was guided by the i-SCR mechanism and the fast i-SCR mechanism. However, the fast i-SCR mechanism was limited by the weak O_2 activation capacity over surface Cu SAC .

To better understand the important role of subsurface Cu , we also performed in situ DRIFTS measurements for the Fe_2O_3 support (Figure S31). The bands at 1192 and 1357 cm^{-1} on the surface of Fe_2O_3 can be assigned to ammonia species bound to Lewis acid sites.^[33] The monodentate nitrate^[36] (1243, 1470 and 1546 cm^{-1}) and bridging nitrate^[37] (1613 cm^{-1}) species were observed on the surface Fe_2O_3 supports, suggesting that the $\text{NH}_3\text{-SCO}$ on Fe_2O_3 follows the conventional i-SCR mechanism.

The bands observed at 2210 and 2238 cm^{-1} on Fe_2O_3 supports were attributed to N_2O species.^[17b] However, the adsorbed NO^+ species and adsorbed NO_2 cannot be observed. These results suggest the Fe_2O_3 supports have restricted O_2 activation but suffer from the formation of N_2O . These findings highlight the effect of the ball milling to modify the surface of the catalysts and enhance the catalytic performance.

Isolated single metal atoms are usually anchored on the surface of metal oxides by strong metal-support interactions.^[3a,46] However, these anchoring strategies may not be effective for reactions at high temperatures, especially in harsh reaction environments.^[10a,47] Fe_2O_3 was reported to have a much higher Tammann temperature (around 1400 K) than Cu ,^[48] and it can act as an inhibitor to prevent the sintering of subsurface Cu atoms. To explore the structural change of the catalyst, the measurement by a HAADF-STEM of the sample after the reaction was performed. As shown in Figure S32, Fe_2O_3 nanoparticles were still remained similar in particle size, which indicated

that Fe_2O_3 could not sinter significantly to cause severe deactivation.

To investigate the chemical state of Cu during the reaction, the high-energy-resolution fluorescence detected X-ray absorption near edge structure (HERFD-XANES) were measured under different conditions. The Cu in the fresh catalyst was in the form of Cu^{2+} from room temperature to 573 K under NH_3 oxidation conditions. The catalyst after ball milling shows a pre-edge peak at 8983 eV, which has been reported to be due to a transition from a 1s to the doubly degenerate $4p_{xy}$ orbitals in a two-coordinate Cu^+ system.^[49] This feature can exist under 473 K and Cu^+ can be oxidized to Cu^{2+} at 573 K (Figure S33). The HERFD-XANES spectra of Cu^{2+} in the subsurface show higher white line amplitudes than that of Cu^{2+} on the surface, suggesting different coordination environments of Cu (Figure 5a). To further verify the DFT model, we simulated the Cu K-edge XANES spectra with different Cu locations. For Cu on the surface of Fe_2O_3 , the Cu K-edge XANES shows lower white line amplitudes compared with Cu in the subsurface (Figure S34), which is in line with the experimental XANES spectra.

Changes in the oxidation state of Cu as well as in the near edge structures were observed for the sample under different conditions at 573 K (Figure 5a–c). NH_3 and $\text{NH}_3 + \text{NO}$ in sequence provide a reduction atmosphere for Cu as expected. Cu in the fresh catalyst is easier to be reduced and can be reduced to Cu^0 cluster, while Cu in the ball-milled catalyst is more stable and even cannot be fully reduced to Cu^+ (Figure 5b). Under the reduction conditions ($\text{NH}_3 + \text{NO}$), the surface Cu single atoms sintered, but subsurface Cu species were still anchored as single atoms in the catalyst subsurface. Cu was then fully oxidized to Cu^{2+} in $\text{NO} + \text{O}_2$ for both catalysts. The reduction and oxidation of subsurface Cu can also be observed in NAP-NEXAFS (Figure S35), which suggests a facile subsurface diffusion of O vacancies around impurity atoms,^[50] enhancing catalytic activities.^[51]

The different stability between the two catalysts can also be shown after continuous reduction and oxidation. After a reduction in $\text{NH}_3 + \text{NO}$ and oxidation in $\text{NO} + \text{O}_2$, the fresh catalyst shows an obvious shoulder, which is similar to the CuO standard (Figure S36), while the Cu coordination environment of the ball-milled catalyst remained unchanged (Figure 5c). Based on our XANES results of Cu single atoms and clusters,^[19–20] the absorption energies for the $1s \rightarrow 4p_z$ (i.e., the shakedown peak from ligand-to-metal charge transfer) and $1s \rightarrow 4p_{xy}$ transition rise with less CuO loading. For subsurface Cu single atoms in Fe_2O_3 , this phenomenon can also be observed (as shown in Figure S37). For Cu single atoms on the surface and in the subsurface, the transition energies of $1s \rightarrow 4p_z$ are 8986.4 eV and 8987.0 eV (Table S3), respectively. The higher $1s \rightarrow 4p_z$ transition energy suggests that the electron-withdrawing effect is strengthened by increasing the number of Fe^{3+} ions per Cu site for subsurface Cu SACs . The absorption energies for the $1s \rightarrow 4p_z$ and $1s \rightarrow 4p_{xy}$ transition decrease (2.3 eV and 0.7 eV, Table S3) for the surface single-atom catalyst after reduction and oxidation. In contrast, the $1s \rightarrow 4p_z$ and $1s \rightarrow 4p_{xy}$ transition energies of subsurface single-atom catalyst are still higher

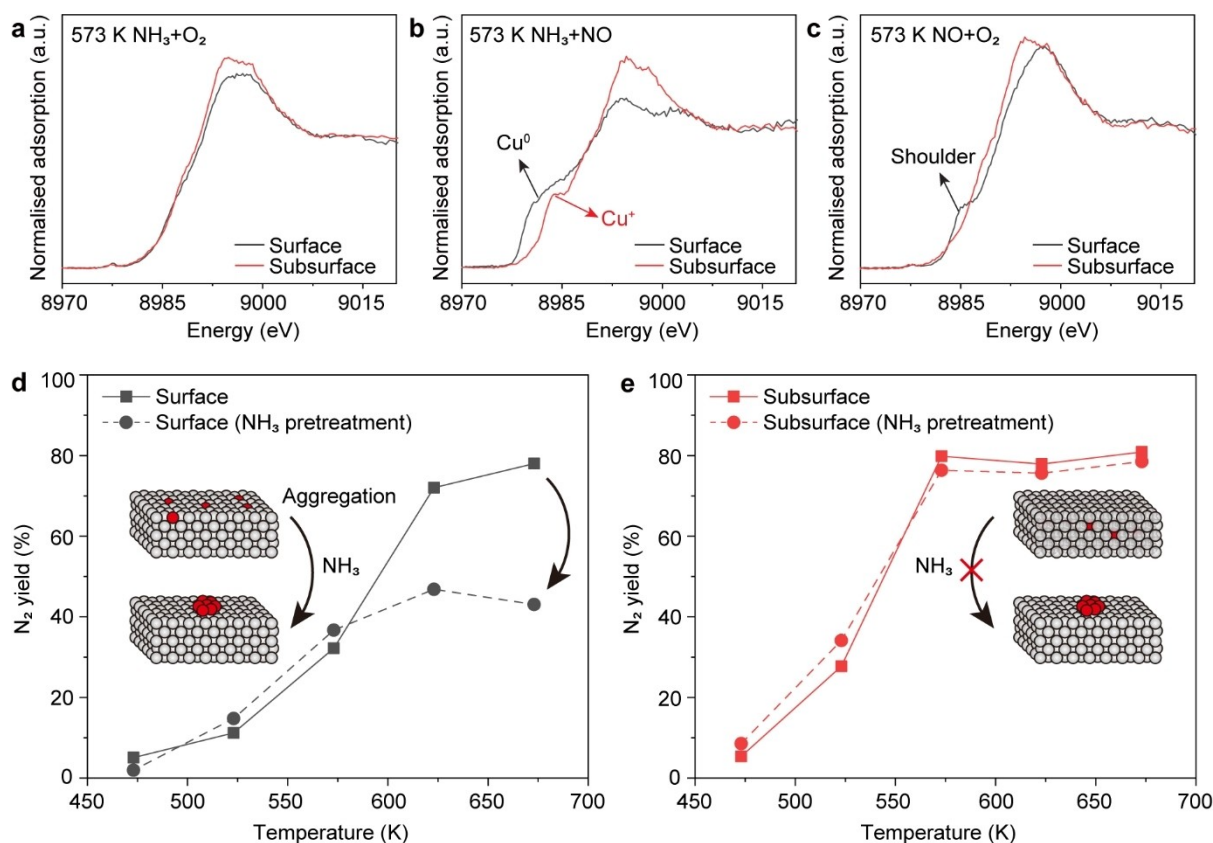


Figure 5. Evaluation of stability under reduction conditions. Cu HERFD-XANES spectra of 1 wt% CuO–Fe₂O₃ (fresh, surface Cu SAC) and 1 wt% CuO–Fe₂O₃ (ball milling, subsurface Cu SAC) samples in (a) NH₃ + O₂, (b) NH₃ + NO and (c) NO + O₂ under 573 K. d, N₂ yield of 1 wt% CuO–Fe₂O₃ (fresh, surface Cu SAC) before and after NH₃ pre-treatment. e, N₂ yield of 1 wt% CuO–Fe₂O₃ (ball milling, subsurface Cu SAC) before and after NH₃ pre-treatment.

than that of CuO (Figure S37). The decrease of $1s \rightarrow 4p_z$ and $1s \rightarrow 4p_{xy}$ energies suggests the formation of CuO clusters for Cu single atoms over Fe₂O₃ after reduction and oxidation.

The operando XAFS experiments show similar results in NH₃ oxidation conditions under high WHSV (Figure S38). The surface Cu single atoms keep in the form of Cu²⁺, while the small amount of Cu⁺ in the subsurface single-atom catalyst gradually oxidised with the increase of temperature. In agreement with HERFD-XANES results, the operando XANES experiments exhibited a higher white line amplitude for subsurface Cu SACs (Figure S39). The reducing treatment, such as NH₃, causes the formation of Cu⁰ clusters for surface Cu single-atom catalysts (Figure S39). In comparison, Cu in the subsurface can only be partially reduced to Cu⁺. The oxidation and reduction of subsurface Cu suggest that Cu can activate the lattice O in Fe₂O₃, thereby improving the redox behaviour of Fe₂O₃. The sintering of surface Cu single atoms leads to substantially decreases N₂ yields (Figure 5d), while subsurface Cu single atoms show robustness under both oxidative (Figure S40) and reductive environments (Figure 5e).

These results suggest that the subsurface single atoms with less mobility are well resistant to sintering.

We found that the electronic structure of subsurface Cu single atoms shows no obvious change after reduction and

oxidation (Figure S37b,d). In contrast, surface Cu sintered after reduction and oxidation, showing similar transition energy of $1s \rightarrow 4p_z$ and $1s \rightarrow 4p_{xy}$ with the CuO standard (Figure S37a,c). The significant agglomeration of surface Cu caused severe deactivation in N₂ productivity (Figure 5d), while subsurface Cu SAC kept the catalytic performance after reduction (Figure 5e). Therefore, both the morphology of catalysts and environment coordination of the subsurface Cu species remain unchanged after the reaction.

Conclusion

These observations illustrate the value of the subsurface single-atom catalysts design strategy, which opens promising perspectives to initiate the lattice distortion and facilitate the activation of the inactive lattice oxygen. The mechanochemical approach has been shown to be an effective method of regulating the position of single atoms. Controlling single atoms at the surface or subsurface results in catalysts with very different adsorption properties, leading to different reaction mechanisms. Such nanoengineering of SAC provides new ideas for preventing the poisoning of SAC and for designing different reaction pathways. For practical applications, the subsurface Cu single atoms in

Fe₂O₃ remained isolated under either oxidation or reduction environments, while surface Cu single atoms on Fe₂O₃ sintered under reduction. Our findings for immersing single metal atoms in the subsurface of supports constitute a promising new target for the development of advanced metal catalysts in oxidation chemistry.

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Conflict of Interest

The authors declare no conflict of interest.

Data Availability Statement

The data that support the findings of this study are available in the supplementary material of this article.

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- [1] B. Qiao, A. Wang, X. Yang, L. F. Allard, Z. Jiang, Y. Cui, J. Liu, J. Li, T. Zhang, *Nat. Chem.* **2011**, *3*, 634–641.
- [2] a) R. Lang, X. R. Du, Y. K. Huang, X. Z. Jiang, Q. Zhang, Y. L. Guo, K. P. Liu, B. T. Qiao, A. Q. Wang, T. Zhang, *Chem. Rev.* **2020**, *120*, 11986–12043; b) J. J. Li, Q. Q. Guan, H. Wu, W. Liu, Y. Lin, Z. H. Sun, X. X. Ye, X. S. Zheng, H. B. Pan, J. F. Zhu, S. Chen, W. H. Zhang, S. Q. Wei, J. L. Lu, *J. Am. Chem. Soc.* **2019**, *141*, 14515–14519; c) H. Zhao, R. Yu, S. Ma, K. Xu, Y. Chen, K. Jiang, Y. Fang, C. Zhu, X. Liu, Y. Tang, L. Wu, Y. Wu, Q. Jiang, P. He, Z. Liu, L. Tan, *Nat. Catal.* **2022**, *5*, 818–831.
- [3] a) Z. Zhang, Y. Zhu, H. Asakura, B. Zhang, J. Zhang, M. Zhou, Y. Han, T. Tanaka, A. Wang, T. Zhang, N. Yan, *Nat. Commun.* **2017**, *8*, 16100; b) L. Nie, D. Mei, H. Xiong, B. Peng, Z. Ren, X. I. P. Hernandez, A. DeLaRiva, M. Wang, M. H. Engelhard, L. Kovarik, A. K. Datye, Y. Wang, *Science* **2017**, *358*, 1419–1423.
- [4] H. Li, L. Wang, Y. Dai, Z. Pu, Z. Lao, Y. Chen, M. Wang, X. Zheng, J. Zhu, W. Zhang, R. Si, C. Ma, J. Zeng, *Nat. Nanotechnol.* **2018**, *13*, 411–417.
- [5] Y. Yao, Z. Huang, P. Xie, L. Wu, L. Ma, T. Li, Z. Pang, M. Jiao, Z. Liang, J. Gao, Y. He, D. J. Kline, M. R. Zachariah, C. Wang, J. Lu, T. Wu, T. Li, C. Wang, R. Shahbazian-Yassar, L. Hu, *Nat. Nanotechnol.* **2019**, *14*, 851–857.
- [6] J. Gu, C.-S. Hsu, L. Bai, H. M. Chen, X. Hu, *Science* **2019**, *364*, 1091–1094.
- [7] D. Liu, X. Li, S. Chen, H. Yan, C. Wang, C. Wu, Y. A. Haleem, S. Duan, J. Lu, B. Ge, P. M. Ajayan, Y. Luo, J. Jiang, L. Song, *Nat. Energy* **2019**, *4*, 512–518.
- [8] B.-H. Lee, S. Park, M. Kim, A. K. Sinha, S. C. Lee, E. Jung, W. J. Chang, K.-S. Lee, J. H. Kim, S.-P. Cho, H. Kim, K. T. Nam, T. Hyeon, *Nat. Mater.* **2019**, *18*, 620–626.
- [9] a) L. DeRita, J. Resasco, S. Dai, A. Boubnov, H. V. Thang, A. S. Hoffman, I. Ro, G. W. Graham, S. R. Bare, G. Pacchioni, X. Pan, P. Christopher, *Nat. Mater.* **2019**, *18*, 746–751; b) F. Maurer, J. Jelic, J. Wang, A. Gänzler, P. Dolcet, C. Wöll, Y. Wang, F. Studt, M. Casapu, J.-D. Grunwaldt, *Nat. Catal.* **2020**, *3*, 824–833.
- [10] a) S. Duan, R. Wang, J. Liu, *Nanotechnology* **2018**, *29*, 204002; b) X. Li, X. I. Pereira-Hernández, Y. Chen, J. Xu, J. Zhao, C.-W. Pao, C.-Y. Fang, J. Zeng, Y. Wang, B. C. Gates, J. Liu, *Nature* **2022**, *611*, 284–288; c) V. Muravev, G. Spezzati, Y.-Q. Su, A. Parastaev, F.-K. Chiang, A. Longo, C. Escudero, N. Kosinov, E. J. M. Hensen, *Nat. Catal.* **2021**, *4*, 469–478.
- [11] a) J. Liu, F. R. Lucci, M. Yang, S. Lee, M. D. Marcinkowski, A. J. Therrien, C. T. Williams, E. C. H. Sykes, M. Flytzani-Stephanopoulos, *J. Am. Chem. Soc.* **2016**, *138*, 6396–6399; b) L. Cao, W. Liu, Q. Luo, R. Yin, B. Wang, J. Weissenrieder, M. Soldemo, H. Yan, Y. Lin, Z. Sun, C. Ma, W. Zhang, S. Chen, H. Wang, Q. Guan, T. Yao, S. Wei, J. Yang, J. Lu, *Nature* **2019**, *565*, 631–635.
- [12] J. Becher, D. F. Sanchez, D. E. Doronkin, D. Zengel, D. M. Meira, S. Pascarelli, J.-D. Grunwaldt, T. L. Sheppard, *Nat. Catal.* **2021**, *4*, 46–53.
- [13] M. Jablonska, A. M. Beale, M. Nocun, R. Palkovits, *Appl. Catal. B* **2018**, *232*, 275–287.
- [14] D. P. Nolan, in *Fire Pump Arrangements at Industrial Facilities (Third Edition)* (Ed.: D. P. Nolan), Gulf Professional Publishing **2017**, pp. 161–167.
- [15] F. Y. Gao, Y. Y. Liu, Z. Sani, X. L. Tang, H. H. Yi, S. Z. Zhao, Q. J. Yu, Y. S. Zhou, *J. Environ. Chem. Eng.* **2021**, *9*, 104575.
- [16] D. A. Svintsitskiy, L. S. Kibis, A. I. Stadnichenko, E. M. Slavinskaya, A. V. Romanenko, E. A. Fedorova, O. A. Stonkus, D. E. Doronkin, V. Marchuk, A. Zimina, M. Casapu, J. D. Grunwaldt, A. I. Boronin, *ChemCatChem* **2020**, *12*, 867–880.
- [17] a) M. Jablonska, R. Palkovits, *Appl. Catal. B* **2016**, *181*, 332–351; b) X. Guan, H. Asakura, R. Han, S. Xu, H.-X. Liu, L. Chen, Z. Yao, J. H. C. Yan, T. Tanaka, Y. Guo, C.-J. Jia, F. R. Wang, *ACS Catal.* **2023**, 13816–13827.
- [18] a) W. Chen, S. Yang, H. Liu, F. Huang, Q. Shao, L. Liu, J. Sun, C. Sun, D. Chen, L. Dong, *Environ. Sci. Technol.* **2022**, *56*, 10442–10453; b) F. Liu, W. Shan, Z. Lian, J. Liu, H. He, *Appl. Catal. B* **2018**, *230*, 165–176.
- [19] L. Kang, B. Wang, Q. Bing, M. Zalibera, R. Büchel, R. Xu, Q. Wang, Y. Liu, D. Gianolio, C. C. Tang, E. K. Gibson, M. Danaie, C. Allen, K. Wu, S. Marlow, L.-d. Sun, Q. He, S. Guan, A. Savitsky, J. J. Velasco-Vélez, J. Callison, C. W. M. Kay, S. E. Pratsinis, W. Lubitz, J.-y. Liu, F. R. Wang, *Nat. Commun.* **2020**, *11*, 4008.

- [20] X. Guan, R. Han, H. Asakura, Z. Wang, S. Xu, B. Wang, L. Kang, Y. Liu, S. Marlow, T. Tanaka, Y. Guo, F. R. Wang, *ACS Catal.* **2022**, 15207–15217.
- [21] a) J. De Bellis, H. Petersen, J. Ternieden, N. Pfänder, C. Weidenthaler, F. Schüth, *Angew. Chem. Int. Ed.* **2022**, 61, e202208016; b) J. De Bellis, C. Ochoa-Hernández, C. Farès, H. Petersen, J. Ternieden, C. Weidenthaler, A. P. Amrute, F. Schüth, *J. Am. Chem. Soc.* **2022**, 144, 9421–9433.
- [22] M. Munoz, P. Argoul, F. Farges, *Am. Mineral.* **2003**, 88, 694–700.
- [23] a) S. Tanuma, C. J. Powell, D. R. Penn, *Surf. Interface Anal.* **1988**, 11, 577–589; b) S. Tanuma, C. Powell, D. Penn, *Surf. Interface Anal.* **2003**, 35, 268–275.
- [24] a) F. Lin, I. M. Markus, D. Nordlund, T. C. Weng, M. D. Asta, H. L. L. Xin, M. M. Doeff, *Nat. Commun.* **2014**, 5; b) J.-P. Deville, C. S. Cojocar, in *Materials Surface Processing by Directed Energy Techniques* (Ed.: Y. Pauleau), Elsevier, Oxford **2006**, pp. 411–441.
- [25] a) H. Idriss, *Surf. Sci.* **2021**, 712, 121894; b) S.-S. Li, W.-J. Li, T.-J. Jiang, Z.-G. Liu, X. Chen, H.-P. Cong, J.-H. Liu, Y.-Y. Huang, L.-N. Li, X.-J. Huang, *Anal. Chem.* **2016**, 88, 906–914.
- [26] a) A. D. Allian, K. Takanabe, K. L. Furdala, X. Hao, T. J. Truex, J. Cai, C. Buda, M. Neurock, E. Iglesia, *J. Am. Chem. Soc.* **2011**, 133, 4498–4517; b) M. Cargnello, V. V. T. Doan-Nguyen, T. R. Gordon, R. E. Diaz, E. A. Stach, R. J. Gorte, P. Fornasiero, C. B. Murray, *Science* **2013**, 341, 771–773.
- [27] a) Y. Lu, J. Wang, L. Yu, L. Kovarik, X. Zhang, A. S. Hoffman, A. Gallo, S. R. Bare, D. Sokaras, T. Kroll, V. Dagle, H. Xin, A. M. Karim, *Nat. Catal.* **2019**, 2, 149–156; b) W.-Z. Yu, W.-W. Wang, S.-Q. Li, X.-P. Fu, X. Wang, K. Wu, R. Si, C. Ma, C.-J. Jia, C.-H. Yan, *J. Am. Chem. Soc.* **2019**, 141, 17548–17557.
- [28] a) M.-H. Liu, Y.-W. Chen, T.-S. Lin, C.-Y. Mou, *ACS Catal.* **2018**, 8, 6862–6869; b) L. DeRita, S. Dai, K. Lopez-Zepeda, N. Pham, G. W. Graham, X. Pan, P. Christopher, *J. Am. Chem. Soc.* **2017**, 139, 14150–14165.
- [29] A. Ruiz Puigdollers, P. Schlexer, S. Tosoni, G. Pacchioni, *ACS Catal.* **2017**, 7, 6493–6513.
- [30] S. Giovannini, F. Boscherini, R. Carboni, L. Signorini, L. Pasquini, N. Mahne, A. Giglia, M. Pedio, S. Nannarone, M. Benfatto, S. Della Longa, *Phys. Scr.* **2005**, 2005, 424.
- [31] S. H. Sun, H. Zeng, D. B. Robinson, S. Raoux, P. M. Rice, S. X. Wang, G. X. Li, *J. Am. Chem. Soc.* **2004**, 126, 273–279.
- [32] L. Gang, B. G. Anderson, J. van Grondelle, R. A. van Santen, *J. Catal.* **2001**, 199, 107–114.
- [33] Y. Zhang, J. Chen, Z. Liu, *Catal. Today* **2022**, 402, 115–121.
- [34] a) Z. Hua, H. Song, C. Zhou, Q. Xin, F. Zhou, W. Fan, S. Liu, X. Zhang, C. Zheng, Y. Yang, X. Gao, *Chem. Eng. J.* **2023**, 473, 145106; b) J. Huang, Z. Tong, Y. Huang, J. Zhang, *Appl. Catal. B* **2008**, 78, 309–314.
- [35] a) H.-Y. Chen, M. Kollar, Z. Wei, F. Gao, Y. Wang, J. Szanyi, C. H. F. Peden, *Catal. Today* **2019**, 320, 61–71; b) J. Szanyi, M. T. Paffett, *J. Catal.* **1996**, 164, 232–245.
- [36] a) X. Xiao, J. Wang, X. Jia, C. Ma, W. Qiao, L. Ling, *ACS Omega* **2021**, 6, 12801–12812; b) Y. Che, X. Liu, Z. Shen, K. Zhang, X. Hu, A. Chen, D. Zhang, *Langmuir* **2023**, 39, 7434–7443.
- [37] L. Liu, Y. Cao, W. Sun, Z. Yao, B. Liu, F. Gao, L. Dong, *Catal. Today* **2011**, 175, 48–54.
- [38] F. Wang, J. Z. Ma, G. Z. He, M. Chen, C. B. Zhang, H. He, *ACS Catal.* **2018**, 8, 2670–2682.
- [39] Y. He, M. E. Ford, M. Zhu, Q. Liu, Z. Wu, I. E. Wachs, *Appl. Catal. B* **2016**, 188, 123–133.
- [40] a) N. Y. Topsoe, H. Topsoe, J. A. Dumesic, *J. Catal.* **1995**, 151, 226–240; b) Z. Liu, S. Zhang, J. Li, L. Ma, *Appl. Catal. B* **2014**, 144, 90–95.
- [41] a) R. Kefirov, E. Ivanova, K. Hadjiivanov, S. Dzwigaj, M. Che, *Catal. Lett.* **2008**, 125, 209–214; b) G. Li, S. C. Larsen, V. H. Grassian, *J. Mol. Catal. A* **2005**, 227, 25–35.
- [42] G. Mul, M. W. Zandbergen, F. Kapteijn, J. A. Moulijn, J. Pérez-Ramírez, *Catal. Lett.* **2004**, 93, 113–120.
- [43] T. Lan, J. Deng, X. Zhang, F. Wang, X. Liu, D. Cheng, D. Zhang, *ACS Catal.* **2022**, 12, 3955–3964.
- [44] T. Lan, M. Gao, J.-y. Hasegawa, Y. Shen, W. Qu, Q. Hu, J. Deng, D. Cheng, D. Zhang, *ACS Catal.* **2023**, 13, 14070–14079.
- [45] Y. Liu, Z. Liu, C. Wang, J. Xu, J. Ai, X. Liu, A. Zhang, Y. Zhao, C. Du, B. Shan, *ACS Catal.* **2023**, 13, 7178–7188.
- [46] a) L. Lin, W. Zhou, R. Gao, S. Yao, X. Zhang, W. Xu, S. Zheng, Z. Jiang, Q. Yu, Y.-W. Li, C. Shi, X.-D. Wen, D. Ma, *Nature* **2017**, 544, 80–83; b) J. Jones, H. F. Xiong, A. T. Delariva, E. J. Peterson, H. Pham, S. R. Challa, G. S. Qi, S. Oh, M. H. Wiebenga, X. I. P. Hernandez, Y. Wang, A. K. Datye, *Science* **2016**, 353, 150–154.
- [47] a) C. Dessal, T. Len, F. Morfin, J.-L. Rousset, M. Aouine, P. Afanasiev, L. Piccolo, *ACS Catal.* **2019**, 9, 5752–5759; b) C. Dessal, A. Sangnier, C. Chizallet, C. Dujardin, F. Morfin, J.-L. Rousset, M. Aouine, M. Bugnet, P. Afanasiev, L. Piccolo, *Nanoscale* **2019**, 11, 6897–6904.
- [48] Y. Wang, J. Zhang, H. Jiang, C. Xu, Z. Liu, R. Xu, Z. Chen, *Mater. Des.* **2022**, 215, 110434.
- [49] a) C. Paolucci, A. A. Verma, S. A. Bates, V. F. Kispersky, J. T. Miller, R. Gounder, W. N. Delgass, F. H. Ribeiro, W. F. Schneider, *Angew. Chem. Int. Ed.* **2014**, 53, 11828–11833; b) T. V. W. Janssens, H. Falsig, L. F. Lundegaard, P. N. R. Vennestrom, S. B. Rasmussen, P. G. Moses, F. Giordanino, E. Borfecchia, K. A. Lomachenko, C. Lamberti, S. Bordiga, A. Godiksen, S. Mossin, P. Beato, *ACS Catal.* **2015**, 5, 2832–2845.
- [50] Z. Łodziana, J. Piechota, *Phys. Rev. B* **2006**, 74, 184117.
- [51] J. Kang, Y. Zhang, Z. Chai, X. Qiu, X. Cao, P. Zhang, G. Teobaldi, L.-M. Liu, L. Guo, *Adv. Mater.* **2021**, 33, 2100407.

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