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ABSTRACT
In GaN:Mg, the MgGaN acceptor is compensated extensively by the formation of nitrogen vacancies (VN) and Mg interstitials (Mgi). However, we show that such compensation can be overcome by forming two kinds of Mg-rich complexes: one that contains VN and the other that contains only MgGaN and Mg. Such complexing not only neutralizes VN and Mgi but also forms better complex acceptors that have lower formation energies and smaller hole localization energies than isolated MgGaN. Our results help explain the different doping behaviors in samples grown by different methods.

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I. INTRODUCTION

The realization of p-type conductivity in GaN:Mg has enabled the application of GaN-based electronic and optical devices. However, such industrial success is achieved under the shadow of a poor understanding of the underlying mechanism, despite decades of academic research. The p-type conductivity of GaN can be achieved only by heavily doping with Mg, with concentrations of Mg above \(10^{19}\) cm\(^{-3}\) in metalorganic chemical vapor deposition (MOCVD) grown samples, for example. According to Hall-effect measurements of MOCVD GaN:Mg layers by Kaufmann et al., the hole concentration reaches a maximum value of \(6 \times 10^{17}\) cm\(^{-3}\) at an Mg concentration of \(2 \times 10^{19}\) cm\(^{-3}\), then decreases at higher Mg concentrations. The photoluminescence (PL) bands in undoped and Mg-doped GaN can help clarify the electronic properties, yet their attribution is still under debate. The ultraviolet luminescence (UVL) band is the dominant PL band in conductive n-type undoped and Mg-doped GaN grown by different techniques, as well as in conductive p-type Mg-doped GaN grown by molecular beam epitaxy (MBE). The blue luminescence (BL) band is observed only in GaN:Mg samples grown by hydride vapor phase epitaxy (HVPE) or MOCVD when the concentration of Mg exceeds \(10^{19}\) cm\(^{-3}\). Kaufmann et al. connected the emergence of the room temperature (RT) PL band that peaks around 2.8 eV with the maximum value of the hole density reached as a function of Mg concentration for Mg concentration of \((1–2) \times 10^{19}\) cm\(^{-3}\) in MOCVD-grown GaN and presented clear evidence for the distant donor-acceptor (D–A) pair character of the 2.8 eV band. The reasons for the absence of p-type conductivity under moderate Mg doping, the onset of p-type conductivity under heavy Mg doping, and the subsequent drop-off in conductivity after reaching a maximum value of the hole concentration, as well as the origin of the UVL and BL in GaN:Mg, have...
been debated intensively for a few decades now, without resolution so far.

To date, the attribution of the electronic and optical properties of GaN:Mg to particular defect states remains unclear. Several models have been proposed. Apart from the commonly calculated deep levels of acceptors in wide bandgap semiconductors that localize a hole at single anion ligand atoms with a Jahn-Teller type symmetry breaking, Lany and Zunger found an effective-mass-like delocalized state of Mg_{Ga} acceptors in GaN, which, in contrast, has a symmetric configuration. Despite the large geometry distortion in the localized state, this dual nature has a small energy difference of 0.03 eV, according to their calculation employing a Koopmans-corrected density functional theory (DFT) method. They predicted that the shallow state would dominate in \( n/\)type samples, whereas in less compensated \( p/\)type samples, the deep state would dominate. Lyons et al.\(^{11}\) with hybrid DFT calculations, computed a relatively shallow (0/\( \pm \)) thermodynamic transition level for Mg_{Ga} (260 meV), which they found to exhibit key features of a deep acceptor: the hole is localized on an N atom neighboring the Mg impurity, inducing a large local lattice distortion and giving rise to broad blue luminescence. Therefore, they attributed the BL to Mg_{Ga} and the UVL to Mg_{Ga}-H. They found no evidence for the existence of other metastable configurations, in contrast to Lany and Zunger. Their calculations were reproduced by Demchenko et al.,\(^{12}\) who attributed the UVL band to the anisotropically delocalized acceptor state. In earlier work, using the hybrid quantum mechanical (QM) and molecular mechanical (MM) embedded cluster method,\(^{13}\) we found that in lightly doped GaN:Mg, isolated Mg_{Ga} strongly trap holes and, therefore, do not contribute to \( p/\)type conduction.

Whether the model involving only isolated Mg_{Ga} defects, which may have a dual nature, is sufficient to account for the observed \( p/\)type conductivity and different PL bands or whether more complicated defects (complexes) have to be included has also drawn controversy in experimental studies. Monemar et al.\(^{21}\) studied bound-exciton spectra in samples for Mg concentrations up to \( [Mg] \approx 10^{19} \text{ cm}^{-3}\) and found evidence for two Mg-related acceptors in GaN. The unstable UVL emission peaking at 3.27 eV upon \( p/\)type activation was thus attributed to one of the two acceptors, which is unstable in \( p/\)GaN. Later, Monemar et al.\(^{10}\) attributed these two acceptors to isolated Mg_{Ga} and a deeper acceptor that arises due to the interaction between basal plane stacking faults and Mg_{Ga} acceptors. The dual nature model of Lany and Zunger was supported by Callisen et al.\(^{11}\) from the optical signature of Mg-doped GaN through the determination of the binding energies of two Mg acceptor states, while their sample was lightly Mg-doped (8 \( \times \) \( 10^{17} \text{ cm}^{-3}\)). Singh et al.\(^{22}\) probed the lattice location of Mg in doubly doped GaN (Mg):Eu and linked two distinct Eu–Mg defect configurations with Lany and Zunger’s model. Electron paramagnetic spin resonance (EPR) and optically detected magnetic resonance (ODMR) measurements, along with PL, have also been performed to investigate acceptors in GaN:Mg. Glaser et al.\(^{11}\) performed ODMR and found a highly anisotropic g-tensor associated with the 3.27 eV PL emission from lightly Mg-doped samples predicted for effective-mass Mg acceptors in GaN. Later, (basal-plane) strain was highlighted as playing a crucial role in influencing the detailed nature of the acceptor states: the shallow effective-mass-like acceptor and deeper acceptor states are attributed to simple substitutional Mg at Ga sites in unstrained and strained regions, respectively.\(^{23,24}\)

It is clear from the above-mentioned discussion that there is no single model that can satisfactorily explain the evolution of the \( p/\)type conductivity or the observed PL signals upon Mg doping. The electrical activation of Mg as a \( p/\)type dopant requires its incorporation into substitutional Ga as a single acceptor, while donor type defects such as N vacancies, Mg interstitials, or impurities such as hydrogen and silicon are likely to compensate, making Mg_{Ga}, a “vulnerable” acceptor dopant (giving rise to the observed doping limit).\(^{25}\)

The nitrogen vacancy \( V_N\), as an intrinsic defect, has long been proposed as a major compensating center in GaN:Mg. By using room temperature Hall-effect measurements as a function of the Mg concentration, Kaufmann et al.\(^{26}\) found evidence for self-compensation, i.e., dopant-driven compensation of the Mg acceptor after reaching the maximum hole concentration by intrinsic donor defects rather than donor impurities, for which they developed a self-compensation model involving nitrogen vacancies. For heavily doped materials, they favored the \( V_N-Mg_{Ga}\) pair model. Applying positron annihilation spectroscopy, Hautakangas et al.\(^{27}\) recorded a positron lifetime of 180 ps in MOCVD grown Mg-doped GaN and suggested that neutral or negative complexes involving the monovacancy \( V_N\), in particular the \( V_N-Mg_{Ga}\) complex, were present in semi-insulating and \( p/\)type GaN. Interestingly, they observed no vacancies in MBE grown GaN:Mg layers. By performing Hall-effect, secondary-ion-mass spectroscopy (SIMS), and Rutherford backscattering spectrometry (RBS)-channeling measurements, Obata et al.\(^{28}\) investigated the lattice properties of heavily Mg-doped GaN and found that the lattice relaxation of nitrogen atoms is much higher than for Ga atoms, suggesting that the inverted \( n/\)type conductivity is caused by self-compensation effects induced by nitrogen vacancies. Using DFT calculations, Myers et al.\(^{29}\) indicated that \( V_N\) compensates Mg acceptors extensively at higher doping levels. By employing a hybrid functional, Yan et al.\(^{30}\) found that both Mg_{Ga}-\( V_N\) and \( V_N\) act as compensating centers in \( p/\)type GaN. Lee et al.\(^{31}\) studied three Mg–H–\( V_N\) complexes in GaN:Mg and found that the inclusion of \( V_N\) in many of the passivated complexes is necessary, given their abundance and energetic favorability in GaN:Mg. Using the QM/MM embedded cluster method,\(^{32}\) we previously determined that the nitrogen vacancy is a shallow compensating center in GaN doped with divalent metals, and consequently, isolated Mg_{Ga} alone cannot account for the \( p/\)type conductivity in GaN:Mg. We calculated the optical level associated with \( V_N\) at 44 meV below the conduction band minimum (CBM), to which we attributed the observed 3.466 eV UVL peak.

Apart from the nitrogen vacancy, there is evidence that the Mg interstitial (Mg\(_i\)) is also a major compensating center in GaN:Mg. Early electronic structure calculations by Reboredo and Pantelides\(^{33}\) showed that Mg\(_i\) and its complexes play a significant role in controlling the \( p/\)type doping process; they found that compensation of Mg_{Ga} occurs mainly through the formation of Mg substitutional-interstitial pairs, among others, while H impurities can passivate them all. More recently, low formation energies for Mg\(_i\), when the Fermi level is located below midgap have been computed.\(^{34}\) Miceli and Pasquarcello,\(^{35}\) using hybrid DFT, found that the Mg,...
shows a noticeably lower free energy of formation than the MgGaN in p-type conditions and, together with nitrogen vacancies, will act as a compensating center for donors. Recently, by implanting radioactive $^{27}$Mg into GaN, Wahl et al.\textsuperscript{25} gave direct experimental evidence for Mg, near octahedral sites and showed a significant concentration of Mg, in p-type GaN. Later, Wahl et al. elucidated the amphoteric nature of Mg, i.e., the concurrent occupation of substitutional Ga and interstitial sites. They concluded that challenges toward high electrical activation of implanted Mg are not related to a lack of substitutional incorporation.\textsuperscript{26} Along with other divalent metals, Be, for example, has also been investigated as an alternative p-type dopant; however, self-compensation induced by interstitials is likely to be considerably more pronounced for Be than for Mg, which would explain why no successful synthesis of p-type GaN:Be has been reported as yet.\textsuperscript{27}

To achieve p-type conductivity, the compensation effects of $V_N$ and Mg, induced inevitably by heavy Mg doping must be controlled. In this paper, we study complexes comprising the acceptor MgGa$_i$ and the donors $V_N$ and Mg. We show that $V_N$ as a triple donor in p-GaN must be passivated; MgGa$_i$ as the only acceptor-type defect has to contribute to p-type conductivity, possibly by forming acceptor complexes; and Mg, as a double donor needs to be passivated in p-type GaN and can account for the doping threshold. We find that a large complex with Mg$_2$N$_2$ stoichiometry passivates $V_N$ and Mg, and with one more MgGa$_i$ as an acceptor—improves the p-type behavior of the dopant, a trend we postulate may continue for even larger clusters, although the small clusters studied here cannot be two different acceptors: Mg-rich complexes with and without an N vacancy, which can explain the different doping behavior in GaN:Mg samples—grown by different methods.

II. COMPUTATIONAL DETAILS

In this work, electronic properties of defects and their complexes in wurtzite GaN:Mg are obtained following a hybrid QM/MM embedded cluster approach, which has been developed to describe localized states in ionic solids where charged and strong polar centers interact via long-range Coulomb and short-range Pauli repulsion forces.\textsuperscript{28,29} This method splits extended systems into the inner region, containing the centrally located defect, described using molecular QM theories, and its surroundings, which are only lightly perturbed by the defect, modeled with MM techniques based on interatomic potentials. Our choice of the QM methodology is DFT; the MM simulations employ polarizable-shell model interatomic potentials. The interface between the inner and outer regions employs cation-centered semi-local pseudopotentials to provide an appropriate embedding potential acting on the electronic subsystem of the defect and to contain the electrons within the inner region, effectively preventing them from spilling over the positively charged, attracting cations residing in the outer MM region. The outer part of the MM region is held fixed at the pre-optimized geometry,\textsuperscript{28,29} and any polarization here is treated \textit{a posteriori}. The entire cluster is surrounded by point charges, whose values are fitted to reproduce the Madelung potential of the infinite crystal within the inner QM region along with its immediate MM environment, all contained within an “active sphere” of a user-defined radius (typically 15–20 Å). Using the embedded cluster approach with a suitably terminated QM cluster (i.e., with cations as terminating ions and an appropriate embedding potential), we can model the well localized valence band maximum (VBM) states. Therefore, by calculating the ionization energy of an electron in the bulk system, we can probe the VBM and determine its position relative to the vacuum. We then shift the CBM at an energy equal to the bandgap (3.5 eV) above the VBM relative to the vacuum. The advantage of our approach is the comprehensive account of short- and long-range polarization effects due to the presence of charged defects, the lack of periodic image interactions, the true infinitely dilute limit when reporting defect energetics, and an unambiguous definition of the vacuum level, facilitating the calculation of ionization energies with an absolute reference.

The method has been implemented in the ChemShell package,\textsuperscript{30,31} employing GAMESS-UK\textsuperscript{32} for the QM single-point energy and gradient calculations and GULP\textsuperscript{33} for the MM calculations. The QM region of 6.8 Å radius (116 atoms of wurtzite GaN) is embedded in an MM region of radius 30 Å. In our QM calculations, we use (i) the hybrid exchange and correlation density functional B97-2,\textsuperscript{34} which is similar to those commonly used in plane-wave supercell calculations (21% exact exchange compared with 25% for PBE0\textsuperscript{35} or HSE06\textsuperscript{36}), (ii) the SBKJC small-core pseudopotentials on Ga\textsuperscript{37} within the cluster and large-core refitted pseudopotentials\textsuperscript{38} on the cations forming the interface, and (iii) the atom-centered Gaussian basis set of def2-TZVP form on N\textsuperscript{39} and a matching quality SBKJC basis on Ga.\textsuperscript{37} For comparison, we use a second hybrid exchange and correlation density functional, BB1K, employing 42% exact exchange\textsuperscript{40} and fitted to reproduce kinetic barriers and thermochemical data, which removes a significant portion of the self-interaction error in exchange and correlation, thus giving a more accurate description of electron and hole localization and consequently deeper transition levels with respect to the VBM.\textsuperscript{41} Other computational details can be found in our earlier publication.\textsuperscript{28,41} We have applied this method previously to treat defects in AgCl,\textsuperscript{42} ZnO,\textsuperscript{43,44} and in earlier studies of GaN.\textsuperscript{8,40,45,46}

III. RESULTS AND DISCUSSION

We first show the formation energies under anion-rich conditions of the three important defects in p-GaN:Mg, as discussed earlier: $V_N$, MgGa$_i$, and Mg$_i$. In Figs. 1(a) and 1(b). We note that $V_N$ has been discussed in detail in our previous work.\textsuperscript{8,40} The formation energies of MgGa$_i$ are reproduced from Ref. 8, but here we include a +/0 transition at 0.37 (0.84) eV above the VBM, determined using the B97-2 (BB1K) functional. Mg stabilizes at an octahedral interstitial site in the hexagonal channel of wurtzite GaN, similar to Ga$_i$.\textsuperscript{40} Mg$_i$ is a donor-like defect for Fermi level values across the bandgap, stabilizing in the +2 charge state, and the formation energy at the VBM is –0.59 (–2.68) eV, computed using the B97-2 (BB1K) functional. For Fermi levels in the lower part of the bandgap, the formation energies of $V_N$ and Mg$_i$ are negative and lower than those of MgGa$_i$, implying spontaneous formation and a compensating nature for both. The low formation energies of $V_N$ and Mg$_i$ in p-GaN have also been reported in studies employing supercell calculations.\textsuperscript{20,23,24}

Reschikov et al. performed hybrid DFT calculations [using a modified Heyd–Scuseria–Ernzerhof (HSE) functional\textsuperscript{30}] with an exact exchange fraction of 0.312 and found low formation energies of $V_N$ and Mg$_i$ near the VBM and the 0/– transition level of MgGa.
FIG. 1. (a) B97-2 and (b) BB1K calculated—formation energies of V_N (black line), Mg_Ga (red line), and Mg (blue line) as a function of Fermi energy above the VBM. (c) and (e) B97-2 and (d) and (f) BB1K calculated—formation energies and binding energies of the complexes Mg_Ga-V_N (black line) and Mg_Ga-Mg (red line). Positive binding energies indicate favorable binding. Anion-rich conditions are assumed in (a)–(f). Structures of (g) Mg_Ga-V_N in a +2 charge state and (h) Mg_Ga-Mg in a +1 charge state. The blue, yellow, and red balls denote the Ga, N, and Mg atoms, respectively.
at 0.326 eV above the VBM.\textsuperscript{22} Miceli and Pasquarello performed similar hybrid DFT calculations (HSE functional with 0.31 exact exchange) and found similar low formation energies of $V_N$ and $Mg_i$ near the VBM and the 0– transition level of $Mg_{Ga}$ at 0.38 eV above the VBM.\textsuperscript{24} Our results are consistent with the trend of the $Mg_{Ga}$ thermodynamic transition levels moving deeper into the gap with charged defects treated by our method compared with supercell calculations and with an increasing exact exchange fraction included in hybrid functionals, as detailed in Ref. 40. We thus confirm the compensating effects of $V_N$ and $Mg_i$, a result largely independent of the functional or model used.\textsuperscript{31,45} Similar compensating effects have been found in GaN:Be.\textsuperscript{25}

As triple- and double-donor defects in $p$-GaN, respectively, we expect $V_N$ and $Mg_i$ to easily bind the acceptor $Mg_{Ga}$ and form complexes $Mg_{Ga-V_N}$ and $Mg_{Ga-Mg_i}$. We calculate three configurations of $Mg_{Ga-V_N}$, where the missing $N$ relative to the $Mg_{Ga}$ is on either an axial site, a basal-plane site, or a next-nearest neighbor site. We find that these configurations all have very similar formation energies, and we present the results only for the axial site configurations.\textsuperscript{25} $Mg_{Ga-Mg_i}$ is composed of $Mg_i$ with a neighboring $Ga$ replaced by $Mg$. Figures 1(c) and 1(d) show the formation energies of these two complexes under anion-rich conditions (under anion-poor conditions, the formation energies will be 1.59 and 2.04 eV lower, respectively). They are donor-like in the $p$-type region (i.e., Fermi level values in the lower half of the gap) and have negative formation energies close to the VBM, so they will act as compensating centers. Comparing Figs. 1(a)–1(d), we see that binding a $Mg_{Ga}$ results in $V_N$ and $Mg_i$ centers with slightly higher formation energies but, crucially, lower positive charge states in the $p$-type region and hence, consequently, less severe compensating centers as complexes than as isolated species. Note that the positive binding energies in Figs. 1(e) and 1(f) of these two complexes imply that they are bound species.\textsuperscript{35} Structures of $Mg_{Ga-V_N}$ in the +2 charge state and $Mg_{Ga-Mg_i}$ in the +1 charge state are shown in Figs. 1(g) and 1(h), respectively. The substitutional $Mg_{Ga}$, as an isolated defect, tends to have associated with it a localized hole on a nearest-neighbor $N$ ion in its neutral charge state and a second localized hole on another nearest-neighbor $N$ ion when the Fermi level is very close to the VBM (the defect is then in the + state). The energy needed to dissociate the localized hole is quite large, 1.404 eV as computed with the BB1K functional; therefore, the activation of $Mg_{Ga}$ and subsequent $p$-type conductivity is difficult to achieve. However, through binding a donor defect, the hole localized by $Mg_{Ga}$ recombines with a donor electron. Note that in both complexes, $Mg_{Ga}$ is in its state of $–1$ throughout the bandgap, except in $Mg_{Ga-Mg_i}$, when the Fermi level is less than 0.07 (0.012) eV above the VBM for the B97-2 (BB1K) functional. Therefore, in $p$-GaN, these complexes will act as donors, but the $Mg_{Ga-Mg_i}$ complex can release a trapped hole with an energy of 0.07 or 0.012 eV. However, crucially, both complexes can serve as nucleation centers for larger complexes, which may act as acceptors, as we show below.

The $Mg_{Ga-V_N}$ complex has been studied several times previously. However, the $Mg_{Ga-Mg_i}$ complex is much less well studied since the role of $Mg_i$ has only recently been appreciated. We compare our calculated formation energies of both complexes in the neutral charge state only with those determined using supercells since the periodic image interactions are particularly problematic for charged defects. For neutral $Mg_{Ga-V_N}$, we determine a formation energy of 2.32 (2.55) eV in N-rich conditions and 0.73 (0.96 eV) in N-poor conditions using the B97-2 (BB1K) functional, compared with supercell-method-calculated values of 2.8 eV (N-rich) and 2 eV (N-poor),\textsuperscript{24} 2.1 eV (N-rich) and 1.25 (N-poor),\textsuperscript{24} determined with HSE, and 0.13 eV (N-poor),\textsuperscript{35} and 1.6 eV (N-poor)\textsuperscript{45} determined in earlier DFT studies. The compensating effects of $Mg_{Ga}$ and $Mg_{Ga-V_N}$ have been investigated using various measurement techniques.\textsuperscript{23,24,26,27,51} Although $V_N$ are very difficult to detect experimentally due to either the small size of the vacancy, the high concentrations of other defects, or the requirements for negative and neutral charge states, depending on the experimental method,\textsuperscript{11} Hautakangas et al.\textsuperscript{11} proposed $Mg_{Ga-V_N}$ complexes as native defects in Mg-doped GaN using positron annihilation spectroscopy (PAS). They attributed their measured 180 ps positron lifetime to $V_N$ or related complexes. As the isolated $V_N$ will not act as a positron trap because of its positive charge, they suggested that neutral or negative complexes involving $V_N$ are present in GaN:Mg. From computed core-electron momentum distributions, they identified $Mg_{Ga-V_N}$ as the source of the observed positron lifetime. $Mg_{Ga-V_N}$ is not neutral or negative in $p$-type GaN, as can be seen from our calculations and those of others referenced earlier. $Mg_{Ga-V_N}$ is in the charge state of +2 for the $+2$/$0$ transition at 1.62 (2.13) eV above the VBM in Fig. 1(c) [and Fig. 1(d)]. Hybrid functional supercell calculations show this transition at $\approx 0.8$ eV\textsuperscript{22,24,50} above the VBM. Like isolated $V_N$, $Mg_{Ga-V_N}$ has a positive charge state in $p$-type GaN and is unlikely to be a positron trap.

$Mg_{Ga-V_N}$, $V_N$, and $Mg_{Ga-Mg_i}$, although less severe, are still compensating centers in $p$-type GaN, just like isolated $V_N$ and $Mg_i$, respectively. Therefore, as double and single donor defects in $p$-GaN, we expect them to continue binding $Mg_{Ga}$ and to form complexes. Moreover, we expect such larger complexes to further reduce the compensating effect of such centers and help free holes from localization to $Mg_{Ga}$, which may finally result in $p$-type conductivity in GaN:Mg.

Therefore, we calculate two Mg-rich complexes, $4Mg_{Ga-V_N}$ and $3Mg_{Ga-Mg_i}$, where $V_N$ and $Mg_i$ are fully neutralized, and these two complexes are acceptors as a whole. $4Mg_{Ga-V_N}$ is composed of a $N$ vacancy with its four nearest $Ga$ all replaced by $Mg$, as shown in Fig. 2(a). In its closed-shell configuration of charge state $–$, the axial $Mg_{Ga}$ moves away from the vacant site by 43% (44%) of the equilibrium $Ga-N$ bond length, while the other three basal $Mg_{Ga}$ move away by 19% (17%), as calculated using the B97-2 (BB1K) functional. When its charge state changes from $–$ to 0 and then to +, $Mg_{Ga}$ moves farther away from the vacancy site by 4% when a hole is localized on a bonded $N$ atom of $Mg_{Ga}$. $3Mg_{Ga-Mg_i}$ is composed of an octahedral site Mg with three of its six nearest Ga replaced by Mg, so we calculate one of six possible configurations, shown in Fig. 2(d). In Figs. 2(b), 2(c), 2(e), and 2(f), we show the formation energies in both conditions for these two complexes. They both have a lower formation energy under anion-poor conditions since the isolated $Mg_{Ga}$, $V_N$, and $Mg_i$ all favor anion-poor conditions. Comparing these two acceptor complexes with the isolated $Mg_{Ga}$ in Figs. 1(a) and 1(b), we find using the B97-2 functional that the 0– transition level has been reduced from the isolated $Mg_{Ga}$ value of 0.863 (1.404) eV to 0.655 (0.997) and 0.690 (1.178) eV above the VBM, respectively, indicating that less energy is needed to dissociate a hole than that localized by isolated $Mg_{Ga}$. Note that the $Mg_{Ga-Mg_i}$ complex has a negative U effect, with a $+/$
transition level at 0.704 (1.237) eV determined using B97-2 (BB1K). Both complexes neutralize a major compensating center in p-GaN and become acceptors that have smaller hole dissociation energies compared to the isolated MgGa acceptor. Moreover, both complexes have lower formation energies in anion-poor conditions than isolated MgGa, indicating that they can form at higher concentrations than isolated MgGa in heavily doped GaN:Mg. We have calculated the equilibrium defect concentrations using the code SC-FERMI, as shown in Fig. 3.

Our calculations have, therefore, shown that the major compensating donor centers $V_N$ and Mg in p-GaN:Mg can be neutralized by binding to multiple MgGa acceptors. Through forming a cluster with Mg$_3$N$_2$ stoichiometry and an additional MgGa, a large Mg-rich acceptor complex forms that is a better acceptor than isolated MgGa, with lower hole localization energies and lower formation energies in anion-poor conditions. Following this pattern, we can expect that even larger Mg-rich complexes can form in p-GaN and contribute to the p-type conductivity. Mg-rich pyramidal defects in p-GaN have been observed previously in experimental studies. For example, in MOVPE Mg-doped GaN, Vennéguès et al. found that these pyramidal defects are Mg rich and present in all studied films, independent of the doping level. Recently, maps of the electrostatic potential provided by off-axis electron holography have confirmed that the highest carrier concentration was achieved in the regions with the highest dopant concentration of $2 \times 10^{20}$ cm$^{-3}$, despite the presence of a high density of Mg-rich clusters revealed by correlating atom probe tomography. We propose that larger complex acceptors can have lower hole localization energies and, with the right size, form at least some of the shallow acceptors observed in conductive p-GaN. Kozodoy et al. investigated heavily Mg-doped GaN through variable-temperature Hall-effect measurements and found that the effective acceptor energy depth decreases from 190 to 112 meV and the compensation ratio increases as the dopant density is increased to the high values typically used in device applications. However, there is likely to be an optimum size of the Mg-rich complex since, in p-GaN, $V_N$ has a charge state of +3, and its formation energy will reduce drastically as the Fermi level moves closer to the VBM. As for Mg, its formation energy is not as low as $V_N$ in p-GaN (though it will still decrease by twice the decrease in Fermi level as it shifts toward the VBM). However, heavy Mg-doping will increase the concentration of Mg. Liliental-Weber et al. determined at atomic resolution of the Mg-rich hexagonal pyramids and truncated pyramids in GaN:Mg, to which they attributed the commonly observed decrease in acceptor concentration in the heavily doped material.

The BL band is probably due to a Mg-rich cluster that contains a $V_N$ since it is observed only in heavily doped GaN:Mg samples grown by HVPE or MOCVD, and Hautakangas et al. suggested...
FIG. 3. The calculated equilibrium concentrations of electrons ($n$), holes ($p$), and defects (complexes) studied in this work as a function of temperature. Results from both exchange and correlation density functionals are shown for (a) N-rich condition and (b) mid-way between the N-rich and N-poor conditions.

that neutral or negative complexes involving isolated $V_N$ exist in semi-insulating and $p$-type MOCVD-grown GaN. In contrast, the BL band is not observed in $p$-type, Mg-doped GaN grown by MBE, and Hautakangas et al.\textsuperscript{17} observed no vacancies in MBE-grown GaN:Mg samples. This trend is consistent with that observed for the $p$-type conductivity; it is reported that the maximum hole concentration can be one order of magnitude higher in MBE-grown GaN:Mg than MOCVD-grown samples. As $V_N$ has a higher concentration in MOCVD-grown samples than in MBE-grown ones, we expect that the dominant compensating isolated donors in MOCVD samples will be $V_N$ and Mg, and mostly Mg, in MBE-grown samples. Therefore, the overall compensating effect is more prominent in MOCVD samples, resulting in a different doping efficiency. The activation of $p$-type conductivity by annealing in MOCVD samples is probably due to the dissociation of large open volume defects into $V_N$ and Mg, and mostly Mg, as complexes we propose here, so that H passivation is probably an additional process rather than a limiting factor in Mg doping.

IV. CONCLUSIONS

In conclusion, we show that to achieve $p$-type conductivity in GaN:Mg, the major compensating centers $V_N$ and Mg, can be passivated by binding to Mg$_{Ga}$ centers. Such complexing can reduce the number of holes localized on isolated Mg$_{Ga}$ and form large Mg-rich acceptor complexes that have smaller hole localization energies and lower formation energies than isolated Mg$_{Ga}$. There are two kinds of such large complexes, which differ as to whether they contain a $V_N$. By comparison with experimental observations of the optical and electronic properties of GaN:Mg, we find that the complexes that contain a $V_N$ are likely to prevail in MOCVD-grown GaN but not in MBE-grown GaN. Therefore, the BL band observed in MOCVD- and not MBE-grown GaN is attributed to Mg$_{Ga}$-containing complexes. The requirement for annealing after growth in MOCVD- but not MBE-grown GaN to activate $p$-type conductivity is attributed to the need to dissociate large open volume defects from N monovacancy complexes. Future work will investigate the influence of complexes that contain more than one $V_N$ on $p$-type conductivity.

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AUTHOR DECLARATIONS
Conflict of Interest
The authors have no conflicts to disclose.

Author Contributions
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DATA AVAILABILITY
The data that support the findings of this study are available from the corresponding authors upon reasonable request.

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The formation energies of the configurations in the neutral and next-nearest neighbor cases; the corresponding $+2/0$ charge state are 2.55, 2.54 and 2.93 eV for the axial, basal-plane transition levels are 2.13, 2.05 and 2.22 eV above the VBM.