

Enhanced luminescence of dyes-decorated ZIF-8  
composite films via D-A interactions for white light  
emission

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## ABSTRACT

Metal organic frameworks (MOFs) constructed by metal ions/clusters and organic linkers are used to encapsulate fluorescent guest species with aggregation-caused quenching (ACQ) effects to enhance fluorescence properties due to their porous structures and high specific surface areas. However, there would be a problem of size matching between MOFs pores and guest molecules sizes. In this paper, amorphous ZIF-8 was modified by carboxyl functional groups (H<sub>3</sub>BTC-ZIF-8) via introducing the 1,2,4-benzenetricarboxylic acid (H<sub>3</sub>BTC) ligand into the ZIF-8 sol system. And the H<sub>3</sub>BTC-ZIF-8 was used for organic fluorescent dyes rhodamine 6G (R6G) and coumarin 151 (C151) loading to prepare R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite films. A white-light-emitting composite film (R6G/C151/H<sub>3</sub>BTC-ZIF-8) with a CIE coordinate of (0.323, 0.347) was successfully prepared by compounding fluorescent dyes (R6G and C151) with H<sub>3</sub>BTC-modified ZIF-8, whose photoluminescence quantum yield (PLQY) can reach 64.0%. It was higher than the PLQY of the composite films prepared by crystalline-ZIF-8 (40.2%) or amorphous ZIF-8 without H<sub>3</sub>BTC (48.0%) compounded with the same concentrations of dyes. The fluorescence enhancement was probably attributed to an increased amount of active sites of H<sub>3</sub>BTC-modified ZIF-8 interacted with dyes C151 and R6G. This can form hydrogen bonds between H<sub>3</sub>BTC-ZIF-8 and C151, and weak electron donor-acceptor (D-A) interactions between H<sub>3</sub>BTC-ZIF-8 and R6G molecules, respectively, thus enhancing the interactions between dyes and ZIF-8 and reducing the ACQ effect existing between dye molecules. Therefore, this strategy could provide an important guidance to develop white light emissive materials.

## 1. Introduction

Solid white-light-emitting-diodes (WLED) are the most promising illumination sources because of their excellent luminescence properties like high luminous efficiency, high cycle durability, and environmental friendliness.<sup>1-3</sup> In general, a blue LED chip with a yellow phosphor  $\text{Y}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$  can fabricate a WLED, and the combination of individual of red, blue and green LED chips into a lamp also can achieve white light emission.<sup>1,4-6</sup> However, the synthesis of phosphors requires rigorous conditions and substantial energy cost. Moreover, the difference in the degradation rate between chips and phosphors may cause chromatic aberration and poor white light performance after reuse for many times.<sup>6</sup> In this sense, it is necessary to develop emerging luminescent materials. Up to now, many luminescent materials such as rare earth materials,<sup>7-9</sup> organic fluorescent dyes,<sup>10-13</sup> perovskites,<sup>14-20</sup> quantum dots,<sup>21-24</sup> metal organic frameworks (MOFs),<sup>4,5,13</sup> and carbon dots,<sup>25-27</sup> show great potentials in structure versatility, wavelength tunability and short luminescence lifetimes, and have been used in white LEDs.

Among these luminescent materials, metal organic frameworks (MOFs) constructing metal ions/clusters with organic ligands, have been promising inorganic-organic hybrid porous materials. Compared with other porous materials, MOFs exhibit high specific surface areas, adjustable pore sizes, and high degrees of functionalization.<sup>28-30</sup> Additionally, luminescent MOFs (LMOFs)<sup>31</sup> have attracted much attention to develop excellent luminescent functional materials in recent years due to their inherent luminescent centers of metal nodes/clusters such as rare earth cations, and chromophoric ligands.<sup>32-35</sup> On the other hand, many researchers have tried to encapsulate other luminescent materials like organic fluorescent dyes using MOFs to obtain composite materials,<sup>1,33,36-40</sup> which can exhibit better fluorescence performance.

Organic fluorescent dyes are one of main groups of luminescent materials, presenting good luminescent performance including large absorption coefficient, high quantum yields, and wide excitation band.<sup>1,41</sup> Unfortunately, most commercial organic fluorescent dyes often suffer from a non-radiative process due to strong aggregation-caused quenching (ACQ) effect between dye intermolecular when they are in solid states or highly concentrated solutions, which would greatly reduce fluorescence properties or generate even no fluorescence.<sup>41,42</sup> In this sense, encapsulating organic fluorescent dyes into MOFs was a good strategy to enhance luminescence performance and weaken ACQ effects between dyes. However, there would be a problem of size matching between MOFs pores and dye molecules. Hence, it is very important to select MOF materials with appropriate pore sizes to encapsulate corresponding dye molecules, which can effectively weaken the ACQ effect between dyes molecules. Therefore, there are still difficulties and limitations in using MOFs to encapsulate organic fluorescent dyes.

In our work, we provided a new method to combine fluorescent dyes with MOF materials, the MOF structure was modified with functional groups to enhance the interaction between dyes and MOFs, thus weakening the ACQ effect between dyes molecules and improving the fluorescence performance. Firstly, an amorphous ZIF-8 (one type of zeolitic imidazole frameworks, belonging to MOFs<sup>43</sup>) film with carboxyl functional groups modified was obtained by adding some H<sub>3</sub>BTC into the ZIF-8 reaction system, and the optical band gap of the H<sub>3</sub>BTC-modified ZIF-8 was decreased to 4.06 eV comparing with the amorphous ZIF-8 ( $E_g=5.17$  eV). And a white-light-emitting composite film (R6G/C151/H<sub>3</sub>BTC-ZIF-8) with a CIE coordinate of (0.323, 0.347) was successfully prepared by compounding fluorescent dyes (R6G and C151) with H<sub>3</sub>BTC-modified ZIF-8, whose photoluminescence quantum yield can reach 64.0%. It was higher than the PLQY of the composite films prepared by crystalline ZIF-8 (40.2%) or amorphous ZIF-8 (48.0%)

compounded with the same concentrations of dyes. The results suggested that the interaction sites **between** dyes (C151 and R6G) and ZIF-8 can be increased by modifying the structure of ZIF-8 using additional ligands (H<sub>3</sub>BTC), enhancing the interactions **between** dyes and ZIF-8, **and** reducing the ACQ effect between dyes molecules. **Finally**, better immobilization and dispersion of dye molecules in **the H<sub>3</sub>BTC-ZIF-8 film** can be achieved to improve the **whole** fluorescence performance **in** the composite film.

## 2. Experiment Section

### 2.1. Materials

Zinc acetate dihydrate ( $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$ ,  $\geq 99.9\%$ , Shanghai China), 2-methylimidazole (Hmim,  $\text{C}_4\text{H}_6\text{N}_2$ , 99%, Sigma-Aldrich), 1,2,4-benzenetricarboxylic acid (H<sub>3</sub>BTC,  $\text{C}_9\text{H}_6\text{O}_6$ , Macklin), Ethanol ( $\geq 99.7\%$ , Shanghai China), 7-amino-4-trifluoromethyl coumarin (C151,  $\text{C}_{10}\text{H}_6\text{F}_3\text{NO}_2$ , 99%, Aladdin), and Rhodamine 6G (R6G,  $\text{C}_{28}\text{H}_{31}\text{N}_2\text{O}_3\text{Cl}$ , WoKai) were used without further purification, and ultrapure water was obtained through the purified water system (He Tai).

### 2.2. Preparation of **H<sub>3</sub>BTC-ZIF-8** film and R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film

#### 2.2.1. Preparation of **H<sub>3</sub>BTC-ZIF-8** precursor sols

The preparation process of the **H<sub>3</sub>BTC-ZIF-8** precursor sol was similar to our previous work,<sup>44</sup> and specific preparation process was as followed. Firstly, H<sub>3</sub>BTC ligands (0.75 mmol) and Hmim ligands (25 mmol) (The mole ratio of Hmim to H<sub>3</sub>BTC was 100:3) were mixed in 2 mL ethanol, and then stirred at room temperature to ensure complete dissolution of H<sub>3</sub>BTC powders. Afterwards, 10 **mmol**  $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$  powders were added into the well-mixed solution under the continuous stirring process. Finally, the steady transparent **H<sub>3</sub>BTC-ZIF-8** precursor sol was obtained.

### 2.2.2. Preparation of H<sub>3</sub>BTC-ZIF-8 films

The H<sub>3</sub>BTC-ZIF-8 precursor sol was coated on a clear quartz glass through a spin-coating method to obtain a H<sub>3</sub>BTC-ZIF-8 wet film. Then, the H<sub>3</sub>BTC-ZIF-8 wet film was cured on a hot plate. The specific curing procedure was at 60 °C for 30 mins firstly, increasing the temperature by 20 °C per 30 mins to 120 °C and maintaining for 5 h. Finally, the H<sub>3</sub>BTC-ZIF-8 film was obtained. It is noteworthy that the amount of H<sub>3</sub>BTC in the H<sub>3</sub>BTC-ZIF-8 film can be controlled by changing the mole ratio of Hmim to H<sub>3</sub>BTC in the H<sub>3</sub>BTC-ZIF-8 precursor sol.

### 2.2.3. Preparation of R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite films

The H<sub>3</sub>BTC-ZIF-8 precursor sol was used as the solvent to disperse organic fluorescent dyes C151 and R6G, in which the mole ratio of Hmim to H<sub>3</sub>BTC was 100:3. Firstly, 0.005 mmol C151 and 0.005 mmol R6G powders were mixed and dissolved in 5 mL of the H<sub>3</sub>BTC-ZIF-8 precursor sol to form a transparent composite sol (The concentration of C151 and R6G were both 0.001 mol/L). After that, the same method of preparation of H<sub>3</sub>BTC-ZIF-8 films was used to prepare C151/R6G/H<sub>3</sub>BTC-ZIF-8 composite films. Among them, the concentrations of C151 and R6G in the obtained composite films were stood for the concentrations in the composite sol due to the homogeneity of the sol. And the colour of emission could be controlled by the concentrations of C151 and R6G dyes in the sol.

## 2.3. Characterizations

The phase structures of obtained films were characterized by X-ray diffraction (XRD) adopting grazing incidence scanning with the Cu K $\alpha$  radiation at a scanning rate of 0.02°·2 $\theta$ /s. The morphology and film thicknesses of films were employed by a field-

emission scanning electron microscope (FESEM) with an equipment of energy disperse spectroscopy analysis (EDS). The functional groups in microstructures of obtained films were detected by Fourier transform infrared spectra (FTIR) in the mid-infrared spectral range from 4000 to 400  $\text{cm}^{-1}$ . To characterize possible structure changes on the surface of the ZIF-8 films before and after compounding fluorescent dyes, X-ray photoelectron spectrometer (XPS) tests accompanied with the ultraviolet photoelectron spectroscopy (UPS) characterizations were performed. The UV-Vis absorption spectra were obtained using a UV-Vis spectrophotometer with the wavelength range from 200 to 800 nm. The photoluminescence spectra (PL) of all films were measured at room temperature with a fluorescence spectrophotometer. The PL quantum efficiency was measured using a UV-NIR absolute PLQY spectrometer (Quantaaurus-QY Plus, Hamamatsu, Japan). The fluorescence lifetime decay curves of films were obtained with a time-resolved fluorescence spectrometer (Newport, USA), and the excitation light was pumped by a femtosecond laser.

### 3. Results and discussions

#### 3.1. Structure characterizations

Figure S1 showed XRD patterns of ZIF-8 film without H<sub>3</sub>BTC, H<sub>3</sub>BTC-ZIF-8 film and R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film synthesized by a sol-gel method. The results showed that all samples had a strong and broad peak at 15°, indicating the formation of amorphous ZIF-8 phase<sup>44</sup> both before and after adding the fluorescent dyes (R6G and C151). This may be that the amount of dyes added was too small to detect the change. In order to further demonstrate the prepared amorphous film was amorphous ZIF-8, the amorphous film without H<sub>3</sub>BTC was heated to 140 °C for 1 h and then characterized by

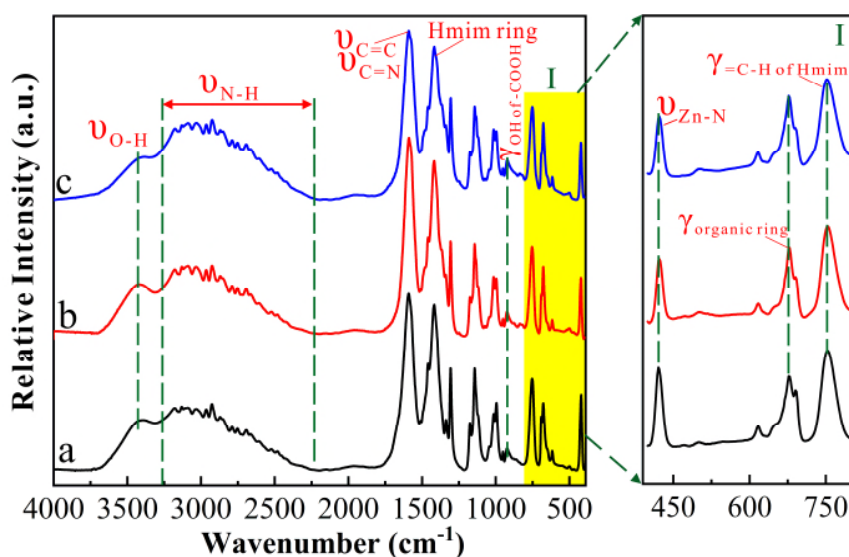
XRD. The result was shown in **Figure S2**. It can be seen that obvious peaks of ZIF-8 crystal can be observed, indicating that the nucleation condition of ZIF-8 was not reached under our experimental temperature of 120 °C in our reaction system and **thus** the amorphous ZIF-8 was formed.

**To observe morphologies of these films**, FESEM characterizations were carried out shown in **Figure S3**. It can be seen from the **Figure S3a that**, the surface of the ZIF-8 film synthesized without H<sub>3</sub>BTC ligand was smooth, and there were many small particles on the surface after magnification and **average size of particles** was approximately 50 nm. **As for the H<sub>3</sub>BTC-ZIF-8 film, particles** on the surface of the **prepared** film are smaller shown in **Figure S3e**. Additionally, it cannot be seen particles on the surface of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film (**Figure S3h**). The reason for this phenomenon may be that the addition of both H<sub>3</sub>BTC and dyes (R6G and C151) may affect the coordination of Zn<sup>2+</sup> with Hmim in the ZIF-8 skeleton, resulting in the difference in particle sizes. **Moreover,** the thicknesses of above three films were 6.91, 6.01 and 8.05 μm, respectively. To further explore the elemental distribution of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film, element mapping has been employed shown in **Figure S4**. It can be seen from the result that elements of C, N, O, Zn and F were exhibited a uniform distribution, suggesting the homogeneity of the composite film.

To investigate the structural difference between ZIF-8 films and the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film, FTIR characterizations were performed shown in **Figure 1**. The absorption peak at 422 cm<sup>-1</sup> was attributed to Zn-N stretching vibration, and the peak at ~678 cm<sup>-1</sup> was the out-of-plane bending vibration of the organic ring. Peaks at ~753 and 1143 cm<sup>-1</sup> belonged to =C-H bending vibrations. The peaks at ~995 cm<sup>-1</sup> and ~1175 cm<sup>-1</sup>



were corresponding to C=C-N twisting and C-N bending vibration, respectively. The peak at  $\sim 1015\text{ cm}^{-1}$  belonged to C-H out-of-plane bending vibration of  $\text{CH}_3$ .<sup>45,46</sup> And a broad absorption band at the range of  $3200\text{ to }2200\text{ cm}^{-1}$  indicated the existence of N-H stretching vibration due to protonated imidazole and the peak at  $\sim 3400\text{ cm}^{-1}$  was associated with O-H stretching vibration.<sup>47-49</sup> Additionally, the peak ( $1591\text{ cm}^{-1}$ ) with the highest relative intensity is attributed to C=C/ C=N or C=O stretching vibration due to the presence of conjugated structures in Hmim or  $\text{H}_3\text{BTC}$  ligands.<sup>50</sup> These results indicated that ZIF-8 frameworks with protonated imidazole structures has been formed (Figure 1a). Moreover, a wide band at the range of  $955\sim 915\text{ cm}^{-1}$  was attributed to the out-of-plane deformation vibration of O-H, which was used to confirm the presence of COOH.<sup>47</sup> In this sense, the ZIF-8 has been modified by carboxyl groups successfully. Although no significant peak changes were observed after forming the R6G/C151/ $\text{H}_3\text{BTC}$ -ZIF-8 composite film, the relative strength of Zn-N, a key structure in the formation of ZIF-8 framework, had been decreased in the  $\text{H}_3\text{BTC}$ -ZIF-8 film in Figure 2I, comparing with out-of-plane bending vibration of =C-H of Hmim ( $\gamma_{\text{C-H}}$ ). The possible reason for this phenomenon was that  $\text{H}_3\text{BTC}$  may affect the coordination environment of ZIF-8 and eventually lead to a change in the number of N active sites used for coordination with Zn.



**Figure 1** FTIR spectra of ZIF-8 film without H<sub>3</sub>BTC (a), H<sub>3</sub>BTC-ZIF-8 film (b), and R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film (c). I. The enlarged FTIR spectra with a wavenumber range from 390 to 800 cm<sup>-1</sup>.

To further investigate the role of the ligand H<sub>3</sub>BTC in the formation of the ZIF-8 structure, sols prepared in different mole ratios of Hmim to H<sub>3</sub>BTC without zinc salts and ZIF-8 films prepared in different mole ratios of Hmim to H<sub>3</sub>BTC in the system in the presence of zinc salts were further characterized by FTIR. Prior to this, ligands Hmim and H<sub>3</sub>BTC were also characterized by FTIR, and results were shown in Table S1 and S2 respectively.<sup>44,51</sup> In general, both Hmim and H<sub>3</sub>BTC are easy to form intermolecular hydrogen bonds in solid state.<sup>45,51</sup> However, the types of intermolecular hydrogen bonds would be changed after adding H<sub>3</sub>BTC into Hmim, results have been shown in Figure S5. It can be seen that the vibrational peaks of both Hmim and H<sub>3</sub>BTC were existed in the sol samples mixed Hmim with H<sub>3</sub>BTC, and the absorption peaks due to the resonance between  $\nu_{\text{N-H}}$  and  $\gamma_{\text{N-H}\cdots\text{N}}$ <sup>52</sup> moved to high wavenumber with the increase of H<sub>3</sub>BTC. Also, the proportion of hydrogen bonds formed between Hmim molecules was decreased

in terms of the peak shape of the formed hydrogen bonds (2200-3200  $\text{cm}^{-1}$ ).<sup>48</sup> The possible reason for this phenomenon was the formation of hydrogen bonding interactions between Hmim and H<sub>3</sub>BTC, creating new hydrogen bonding species, i.e. N-H...O or O-H...N hydrogen bonds. Additionally, FTIR spectra (Figure S6) of ZIF-8 films prepared with different mole ratios of Hmim to H<sub>3</sub>BTC showed that the relative intensity of stretching vibration of Zn-N decreased with the increase of H<sub>3</sub>BTC (Figure S6a). And as the mole ratio increased, the deformation vibration of =C-H from benzene ring gradually became obvious (Figure S6b). In Figure S6c, the vibrations of characteristic functional groups from H<sub>3</sub>BTC increased with the increase of the mole ratio, and the absorption peak at  $\sim 1590 \text{ cm}^{-1}$  widened due to the increased conjugated system. These results have indicated that H<sub>3</sub>BTC modified ZIF-8 was indeed formed and the interaction between H<sub>3</sub>BTC and Hmim inhibited the formation of the ZIF-8 structure to a certain extent. On the other hand, the presence of a certain number of carboxyl functional groups on the H<sub>3</sub>BTC-modified ZIF-8 provided some convenience for following dyes loading. In a word, the addition of H<sub>3</sub>BTC ligands to the ZIF-8 synthesis could modulate the coordination of Zn with Hmim and also modify the ZIF-8 structure with functional groups of -COOH. It was worth noting that the addition of H<sub>3</sub>BTC shouldn't be too much, which would lead to serious inhibition of Zn coordinated with Hmim, and eventually failed to form the ZIF-8 skeleton resulting in poor stability.

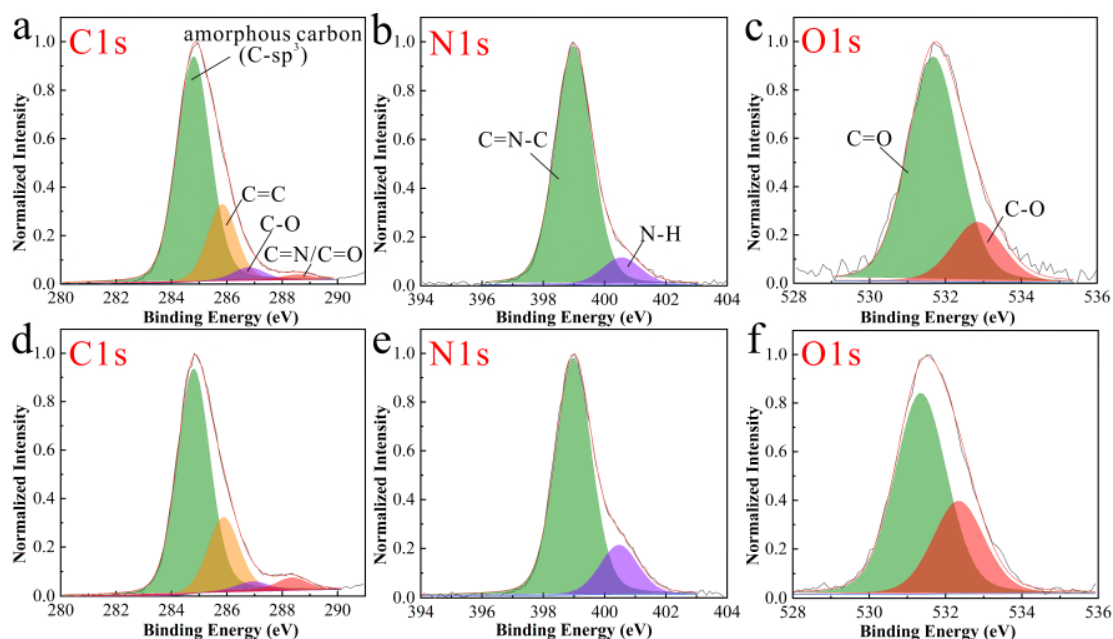
Additionally, ZIF-8 sols and film samples at different curing temperatures after film formation were also characterized by FTIR, as shown in Figure S7. It can be seen from the figure that both in the system without H<sub>3</sub>BTC (Figure S7a) and the system containing H<sub>3</sub>BTC ligands (Figure S7b), the relative intensity of Zn-N stretching vibration in formed

sols was very small, marked yellow **area 1 and 5**, indicating that the ZIF-8 skeleton was almost not formed in sols. As the increasing of the curing temperature, the intensity of the Zn-N stretching vibration was significantly enhanced, implying the **gradual** formation of the ZIF-8 structure. The **area 2 and 6** in **Figure S7** showed the out-of-plane deformation vibration of O-H from -COOH and the relative strength was decreased with the increasing of the curing temperature. Also, the absorption peaks due to the resonance between  $\nu_{\text{N-H}}$  and  $\gamma_{\text{N-H}\cdots\text{N}}$  (**area 3 and area 7**), and protonated imidazole (**area 4 and area 8**) were disappeared and decreased, respectively, indicating reduced interactions between imidazole molecules. The reason for these results may be that increasing temperature can not only promote the volatilization of solvent molecules in wet films, but also destroy part of hydrogen bond interactions between imidazole molecules or between imidazole molecules and acidic small molecules (acetic acid or H<sub>3</sub>BTC), which would promote the coordination of Zn and imidazole, and resulting in the formation of the ZIF-8 structure. In this sense, **increasing** temperature was favourable to the formation of ZIF-8. Furthermore, the relative intensity of Zn-N in the ZIF-8 film without H<sub>3</sub>BTC was higher than that of the ZIF-8 film with H<sub>3</sub>BTC, suggesting that the ligand H<sub>3</sub>BTC modulated the coordination environment of ZIF-8.

To demonstrate that Zn<sup>2+</sup> didn't coordinate with H<sub>3</sub>BTC and to verify our previous results, XPS **characterizations** were performed on ZIF-8 films synthesized with/without **H<sub>3</sub>BTC and H<sub>3</sub>BTC powders**. **XPS survey spectra showed the presence of Zn, C, N, O elements in both the ZIF-8 film without H<sub>3</sub>BTC (Figure S8c) and H<sub>3</sub>BTC-ZIF-8 film (Figure S8a)**. Because there was no O element in the ZIF-8 structure, the O element was mainly derived from acid molecules. Additionally, the contents of C and O relative to Zn

increased significantly, suggesting that H<sub>3</sub>BTC-modified ZIF-8 structures have been formed. And Zn 2p spectra of ZIF-8 without H<sub>3</sub>BTC (Figure S8d) had two peaks centred at ~1045 and ~1022 eV, corresponding with binding energies of Zn 2p<sup>1/2</sup> and Zn 2p<sup>3/2</sup>, respectively.<sup>53</sup> It was worth noting that the two peaks of Zn 2p moved towards the higher binding energy by ~1 eV for the H<sub>3</sub>BTC-ZIF-8 film comparing with the ZIF-8 film without H<sub>3</sub>BTC modified, indicating that there may be some interaction between ZIF-8 and H<sub>3</sub>BTC resulting in the peak offset of Zn 2p. The XPS high resolution of C1s spectra of ZIF-8 films (Figure 2a and 2d) through peak fitting contained four types of the carbon environment with binding energy values of ~284.8 eV, ~285.8 eV, 286.8 eV and ~288.5 eV, which mainly originated from the graphitic carbon/C-sp<sup>3</sup>, C=C, C-O and C=N/C=O, respectively.<sup>44,54</sup> When adding H<sub>3</sub>BTC ligand to the ZIF-8 system, the relative intensity of C=N/C=O in Figure 2d was increased, indicating the successful synthesis of carboxyl-modified ZIF-8. Furthermore, XPS high resolution of N1s (Figure 2b and 2e) and O1s (Figure 2c and 2f) spectra were important for the proof of the coordination in ZIF-8, and the results indicated that the relative intensity of protonated N (N-H) in the H<sub>3</sub>BTC-ZIF-8 film was significantly enhanced, comparing with the ZIF-8 film synthesized without H<sub>3</sub>BTC. This suggested that adding H<sub>3</sub>BTC inhibited the deprotonation of Hmim, which eventually led to a decrease in N sites coordinated with Zn. At the same time, peaks of O1s indicated the presence of oxygen only from small molecules such as H<sub>3</sub>BTC, CH<sub>3</sub>COOH, H<sub>2</sub>O and CO<sub>2</sub>, and no coordination of O with Zn was found, which proved that H<sub>3</sub>BTC only changed the N environment of Zn coordinated with Hmim and did not participate in the formation of ZIF-8 structures.

In summary, the addition of H<sub>3</sub>BTC ligand to the ZIF-8 synthesis system can regulate the N environment of ZIF-8 and modify carboxyl functional groups on ZIF-8 structure. Furthermore, it does not participate in the coordination with Zn.



**Figure 2** XPS high resolution C1s spectra (a and d), N1s spectra (b and e) and O1s spectra (c and f) of ZIF-8 films without H<sub>3</sub>BTC (a-c) and H<sub>3</sub>BTC-ZIF-8 film (d-f).

Next, interactions between fluorescent dyes (R6G and C151) and H<sub>3</sub>BTC-ZIF-8 in the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film was investigated. Based on others' works previously, there were several main interactions among MOF hosts and guest species, including electrostatic interaction,  $\pi$ - $\pi$  interaction, ion exchange, hydrogen bonds, and so on.<sup>55</sup> Firstly,  $\pi$ - $\pi$  interactions have been formed between H<sub>3</sub>BTC-ZIF-8 and dyes due to the existence of  $\pi$  conjugation systems in both H<sub>3</sub>BTC-ZIF-8 and dye molecules (R6G and C151).<sup>47,56</sup> On the other hand, to investigate whether there were other interactions between H<sub>3</sub>BTC-ZIF-8 and dye molecules, FTIR characterizations were performed on

composite films with different concentrations of single fluorescent dyes and results have shown in Figure S10. As seen from Figure S10a, when the concentration of R6G was 0.001 mol/L, the relative intensity of Zn-N had little changes compared with the H<sub>3</sub>BTC-ZIF-8 film without dyes. While the concentration of R6G was up to 0.01 mol/L, the relative intensity of Zn-N was increased. And the stretching vibration of O-H also weakened with the increase of R6G. Since R6G is a cationic dye (R6G<sup>+</sup>), it can serve as an electron acceptor, and there are carboxyl functional groups in H<sub>3</sub>BTC-ZIF-8 structure acting as an electron donor.<sup>57</sup> In this sense, the donor-acceptor (D-A) interactions may exist between H<sub>3</sub>BTC-ZIF-8 and R6G, which weakened the inhibition of H<sub>3</sub>BTC on ZIF-8 structure resulting in an enhanced coordination degree between Zn and Hmim. However, when the concentration of R6G was too high, its fluorescence performance would be decreased because of ACQ effects. In this sense, the concentration of R6G incorporation should not be too high.

A similar phenomenon was observed in C151/H<sub>3</sub>BTC-ZIF-8 composite films shown in Figure S10b, and results showed that the relative intensity of Zn-N stretching vibrations increased more obviously with the increasing concentration of C151 comparing with the R6G/H<sub>3</sub>BTC-ZIF-8 composite film. Moreover, the stretching vibration intensity of O-H was weakened more clearly, suggesting that the fluorescent dye C151 was more easily to promote the coordination of Zn with Hmim in dyes/H<sub>3</sub>BTC-ZIF-8 composite film comparing with R6G. Since C151 has three types of structures that can form hydrogen bonds which include participation of the amino nitrogen (N), the carbonyl oxygen (C=O), and amino hydrogen (N-H),<sup>58</sup> hence, the possible reason for this phenomenon was the existence of hydrogen bonding interactions between N-H functional groups in C151 and

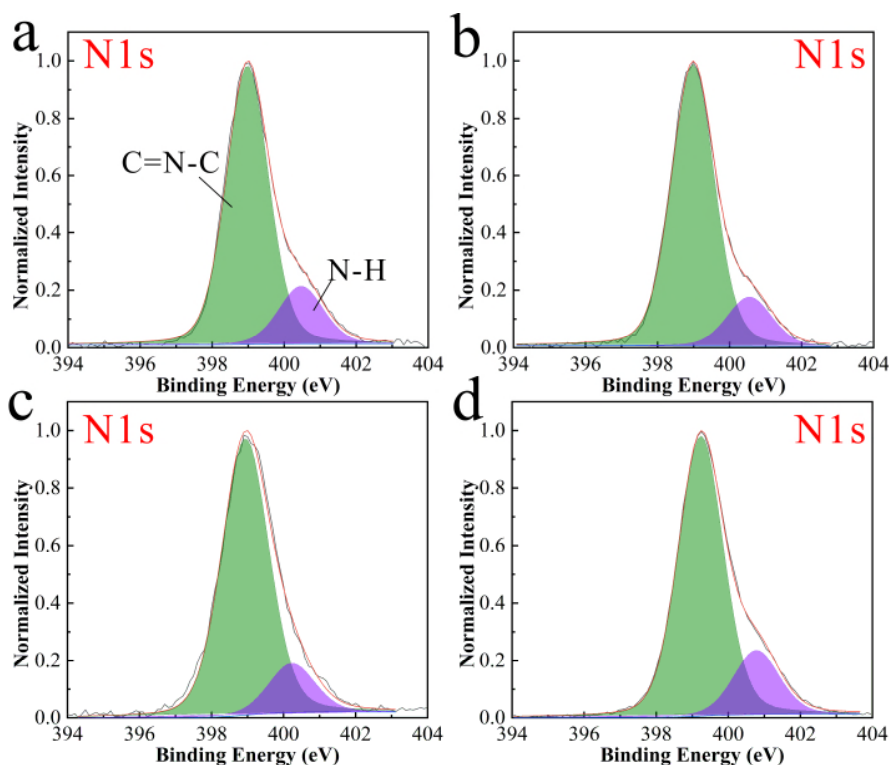
carboxyl groups (-COOH) in H<sub>3</sub>BTC-ZIF-8, and protonated N sites (N-H) in ZIF-8 could also form N-H...O=C hydrogen bonds with the carbonyl oxygen of C151, which both can weaken the inhibitory effect of H<sub>3</sub>BTC on the formation of the ZIF-8 framework, and finally the interplay of multiple roles makes the whole system more stable.<sup>56,59</sup>

In order to further prove that dyes R6G and C151 have attenuated the inhibitory effect of H<sub>3</sub>BTC on ZIF-8, XPS high resolution N1s characterizations of dyes/H<sub>3</sub>BTC-ZIF-8 composite films were performed, as shown in Figure 3. It can be seen from the figure that when the concentration of R6G was 0.001 mol/L, the relative strength of protonated N was decreased compared with the H<sub>3</sub>BTC-ZIF-8 film (Figure 3a and 3b), indicating that R6G molecules formed weak D-A interactions with the electron donor O in H<sub>3</sub>BTC-modified ZIF-8 that weakened the inhibition effect of Hmim deprotonation by H<sub>3</sub>BTC. This interaction resulted in a decrease in the proportion of protonated N sites, corresponding to a relative decrease in the proportion of N-H in Figure 3b. Additionally, when the concentration of C151 was 0.001 mol/L (Figure 3c), the relative intensity of N-H was also significantly weakened compared with that of the H<sub>3</sub>BTC-ZIF-8 film, indicating that the hydrogen bonds formed between C151 and H<sub>3</sub>BTC-ZIF-8 could effectively weaken the inhibitory effect of H<sub>3</sub>BTC on deprotonation of Hmim, consistent with the FTIR result (Figure S10b).

When the concentration of both R6G and C151 used to prepare a R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film was 0.001 mol/L, there was a certain competition between them due to the limited and overlapped action sites of H<sub>3</sub>BTC-ZIF-8, which made the degree of the interaction between dyes (R6G and C151) and H<sub>3</sub>BTC-ZIF-8 changed. Therefore, it was possible to optimize the distribution of R6G and C151 in the H<sub>3</sub>BTC-ZIF-8 structure



through modulating the amount carboxyl functional groups of ZIF-8 after H<sub>3</sub>BTC modification, which can effectively reduce the ACQ effect of dye molecules and finally obtain a composite film with excellent fluorescence performance.

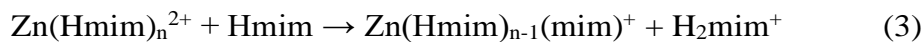


**Figure 3** XPS high resolution N1s spectra of the H<sub>3</sub>BTC-ZIF-8 film without dyes (a), the R6G/H<sub>3</sub>BTC-ZIF-8 film with a concentration of R6G of 0.001 mol/L (b), the C151/H<sub>3</sub>BTC-ZIF-8 film with a concentration of C151 of 0.001 mol/L (c), and the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film with dyes (both R6G and C151) concentrations of 0.001 mol/L (d).

### 3.2. Structure formation mechanism

In this work, the solid-state white light composite film with high fluorescence quantum efficiency was successfully prepared by the interaction of the H<sub>3</sub>BTC-modified ZIF-8 structure with fluorescent dyes R6G and C151. The possible mechanism of interactions

between dyes (R6G and C151) and H<sub>3</sub>BTC-ZIF-8 was shown in **Figure 4**. In a ZIF-8 sol system without H<sub>3</sub>BTC ligands, the mole ratio of Hmim to Zn source (Zn(CH<sub>3</sub>COO)<sub>2</sub>·2H<sub>2</sub>O) was relatively low (2.5:1), and small acidic molecules CH<sub>3</sub>COOH produced by the hydrolysis of Zn source inhibited the deprotonation process of Hmim, which would make a low degree of coordination between Zn and Hmim in the sol. Specific reaction equations are as follows:<sup>44</sup>

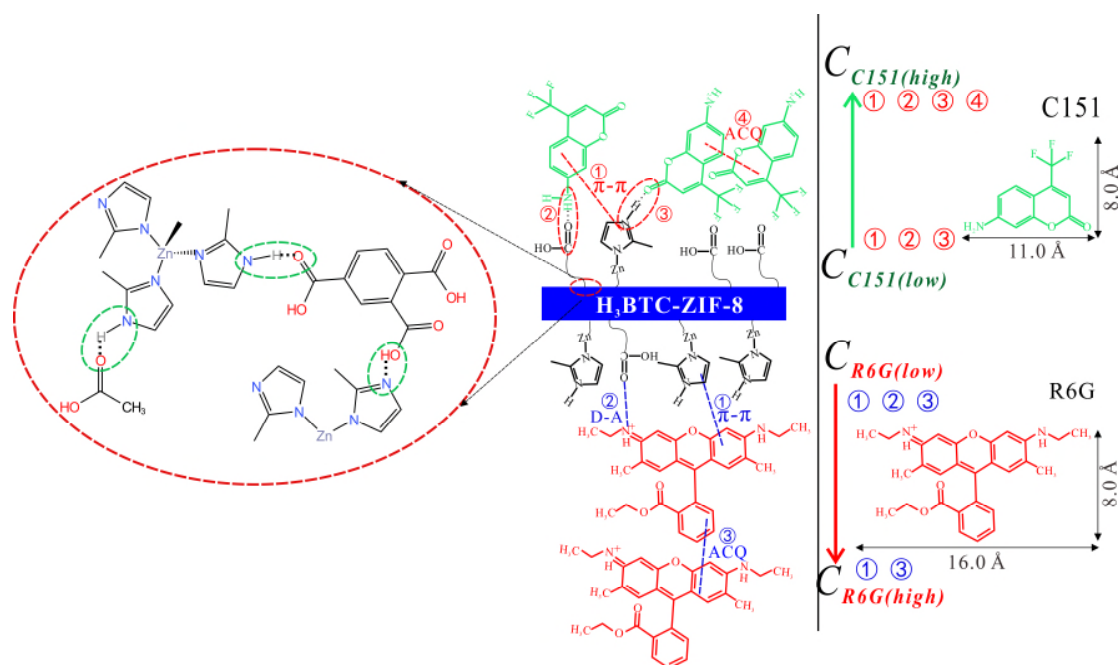


**When the sol forms solid films through a spin-coating method and curing,** the inhibition effect on Hmim deprotonation was weakened because of the volatilization of some small molecules acetic acid during the curing process, which promoted the coordination of Zn with imidazole to form the ZIF-8 skeleton. However, it is difficult to remove all acetic acid molecules due to the existence of hydrogen bonds between CH<sub>3</sub>COOH and protonated N (N-H) of imidazole.<sup>44</sup> Hence, **some** acetic acid existed in the ZIF-8 film, which was consistent with **Figure 1** and **Figure S7a**. When a small amount of H<sub>3</sub>BTC was added to the ZIF-8 sol, more carboxyl functional groups present in H<sub>3</sub>BTC formed a strong hydrogen bonds (N-H···O and N···H-O) with N sites in Hmim, and these were not easily destroyed during curing at low temperatures (120 °C). Therefore, some hydrogen bonds and carboxyl functional groups have been preserved in the H<sub>3</sub>BTC-ZIF-8 after curing and forming the film. Since the mole ratio of H<sub>3</sub>BTC relative to Hmim was very small (3:100), and also, the system was under acidic conditions, in this sense, there would

be protonated N and carboxyl functional groups in the film, which facilitated interactions with fluorescence dye molecules. In a R6G/H<sub>3</sub>BTC-ZIF-8 composite film at a low concentration of R6G, R6G molecules presented a better dispersion in the ZIF-8 structure, and the interaction between them was dominated by  $\pi$ - $\pi$  stacking because of the presence of a large number of  $\pi$  conjugation systems both in ZIF-8 and R6G molecules.<sup>55</sup> Moreover, since R6G molecules belong to cationic dyes, they can act as electron acceptors to form D-A interactions with electron-rich structures (-COOH) in H<sub>3</sub>BTC-ZIF-8, as shown in Figure 4. However, the relatively large size ( $16.0 \times 8.0 \text{ \AA}$ )<sup>60</sup> of single R6G molecule made the D-A interaction weak, and R6G molecules were not completely highly dispersed in the H<sub>3</sub>BTC-ZIF-8 structure, so there was also H-aggregate between R6G molecules resulting in ACQ effects.<sup>39,61</sup> When the concentration of R6G gradually increased, its  $\pi$ -conjugated structures gradually increased, which led to an enhanced  $\pi$ - $\pi$  stacking interaction between H<sub>3</sub>BTC-ZIF-8 and R6G. At the same time, the limited numbers of active sites on ZIF-8 molecules and the existence of stereo-hindrance effects would enhance R6G intermolecular interactions and gradually enhance the ACQ effect, and finally the fluorescence performance would be weakened.

On the other hand, in the C151/H<sub>3</sub>BTC-ZIF-8 composite film, when the concentration of C151 was low, it could keep well interaction with the H<sub>3</sub>BTC-ZIF-8 due to its small molecular size ( $8.0 \times 11.0 \text{ \AA}$ )<sup>13</sup> and forming hydrogen bonds, and the existence of  $\pi$  conjugation system. These interactions would make C151 highly and stably dispersed in the H<sub>3</sub>BTC-ZIF-8 film. When the concentration of C151 was increased, part of C151 could only exist in the ZIF-8 structure by intermolecular interaction due to the saturation of the interaction between C151 and H<sub>3</sub>BTC-ZIF-8, which would enhance the ACQ effect

between C151 molecules shown in Figure 4. When adding the two dyes into the H<sub>3</sub>BTC-ZIF-8 synthetic system, they would compete with each other for the overlapped H<sub>3</sub>BTC-ZIF-8 active sites. Since H<sub>3</sub>BTC ligands can regulate numbers of interaction sites with dyes, the reasonable regulation of ZIF-8 structures by H<sub>3</sub>BTC may reduce the ACQ effect of C151 and R6G in the composite structure to bring out their respective maximum fluorescence performance.



**Figure 4** A possible structure formation mechanism illustration of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film.

### 3.3. Luminescent properties

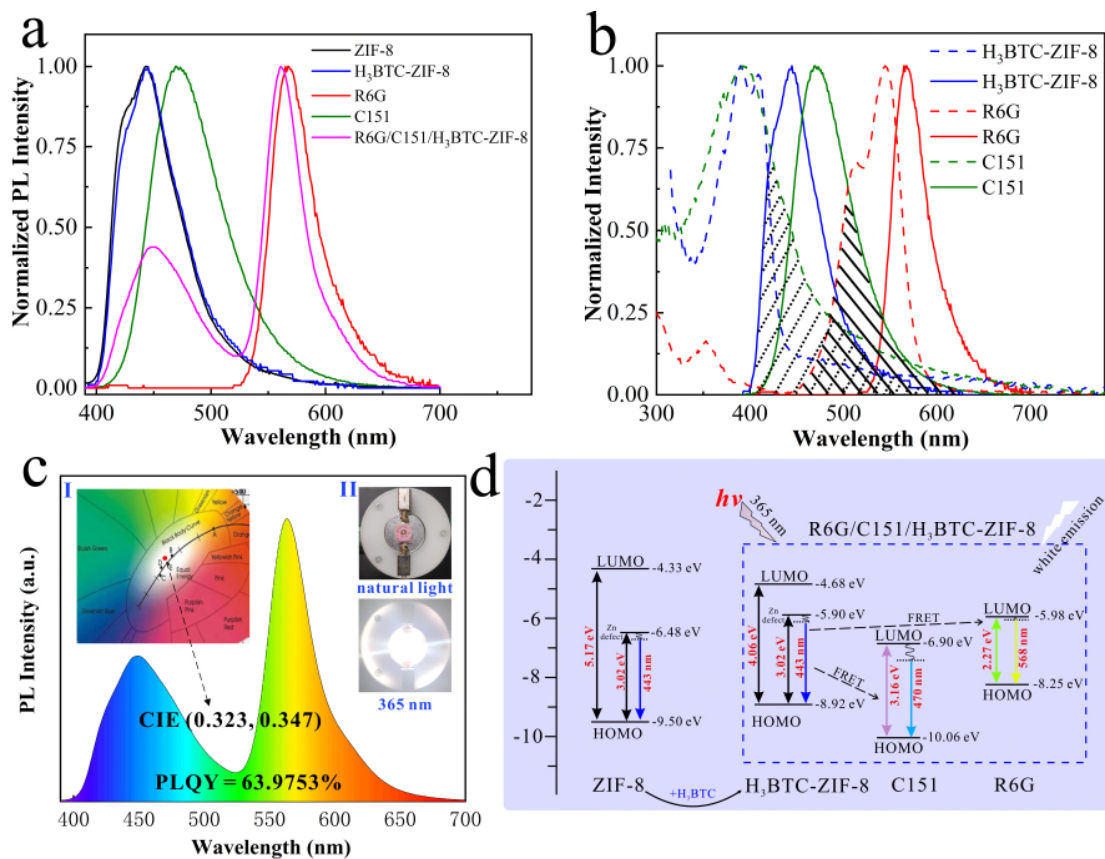
In order to explore the luminescent properties of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film, firstly, the UV-Vis absorption spectra of ZIF-8 films and the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film have been employed shown in Figure S11. It

can be seen from the results that all films exhibited strong absorption edges in the ultraviolet region, and it was obvious that the absorption edge of the **H<sub>3</sub>BTC-ZIF-8** film moved from ~240 nm to ~305 nm comparing with the **ZIF-8** film. Furthermore, the intensities of the absorption peaks at ~390 nm and ~540 nm of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film increased slightly after the dye molecules were compounded. In order to further investigate origins of these absorption differences among these films, the UV-Vis absorption spectra of reactants including ligands and dyes used have been characterized (**Figure S12**). Results have shown that the absorption edge of the ZIF-8 film without H<sub>3</sub>BTC at ~240 nm was from the imidazole Hmim ligand. And the absorption edge of H<sub>3</sub>BTC was approximately 350 nm, which may be the reason for the movement of absorption edge between ZIF-8 film without H<sub>3</sub>BTC and H<sub>3</sub>BTC-ZIF-8 film. Similarly, the peaks at ~393 nm and ~545 nm in the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film originated from the organic fluorochromes C151 and R6G, respectively. Interestingly, in the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite, the relative intensity of the electronic vibrational shoulder at 500 nm (0.305) **originated from** the formation of different types of dimers<sup>62</sup> comparing with the peak centred at 545 nm was significantly smaller than that of the single dye R6G (0.732), indicating that the formation of the composite film combining dyes with ZIF-8, to a certain extent, can weaken the ACQ effect of dyes and improve fluorescence performances.

On the other hand, the emission spectra of ZIF-8 films, the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film, and fluorescent dyes R6G and C151 at room temperature have been examined to determine the luminescent property of the **dyes/H<sub>3</sub>BTC-ZIF-8** composite film in **Figure 5a**. It can be seen that fluorescence emission peaks of C151 and R6G at an

excitation wavelength of 365 nm were at 470 and 568 nm, respectively. There was almost no change in blue emission (443 nm) between the ZIF-8 film and H<sub>3</sub>BTC-ZIF-8 film, indicating that H<sub>3</sub>BTC did not change the main structure of ZIF-8.

Furthermore, the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film was obtained by adding fluorescence dyes R6G and C151 to the H<sub>3</sub>BTC-ZIF-8 synthesis. And two fluorescent emission peaks (~450 and ~570 nm) are attributed to the emission peaks of ZIF-8, C151 and R6G superimposed on each other.



**Figure 5** (a) Normalized PL spectra ( $\lambda_{ex}=365$  nm) of dyes (R6G and C151), ZIF-8 films H<sub>3</sub>BTC-ZIF-8 film and R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film; (b) Emission spectra (solid lines) and absorbance spectra (dotted lines) of H<sub>3</sub>BTC-ZIF-8 and dyes. (The black dotted line shaded area for the overlap of the emission spectrum of H<sub>3</sub>BTC-ZIF-8 and the absorbance spectrum of C151, and the black solid line shaded area for the overlap of the emission spectrum of H<sub>3</sub>BTC-ZIF-8/

C151 and the absorbance spectrum of R6G.) (c) The emission spectrum of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film ( $\lambda_{\text{ex}}=365$  nm). I: CIE chromaticity coordinates (0.323, 0.347), II: photographs of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film upon the natural light and UV-365 nm, respectively. (d) A schematic diagram of R6G/C151/ZIF-8 composite film with white light emission prepared with H<sub>3</sub>BTC-modified ZIF-8.

To investigate the possible resonance energy transfer phenomenon among H<sub>3</sub>BTC-ZIF-8, C151 and R6G, their absorption and emission spectra were characterized as shown in Figure 5b. The shaded parts present in the figure between H<sub>3</sub>BTC-ZIF-8 and C151, and between H<sub>3</sub>BTC-ZIF-8/C151 and R6G may have satisfied the Förster resonance energy transfer (FRET)<sup>44,63</sup> from ZIF-8 to C151, and from ZIF-8/C151 to R6G, respectively, which was beneficial to improve the fluorescence performance of dyes. However, the overlap of the absorption and emission spectra of the individual material indicated the existence of self-absorption phenomena, which would cause an unavoidable weakening of the fluorescence performance.

Moreover, absolute photoluminescence quantum yields (PLQYs) of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 and R6G/C151/ZIF-8 composite films without H<sub>3</sub>BTC modified have been measured and results were shown in Figure S13. It can be seen that the PLQY of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film (64.0%) was higher than that of the R6G/C151/ZIF-8 composite film (48.0%) without H<sub>3</sub>BTC under the condition of the same concentrations of dyes. Also, a composite film consisting of crystalline ZIF-8 and the same dyes was prepared using the same method and characterized by XRD and PLQY, as shown in Figure S14, suggesting its low quantum yield (40.2%). These results demonstrated that the obtained R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film prepared by using H<sub>3</sub>BTC-modified ZIF-8 was more favourable to the interaction between ZIF-8 and dyes, which effectively reduced the ACQ effect between dyes molecules and improved the

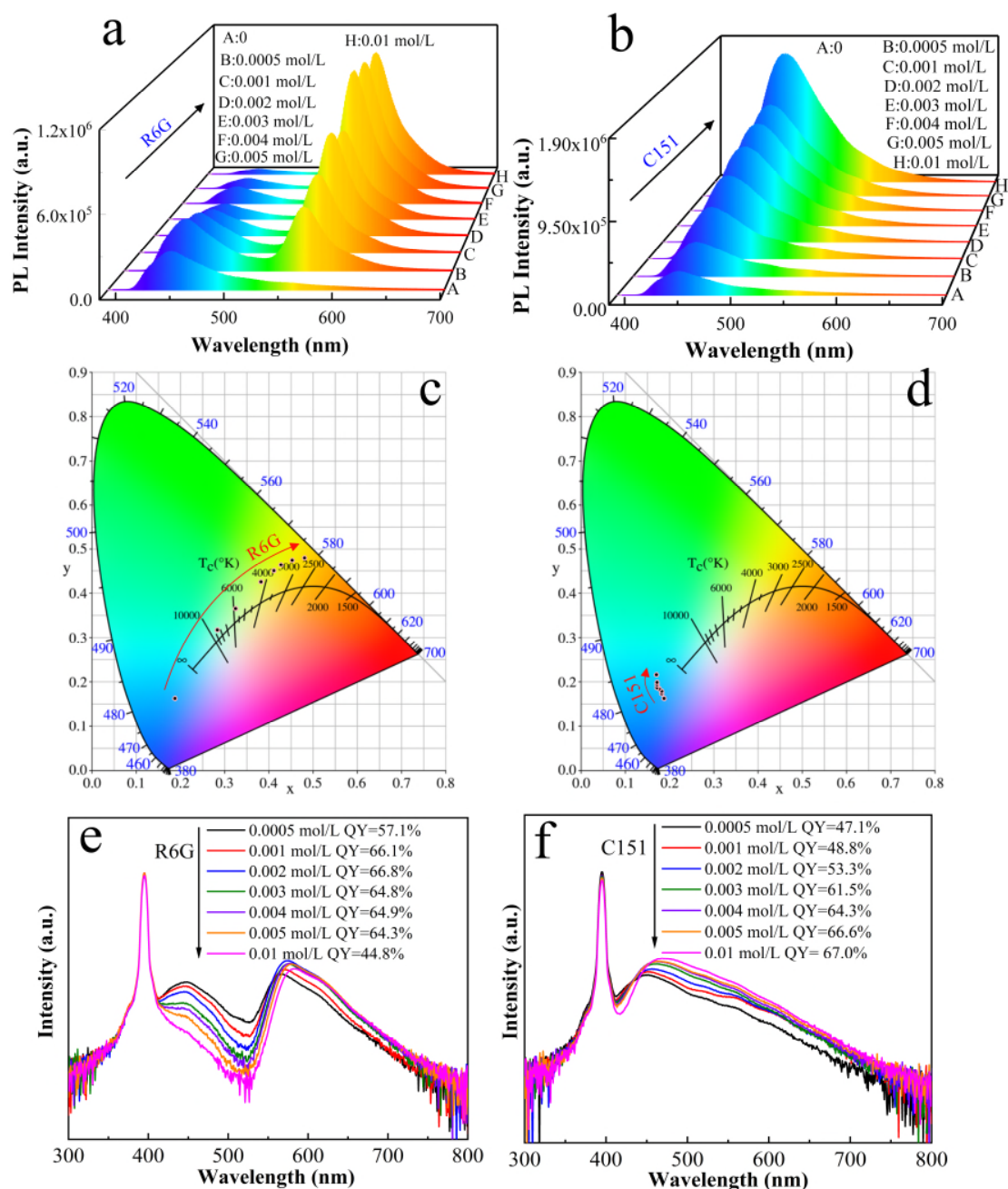
fluorescence performance of the composite film. And the composite film showed a cold white light emission with CIE coordinates of (0.323, 0.347) shown in Figure 5c, and its fluorescent lifetime (Figure S15) exhibited 5.41 and 2.46 ns for the peaks of 560 and 450 nm, respectively. Additionally, the correlated colour temperature (CCT) of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film was 5908 K according to the relevant calculation formula of CCT ( $CCT = 437 \times N \times N \times N + 3601 \times N \times N - 6861 \times N + 5514.31$ ,  $N = (x - 0.332) / (y - 0.1858)$ ,  $x$  and  $y$  are CIE coordinates.). Moreover, some previous works on fluorescence properties of composite materials prepared based on MOFs encapsulating fluorescent dyes, are summarized in Table 1. It could be seen that this approach of using functional groups modified amorphous MOF structures compounded with fluorescent dyes has an obvious advantage in improving fluorescence efficiency for the development of solid-state white light materials.

To evaluate the applicability of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film for practical application, its thermostability and photostability were investigated. The results of thermal stability indicated that the fluorescence relative intensities of the emission peaks at 562 and 445 nm) of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film can be maintained basically after heating at 100 °C for different times (Figure S16). However, the initial values of the spectrum differed significantly from other values due to insufficient preheating time of the light source. Meanwhile, the emission spectra of the R6G/C151/H<sub>3</sub>BTC-ZIF-8 composite film was recorded after different exposure times irradiated by 365 nm UV light (Figure S17), indicating good photostability.

In addition, the luminescent properties of R6G/H<sub>3</sub>BTC-ZIF-8 and C151/H<sub>3</sub>BTC-ZIF-8 composite films prepared by compounding H<sub>3</sub>BTC-ZIF-8 with different concentrations of single dye were investigated and results were shown in Figure 6. In Figure 6a, as the



concentration of R6G in the R6G/H<sub>3</sub>BTC-ZIF-8 composite film increased, the fluorescence intensity derived from R6G was stronger **excited** at 365 nm, corresponding to the CIE spectrum colour gradually shifting from blue to orange-yellow (**Figure 6c**). And fluorescence quantum yields (**Figure 6e**) of R6G/H<sub>3</sub>BTC-ZIF-8 composite films showed a trend of increasing (from **57.1%** to **68.8%**) and then decreasing (from **68.8%** to **44.8%**) with the increase of the R6G concentration, indicating that the ACQ effect of R6G molecules **became** obvious at higher concentrations. Meanwhile, a similar result was also observed in C151/H<sub>3</sub>BTC-ZIF-8 composite films. As the C151 content gradually increased, the blue-green emission intensity of C151 displayed an incremental enhancement (**Figure 6b**). **And** PLQYs of C151/H<sub>3</sub>BTC-ZIF-8 composite films were ranged from **47.1%** to **67.0%** (**Figure 6f**). In this sense, H<sub>3</sub>BTC-modified ZIF-8 compounding fluorescence dyes R6G and C151 can effectively improve the fluorescence performance of **the dyes**.



**Figure 6** The emission spectra ( $\lambda_{\text{ex}}=365$  nm) of R6G/H<sub>3</sub>BTC-ZIF-8 composite films with different concentrations of R6G (a) and C151/H<sub>3</sub>BTC-ZIF-8 composite films with different concentrations of C151 (b). CIE coordinates ( $\lambda_{\text{ex}}=365$  nm) of R6G/H<sub>3</sub>BTC-ZIF-8 composite films (c) and C151/H<sub>3</sub>BTC-ZIF-8 composite films (d) at ambient temperature. **PLQYs** of

R6G/H<sub>3</sub>BTC- ZIF-8 composite films (e) and C151/H<sub>3</sub>BTC-ZIF-8 composite films (f), ( $\lambda_{\text{ex}}$ =395 nm).

**Table 1** Summary of the fluorescence properties of various dyes@MOF composite materials.

materials	CIE (x, y)	emission color	PLQY	CCT (K)	references
C151@ZIF-8 <sup>2</sup> :F@ZIF-8 <sup>2</sup> :RB@ZIF-82 composite powders	0.32, 0.34	white	—	—	13
NKU-114@DSM/AF/9-AA	0.34, 0.32	white	42.07%	5101	41
HSB-W $\supset$ DCM/C6a/CBS-127	0.31, 0.32	white	26.0%	6638	11
ZIF-8 $\supset$ pm546/pm605/SRh101	0.465, 0.413	white	52.5%	2642	64
CD-MOF $\supset$ 7-HCm@FL@RhB	0.35, 0.32	white	—	—	36
ZJU-28 $\supset$ Cou-6/R6G/R101	0.36, 0.34	white	82.9%	4446	65
DSM@PCN-128W	0.34, 0.33	white	21.2%	5525	12
Rh110@bio-MOF-1	0.3111, 0.3382	yellow-green	79%	6981	1
RhB <sup>+</sup> @LIFM-WZ-6	0.33, 0.35	white	9.8%	4745	57
TPPA+ R6G@NKU-110	0.33, 0.34	white	49.1%	—	66
R6G/C151/H <sub>3</sub> BTC-ZIF-8 film	0.323, 0.347	white	64.0%	5908	this work

### 3.4. The possible luminescent performance improvement mechanism

To uncover the relationship between fluorescent dyes (R6G and C151) and H<sub>3</sub>BTC-ZIF-8, and the energy transfer among them, UPS characterizations of the dyes and ZIF-8 were performed, as shown in **Figure S18**. Additionally, the optical band gaps of dyes (R6G and C151) and ZIF-8 were calculated combining the corresponding absorption spectra. The specific schematic diagram was shown in **Figure 5d**. As can be seen in **Figure 5d**, the highest occupied molecular orbital (HOMO) levels located at -9.50, -8.92, -10.06, -8.25 eV and the lowest unoccupied molecular orbital (LUMO) levels located at -4.33, -4.68, -6.90, -5.98 eV for the ZIF-8, H<sub>3</sub>BTC-ZIF-8, C151, and R6G, respectively. According to the optical band gap of the ZIF-8 ( $E_g=5.17$  eV), 365 nm light can provide a photo energy of  $\sim 3.40$  eV, not enough to excite electrons in the ZIF-8 to transfer. In this sense, the blue emission produced by 365 nm excitation was derived from the defect luminescence (Zn defect) in the ZIF-8 structure<sup>44</sup>. When the ZIF-8 was modified by H<sub>3</sub>BTC with carboxyl functional groups, the HOMO-LUMO gap of the obtained H<sub>3</sub>BTC-ZIF-8 was reduced from 5.17 eV to 4.06 eV, and also, the HOMO level was increased by 0.58 eV, indicating that the electrons were more easily excited. After **forming** a composite structure of R6G/C151/H<sub>3</sub>BTC-ZIF-8 by combining the fluorescent dyes C151 and R6G with H<sub>3</sub>BTC-ZIF-8, the inherent HOMO-to-LUMO energy transfer of **H<sub>3</sub>BTC-ZIF-8**, C151, and R6G **shows** fluorescence emissions of 443, 470, 568 nm, respectively. Additionally, two types of possible FRET may exist in the composite structure, i.e. energy transfer from H<sub>3</sub>BTC-ZIF-8 to C151 and from H<sub>3</sub>BTC-ZIF-8 to R6G **according to positions of LUMOs**.<sup>67</sup> When the fluorescence produced by these energy transfers reaches the equilibrium emission of forming white light, white light would be obtained.

#### 4. Conclusions

In summary, the R6G/C151/H<sub>3</sub>BTC-ZIF-8 amorphous composite film with white light emission was successfully synthesized by using H<sub>3</sub>BTC-modified amorphous ZIF-8 in combination with fluorescence dyes R6G and C151, exhibiting a high PLQY of ~64%. It was higher than the PLQY of the composite films prepared by crystalline ZIF-8 (40.2%) or amorphous ZIF-8 (48.0%) compounded with the same concentrations of dyes. The possible reason for this phenomenon is that there are carboxyl functional groups existed in H<sub>3</sub>BTC-ZIF-8 structures increasing effectively active sites for interaction with dyes, which can form hydrogen bonds with C151 and weak D-A interactions with R6G molecules, thus enhancing the interactions between dyes and ZIF-8, and reducing the ACQ effect between dye molecules. Ultimately, the overall fluorescence performance of the composite film has been improved. This strategy by using functional groups modified amorphous MOF structures to synthesize composite materials has a significance for the development of luminescent materials in the application of solid white light.

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#### **Author Contributions**

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

#### **Conflicts of interest**

There are no conflicts to declare.

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