





Electrochemical processing in molten salts – a nuclear perspective

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Nuclear power provides a sustainable, long-term energy source that is an important technology in efforts to curb greenhouse gas emissions. However, issues related to construction, safety, risk of proliferation, poor public perception and the perpetual issue of end-of-life waste management remain of serious concern. A large body of work has employed molten salt-based electrochemical techniques as practical and viable methods to reprocess spent nuclear fuel and nuclear materials, in addition to producing refractory metals of broader technological relevance. This review summarises the essential aspects of the design of electrochemical reactors for electroreduction, -winning and -refining technologies. Specifically, a critique of materials choice in relation to temperature is presented alongside the effects of different metal oxide feed precursors and molten salt selection. A thorough discourse is provided on each of the actinides (Am, Cm, Np, Pu, Th and U) and lanthanides (Eu) with an evaluation of the performance of several precursor cell types and designs. The type of feed precursor, metal oxide-to-salt ratio and oxide ion concentration are factors that must be considered with regards to current efficiency and are critical to the scaling of current laboratory technologies to pilot-scale. However, a challenge remains in this transition due to potential irradiation hazards of materials under a neutron flux. There is still a need to understand the mechanism by which different radioisotopes are formed and their interaction with metallic compounds in molten salts, yet data in this area is scarce and further work is needed.

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Broader context

From broad global consensus on the negative impact of CO₂ emission, the world's industrial nations are moving to tackle climate change issues and transition to a low carbon energy era. As the global population continues to grow, so too does our need for more electricity and water, potentially leading to steep increments in anthropogenic greenhouse-gas emissions if alternative, low-carbon, sources of power are not adopted. Of all low carbon energy generation forms, nuclear power remains one of the few robust technologies that can be deployed on a large-scale to deliver ample 'green' electricity at the scale required. Arguably the greatest challenge for civilian nuclear power is the need to deal with spent fuel. This review examines the recycling strategies that have been employed from a civilian and defence perspective, with a focus on the use of molten salt technologies as a recycling process for the most significant lanthanides and actinides. Different models are explored to provide a comprehensive understanding of electrochemical reduction pathways and advanced experimental techniques to understand and optimise electrochemical processing in molten salts. The work shows that advances in molten salts research and development lie at the intersection between computation, electrochemistry, materials science and engineering and it is proposed that with a strong interdisciplinary approach we can continue to make headway in our commitment to achieving a net-zero economy.

1 Introduction

Ensuring the sustainable supply of future energy demands remains one of our greatest global challenges. With a growing

world population aspiring to higher living standards and the expectation to achieve increasingly ambitious emissions targets, this places immense pressure on the transport, industry and power sectors. Nuclear energy is a well-established and reliable low-carbon energy source. The current 440 power reactors in operation generate 10% of the world's electricity supply.¹ To increase this capacity, more must be done to maximise the operating lifetime and economics of nuclear power plants, as well as effectively and safely managing the spent fuel generated.² However, there continues to be controversy surrounding the use

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Table 1 Worldwide commercial spent nuclear fuel reprocessing capacity¹⁴

Type of reprocessing capacity	Location (design)	Reactor	Capacity (t year ⁻¹)
Light-water reactor (LWR) fuel	Russia, Ozersk (Mayak) ^a	Thermal, oxide	400
	France, La Hague	Thermal, oxide	1700
	Japan (Rokkasho)	Thermal, oxide	800 ^b
Other nuclear fuels	India (PHWR, 4 plants)	Thermal, metal	260
Total civilian capacity in operation			2360

^a N.B. The reprocessing facilities based in Mayak, Russia are not strictly civilian as it was and still is a storage site for weapons-grade plutonium and highly-enriched uranium.¹⁵ ^b Operation to start in 2024.

of nuclear energy, namely due to operational safety, production and disposal procedures for radioactive waste, and public perception and acceptance. To mitigate some of these factors, there has been considerable work undertaken to understand how spent fuel can be reused using a combination of both chemical and electrochemical processes to reduce and/or refine the actinide metal oxides. The ability to reprocess spent fuel may lead to a much more sustainable and publicly-acceptable energy technology offering. But this relies on the careful selection of reactor systems and complementary fuel cycles which together can advance future nuclear fuel cycle strategies.

During the Cold War era, the US was regarded as the leader in spent fuel reprocessing. This included the extraction of weapons-grade plutonium from spent nuclear fuel; and took place *via* one of two processes: either precipitation or solvent extraction. Previously, during World War II, the bismuth phosphate (BiPO₄) process was employed as part of the Manhattan Project as a method of coprecipitating pure plutonium with bismuth from irradiated nuclear fuel.³ Much of the large-scale work was carried out at the Hanford site near Richland, Washington.⁴ The principle behind this process lies in the dissimilar solubilities of uranium and plutonium. U⁶⁺ is far more soluble in sulfuric acid and will form sulfates, whereas plutonium (Pu³⁺ and Pu⁴⁺) is insoluble and co-precipitates with BiPO₄ to form a compact cake.⁵ This product is then dissolved in nitric acid and oxidised by the addition of sodium or potassium dichromate to separate Pu⁶⁺ from BiPO₄.⁶ Thereafter, several more purification steps are required to separate plutonium to a suitable degree of purity. Although a successful process it had the disadvantage of not allowing the recovery of uranium and generated a large amount of uranium containing radioactive waste.

In a later development, Herbert H. Anderson and Larned B. Asprey, who were participants of the Manhattan Project; an R&D programme comprising 150 000 workers to develop the first nuclear weapons during World War II,^{7,8} co-patented the plutonium–uranium reduction extraction (PUREX) process.⁹ The PUREX process is a mature liquid–liquid extraction technique that works by using an extractant, tri-*n*-butylphosphate (TBP) diluted in paraffin (kerosene) and a salting agent, nitric acid, from which separated uranium and plutonium can be recovered and purified.¹⁰ The desired actinide species are separated through a complexation process; in the aqueous phase (nitric acid) the preferential solubility of actinide complexes decreases with respect to the organic phase (TBP) as the molarity of the nitric acid increases.¹¹ The facile recovery

of nitric acid *via* distillation as well as the low volatility and high flash point of TBP offers two key advantages with respect to safety and waste requirements.¹⁰ The PUREX process has since been modified and adapted by the UK, France and Japan, who have implemented similar designs, resulting in the development of second-generation reprocessing plants.¹² Third-generation civil processing plants began in the 1980s when the PUREX process went through another transformation resulting in the reduction of radioactive effluent waste.¹¹ More recently both reprocessing facilities in Sellafield, UK, namely the Magnox and thermal oxide reprocessing plant (THORP) have now closed following the end of their design life.¹³ Table 1 shows the current worldwide commercial spent nuclear fuel reprocessing capacity.

2 Civilian nuclear power

2.1 Nuclear reactors and pyroprocessing

A nuclear reactor is a construction or device designed to sustain a long-term nuclear fission (chain reaction) process with a fissile material (²³⁵U and/or ²³⁹Pu) typically enriched to 3–5%. A set of conditions must be met for a required reaction to occur, these include: mass of fissile material (near to critical mass is optimal), geometry (reactor cores tend to approximate to spherical particles) and an appreciation of the neutron spectrum and the required kinetic energies for neutrons.¹⁶ Separate procedures are also in place to control the fission process according to the reactor design (for example: graphite moderators or chemical shim with boric acid) and the reactor is typically enclosed in a concrete edifice.^{17,18} The primary aim of a nuclear reactor is to therefore harness the heat energy generated from the fission process and convert it into electricity. The reactor types are designated according to the type of coolant, enrichment of the nuclear fuel, energy of the neutrons used in the fission reactions, and the purpose of the reactor. Three of the candidate reactors to be used in the next generation, Generation IV, of reactors use metal fuels, three use metal oxide fuels and one uses fluoride-based fuel. A thorough description of these designs is summarised in Table 2. The advantages of Generation IV reactors and the associated nuclear fuel reprocessing processes are that they are significantly more fuel-efficient compared with the previous generations and make the conversion of fissile material to weapons manufacture less attractive. This is because both uranium and plutonium are left mixed with several other elements rather than being purified as pure elements and while one might readily apply the PUREX



Table 2 A summary of generation IV reactor designs²⁰

Type of generation IV reactors	Neutron spectrum	Coolant	Fuel	Core outlet temperature (°C)	Size (MWe)
High-temperature gas	Thermal	Helium	UO ₂	900–1000	250–300
Sodium-cooled fast	Fast	Sodium	U-238	500–550	50–1500
Supercritical water-cooled	Thermal/fast	Water	UO ₂	510–625	300–1500
Gas-cooled fast	Fast	Helium	U-238 ^a	850	1200
Lead-cooled fast	Fast	Lead or Pb/Si	U-238 ^a	480–750	20–1200
Molten salt fast	Fast	Fluoride salts	UF ₄	700–800	1000
Molten salt – advanced high temperature	Thermal	Fluoride salts	UO ₂	750–1000	1000–1500

^a Fuel contains small quantities of ²³⁵U or ²³⁹Pu.

process; this leads to the need for long-term storage, offsite transportation and the undesired proclivity of weapons manufacture.¹⁹

Of all the Generation IV designs (Table 2), molten salt reactors are arguably the most promising, with the ability to operate at low pressures and high temperatures. The carrier molten salts are well known for their high specific heat capacity per unit volume and relative radiation-resistant reprocessing media. Molten salt reactors have been designed such that they avoid the build-up of neutron poisoning gases, xenon and krypton, in the molten salt by employing gas extraction techniques.²¹ What is more, it is predicted that the civilian nuclear power sector will experience renewed interest in the thorium fuel cycle, spurred on by R&D progress that includes spent fuel reprocessing (reduction of metal oxides), and the process of electrochemical refining.²²

With a wide range of reactor groups, it is essential that all, despite their technological advancements, size and capacities are developed to take into consideration the minimisation of radiotoxic elements produced as by-products of the nuclear fission process. After discharge from a reactor, the fuel rods contain a variety of natural and synthetic radiotoxic elements (Cm, Pu, U, Am, Np), which release between 50 (Magnarox) to 5000 (Fast reactor) kW of heat per tonne in the first six months and contain radioactivity that lasts over many years;²³ hence, a number of different strategies have been employed to ensure these elements are partitioned effectively both for storage and recycling through an appropriate reactor design. Overcoming these challenges remains difficult; and any of the aforementioned technologies which are implemented by different countries should aim to be environmentally friendly, resistant to nuclear weapons proliferation and economically viable.

Moreover, the spent metal fuels generated from nuclear reactors are incompatible with aqueous processing unless they are first oxidised followed by separation *via* the PUREX process, before ultimately being reduced. This, however, is needlessly circuitous and generates large volumes of waste. Pyroprocessing of spent nuclear fuel is the preferred option, incorporating a series of steps that utilises a high-temperature process and electrochemical reduction steps using molten salts, liquid media and/or gases to provide a safe and efficient route to metal production that is proliferation-resistant. It presents several advantages over conventional aqueous processes as it is able to handle a greater quantity of fissile materials. Unlike aqueous/organic reprocessing where the reagents can suffer from radiation damage there is no such problem with molten salts. The metal can be further purified by processes such as electrorefining.

Depending on the material to be treated, there are several different types of pyroprocessing techniques. Many of these processes try to avoid the troublesome use and handling of lithium metal, which is commonly found within metallothermic reduction (a displacement reaction of a reactive metal to form pure metals or alloys) using a molten lithium chloride pool.²⁴ Additionally, the application of chloride and fluoride melts is far more beneficial than aqueous systems as it avoids the possibility of radiolysis-induced degradation and reduces the overall cooling time required before high-burnup spent nuclear fuel is processed when using aqueous solvents.²⁵ Furthermore, the reduction process has drawbacks too; for example, the conventional 'bomb reduction' requires the use of hazardous fluorine, where there are high neutron doses associated with alpha-n reactions and inherently high temperatures and pressures.

Pyrochemical processing of spent nuclear fuel can be conducted such that pure plutonium metal or compounds are not isolated. The final steps in a typical pyro-process involve the recovery of uranium and/or plutonium, which are usually removed along with the other actinides (neptunium, americium and curium) as their reduction potentials lie close together. Plutonium, in this case, can form an intermetallic compound with its cathode crucible, *e.g.* when made from cadmium there is the formation of PuCd₆ and the presence of lanthanide fission products may lead to contamination. The formation of PuCd₆ occurred due to the reduction with lithium metal and an applied cathodic current density of 66 mA cm⁻², leading to its deposition in the solid phase at the bottom of the crucible. The reason for the solid phase deposition may arise from a solubility limit or because of a vertical temperature gradient due to slower cooling above the molten salt.²⁶ As a result, this intermetallic reaction process has two advantages; the first is that it prevents nuclear proliferation and the second is that the overall heat and radiation produced from the process is far less.²⁵

3 Molten salts and their properties

Molten salts are inorganic compounds that are solid at standard room temperature and when heated to elevated temperatures, dissociate into cations and anions with high ionic conductivities and wide electroactivity ranges. They have attracted interest for a broad range of applications, and continue to find diverse new uses.²⁷ For example, in the solar thermal industry, molten salts have been used as heat transfer media at high temperatures,^{28,29}



as electrolytes for fuel cells and in the extraction of metals from spent lithium-ion batteries.^{30–32} Molten salts are used in metallurgy to reduce refractory metal oxides to metals where other more conventional methods of reducing oxides are impractical and difficult to commercialise.^{33–35} They have also been applied to the production of alloys with advanced properties.^{36,37} In most of these applications, molten salts are used as binary or ternary mixtures that may form a eutectic mixture with a lower combined melting point than each independent component.

With regards to the treatment of spent nuclear fuels, the field of pyroprocessing has changed dramatically. Since the 1980s and 1990s, countries across the world have employed different pyroprocessing ('pyro' or 'high temperature') approaches, with broad agreement that molten salt electrorefining is a promising approach for the processing and recovery of spent nuclear fuel and radioactive metals.³⁸ In general, molten salts possess several key advantages in nuclear applications. As non-aqueous media, they are less likely to pose criticality hazards which are often found in aqueous solutions. Molten salts are able to operate at high temperatures without a direct fire risk. Furthermore, they have useful properties in processing high burn-up nuclear fuels with fewer risks post-irradiation, there is also reduced susceptibility to radiolysis compared to aqueous methods and they have the capability of dissolving large quantities of metal oxides compared with aqueous extraction methods.³⁹ Despite these promising advantages, their inherently high operating temperature and corrosive nature (particularly fluorides) have acted as an impediment to technological progress and the subsequent advancement of industrial systems.

In Table 3, a variety of molten salt systems that have been used in the nuclear industry are considered. The reasons for the selection of one particular molten salt over another is often governed by temperature requirements, the solubility of oxide ions, issues of criticality limits and stability.³⁸ For example, LiCl-KCl, NaCl-KCl and CaCl₂ have vastly different working temperatures of 450 °C, 750 °C and 900 °C, respectively. LiCl has been used as a medium to extract metallic nuclear fuels, whereas NaCl-KCl (given its higher operating temperature) as well as LiCl-KCl eutectic, have been preferred for the electro-winning of spent oxide fuels. The method by which each of these molten salts has been used to reprocess the respective metal oxides/chlorides is described in this review. Moreover, molten salts are excellent heat transfer fluids and due to their advantageous characteristics, their application to industrial use, namely as a coolant and/or fuel in molten salt reactors

and in pyroprocessing, appears promising. Given the wide array of molten salts on offer, research efforts must continue to explore their benefits and progress from the lab bench to industry.

4 Process chemistry

4.1 Electrochemical reduction reaction mechanisms

The FFC-Cambridge Process has demonstrated the process by which the direct electrochemical reduction of metal oxide to metal in a molten chloride system occurs.³⁴ This process has been useful in enabling further understanding of the electrochemical reduction of spent oxide nuclear fuel to metal *via* an oxygen ionisation mechanism which is now one of the theories used to describe pyro-electrochemical processes. In this electro-deoxidation process, and upon the application of a suitable potential, electrochemical reduction of the metal oxide occurs at the cathode and the oxide ions generated migrate through

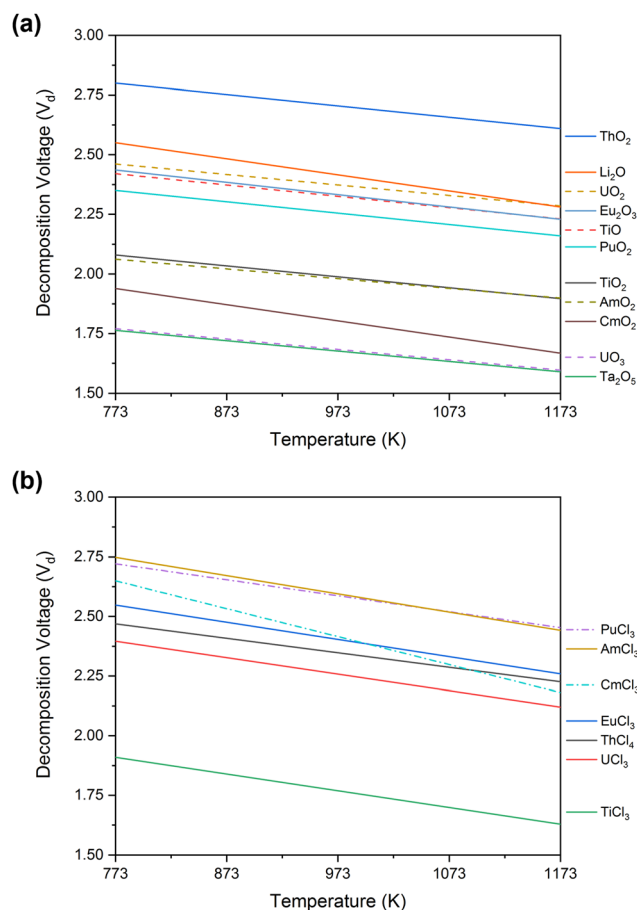


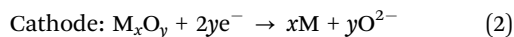
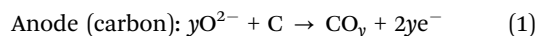
Fig. 1 Calculated values of decomposition voltage defined as the minimum voltage (V) vs. standard chlorine electrode needed for the reduction of the metal chloride or oxide systems stated above as determined from the Gibbs Energy of Formation derived between 773 and 1173 K for (a) metal-oxide and (b) metal-chloride systems; this can then be compared to the expected decomposition of inorganic salt systems to identify the feasibility of an electrochemical process. All data, relating to the Gibbs free energies, has been acquired from HSC Chemistry 6.12.

Table 3 Molten salts and their properties^{40–42}

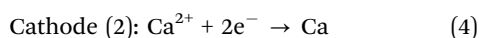
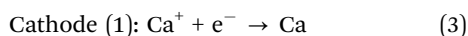
Properties	LiCl-KCl	NaCl-KCl	CaCl ₂	LiF-CaF ₂
Composition (A-B, mol% A)	59	50	100	23
Melting temperature (°C)	352	658	772	769
Molar mass (g mol ⁻¹)	55.3	66.5	111	37.9
Density at melting point (g cm ⁻³)	1.64	1.60	2.15	2.07
Viscosity at melting point (g cm ⁻¹ s ⁻¹)	2.80	2.13		
Electrical conductance (Ω ⁻¹ cm ⁻¹)	0.637	2.20	2.48	6.25



the molten salt system to the anode. Here, the oxide ions react with (typically) a carbon anode to form CO or CO₂ (eqn (1)). If a suitable overpotential is applied to the metal oxide at the cathode for a sufficient period, this leads to the metallisation of the surface of the cathode (eqn (2)).⁴³ The exact process that occurs depends on the metal oxide, type of precursor, choice of molten salt and the activity of O²⁻ and Cl⁻ in the melt. Fig. 1 shows the decomposition voltages of M–O and M–Cl systems between a temperature range of 773–1173 K. The decomposition voltages of these systems must remain below that of molten salt systems to ensure a successful electrochemical reduction.



Shortly after the invention of the FFC-Cambridge Process, the Ono-Suzuki process was presented.⁴⁴ This study was originally demonstrated for TiO₂ and can be applied to the study of nuclear materials reprocessing especially in terms of understanding the electrochemical reduction of UO₂. Ti metal is recovered by a process called ‘calciothermic’ reduction, this is where, calcium metal is produced *in situ* from the reduction byproduct CaO. CaO is soluble in CaCl₂ molten salt and leads to the formation of Ca⁺, liquid calcium (eqn (3) and (4)) and carbon dioxide (identical to the reaction described eqn (1) on the application of a suitable overpotential. Ti metal is formed by the thermochemical reaction between TiO₂ and liquid Ca as shown in eqn (5). Further description of other electrolytic processes are described in Table 4 although they lie beyond the scope of this review article.



4.2 Electrochemical reduction and refining of nuclear materials relevant to the pyrochemical process

The electrochemical recovery processes mentioned before (*e.g.* FFC-Cambridge, Ono-Suzuki) are important processes that may be useful in closing the back end of the fuel cycle. Typically, this is where some or all of the spent nuclear fuel is reprocessed. It comprises the following steps: mining, milling, chemical conversion of uranium to an enriched version suitable for fuel rod fabrication and the subsequent disposal post-irradiation of the used fuel, including plutonium and fission products.¹⁶ There are several stages in the nuclear fuel cycle where electrochemical processes can feature (Fig. 2). Prior to electrochemical processing, the as-received fuel rods must be converted into a suitable form appropriate for use as a feed material which involves the disassembly and decladding of spent fuel assemblies into smaller segments and the subsequent voloxidation step where a high-temperature treatment (typically greater than 1000 °C) is applied to convert UO₂ to U₃O₈. This latter process is imperative in improving the current efficiency for electrochemical reduction of the uranium and plutonium oxides generated and is an important fabrication step in controlling the stoichiometry and atomic arrangement of particles for subsequent steps. Heating the used fuel to high temperatures prior to addition to the molten salt removes the salt-soluble fission products such as caesium, iodine, rubidium and technetium, therefore preventing them from accumulating within the reduction process, although it does generate a separate waste stream containing fission products such as ¹³⁷Cs and ⁹⁹Tc. The electroreduction procedure involves the dissolution of the uranium and plutonium oxide granules or powder typically into a molten chloride system such as LiCl–Li₂O at 650 °C (Fig. 3); the careful control of salt-soluble fission products is needed to prevent the slowing of the reduction process.^{46,47} It has been demonstrated that this initial step in the electrochemical reduction reduces the heat load (which is also described as the specific decay heat of fission products) and leads to a volumetric reduction by conversion of metal oxide to metal. For optimal results, this process

Table 4 A summary of the various electrochemical reaction schemes of molten salt processes. table reproduced from Xiao and Wang⁴⁵ with permission from the Royal Society of Chemistry

Reaction mechanism	Cathodic products
Direct electro-deoxidation of solid oxides	
One-step electro-deoxidation	Metals
Continuous or stepwise electro-deoxidation	Oxides or metals
Simultaneous electro-deoxidation and de-calcification	Metals
Immobilisation of active liquid metal with formation of alloys	Alloys
Cathodic passivation	Oxides and metals
Direct electro-desulfidation of solid sulfides	
	Metals or alloys
Deposition of active metal together with electro-reduction of solid oxides	
Deposition of active metal followed by metallothermic reduction	Metals
Electro-deoxidation under active metal deposition	Metals
Under-potential deposition of active metals	Alloys
Electro-inclusion of cations from molten salts into oxides	
	Oxide composite
<i>In situ</i> dissolution–electrodeposition process	
	Nanostructured semiconductors



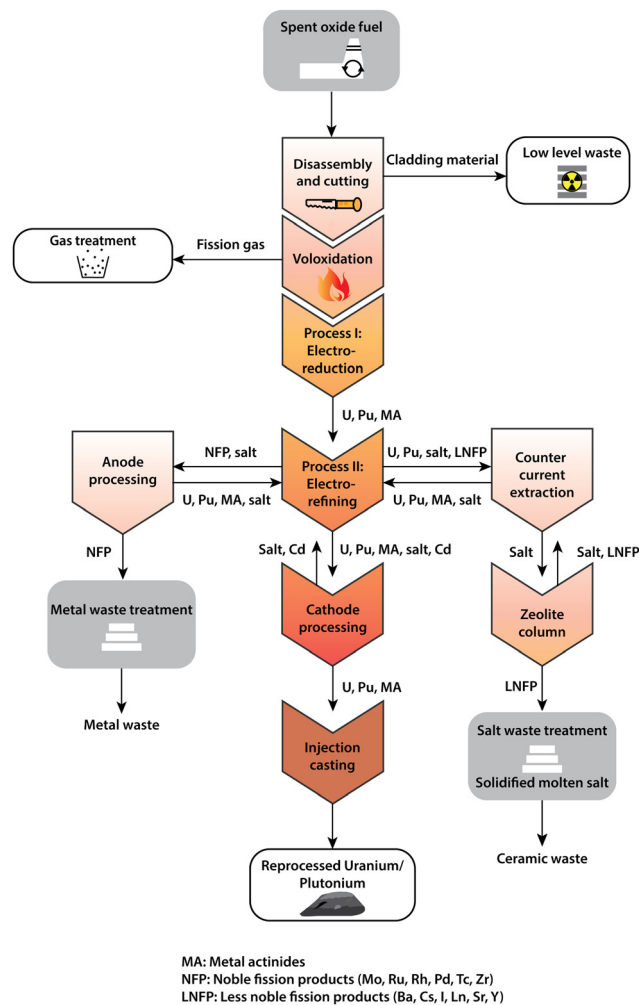


Fig. 2 Schematic recycle flow process showing the various steps involved in the reprocessing of spent nuclear (oxide) fuel. There are several different steps involved in the electrochemical processing of nuclear materials but the two most critical in the recovery of high purity uranium and plutonium are electrorefining and electroreduction. This schematic diagram has been reprinted from Koyama, with permission from Elsevier.⁴⁷

must separate all fission products such as barium, caesium, and strontium from the fuel. The overall cathodic and anodic reduction is presented for the oxide fuel where M represents the actinide as described in eqn (1) and (2).

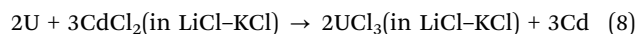
In general, the cathodic form of the process has been found to vary greatly and various studies have considered the performance when using rod-cut, powder, pellet form or crushed particles of actinide oxides which are loaded into a cathode basket.^{48–51} To improve the rate of reduction, an appropriate quantity of lithium oxide is added to prevent the anodic dissolution of the inert platinum anode enshrouded in magnesium oxide – a concentration which lies between 1–3 wt% is typically used to ensure there is enough present to accelerate the reduction without leading to any corrosion resistance or impediment to the extent of reduction.⁴⁷ Uranium remains the most widely studied actinide in this field and comparisons have been made using differing oxidation states of feed materials

(UO_2 and U_3O_8); these studies have demonstrated that in an LiCl-KCl system the likelihood of other intermediates such as LiUO_3 , U_4O_9 and UO_2 (if U_3O_8 is used as the feed material) forming in the initial stages is highly probable.^{52,53} The reduction of UO_2 is therefore likely to proceed through two different routes, one of which involves the formation of a lithium-intercalated compounds and the other through direct cathodic reduction.



Electrorefining is the second process with potentially two feed streams. One of these streams comprises a spent metal feed and the other is a metal stream from the electrochemical reduction of the spent oxide fuel. For the former scenario, impure uranium metal is selectively separated from transuranium (TRU) and lanthanide elements at a solid cathode in various molten salt systems from the electrolytic reduction process. It is important to note that the steps described forthwith are in reference to the actinide separation process only and do not account for the initial decladding and final waste treatment processes which follow.

Electrorefining technology, since the early 1960s, has gone through several developmental phases and whilst most of the studies on UCl_3 utilised $\text{BaCl}_2\text{-KCl}$ as the most appropriate molten salt, the realisation that LiCl-KCl offered a more practical and advantageous route, with a considerably lower working temperature, without compromising product or recovery, was found to be more suitable.⁵⁴ There have been several reports on the electrorefining process for the LiCl-KCl-UCl_3 eutectic solution. The uranium rod from the electroreduction step is subsequently reacted with cadmium chloride (in LiCl-KCl) to produce UCl_3 .⁵⁵ The formation of UCl_3 is advantageous because it stabilises the initial cell voltage according to eqn (8).⁵⁶



The electrorefining cell typically comprises a stainless-steel anode basket that contains the final reduction products, a solid iron cathode and a Ag/AgCl reference electrode. This setup is immersed in the aforementioned LiCl-KCl-UCl_3 system and since cadmium has a greater density this metal sinks to the bottom and captures any residual uranium deposit falling from the surface of the cathode. For the duration of the process, a current is supplied to avoid anodic dissolution of the stainless-steel, whilst both the cathode and anode assembly are simultaneously agitated between 10–20 rpm to enhance symmetrical uranium deposit growth at the cathode as well as improve the rate of anodic dissolution. The completion of the process occurs once the applied current decreases. Afterwards, the uranium metal in the anode basket is raised from the molten salt and left to cool. Then it is rinsed with distilled water to remove any adhering lithium chloride salt. This process, as explained in detail in Fig. 2 and 3 (Mark-IV electrorefiner), results in the separation of pure uranium directly from the nuclear fuel onto the surface of the cathode and the



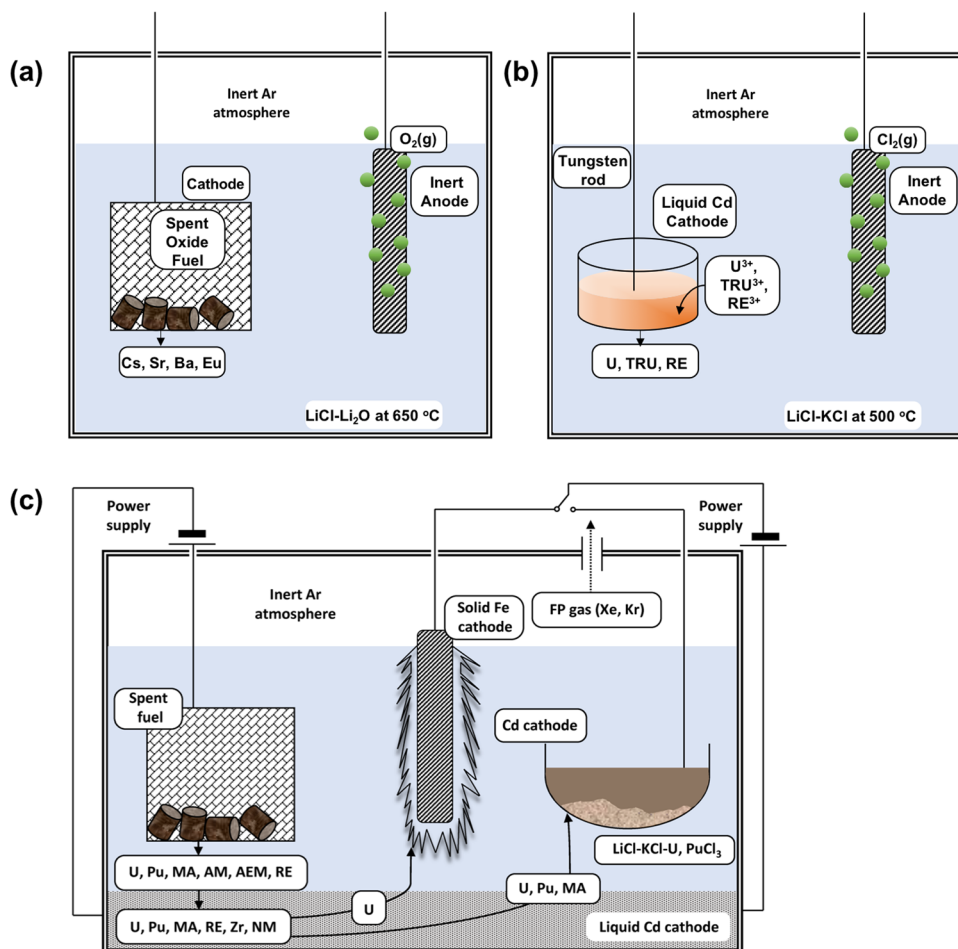
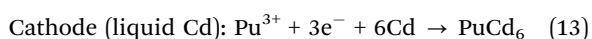
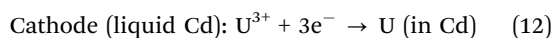
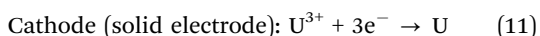
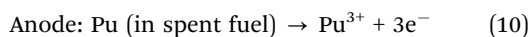
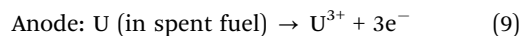


Fig. 3 Schematic of (a) electroreduction, (b) electrowinning and (c) electrorefining process. (a) Represents a typical schematic diagram for the electroreduction as a first step to reducing the feed material reprinted from Koyama with permission from Elsevier.⁴⁷ (b) Represents a simplified drawing of the liquid cadmium cathode (LCC) electrowinning process. (c) Represents the Mark-IV electrorefiner developed by the Argonne National Laboratory operating at 500 °C, where AM is alkali metal, AEM is alkaline earth metal, MA is Np, Am and Cm, NM are the noble metals, RE are the rare earth elements and FP refers to the fission products. Reprinted by permission of the publisher, Taylor and Francis, on behalf of Atomic Energy Society of Japan from Koyama *et al.*⁵⁹

resultant deposition of residual uranium, transuranic elements and lanthanides. The Mark-IV electrorefiner consists of four rectangular anode baskets and a stainless-steel cathodic rod. The reaction schemes for the electrorefining process are described as:



A large body of research on spent nuclear materials has been performed at the Argonne National Laboratory (ANL).^{54,57,58} Their molten salt electrorefiner, illustrated in Fig. 3, represents a striking innovation. Both cathodes are made from stainless-steel/iron and liquid cadmium. Initially, spent nuclear fuel is

cut into metal ingots following a conversion and placed in a basket that forms the anode and undergoes electrochemical oxidation. The cadmium pool found at the bottom is where the reduction of actinides occurs. The uranium metal is transported to the stainless-steel cathode from the anode. All actinides migrate towards the cadmium electrode, where they are oxidised. The deposit of uranium in particular takes the form of a dendrite on the stainless-steel cathode which facilitates retrieval at the end of the process; however, when there is a mixture of fission products, especially with plutonium, the deposit does not adhere to the cathode surface and consequently use of a liquid cadmium cathode becomes far more fruitful than in uranium electrorefining cells.

Koyama *et al.* extensively studied the application of an electrorefiner and his work led to the replacement of stainless-steel electrodes with an iron cathode (Fig. 3);^{59–61} Sakamura *et al.* and Roy *et al.* were able to determine the redox potentials which led to the reduction of actinides and cladding materials, such as



Table 5 Redox potentials of elements in LiCl–KCl eutectic salt (V vs. Ag/AgCl)^{59,62,69,70}

Elements	Working temperature (°C)		
	400	450	500
Pu(III)/Pu(0)	–1.591	–1.543	–1.497
Np(III)/Np(0)	–1.472	–1.434	–1.390
Am(III)/Am(0)		–1.592	
Zr(II)/Zr(0)		–0.693	
U(III)/U(0)	–1.274	–1.233	–1.190

those made from zirconium, from their chlorides after undergoing oxidation at the anode. Table 5 summarises these reported potentials.^{62,63} There is a discrepancy associated with each of the observed potentials (e.g. for the U³⁺/U⁰ couple, from Inman *et al.*,⁶⁴ Kuznetsov *et al.*⁶⁵ and Koyama *et al.*⁵⁹). This occurs because of the dependence of a molten salt system on the concentration of O^{2–} ion activities in molten salt systems.^{66–68}

After electrorefining, the obtained deposits can go through a subsequent purification step, cathode processing, where the deposit is heated to separate the actinide metal from the adhering cadmium and salt. As the vapour pressures between cadmium and the actinides differ quite significantly the separation between these species can proceed with ease. The final step is referred to as ‘injection casting,’ and is a method where the composition of the actinide metals is adjusted by melting U–Zr alloys at 1560 °C and Pu–U–MA–Zr alloy at 1480 °C under a vacuum.⁴⁷ This is then injection-cast into a mould cavity which cools and hardens according to the shape of the mould (rod-shape). These rod-shaped moulds are then fabricated inside a steel cladding for use in a nuclear reactor.

Electrowinning is a separate procedure for the recovery of uranium and plutonium metal from TRU using a liquid cadmium cathode.⁷¹ This technology has gained traction through the Russian Institute of Atomic Reactors (RIAR) and the successful recovery of 7 tonnes of unirradiated fuel has been demonstrated.⁷² This system works by the electro-co-deposition of PuO₂ and UO₂ under the influence of a chlorine–oxygen sparge. In oxidising conditions, the conversion of PuO₂ to PuO₂²⁺ becomes possible and by changing the redox potential of the system, all oxidation states of plutonium can form PuO₂. Under the same oxidising conditions, uranium remains in the melt and plutonium is precipitated out. As the reduction of PuO₂²⁺ and UO₂²⁺ takes place at more positive potentials, purification takes place during electro-deposition. In other systems, the success of this technology lies in the distinct difference in reduction potential of the uranium and TRU, a diagram for which is shown in Fig. 3(c); it has been shown in studies that uranium metal has a preferentially close redox potential to solid working electrodes, some of which include tungsten and graphite, whereas the TRU have very close redox potentials to liquid metals such as cadmium and bismuth.^{59,73,74}

5 The lanthanide and actinide materials

The exploratory work concerning the reduction of nuclear materials in molten salts began in the early 1960s. At this time nuclear materials had been used primarily for weapons production

purposes and with this came a build-up of associated nuclear waste. For most of the elements detailed, consideration is given to the metal chloride systems which comprises a large part of the electrorefining process. A review of these elements is necessary because of the interesting outcomes observed in alkali halide eutectics and because they are commonly considered as part of the burn-up credit analysis conducted by the Nuclear Energy Agency with the exception of thorium metal.⁷⁵ Additionally, progress is being made in the processing of metal oxide systems, although such systems are far less studied. Not all reduction processes described beforehand are mentioned in the following sub-sections. A general description of each chemical element, including its occurrence and earlier uses, is provided to help orientate readers.

5.1 Uranium

Uranium, atomic number 92, is one of the few naturally occurring actinides found within the earth’s crust (<3 ppm) and is a constituent within many minerals such as: autunite, carnotite, monazite, samarskite and uraninite. Several extraction processes have been developed to industrially recover uranium from its ores. Uranium displays fluorescent properties, and was first used as a yellow-green colourant in glass from as early as the Roman period.⁷⁶ This use remained unchanged for the next two millennia. In 1896, Henri Becquerel was the first to realise that uranium emitted penetrating rays;⁷⁷ the important and ground-breaking discovery of nuclear fission was made in 1939 by Hahn and Strassman.^{78,79}

Uranium dioxide, UO₂, was first reported erroneously as an element in literature but this finding was corrected by Pélégot who realised that it was in fact a pure oxide.⁸⁰ For molten salt processes, UO₂ is produced *via* the oxidation reaction of the higher oxides, U₃O₈, under the flow of hydrogen gas. This is commonly the feed material along with UCl₃. To understand the electrorefining pathway, Inman *et al.* were the first to study this particular uranium species, UCl₃, in LiCl–KCl eutectic.⁸¹ This study focused on uranium metal deposition on a uranium working electrode in accordance with the redox process described in eqn (14). The process was found to have a high faradaic efficiency (when current densities <100 mA cm^{–2}) on a single electrode reaction.



In a separate experiment, the reduction potential was measured to be –1.398 V (vs. Ag/AgCl). It was understood that uranium metal was formed by a chemical reduction with lithium present in the salt. Because of this, some of the final product appeared powdery whilst other product material formed at the working electrode (at lower working potentials) lead to dendritic growth. In comparison, Kuznetsov *et al.* conducted their own study and determined that the reduction potential in a LiCl–KCl system was ~–1.5 V vs. Ag/AgCl.⁸² Further details of redox potentials are described in Table 6.

There has also been a multitude of work conducted on spent fuel oxides in molten salts. Herrmann *et al.* placed crushed spent nuclear fuel into a LiCl-1 wt% Li₂O at 650 °C.⁸⁷ In this

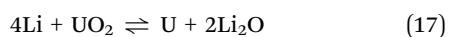
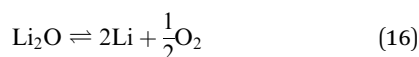


Table 6 Summary of reduction potentials of U^{3+}/U^0 in LiCl–KCl molten salt eutectic at various temperatures

<i>T</i> (K)	Apparent experimental potential (V) of U^{3+}/U^0							
	Martinot and Caligara ⁸³	Roy <i>et al.</i> ^{69 a}	Kuznetsov <i>et al.</i> ^{84 b}	Shirai <i>et al.</i> ^{85 b}	Masset <i>et al.</i> ^{86 b}			
673	−2.522	−2.528			−2.5741			
	−2.506							
	−2.598							
694	−2.527	−2.509						
	723					−2.498	−2.541	−2.4841
						−2.476		
726	−2.522							
	748					−2.557		
						−2.548		
748	−2.483		−2.514	−2.4689	−2.5088			
	773			−2.453		−2.4533		
				−2.491				
823	−2.527		−2.487	−2.4224	−2.4627			
				−2.460		−2.4224		

^a Reference electrode: Ag/AgCl. ^b Reference electrode: $Cl_2/2Cl^-$.

study, a platinum anode and a Ni/NiO reference electrode were used. A potential of -2.40 V *vs.* Ag/AgCl was calculated from the electrochemical reduction of uranium dioxide (eqn (15)) and the lithium deposition potential was -2.47 V *vs.* Ag/AgCl (eqn (16)). Given that the cathodic reduction potential of uranium dioxide and the decomposition of the salt lie very close together, this resulted in the *in situ* formation of lithium metal and presented an advantage in enabling the chemical reduction of uranium dioxide *via* eqn (17). However, a disadvantage of this is that the formation of lithium resulted in the attack of the platinum anode and its dissolution. Similarly, Choi *et al.* studied uranium dioxide on a much larger scale reducing 17 kg of UO_2 in LiCl– Li_2O .⁸⁸ Through their study they realised that to achieve higher current efficiencies a small pellet precursor should be suitably complemented with a larger anode surface area.



There has also been work accomplished on the higher oxides of uranium such as U_3O_8 . Seo *et al.* studied the reduction mechanism of U_3O_8 within a porous magnesium oxide membrane and found the potential to be -2.27 V *vs.* Pt.⁸⁹ Jeong *et al.* studied this process on a larger scale reducing 20 kg of U_3O_8 and achieving a 99% conversion.⁵² The reduction potentials of this process were found to vary from -2.47 to -3.46 V due to differences in O^{2-} ion concentration. These results were in agreement with Choi *et al.* where they found the size of the pellet to be of hindrance. The larger the size of the pellet the more likely it was to inhibit the diffusion of the electrolyte leaving the pellet partially reduced.⁸⁸ The three-phase interline (3PI) theory illustrated in Fig. 4 can be used as an explanation of this occurrence. This model was initially conceived by Chen *et al.* who were the first to propose a model for the 3PI before it

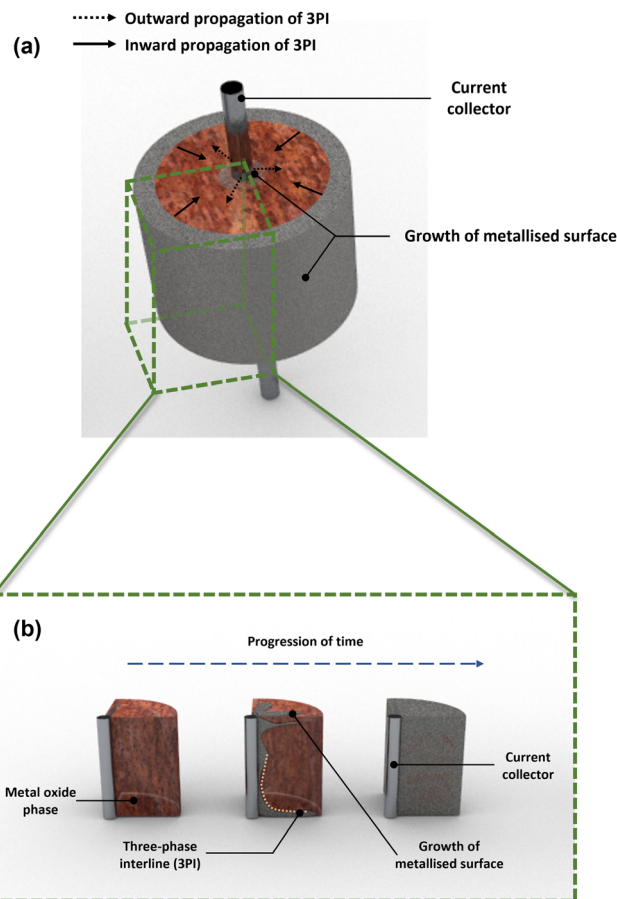


Fig. 4 3D rendering of the three-phase interline (3PI) or three-phase boundary (TPB) which highlights the metallization of the pellet surface showing inward and outward propagation of TPB in (a) and (b) where a cross-sectional area of the metal oxide phase changes with respect to time (blue dashed line). The 3PI (indicated by the orange dotted line) occurs between the current collector (central metal rod), metal oxide (red hue around the surface of a metal rod) and molten salt (surrounding region).

being further developed by Deng *et al.* to consider both static and dynamic 3PIs.^{90,91} Kar and Evans then developed their own mathematical model accounting for metal oxide porosity, pellet radius and particle size distribution.⁹² A major assumption made within this model is that microstructural changes during reduction are negligible. Furthermore, if we assume that there are two phases within a molten salt system then they can only be connected by a two-dimensional plane, however the majority of processes encompass three distinct phases where the point of intersection occurs at a single point (in one dimension), otherwise known as the 3PI. If we then take the example of a typical electrochemical system where we are reducing a metal oxide to a metal, the phases are as follows: metal phase (current passage path), metal oxide phase (reactant) and the molten salt phase (electrolyte). For pellet-based studies, the progress of forming a metal surface occurs inwardly from the 'outer' metal oxide surface and outwardly from the current collector. This relies on there being sufficient contact between the conductor and insulator for the transfer of electrons; transfer between oxide ions and the surrounding electrolyte.



Sakamura *et al.* have also studied UO_2 in both CaCl_2 at 800 °C and LiCl at 650 °C. In the former system, the reduction occurred at <0.6 V vs. Ca/Ca^{2+} .⁹³ The reduction process however led to the formation of calcium metal on the metal oxide surface and prevented the formation of the TPB on the inside of the UO_2 disk sample and impeding the reduction progress. In the latter system, the reduction potential was found to be <0.15 V vs. Li/Li^+ . This process yielded a much greater current efficiency as well as improved electrolyte diffusion in the precursor. Furthermore, Hur *et al.* then selected a system where a lower working temperature was needed as realised in $\text{LiCl-KCl-Li}_2\text{O}$ at 520 °C.⁵¹ The reduction potential of UO_2 was determined to be -1.27 V vs. Li-Pd . The conversion of uranium dioxide to uranium metal was attributed directly to the chemical reduction facilitated by lithium as previously mentioned. The fluctuating activity of O^{2-} ions in the molten salt is regarded as the reason behind the differences reported in reduction potentials and the same is found to be true here. The activity of O^{2-} ions also has far-reaching effects when the reduction and decomposition potential are found to be in close proximity as this affects the kinetic pathways and rate-determining steps of a reaction. If the local O^{2-} ion concentration increases, this impedes the withdrawal of the O^{2-} ion from the electrolyte and thus leads to a lower reduction potential. However, the fluidised process provides a constant agitation of argon gas as well as sufficient electrical contact between the metal and metal oxide; it further enhances the mixing between these two phases and enables the process to take place at a more positive potential. Further discussion of the different electrochemical reactor types and the importance of the three-phase interline is presented in Section 7.⁴⁵

Much of the headway made in improving the electrochemical reduction process of uranium chlorides is because the reaction proceeds far more easily at higher O^{2-} ion concentrations. It is worth noting that uranium has three thermodynamically stable oxidation states, namely U^{3+} , U^{4+} and U^{6+} , and more often than not work on uranium oxides has attempted to determine the oxidation states of U^{4+} (UO_2) and U^{6+} (UO_3). The reduction potentials for the uranium chlorides has been summarised in Table 6 with small differences which have likely arisen due to differences in reference electrodes used and in uranium concentrations.

Progress on the electrochemical reduction of uranium has involved a range of molten chloride and fluoride systems. Nourry *et al.* considered the electrochemical behaviour of U^{3+} and U^{4+} in LiF-CaF_2 to demonstrate a viable reduction pathway and confirmed that U^{3+} ions were reduced to uranium in a single step.⁹⁴ Previous work has applied the same reduction to different molten salt systems such as LiF-NaF and $\text{LiF-BeF}_2\text{-ZrF}_4$.^{95,96} Using a nickel plate cathode, their group was able to separate U^{3+} from Gd^{3+} successfully, on the notion that the potential was carefully controlled between -0.16 and -0.38 V vs. Pt.

5.2 Plutonium

Plutonium-239, atomic number 94, is a synthetic isotope produced in nuclear reactors by neutron capture of ^{238}U to ^{239}U . This step is then followed by two successive beta emissions as a result of the natural instability of the radioisotopes. ^{239}U has a

half-life of approximately 20 minutes, decaying to ^{239}Np with a half-life of approximately three days before decay to ^{239}Pu . ^{239}Pu has a relatively long half-life, 2.4×10^4 years and undergoes alpha decay to ^{235}U . It exhibits unique chemical properties that have sparked interest particularly in light of pyroprocessing applications. Several studies have explored different methods of reduction that plutonium and its respective halides would follow given the rising concerns around plutonium metal production and its associated nuclear weapons proliferation risk.⁵⁴ In a similar fashion to uranium studies, much of the earlier work was concerned with reducing PuCl_3 as part of the Manhattan Project in the 1940s; this was done to re-use the large quantities of waste that had been generated from the production of the atomic bomb.

The correct determination of the melting point of plutonium metal was reported by comparison of different morphologies (droplets at higher temperature and powdery deposits at lower temperatures) before chemical determination at Los Alamos National Laboratory. Kolodney *et al.* were the first to successfully obtain plutonium metal from a melt containing PuCl_3 .⁹⁷ A working temperature of 660 °C was applied to a Pyrex cell using a tantalum cathode and graphite anode. Once the process had been completed the final product was shown to contain minute quantities of various contaminants, including: beryllium, lithium, sodium and silicon, as a result of the interaction between the plutonium metal and Pyrex cell. Silicon has been shown to stabilise the lower density delta phase of plutonium. But no further tests were done to establish whether the addition of silicon led to an increase in plutonium stability. The Manhattan Project developed a process to extract plutonium from PuF_4 using calcium metal, leaving CaF_2 as a by-product, this technique being more commonly known as the 'bomb reduction' process.⁹⁸

Subsequent work focused on obtaining higher purity depositions of plutonium without leading to the contamination that can occur from either an impure feed material or corrosion due to the electrochemical reactor design. In 1960, Blumenthal and Brodsky produced data suggesting that surface contamination was likely to occur with the molten plutonium and lead to hydrogen, carbon and uranium impurities.⁹⁹ Their work focused on depositing plutonium as a solid. As a useful comparison, work on the reduction of U_3O_8 has found that the choice of electrochemical reactor material is important in avoiding corrosion which likely occurs in the cathode basket. In this study, an FeU_6 peak was reported in an XRD pattern due to the interaction between iron present in the steel cathode basket and uranium.¹⁰⁰ Therefore, the use of a Vycor cell (high-temperature silica resistant design) compared with Hastelloy (a superalloy of nickel, molybdenum and chromium) would prove to be far more appropriate in the design.⁵⁴

Similarly, it was found that using tungsten rather than a tantalum cathode, leads to much less contamination in the final product despite earlier work indicating that a tungsten electrode was a far better choice.¹⁰¹ Work accomplished by Leary and Muslin found carburised tantalum cathodes to be far more promising because their heats of formation, 161 kJ mol^{-1} ,



are much greater than using plutonium carbide (PuC), 7 kJ mol^{-1} . This is indicative of stability as a cathode.¹⁰² These findings were directly useful to the Los Alamos molten plutonium electrorefining process (LAMEX) project which began in the late 1950s. Their preliminary work was channelled towards electrorefining a Pu–Fe alloy, otherwise known as ‘fissium’,¹⁰³ to recover a high purity plutonium metal. In these experiments, impure plutonium was selected as the anode material and the high-purity metal formed was deposited at the cathode. Mullins, Leary and Bjorklund scaled up their process to operate with 100–300 g samples of Pu and Pu–Fe alloy before further development to a 3.5 kg scale.¹⁰¹ These experiments demonstrated a current efficiency of $>70\%$ and definitive reduction of impurities compared with previous reports. The final Fe concentration was found to be 0.02% compared with 2.54%. Their subsequent work involved selecting different crucible compositions such as alumina and magnesia (magnesium oxide) which were both able to tolerate high temperatures and were corrosion-resistant.

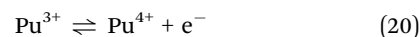
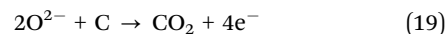
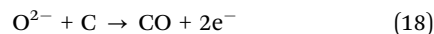
The choice of a suitable crucible material is vital to successfully processing and recovering plutonium metal. Most of these designs utilise alumina and magnesia due to their strong chemical and thermal resistance.¹⁰⁴ Despite these two important qualities, they are only suitable for single use because their mechanical strength is far too low. Plutonium metal adherence to the surface of a ceramic is another significant issue; this may occur in the presence of other metals which changes the chemical characteristics of plutonium metal. Paget *et al.* reported this phenomenon in the presence of 2000 to 5000 ppm of titanium.¹⁰⁵ It was later found that this could be overcome, on a smaller-scale, with the addition of CaCl_2 .¹⁰⁶ A study has also been conducted on Si_3N_4 with results suggesting that this material has far superior corrosion and thermal resistance characteristics compared with magnesia, tantalum, tungsten, and other alloys.¹⁰⁷

Considering the design of crucibles more closely, it was found that use of magnesia alone was far more stable and reduced hydrogen contamination. Magnesia is much more chemically inert when stabilised with yttria (Y_2O_3) and the same was also found to be true for fluoride molten salt systems. Yttria-stabilised magnesia was later found to have both mechanical and thermal shock-resistant properties which made it more suitable in electrolytic cell applications. The product was collected as an annular ring at the bottom of the crucible. Minor modifications to the crucible in the form of a magnesium titanate addition would lead to the formation of a flat disk of plutonium rather than a ring structure.¹⁰⁸ Work conducted by Curtis considered different molten salt systems, where a mix of 30 wt% PuCl_3 , 28 wt% BaCl_2 and 42 wt% KCl with the addition of PuCl_3 as feed material.¹⁰⁹ Their findings revealed several desirable properties pertaining to BaCl_2 ; first, that it separated the oxides from the cathode ensuring the final product remained coalesced and second that it caused the oxides and less dense impurities to have a greater buoyancy induced by BaCl_2 .

Much of the later work focused on the electrochemical reduction of plutonium oxide with an overall aim of determining the feasibility of the process. In these experiments, the feed material, PuO_2 , was directly reduced in a molten fluoride salt

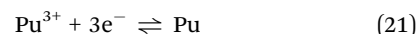
system which was comprised of 50 wt% LiF , 20 wt% BaF_2 and 30 wt% PuF_3 . The low current efficiency of 10–20% reported was most likely due to unwanted parasitic reactions taking place at the surface of the anode structure. These parasitic reactions could either have been the oxidation reaction of carbon with oxide anions or the reduction of Pu^{3+} and Pu^{4+} as described in eqn (18)–(20).

Anode reactions



It was found that a slow and constant addition of PuO_2 avoided the accumulation of the feed material at the bottom of the crucible. Addition of PuO_2 in bulk proved detrimental because it inhibited the coalescence of the final product and a substantial proportion of the oxide was not electrolysed. Brodsky and Carleson further investigated the potentially deleterious effects caused by a molten fluoride system.¹¹⁰ Their work led to the understanding that the proximity of fluoride to plutonium would lead to a neutron flux in a NaCl-KCl system due to the $^{19}\text{F}(\alpha, n)^{22}\text{Na}$ reaction. In a LiCl-KCl system, such (α, n) reactions are less common. Alternatively, the use of PuCl_3 would also prove detrimental, given its highly hygroscopic nature resulting in the absorption of any moisture in the molten salt system. To avoid this, their molten salt system was chosen to be a dichloride Cs_2PuCl_6 salt which is non-hygroscopic, prevents any occurrence of neutron flux in the molten salt system and is readily produced.¹⁰⁴ They found that electrodeposition took place at the tantalum cathode at a temperature of 700°C , albeit at a lower yield, but there was substantially less contamination by calcium, molybdenum and nickel in the final product. In the following years, there was much greater throughput of electrorefined plutonium alloy; overall, 653 batches, with the anode metal feed each consisting of approximately 3 kg of plutonium alloy, led to 1568 kg of plutonium metal recovery.¹¹¹ These batch weights were increased to 4 kg to increase product throughput and then finally to 6 kg after further anode feed optimisation in 1982.¹¹²

Shortly after, Poa *et al.* conducted electrochemical studies into the deposition of UCl_3 and PuCl_3 within a melt consisting of $\text{BaCl}_2\text{-CaCl}_2\text{-LiCl-NaCl}$ at 480°C .¹¹³ This study used a three-electrode system where the working potential was compared to Ag/AgCl . Their work shed light on the effects of changing temperature and Pu^{3+} ion concentration on the reaction kinetics. It was reported that the reduction of Pu^{3+} took place in a single step, as indicated in eqn (21).



A similar study was conducted focusing on the reduction pathway adopted by Shirai *et al.* using PuCl_3 in a LiCl-KCl eutectic using 0.544 wt% PuCl_3 at 450°C .¹¹⁴ A study where the temperature was varied revealed that there was a change in the magnitude of the cathodic current depending on the working electrode material and this was complemented by an increase in



Table 7 Summary of reduction potentials of $\text{Pu}^{3+}/\text{Pu}^0$ at various temperatures in molten LiCl–KCl

T (K)	Apparent experimental potential (V) of $\text{Pu}^{3+}/\text{Pu}^0$					
	Campbell and Leary ^{118 a}	Martinot ^{119 a}	Masset ^{120 b}	Roy <i>et al.</i> ^{63 a}	Sakamura <i>et al.</i> ^{121 a}	Shirai <i>et al.</i> ^{122 b}
673	−2.1224	−2.0539		−2.845	−2.8227	
698	−2.1199	−2.0273		−2.827	−2.8061	
723	−2.1174	−2.0007		−2.808	−2.7895	−2.748
733	−2.1164	−1.9901	−2.7962		−2.7828	
773	−2.1124	−1.9475	−2.7699	−2.775	−2.7562	−2.711
823	−2.1073	−1.8943	−2.7361		−2.7230	−2.673

^a Reference electrode: Ag/AgCl. ^b Reference electrode: $\text{Cl}_2/2\text{Cl}^-$.

the diffusion coefficient with increasing temperature. Initially, Nissen reported a diffusion coefficient of $0.51 \times 10^{-5} \text{ cm}^2 \text{ s}^{-1}$ at 400 °C, $1.11 \times 10^{-5} \text{ cm}^2 \text{ s}^{-1}$ at 500 °C and $1.62 \times 10^{-5} \text{ cm}^2 \text{ s}^{-1}$ at 600 °C.¹¹⁵ Serp *et al.* then determined that the diffusion coefficient of PuCl_3 at 460 °C fell between 1.3×10^{-5} and $1.85 \times 10^{-5} \text{ cm}^2 \text{ s}^{-1}$ agreeing with previous findings.¹¹⁶ Due to difference in the working electrode surface area, diffusion coefficients are likely to vary. Although no observable difference was found by changing materials, the temperature had a direct effect on the potential of the cathode. The data indicated that the process occurred *via* a reversible diffusion-controlled process; however, a

chronopotentiometry study performed by Martinot and Duyckaerts indicated a retardation due to the formation of a coordination compound formed between Pu^{3+} and the Cl^- ions in the electrolyte.¹¹⁷ Table 7 summarises the apparent standard potentials for the most stable oxidation state, Pu^{3+} , as reported in early studies.^{63,118} The results are generally consistent, with the exception of the data obtained by Campbell and Leary and Martinot.^{118,119}

Fig. 5(a) and (b) are cyclic voltammograms of PuCl_3 in equimolar NaCl–KCl and CaCl_2 systems. In Fig. 5(a) there are two cathodic peaks, namely Ic and IIc. The first has been attributed to the direct reduction of Pu^{3+} ions to Pu and the

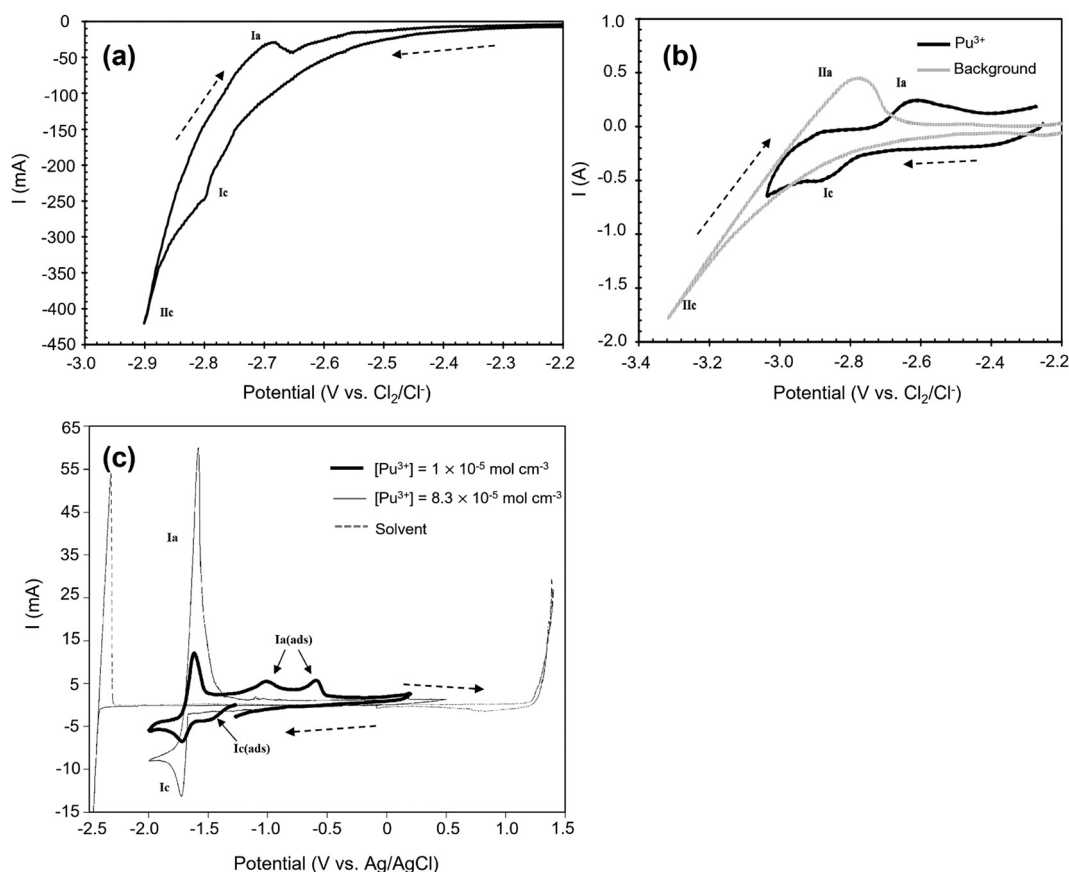


Fig. 5 Cyclic voltammetry of Pu^{3+} ion species in three different molten salt systems conducted on the tungsten working electrode. (a) Equimolar NaCl–KCl at 1073 K, scan rate 0.1 V s^{-1} , surface area (SA) = 0.31 cm^2 . $[\text{PuCl}_3] = 1.1 \times 10^{-3} \text{ mol}$. (b) CaCl_2 at 1073 K, scan rate: 0.1 V s^{-1} , SA = 0.31 cm^2 . $[\text{PuCl}_3] = 1.1 \times 10^{-3} \text{ mol}$. (c) LiCl–KCl eutectic at 733 K, SA = 0.2 cm^2 . (a and b) Have been reprinted from Lambertin *et al.* © The Electrochemical Society. Reproduced by permission of IOP Publishing Ltd.¹²⁵ (c) has been reprinted from Serp *et al.* with permission from Elsevier.¹¹⁶



second is ascribed to the electroplating of Na^+ ions. However, the voltage separation between both processes is very close which makes them difficult to distinguish; whereas in the CaCl_2 , a difference of approximately 0.4 V makes the two processes easier to differentiate. McCurry *et al.* report that the addition of sodium chloride to a potassium chloride melt reduces the voltage separation difference making it difficult to separate the two processes without the possibility of electroplating sodium on the surface of the cathode.¹²³ Fig. 5(c) may be compared to the predominance diagram illustrated in Fig. 7(b) where the reduction potential is found to agree with theoretical prediction and occurs at -1.76 V. Predominance diagrams, as illustrated in Fig. 7, shed light on the electrochemical processes for various metal oxide, chloride and metal ion species (highlighting regions of ion stability) and are acknowledged as a preliminary step to understanding their reduction pathways in ideal conditions. Work by Brown *et al.* and Abdulaziz *et al.* has been undertaken on developing a comprehensive understanding of both uranium and plutonium oxide reduction steps and are described in Fig. 7(a) and (b).^{68,124} As before, this study confirms the findings in Fig. 5(a) and (b) where there is only one reduction step; it also proves that this process is feasible at a lower reduction potential which incurs a reduced energy penalty. However, at lower concentrations, there is an additional cathodic peak found at ~ -1.5 V which has been ascribed to the underpotential associated with the interaction of plutonium with the tungsten rod.¹¹⁶

A small-scale study examined the electrochemical reduction of PuO_2 in $\text{CaCl}_2\text{-CaO}$ by Jones *et al.* at 900 °C.¹²⁶ Whereas previous studies had demonstrated that the reduction of MOX fuels (where a mixture of plutonium and uranium oxides are pressed into a nuclear fuel pellet) was viable, none had solely considered the use of PuO_2 as a feed material. One of the reasons for not considering pure PuO_2 for this was that the processing of feed material into nuclear fuel pellets could lead to increased radiation doses for operators as a result of ingrowth of the gamma emitting daughter radionuclide ^{241}Am . These studies had considered different ratios of uranium and plutonium; Iizuka *et al.* showed that using $\text{LiCl-0.5 wt\% Li}_2\text{O}$ ensured that the materials (47 wt% U, 35 wt% Pu, 4 wt% Np) O_2 would eventually form an alloy.¹²⁷ Kurata *et al.* demonstrated the reduction of (90.55 wt% U and 9.45 wt% Pu) O_2 in $\text{LiCl-1.0 wt\% Li}_2\text{O}$.¹²⁸ Both molten salts had a slightly lower working temperature of 650 °C. The findings revealed that the current efficiency of the process was poor, yielding 12.9% owing to charge losses in the reduction of calcium oxide and furthermore, the concentration of plutonium was found to be 3–5 wt%. Additionally, there were significant amounts of contamination, likely caused by the choice of basket material. The lack of cyclic voltammetry (CV) data also made it difficult to optimise the potential and overall charge that needed to be applied. From this, it was clear that there were no intermediate species and the formation of plutonium occurred in a one-step deoxidation from PuO_2 with four electrons exchanged.

Recently, Rappleye *et al.* attempted the plutonium electrorefining process with a focus on reducing sub-kilogram

quantities in CaCl_2 .¹²⁹ Using NaCl-KCl eutectic can lead to pyrophoric sodium and potassium due to the co-reduction of these solvent ions, whereas the use of CaCl_2 eliminates this. The separation potentials between Ca^{2+} and Pu^{3+} are greater than Na^+/K^+ and Pu^{3+} which avoids co-deposition with Pu^{3+} . The reported efficiency and yield of this process was 71.4% and 78.2%, respectively, which is in line with previous studies by Mullins.¹¹² Differences in results may arise because previous researchers used a lower concentration of MgCl_2 which serves as an extractant (1.9 wt% compared with 3 wt%). Additionally, the use of a new tungsten cathode annulus led to a high concentration of tungsten in the final product (1200 ppm); this may have been caused by loosely adhering tungsten particles coalescing with the plutonium deposits or by the very nature of a small-scale process where there is a greater surface area of contact with the deposited plutonium.

5.3 Neptunium

Neptunium, atomic number 93, is a synthetic element that has been produced in significant quantities by the nuclear industry and may occur as a constituent in MOX fuel. It is commonly found as a product within radioactive waste streams and so understanding its interaction with other metallic species is critical. The development of compact and economic reprocessing plants is especially important and relies on an understanding of the actinide chemistry concerning neptunium species. The main production route for ^{237}Np in a nuclear power plant is the capture of a neutron by ^{235}U to form ^{236}U . ^{236}U subsequently captures a neutron to form ^{237}U , which subsequently decays to ^{237}Np with an approximately 7 day half-life. There is a minor production route *via* the n, 2n reaction with ^{238}U to ^{237}U , but this is more significant in a fast reactor than a thermal reactor. An n, 2n reaction is where 2 neutrons are emitted after the capture of one incoming neutron. The subsequent beta-decay leads to the formation of ^{237}Np .¹³⁰ Neptunium was the first of the actinide elements to be synthesised using a cyclotron in 1940 through the simple process of neutron bombardment of UO_3 . McMillan and Abelson,¹³¹ later questioned whether a decay product had formed from their new find and a year later Seaborg *et al.* published their work on the discovery of element 94 or plutonium.¹³² There are 22 known isotopes of neptunium that have been characterised; however, most radioanalytical chemists will be concerned with the three whose half-lives make them obtainable: ^{237}Np (2.14 million years), ^{238}Np (2.12 days) and ^{239}Np (2.35 days).¹³³ Tens of kilograms of neptunium, alongside associated waste, have been produced worldwide. Since the early 1980s, many of the electrochemical studies, in particular, voltammetric (where a potential is applied and a current is measured) and potentiometric (where a constant current is applied and the potential is measured) have been used to elucidate the governing kinetics and mass transfer processes and also to determine the formal working potential against a standard reference electrode. To complement this work, there has been a surge in effort towards reprocessing nuclear fuels, some of which have been reported in Table 8 and described in Fig. 7(c).



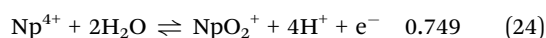
Table 8 Summary of reduction potentials of $\text{Np}^{3+}/\text{Np}^0$ at various temperatures

T (K)	Apparent experimental potential (V) of $\text{Np}^{3+}/\text{Np}^0$					
	Masset <i>et al.</i> ^{134 b}	Shirai <i>et al.</i> ^{135 a}	Martinot ^{136 a}	Roy <i>et al.</i> ^{63 a}	Krueger <i>et al.</i> ^{137 b}	Fusselman <i>et al.</i> ^{138 b}
673	-2.767		-2.39	-2.725	-2.725	-2.725
698	-2.765			-2.714	-2.715	-2.714
723	-2.733	-2.716	-2.35	-2.698	-2.697	-2.698
748	-2.700	-2.703				
773	-2.682	-2.690	-2.31	-2.668	-2.660	-2.668
798		-2.677				
823	-2.568	-2.662	-2.27			

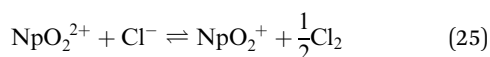
^a Reference electrode: Ag/AgCl. ^b Reference electrode: $\text{Cl}_2/2\text{Cl}^-$.

Neptunium has well-reported valence states in aqueous solution, which include: $\text{Np}(\text{H}_3\text{O})_6^{3+}$ (III), $\text{Np}(\text{H}_2\text{O})_8^{4+}$ (IV), $\text{NpO}_2(\text{H}_2\text{O})_6^+$ (V), $\text{NpO}_2(\text{H}_2\text{O})_6^{2+}$ (VI) and NpO_5^{3-} (VII). The first four oxidation states (III to VI inclusive) are stable in water and form hydrated ions. However, VII is only stable in dilute alkaline solution. From an electrochemical perspective, Np ions possess interesting redox properties where transfer of a single electron results in a fast equilibrium (eqn (22) and (23)) but a process where a Np–O bond is formed (eqn (24)) is much slower.

Standard potential (V)



Cassol *et al.* were the first to study the reaction process in the presence of NaClO_4 at 298 K by potentiometric titrations.¹³⁹ In accordance with other literature, they found that the greater level of acidity was in the following order: $\text{PuO}_2^{2+} > \text{NpO}_2^{2+} > \text{UO}_2^{2+}$, *i.e.* in reverse order of atomic number.^{140,141} The unstable nature of NpO_2^{2+} was confirmed by Martinot *et al.* who reported a similar lack of stability in the presence of chloride ions in a LiCl–KCl eutectic at 450 °C.¹³⁶

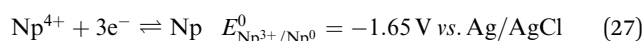
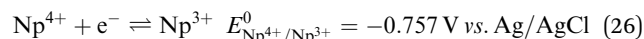


The first two reduction schemes from hexavalent NpO_2^{2+} to pentavalent NpO_2^+ in eqn (25) and subsequently to tetravalent Np^{4+} have been determined from *in situ* absorption electron spectroscopy.^{142,143} The reduction of Np^{4+} to Np^{3+} was reported by both CV and differential pulse polarography. It was found that Np^{3+} and U^{3+} share similar properties to Np^{4+} and U^{4+} in a LiCl–KCl molten salt system.

One of the first studies on neptunium, conducted in 1969, focused on the diffusion coefficients for the higher valence states.¹³⁶ The starting material was made by the slow oxidation of Cs_2NpCl_6 to NpO_2 (5+) before undergoing chronopotentiometry. The work revealed that the higher valent NpO_2 (4+) had a lower diffusion coefficient of $2.0 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$ compared with $4.8 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$ for NpO_2 (5+). This was because the ion in solution is more complexed when it comes into contact with chloride ions and thus it hinders the movement of the then

formed coordination complex.¹³⁶ Further work by Martinot revealed that the single-step reduction from NpCl_3 and NpO_2 was equally possible and highly reversible however the reduction of NpCl_4 did not occur directly requiring reduction first to NpCl_3 .¹⁴⁴ Use of higher valence states of neptunium was avoided because in the presence of chloride ions it would be reduced. The results proved positive with a powdery deposit of neptunium adhering to the surface of the tungsten cathode with some degree of impurity. These impurities mainly arose from the feed material itself with additional elements being drawn from the eutectic and silica crucible similar to the plutonium work.

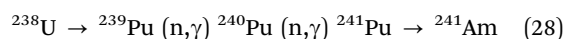
Later, a more comprehensive study was conducted on the reduction of NpCl_4 to Np in LiCl–KCl eutectic at 475 °C.¹³⁴ It was concluded that the reduction took place in two separate steps with the single electron transfer of Np^{4+} to Np^{3+} taking place at a positive potential (eqn (22)). This experimental data is in good agreement with the predominance diagram for Np species in Fig. 7(c) where the electrochemical reduction of Np^{4+} to Np proceeds in two steps; the first is where one electron is exchanged and the second is where three electrons are required at a more negative cathodic potential (eqn (26) and (27)).



Square-wave voltammetry (SWV) was also used to determine the electron transfer and to compare the reduction potentials measured previous to the work undertaken by Masset.¹³⁴ The first peak indicated a single-electron transfer process at $-0.757 \text{ V vs. Ag/AgCl}$ (surface area (SA) = 0.24 cm^2). However, the second cathodic peak was significantly sharper with a potential of $-1.65 \text{ V vs. Ag/AgCl}$ associated with a three-electron transfer from Np^{3+} to Np^0 . The CV revealed that, at incremental scan rates, there was a noticeable negative shift in the cathodic peak (Np^{3+} to Np^0). This finding is evidence that at higher scan rates the process is not reversible; a plot of current (*I*) against scan rate potential ($\nu^{1/2}$) further reveals that this is a diffusion-controlled process. No morphological or materials characterisation studies were performed in this paper. Other studies on NpCl_3 have considered its electrochemistry at the molybdenum and liquid metal electrodes with more positive potentials reported for the liquid metal electrodes.⁷³

5.4 Americium

Americium, atomic number 95, is a synthetic element, with two commonly occurring isotopes ^{241}Am and ^{243}Am . Applications of this radioisotope vary from those found in the nuclear industry to domestic applications *e.g.* smoke detectors. Americium was first discovered in 1944 by Seaborg *et al.* who realised that through the controlled bombardment of neutrons with ^{238}U and ^{239}Pu atoms, it was possible to produce other heavy elements.¹⁴⁵ Americium was generated as a result of the irradiation of ^{238}U with neutrons to form ^{241}Pu . This subsequently led to the observation that ^{241}Pu undergoes beta-decay according to the following process.



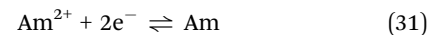
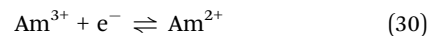
For domestic applications, americium, ^{241}Am , is used in sub microgram quantities (approx. $0.3\ \mu\text{g}$) in domestic smoke alarm detectors. The ^{241}Am radioisotope occurs as an oxide which is housed within a ventilated metal box and emits a small but constant stream of alpha particles which ionises the surrounding air molecules. A small charge is produced and is detected between two plates at different voltages in the metal box. In the presence of smoke, the ionised molecules will absorb the smoke particulates, impeding current flow and setting off the alarm.¹⁴⁶ For industry applications, ^{241}Am is mixed with beryllium and pressed into a pellet to provide a practical neutron source for purposes such as measuring the thickness of glass as it exits the production line. The neutrons are formed by (α, n) reactions with the beryllium. ^{241}Am is also used in medical applications, such as radiography, and as a source for X-ray fluorescence spectroscopy.

Relatively little is known about the reduction pathway of AmCl_x (where $x = 3$ or 4) and published work on AmO_2 is even more scarce. Fig. 6(d) describes the apparent standard potentials of Am^{2+} in LiCl-KCl eutectic, which is known to be stable along with Am^{3+} . However, there is very little data published on the standard potential of Am^{3+} , which undergoes a two-step reduction to Am^0 .¹²⁰ Fig. 6(d) indicates that there is good agreement in work conducted after the year 2000 and that the potential lies between -2.92 and -2.87 V in the range 723 – 823 K. There has been some headway made by Serp *et al.* who found, using CV, that the reduction pathway undergoes two separate steps.¹⁴⁷ Fig. 6(d) indicates that the reduction of Am^{3+} first involves a single electron exchange and the second involving the exchange of two electrons. Each of the two cathodic peaks were associated with two corresponding anodic peaks, the first cathodic peak being much flatter with a peak current of -6 mA as evidenced in the SWV. The reported peak current is associated with the exchange of a single electron during the reduction of Am^{3+} to Am^{2+} at the tungsten working electrode ($\text{SA} = 0.2\ \text{cm}^2$). The second cathodic peak with a current of -16 mA indicated a two-step electron transfer. The relationship between the number of exchanged electrons and the half-wave (Gaussian) peak can be determined by eqn (29).¹⁴⁷

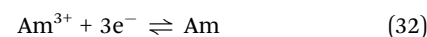
$$W_{1/2} = 3.52 \frac{RT}{nF} \quad (29)$$

The transitions, which take place between the respective oxidation states of Am^{3+} and Am^{2+} , have been confirmed in previous work.¹⁴⁸ Both of these species are the most stable in LiCl-KCl eutectic with the latter, Am^{2+} , being predominant. It must also be made clear, that the separation of americium from the lanthanides is the most difficult to achieve potentiometrically.¹³⁸ Additionally, using chronopotentiometry, the diffusion coefficients of species have been reported at two different temperatures (723 K and 743 K). Both studies report that $\text{Am}^{3+}/\text{Am}^{2+}$ remains stable at temperatures between 733 – 823 K in LiCl-KCl eutectic. With regard to understanding the reduction process of americium in LiCl-KCl eutectic the reduction proceeds in two separate steps on a tungsten working electrode. The first reduction of Am^{3+} to Am^{2+} occurs at -2.74 V vs. Cl_2/Cl^- shortly followed by the

reduction Am^{2+} to Am at -2.82 V vs. Cl_2/Cl^- at $504\ ^\circ\text{C}$.¹⁴⁹ Further details of the reduction potentials of Am^{2+} to Am^0 may be found in Table 9.



In addition to studies on an inert working electrode, there has been a focus on using liquid cadmium and results suggest that the reaction occurs in a single-step with the reduction of Am^{3+} to Am ; this occurred with an apparent potential of -2.56 V vs. Cl_2/Cl^- at $504\ ^\circ\text{C}$ (eqn (32)).¹⁴⁹ Because of the small potential range (<100 mV), the complete separation of americium from lanthanum in this study would not yield sufficient recovery of americium. In general, greater current densities can be induced by installing a blade stirrer (impeller) to improve the mass transfer between the molten salt and metal chloride and ensuring that the evolution of chlorine gas does not contaminate the reduced product are steps that can be taken to increase the quantitative recovery of americium. Americium dispersion can be reduced by appropriate selection of cathode material.



The discoveries of the higher radiotoxic actinides (neptunium, plutonium and americium) have fuelled considerable work into partitioning and transmutation (P&T) studies, although the work has mostly focused on plutonium and uranium. As outlined in previous sections, the focus has been on the availability of thermochemical and electrochemical data in literature. Up till now, the rate of development and understanding of actinides such as americium and curium (as will be described in the following section) has been slow.

5.5 Curium

Curium, atomic number 96, was first discovered in 1944 by Seaborg *et al.* who applied a similar method to that used for the discovery of americium and plutonium.¹⁵¹ In this case, rather than bombarding plutonium, ^{239}Pu , with neutrons, alpha particles (helium cations) were used and the characterisation of curium was confirmed *via* the energy of the alpha particle emissions (eqn (33)). To achieve this, a 60-inch cyclotron at Berkeley was deployed to provide a controlled environment for such high-energy bombardment. Due to its size a much larger target was bombarded and a greater quantity of products formed. These products were then isolated and investigated to determine both chemical and physical properties. The isotopes of curium occupy a wide range in mass which extends from ^{238}Cm to ^{251}Cm . In general, the curium radioisotopes are difficult to work with, but two of its isotopes, ^{242}Cm and ^{244}Cm , have been used in power sources with amenable results.¹⁵² Moreover, ^{244}Cm demonstrates remarkable incandescence, yielding $122\ \text{W g}^{-1}$, on the other hand, ^{238}Pu , which is more readily available, has been preferred in applications requiring this attribute.¹⁵³ The third most commonly studied isotope, ^{248}Cm , has also been used as an enriched target for the



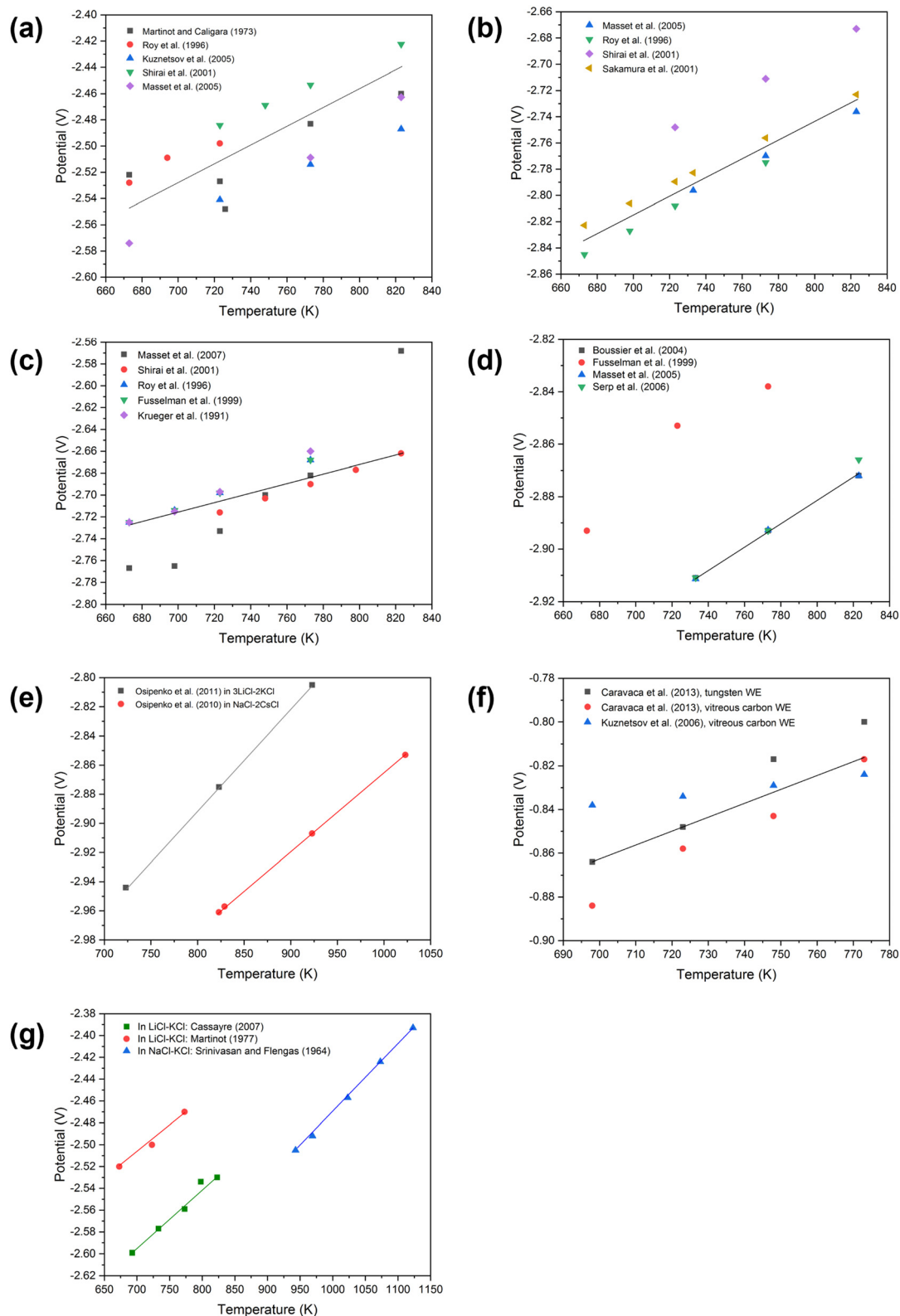


Fig. 6 Comparison of experimental data for certain actinide and lanthanide species. The following have been compared (the value in brackets refers to Tables 6–12 with accompanying references): (a) U (6), (b) Pu (7), (c) Np (8), (d) Am (9), (e) Cm (10), (f) Eu (11) and (g) Th (12). Reference electrodes are specified in the footnotes below Tables 6–12.

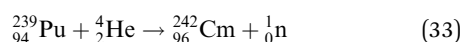


Table 9 Summary of reduction potentials of $\text{Am}^{2+}/\text{Am}^0$ at various temperatures in molten LiCl-KCl

<i>T</i> (K)	Apparent experimental potential (V) of $\text{Am}^{2+}/\text{Am}^0$					
	Boussier <i>et al.</i> ^{150 b}	Fusselman <i>et al.</i> ^{138 b}	Roy <i>et al.</i> ^{69 a}	Masset <i>et al.</i> ^{120 b}	Serp <i>et al.</i> ^{147 b}	Lambertin <i>et al.</i> ^{148 b}
673		-2.893				
698						
723		-2.853	-2.843			
733	-2.911			-2.9114	-2.911	
743						-2.945
773	-2.893	-2.838		-2.8927	-2.893	
823	-2.872				-2.872	-2.866

^a Reference electrode: Ag/AgCl . ^b Reference electrode: $\text{Cl}_2/2\text{Cl}^-$.

production of the transfermium nuclides (elements with atomic numbers greater than one hundred).¹⁵⁴

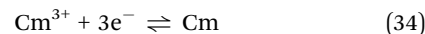


Curium has a much lower volatility than americium. ${}^{242}\text{Cm}$ exhibits high levels of alpha, beta and gamma radiation rendering any laboratory-based work hazardous. As a result, extremely small quantities (on a milligram scale) of curium are used when conducting analytical investigations, with crystallographic examination being hampered because of the damage caused by the gamma rays to X-ray detectors.¹⁵⁵ The high specific activity of curium isotopes make their partition from radioactive waste desirable. Isotopes of curium have large neutron-capture cross-sections which necessitate their partition from materials prior to processing to prevent contamination from capture products of curium in further experimental work.¹⁵⁶ The high neutron capture cross sections of curium isotopes make them ideal for destruction by transmutation to radioisotopes with shorter half-lives.

In the aqueous phase, curium assumes a trivalent state as this is the most stable oxidation state. This stability is a result of the half-filled ($5f^7$) configuration. In strictly anhydrous conditions, the 4+ oxidation state of curium occurs frequently as a compound, examples being CmO_2 and CmF_4 . Much of the electrochemical work is focused on the 3+ oxidation state and there are very few studies that have been conducted in molten salt systems, with only a single report on the standard potential in Cm-Zn alloys.^{54,157,158} Two separate studies have been undertaken using different molten salt systems focusing on the reduction pathway adopted by Cm^{3+} in 3LiCl-2KCl and NaCl-2CsCl ,^{156,159} the apparent standard potentials are shown in Table 10. Fig. 6(e) shows the difference between reduction potentials in two different molten salt systems. As the temperature increases, the reduction potential becomes more positive, suggesting the practical viability of this process at lower temperatures. In both, techniques such as CV and SWV, were used to elucidate the reduction pathway and determine the diffusion coefficient. It was found in both cases that the process remained reversible when the sweep rate was below 0.1 V s^{-1} . Thus, the process was governed by mass transfer and was proven to be irreversible as shown by the observed cathodic

shift (according to a higher sweep rate suggesting the process was charge transfer controlled).¹⁶⁰

SWV of the NaCl-2CsCl system revealed a single independent peak indicating a single-step three electron transfer between -1.0 and -2.1 V vs. Ag/AgCl . According to eqn (29), the number of electrons exchanged is: 2.99 ± 0.15 electrons. Thus, eqn (34) describes the reduction found in this experiment.



The number of electrons exchanged was determined through SWV in both molten salt systems measuring the half-width of the peak recorded over a range of different frequencies: 12–30 Hz for the NaCl-2CsCl system and 6–80 Hz for the 3LiCl-2KCl system. There was good agreement between both systems; from the first, a total electron exchange of 2.99 ± 0.15 was reported compared with the second system where an exchange of 2.98 ± 0.06 electrons was calculated. Inspecting the cyclic voltammograms more closely reveals a difference in the cathodic potential between the two systems; in the 3LiCl-2KCl system, the single cathodic peak appears at $\sim -1.95 \text{ V}$ compared with $\sim -2.05 \text{ V}$, at 723 and 823 K, respectively.¹⁶¹ In both studies, the diffusion coefficient was found to increase with temperature according to Arrhenius-like behaviour. However, the associated activation energy was shown to be much lower in the 3LiCl-2KCl system (28.2 kJ mol^{-1}) compared with the NaCl-2CsCl system (44.5 kJ mol^{-1}); this is a result of the nature of the ionisation potential of the LiCl salt which is the smallest of the alkali metal cations.¹⁶²

5.6 Europium

Europium, atomic number 63, is a member of the lanthanide series and occurs naturally in minerals such as bastnaesite ($\text{La, CeCO}_3\text{F}$) and monazite (La, CePO_4). Typically, the concentration of europium is below 0.1%. This element is regarded as one of the most reactive metals of the lanthanide series being stored under an inert fluid to prevent oxidation. As a metal, it can fluctuate between a divalent and trivalent state with application of pressure, existing primarily in the 3+ oxidation state.¹⁶³ Europium is not widely available and the method by which it is extracted from any mineral is dependent on its concentration in relation to other lanthanide like elements (cerium, yttrium and ytterbium) and the composition of the ore. This has led to the employment of different processing strategies. McCoy initially used zinc to extract europium from other lanthanide elements followed by the precipitation of europium (previously known as europous) sulfate with high recovery efficiency.¹⁶⁴ However, europium is a commonly found byproduct of irradiated nuclear fuel and efforts have been made to focus on an extraction method in molten salt media focusing on the reduction of EuCl_3 to EuCl_2 .¹⁶⁵ Of all the lanthanide metals, there has been relatively little progress made in understanding the electrochemistry of the europium species, with most known about the trihalides.^{166,167}

One of the first molten salt studies performed on europium was in a LiCl-KCl salt to determine the standard electrode potential in comparison with samarium and ytterbium; in this



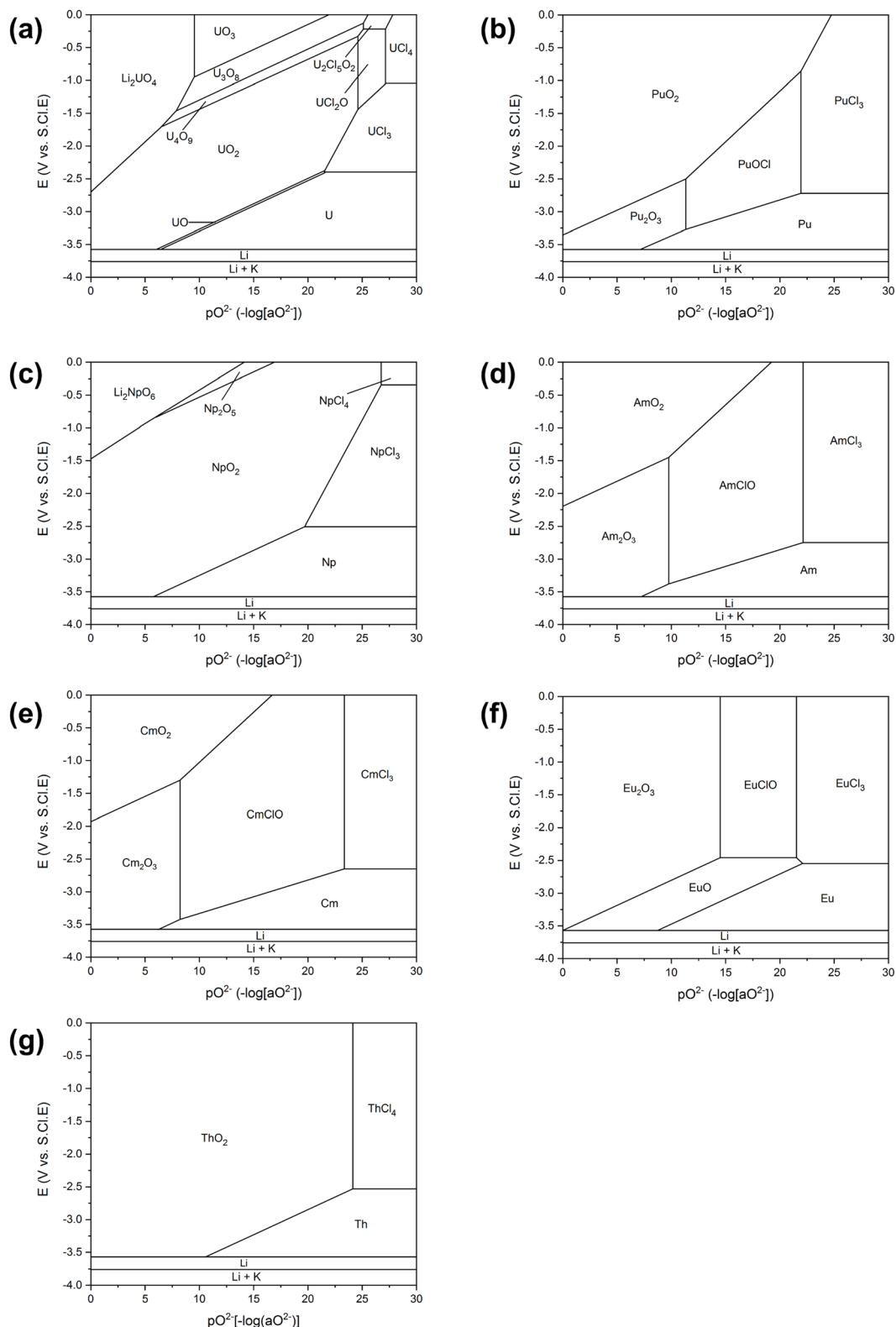


Fig. 7 Predominance diagrams for the most relevant actinide elements; (a) U, (b) Pu, (c) Np, (d) Am, (e) Cm and (g) Th and lanthanide element; (f) Eu species in LiCl–KCl eutectic at 500 °C. All diagrams have been drawn using HSC Chemistry 6.12.

study europium was found to have the most positive potential.^{166,167} Other work has measured the redox potential for the $Eu(II)/Eu(0)$ process, considering Eu – Pb alloys in $NaCl$ –

$CsCl$ and the electrochemical behaviour of $Eu(II)$ at low temperatures.¹⁶⁸ Additional work has attempted to explain the behaviour of Eu^{3+}/Eu^{2+} in LiF – NaF – KF molten salt;

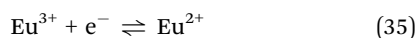


Table 10 Summary of reduction potentials of $\text{Cm}^{3+}/\text{Cm}^0$ at various temperatures

<i>T</i> (K)	Apparent experimental potential (V) of $\text{Cm}^{3+}/\text{Cm}^0$		
	Osipenko <i>et al.</i> ¹⁵⁶ in 3LiCl–2KCl ^a	Osipenko <i>et al.</i> ¹⁵⁹ in NaCl–2CsCl ^b	Osipenko <i>et al.</i> ¹⁶¹
723	–2.944		–2.924 (3LiCl–2KCl)
823	–2.875	–2.961	
829		–2.957	–2.996 (NaCl–2CsCl)
923	–2.805	–2.907	
1023		–2.853 ^c	–2.727 (NaCl–KCl)

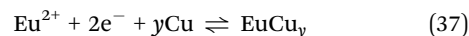
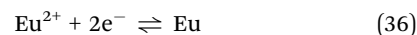
^a Reference electrode: Ag/AgCl. ^b Reference electrode: $\text{Cl}_2/2\text{Cl}^-$. ^c Value has been extrapolated between 829–1023 K using correlation: $E_{\text{Cm}^{3+}/\text{Cm}^0} = -3.285 + 5.48 \times 10^{-4}T$.

Kuznetsov *et al.* focused on the electrochemical behaviour of Eu^{3+} species in chloride melt, with studies in NaCl–KCl^{84,169} and CsCl at 973–1173 K.¹⁷⁰ The apparent standard potentials measured are summarised in Table 11. A report of standard potential in NaCl–KCl was the first comprehensive study attempted by Kuznetsov *et al.* on $\text{Eu}(\text{III})$. It was found that the reduction to $\text{Eu}(\text{II})$ proceeded in a single step with the exchange of one electron.

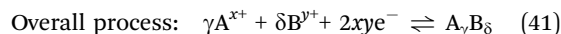
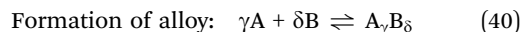
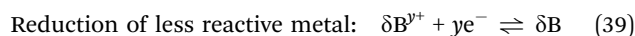
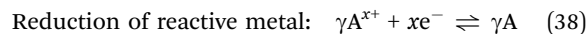


Gibilaro *et al.* studied the extraction of EuF_3 in LiF-CaF_2 at 840 °C.¹⁷¹ They demonstrated that the reduction proceeds in two steps, the first is in accordance with eqn (35); the second reduction occurs with the exchange of two electrons, as described in eqn (36). However, this process had a significant issue; the Eu^{2+} reduction potential was found to occur at a more negative potential than the deposition potential of lithium meaning that the final (direct) recovery of europium was difficult. To circumvent this issue, a reactive cathode substrate, copper, was employed leading to an intermetallic compound of the nature MN_y , where M is the metal substrate (copper) and N is the deposited metal species (europium). Compared with the inert metal electrode, the appearance of a large peak in the cyclic voltammogram is associated with the formation of an intermetallic compound shown in eqn (37). The application of

copper as a reactive cathode substrate was also reported, with success, in the extraction of neodymium and gadolinium.¹⁷²



Linear sweep voltammetry and chronopotentiometry techniques were used to find the diffusion coefficient for both oxidation states and they showed good agreement. The diffusion coefficient was found to vary inversely with the oxidation state. Subsequent work considered the co-electrodeposition behaviour of both Eu^{2+} and Al^{3+} in a LiCl–KCl eutectic salt; it was found to be essential to maintain the concentration of Al^{3+} below that of Eu^{2+} to increase the size of the potential window and to prevent the dissolution of aluminium.¹⁷³ In the reaction process, however, the Al^{3+} and Eu^{2+} ions are reduced consecutively which leads to the formation of Al_xEu_y (eqn (38)–(41)). On the counter electrode, aluminium metal is continuously oxidised; this then migrates to the working electrode to replenish depleting levels of Al^{3+} on the working electrode. This subsequently reacts with Eu^{2+} to form Al_xEu_y . Gibilaro *et al.* also published results on the co-deposition process of europium and aluminium ions using an inert electrode.¹⁷¹ The main principle behind forming an alloy is that there must be a distinct difference in the reactivities of both metals.^{174,175} Using A to denote a reactive metal species and B to denote a less reactive metal; then the following can be stated:



The work undertaken on the reduction of Eu^{3+} to Eu^{2+} in chloride melts revealed that the process occurs as a single electron transfer (Fig. 6(f)) similar to the pathway outlined in the previous processes.¹⁷⁶ From this work, it was found that the extent of reduction was directly proportional to the square root of the potential scan rate and similarly to neptunium the potential was found to be dependent on sweep rate up to 0.1 V s^{-1} only. Between 0.1 – 0.3 V s^{-1} , the process adopted a mixed diffusion and electron-controlled process and beyond this sweep rate the process observed was solely controlled by electron transfer kinetics.¹⁷⁷ However, the work by Schroll *et al.* exhibited large discrepancies in data associated with differences in electrode surface area which induce different measured currents.¹⁷⁶ There are also differences associated with lower operating temperatures where the peak potential varies at sweep rates up to 0.6 V s^{-1} ; up to this point, the reduction is limited by mass transfer, and beyond this point the process tends to be mostly electron-controlled.¹⁷⁸ In both of these experiments, the peak current was found to be dependent on the concentration of EuCl_3 , the same results being found in fluoride melts.¹⁷⁹ In this process, at a temperature of 1073 K, the CV indicated that the reduction of Eu^{3+} to Eu^{2+} occurs at $\sim 0.18 \text{ V}$ vs. a Pt electrode.

Table 11 Summary of reduction potentials of $\text{Eu}^{3+}/\text{Eu}^{2+}$ at various temperatures

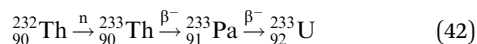
<i>T</i> (K)	Apparent experimental potential (V) of $\text{Eu}^{3+}/\text{Eu}^{2+}$			
	Schroll <i>et al.</i> ^{176a}	Caravaca <i>et al.</i> ¹⁸⁰ (tungsten WE) ^b	Caravaca <i>et al.</i> ¹⁸⁰ (vitreous carbon WE) ^b	Kuznetsov <i>et al.</i> ^{181b}
698		–0.864	–0.884	–0.838
723	–0.355	–0.848	–0.858	–0.834
748		–0.817	–0.843	–0.829
773	–0.333	–0.800	–0.817	–0.824
823	–0.307			
873	–0.282			
923	–0.252			
973	–0.228			
1023	–0.203			

^a Reference electrode: Pt. ^b Reference electrode: $\text{Cl}_2/2\text{Cl}^-$.



5.7 Thorium

Until recently, advances in actinide chemistry were mainly concerned with plutonium and uranium species. More effort is now being given to understanding the reduction pathway adopted by the oxides and halides of thorium. Thorium, atomic number 90, has a high melting point (1755 °C) and good resistance to corrosion. Thorium has been widely used by industry compared to other actinides and lanthanides. The first most useful application, in the late nineteenth century, occurred as a result of its incandescent properties within a gas mantle.¹⁸² Thorium has been used in alloys such as Mg–Th and Mg–Th–Zr with markedly improved thermal resistance.^{183,184} More recently thorium has come to the fore as it exhibits several advantages as a nuclear fuel: (1) it can withstand greater burnup compared to uranium and plutonium fuel; (2) it is three to four times more abundant than uranium in the earth's crust, (3) it results in lower levels of waste compared to uranium and plutonium fuels, and (4) it promotes proliferation resistance due to the complexity in separating fissile isotopes. Thorium itself is not fissile in a thermal nuclear reactor but upon bombardment of thermal neutrons natural ²³²Th is converted to ²³³Th. ²³³Th is known to have a half-life of approximately 20 minutes and decays quickly to ²³³Pa. ²³³Pa has a half-life of approximately 30 days and decays to ²³³U (a fissionable radioisotope) according to eqn (42) where n refers to a neutron and β[−] refers to beta decay. For this reason, ²³²Th is described as 'fertile' as it does, upon neutron bombardment, form ²³³U, which is a fissile isotope.



²³²Th is the most abundant naturally occurring isotope of thorium, with trace quantities of other more radioactive isotopes. Therefore, it is a high value material and of significant interest as a nuclear fuel. There has been a wealth of work focused on understanding the reduction of thorium halides *via* both metal-thermic and electrolytic processes using different feed materials. Some of the early work for thorium considered the reduction of ThCl₄ in sodium with a focus to produce large quantities of high purity thorium metal; this method relied on a chlorination step prior to the formation of the feed material ThCl₄.¹⁸⁵

The production of ThCl₄ incorporated either one of two reagents: sulfur monochloride or phosgene chloride. It was found after using sulfur monochloride that the product had large quantities of sulfur contaminant, in excess of 300 ppm, whereas phosgene chloride had much lower levels of impurity but needed nearly 60 h to produce the feed material, ThCl₄. The reaction between hydrogen chloride gas with ThO₂ has been reported to produce ThCl₄ at 400 °C.¹⁸⁶ In molten chloride systems, only Th⁴⁺ is stable.¹⁸⁷ There is still limited literature on the stability of divalent, Th²⁺, and there has been no verification that divalent thorium occurs in molten chloride systems either.^{83,188} Furthermore, the reaction between Th⁴⁺ and hydrogen azide in an aqueous solution leading to the formation of Th³⁺ is still under question despite an experimental investigation by Klapötke and Schulz.¹⁸⁹ Their means of demonstrating its existence was *via* UV-vis and electron

Table 12 Summary of reduction potentials of Th⁴⁺/Th⁰ at various temperatures in molten LiCl–KCl and NaCl–KCl

T (K)	Apparent experimental potential (V) of Th ⁴⁺ /Th ⁰		T (K)	Apparent experimental potential (V) of Th ⁴⁺ /Th ⁰
	Cassayre <i>et al.</i> ^{188b}	Martinot ^{192b}		Srinivasan and Flengas ^{191a}
673	−2.599	−2.52	943	−2.505
693			969	−2.492
723		−2.50	1023	−2.457
733	−2.577		1073	−2.424
773	−2.559	−2.47	1123	−2.393
798	−2.534			
823	−2.53			
798				
823				

^a Reference electrode: Ag/AgCl. ^b Reference electrode: Cl₂/2Cl[−].

paramagnetic resonance (ESR) spectra. Ionova *et al.* dispute the findings on grounds of the process being thermodynamically impossible and based upon a misinterpretation of the absorption spectra.¹⁹⁰ The absorption band of the azide anion ligand lies between 320–680 nm and this complex would only exist with thorium in the 4+ oxidation state. Fig. 6(g) describes the range of potentials found experimentally for the reduction of Th⁴⁺ with a difference of 0.1 V between the temperatures for both LiCl–KCl systems. Table 12 summarises this data and there is some agreement in the potentials reported by Srinivasan and Flengas.¹⁹¹

Work had previously considered eliminating the halogenation step and instead focused on thorium oxide which would be easier to prepare. To the authors' knowledge, Meyer's study is the first known example of the electrolytic reduction of thorium oxide in a fused metal chloride/fluoride system.¹⁹³ Meyer's findings reveal that ThO₂ responded far more amenable in a molten fluoride system, KF–ThF₄, yielding a higher purity product and a large particle size <7–8 μm compared with a NaCl–KCl–ThF₄ system, although this is one study alone.¹⁹³ Even though the working temperature of the chloride melt was 200 °C less than the fluoride melt and promoted a greater solubility for the constituents in the leaching process, this led to the formation of an impure product with a smaller particle size. Clayton, Mamantov and Manning considered the reduction of both thorium and uranium in a mixed chloride/fluoride melt system at 500 °C.¹⁹⁴ Their findings revealed a poorly defined peak at −2.03 V *vs.* a Ni(n)/Ni(0) reference electrode followed by a much larger and well-defined re-oxidation peak of the lithium metal at −1.83 V. This study implies that the reaction proceeds in a single step from Th(IV) to thorium metal with an exchange of four electrons, although there was no detailed CV/SWV work to support it. Comparison of different electrodes, in particular nickel, platinum and tungsten, revealed a discrepancy in the resolution of the peaks. The occurrence of this has been ascribed to the ability of platinum to form an alloy with lithium metal, thereby reducing the overall cathodic range.¹⁹⁵

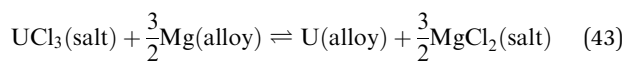
The electrochemical reduction of ThCl₄ in LiCl–KCl,¹⁹² where a pure chloride electrolyte is used, indicates that the reduction process



is electrochemically feasible at 500 °C (see Fig. 6(g)).¹⁹⁶ This implies that it would be more sensible to work in a pure chloride electrolyte; furthermore, the reduction of ThF₄ in NaCl–KCl at 800 °C yielded thorium metal with a reported purity of 96.6 to 97.8%. Later, Martinot studied the behaviour of ThCl₄ in a pure chloride electrolyte finding that the process was near reversible.¹⁸⁶ Compared with work accomplished on other actinides, the feed material varied in concentration between 5 and 20%; due to the much lower radio-toxicity of thorium, higher concentrations were safely handled, leading to an overall impurity of no more than 250 ppm.

6 Developing computational models for electrorefining

Computer-aided simulation can be used to model electrochemical processes taking place in a molten salt electrolytic cell. Experimental studies are time-consuming, difficult to monitor and control, especially with radioactive components and hence expensive to conduct and provide data. In comparison, computational models offer an accelerated/augmented means of understanding relevant processes. Models are also useful in steering the direction of future experimental work in more efficient directions. At present, various models exist in relation to molten salt work incorporating a variety of parameters and variables deemed suitable for each simulation. One of the very first models to be designed was that by Johnson *et al.* at the ANL; this thermodynamic model was developed based upon the distribution between uranium metal in a molten salt system and a binary alloy (eqn (43)).¹⁹⁷



This theory can be applied to a uranium/plutonium system, where the transfer of uranium and plutonium occurs simultaneously in a single-phase liquid alloy, based upon the following equilibrium process (eqn (44)). It was realised that an alloy acceptor could be used in this process to result in two product streams, which are rich in uranium and plutonium, respectively.



Several other simulation codes were subsequently developed including the so-called TRAIL code. Kobayashi and Tokiwai incorporated data on the diffusion layer thickness which varied according to the uranium concentration in a LiCl–KCl eutectic.¹⁹⁸ This study was designed to simulate the reduction process for a small-scale (10 g) electrorefiner (see Fig. 3). In a similar study, the same group then improved their model to consider how the various overvoltage components influence cell resistance and operating cell voltage.¹⁹⁹ The model predicted that at a concentration of only 0.5 wt% for uranium the resistance should increase with cell voltage. Beyond 2 wt%, no difference in the resistance of a cell with change of voltage was anticipated. The DEVON model factored in the resistance of

molten salt systems in two dimensions but failed to account for the activation overpotential.²⁰⁰

Fundamental studies considering current density distribution and the nature of metal ion movement were first reported by Lee *et al.*²⁰¹ These studies used a primary current density model to investigate the nature of uranium deposition and molten salt flow through an anode basket. From this study, a cross-sectional area factor (CSAF) was implemented to predict the shape and size of each of the uranium particles where a value of unity corresponded to a perfectly spherical particle. In practice, estimating the CSAF value of uranium deposit was difficult due to the irregular dendritic geometry. This geometry was also found to change according to the speed of the stirrer which changes the volume fraction of the metal within the molten salt system. By increasing the degree of agitation (>100 rpm) and CSAF, which is proportional to the frictional force, the quantity of uranium deposited increased in the ‘stirrer’ domain. Here, the frictional force determines the motion of uranium dendrites once they spall from the surface of the current collector. Therefore, to recover high-purity uranium effectively an accurate value of the CSAF must be used together with knowledge of the friction force as this is presumed to affect the purity of the uranium product. Kim *et al.* developed a tertiary current density model to evaluate the separation behaviour and the nature of uranium deposition in a complex electrode system.²⁰²

Ghosh *et al.* modified the entire theory leading to the PRAGAMAN code which may be used to determine the results of changes in the solubility arising at the interface of the cathode and anode – leading to 16 different potential solubilities.²⁰³ These simulations were useful for making thermodynamic predictions of the outcome of a process but failed to consider the kinetic effects in relation to the reactant and product species that are found within an experimental process. Ghosh *et al.* then considered the nature of anodic dissolution for uranium and zirconium for a three-electrode system but failed to report the electrode surface area for the working electrodes meaning that the results are not comparable.²⁰⁴ Sequential work has focused on modelling the Mark-IV electrorefiner which works by simultaneously oxidising uranium at the anode and reducing uranium metal at the cathode (for further details see Section 4.2). Several components of the electrorefiner have been simulated such as the cathode, vessel wall and fuel basket using fundamental electrochemical theory.²⁰⁵ Prior to this, Ahluwalia *et al.* designed the General Purpose Electrochemical Simulator (GPEC) based upon the Butler–Volmer Equation and thermodynamically available data.^{206–208} This model was used to consider the behaviour of zirconium in the Experimental Breeder Reactor (EBR)-II. The REFIN model incorporates the diffusion layer model reported by Kobayashi and Tokiwai¹⁹⁸ and simplifies the Butler–Volmer kinetics model.²⁰⁹ Moreover, in line with producing a multi-component model which was able to simulate kinetic effects and fluid flow, Zhang’s model captures the partial current of elements and their distribution within the cell.²¹⁰ This model has several shortcomings; it does not consider nucleation



during deposition or the process of diffusion at solid electrodes. It also does not enable the control of species at the interface between the molten salt and electrode and does not have the capability to model multiple oxidation states for different species.

From this, the Faradaic current for each element has been determined for a one-dimensional model where a solid cathode is used. Execution of this code is troublesome due to the long run times but a model developed by Cumberland and Yim, ERAD, was developed to extend the REFIN model and to focus specifically on simulating transient phenomena such as cyclic voltammetric peaks and rate-limiting steps in electrorefining (ER) processes.²¹¹ The same group used a least-squares fitting method to derive values for the apparent standard potential *vs.* Cl₂/Cl⁻, diffusion coefficient and the exchange current density based on their ERAD model.²¹² This model does not account for kinetic factors, fluid flow characteristics, ohmic drop or metal deposition; it is purely one dimensional and is a useful method to fit/compare data and test assumptions without needing a prolonged period to converge.

7 Precursor types and electrochemical reactor design

7.1 Electroreduction techniques

Much of the success in molten salt technology lies in choosing appropriate operating parameters, electrode materials, electrochemical reactor designs and material selections. Choosing an appropriate molten salt, whether as a binary mixture or not, can also warrant a specific choice of electrochemical reactor to prevent vessel degradation, or electrode material to avoid corrosion. It is also well-known that each molten salt will have unique properties (summarised in Table 3) and thus govern the operation parameters and operational temperature. Moisture, for example, is an unavoidable contaminant within any cell design and different steps are taken to eschew this issue; for example, by vacuum drying salts for prolonged periods or conducting a pre-electrolysis against a sacrificial, inert electrode which might, in turn, lead to unwanted parasitic reactions.²¹³ This section focuses specifically on precursor types of cell designs which lies within the experimentalist's control, offering a fresh outlook on previous work by exposing some of the advantages and disadvantages of each technique.

Several precursor types have been explored in molten salt processes including uniaxially pressed (sintered) pellets, metal cavity electrodes (MCEs), thin-film electrodes, the fluidised cathode and liquid metal cathodes. A pellet of metal oxide (MO_x) is formed by combining the metal oxide powder with an appropriate lubricant or binder (often this is polymeric), such as bis-stearamide, before pressing the powder uniaxially at a pressure between 125 to 135 MPa. The addition of a lubricant is to improve the consistency and stability of the pellet during the reaction. The pressing step shortly precedes the sintering of the pellet which subsequently improves its mechanical strength for handling and integrity and removes any residual moisture.

To date, there has been considerable work accomplished on TiO₂,^{34,214–216} and Ta₂O₅,^{217,218} as part of pellet-based studies. In most studies, the authors have reported that the metal oxide pellet is left partially reduced; by this it is found that the outer surface of the pellet preferentially participates in the reaction and is reduced due to the O²⁻ migration within the neighbouring electrolyte, leaving the inner-most part of the pellet unreduced or otherwise partially reduced. Studies have also explored the consequence of varying the porosity, density and pellet size, finding that a limit of 45% porosity normally yields the most beneficial results.²¹⁸

To explore this outcome further, pellet reduction studies on TiO₂ have been undertaken by Schwandt *et al.* who present a stepwise time-dependent analysis of the slow, progressive reduction of the TiO₂ pellet.²¹⁹ In the same study, due consideration is given to the volume of the pellet itself, with larger or denser pellets limiting the current efficiency and thus the degree of completion of the process. On the other hand, Fray, Farthing and Chen report that whilst the density of the pellet can impede the completion of the process, the use of a pellet can ensure oxidation inhibition after cooling.⁴³ Because the pellet is withdrawn from the molten salt system, the residual salt that has penetrated within the pores will solidify, preventing any re-oxidation of the as-produced metal. Post-treatment processes are relatively simple, especially for CaCl₂ molten salt systems where the reported solubility in water is 1.59 g ml⁻¹ which results in a significantly more efficient process.⁴³ Increasing the metal oxide-to-salt ratio results in a rise in measured current because there are more frequent interactions between the electrode and metal oxide particles. This may incur an iR drop, *i.e.*, where voltage losses occur through the resistive elements, which manifests itself in cyclic voltammetric and chronoamperometric studies where greater overvoltage potentials are usually reported.

Recent work has shed light on some of the electrochemical reduction processes of NiO/Cr₂O₃,²²⁰ Tb₄O₇,²²¹ Ca²⁺/Ca,²²² and MoS₂.²²³ Studies on Nb₂O₅ in CaCl₂ melt have found that although reduction of the oxide is able to take place, the subsequent formation of intermediate states as a result of intercalation between the salt and the metal oxide results in an expansion, due to formation of CaNb₂O₆ and Ca₃Nb₂O₈, which impedes the subsequent reduction to the metal.²²⁴ In this study and other reports, MCEs are prepared by pressing a series of holes or cavities either by laser or drilling or through chemical etching within a molybdenum sheet before affixing to a working electrode such as tungsten using a conductive wire.^{220–223,225} This process alone can inhibit the number of pores available for the migration of O²⁻ ions, and coupled with a potential that is too positive, explains the poor current efficiency and the long period required to reduce the metal oxide.

Moreover, experimental studies have confirmed that during the electrolysis it is quite likely that the metal oxide may fall from the cavity as a result of an associated volume change, thereby reducing the overall current efficiency.²²⁶ This can also cause the dissolution of the metal within the melt, depending on its solubility, which would affect the current efficiency. Furthermore, molybdenum metal cannot be re-used as frequently due to the high



temperature and voltage in a harsh molten salt environment reducing its overall mechanical integrity and thereby increasing overall costs.

On a separate note, some drawbacks have been reported with the use of MCEs. The first concern relates to the control of the surface area in contact with the molten salt. Most experimental work will have a wire of low surface area in contact; however, for MCEs there is a much greater applied surface area which can often lead to smaller charging–discharging currents during double-layer charging.²²⁰ The second concern arises due to background current that may result through contact of the molten salt and substrate, hence the importance of selecting an appropriately inert electrode material. These currents may then mistakenly be reported as intrinsic electrode processes requiring that control experiments are undertaken to ensure robust results.²²⁷

‘Fluidisation’ or fluidised bed processes are where particulate solids come into close contact and agitation with fluids or gases results in a transformation where a pseudo-mixture is formed enhancing and improving the kinetics of a given process. Some of the earliest work in this field dates back to the 1960s and considers the use of fluidised bed electrodes for fuel cells.²²⁸ More recently, this technology has been established in several areas such as the synthesis of alkaline peroxide solutions,^{229,230} improving water purity by nitrate extraction and to facilitate copper deposition.^{231–234} It has had wide-reaching impact with applications to catalytic cracking and has been extended to consider the combustion phenomena of hydrocarbon fuels with inert particles within a fluidised bed system.^{235–238} Needless to say, this technology is versatile and offers many advantages: a large specific surface area, increased mass transfer, free-flowing character of the fluidised bed, constant temperature distribution (enhanced heat transfer) and uniformity of mixing. Progress has already been made in applying the theory of ‘fluidised bed electrodes’ to electrochemical reduction processes. Abdulaziz *et al.* first implemented this in the study of tungsten oxide in LiCl–KCl and in reducing UO₂ (see Fig. 8 for details of the fluidised cathode process).^{239,240} In this context, argon gas is used to ‘fluidise’ the mixture through constant bubbling. In the reduction of tungsten oxide, several parameters were varied including metal oxide-to-salt ratio as well as the argon gas flowrate. It was found that increasing the metal oxide-to-salt ratio yielded larger currents because of an increase in particle collisions with the current collector; however, there was a limit to its efficiency as exceeding 100 g of tungsten oxide in 150 g LiCl–KCl proved difficult to sustain. This might have been due to the size of the system which prevented all the tungsten oxide particles from fluidising. A larger system with a greater working temperature may have yielded better results. More recently, this technique has been successful in the recovery of cobalt from both commercial and spent Li-ion battery materials with reported faradaic efficiencies of >70%.²⁴¹

Fig. 9 illustrates the nature of particle collisions with a current collector for a fluidised cathode process. In this example, UO₂ has been selected as the metal oxide particle. The success of

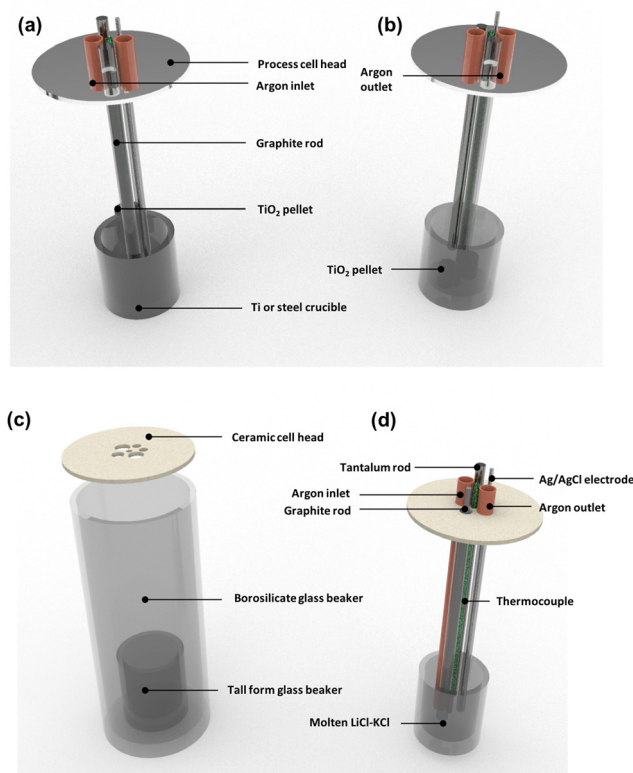


Fig. 8 3D rendering of the electrochemical cell for the FFC–Cambridge Process: in (a) the current collector is attached to the cathode and in (b) the pellets are electroreduced *in situ* at the bottom of the crucible. In (c) and (d) the fluidised cathode process is illustrated with the argon inlet sitting within the molten salt system which agitates the solution continuously throughout the process.

this process, as demonstrated by the figure, lies in the initial contact between the uranium oxide and current collector surface; thereby facilitating the reduction process. The initial collision led to the formation of a deposit on the surface of the current collector (Fig. 9). The scanning electron micrograph in Fig. 9(b) indicates that after the process the particles agglomerate. Fig. 9(c) illustrates the initial increase in current *versus* time; this is in line with the reduction of uranium oxide where we expect the formation of extensive three-phase interlines (3PIs) enabling greater salt penetration and therefore an increased current density.

7.2 Applications of liquid metal cathodes

Liquid metal cathodes, shown in Fig. 10, are widely used in the field of molten salts, displaying excellent electrochemical properties: a constant surface area of contact with the active material, greater diffusion coefficients into the salt, typically a lower applied cell voltage and they enable the reduction of highly electronegative elements.^{59,242} The latter two properties are explained by the deposition potential, where the target metal ions behave differently depending on the state of the cathode (solid/liquid), thereby resulting in a more negative applied potential, which applies to the electrorefining process.



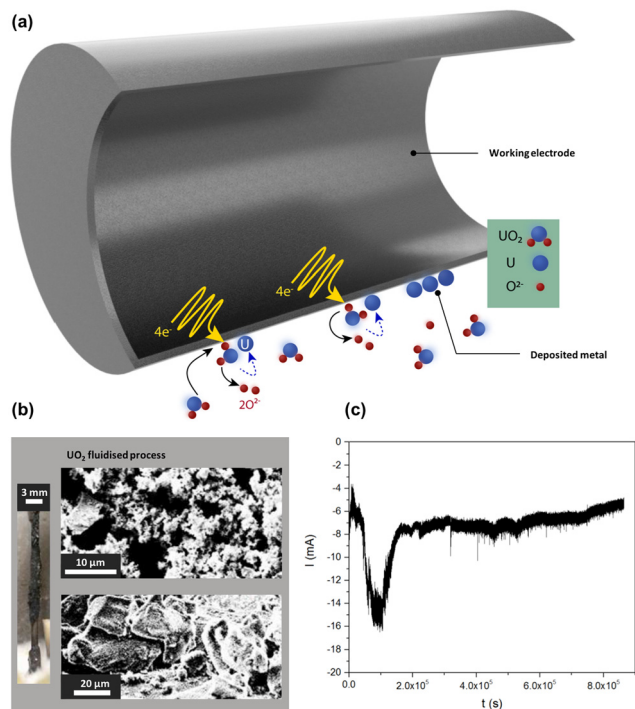


Fig. 9 A schematic diagram of the working electrode surface. (a) The reduction begins with the initial impingement of the metal oxide particles against the surface of the current collector. Thereafter, a direct transfer of electrons occurs resulting in the formation of uranium atoms and oxide ions which subsequently migrate towards the surface of the anode. (b) Scanning electron microscopy reveals the agglomeration of uranium particles following the fluidised cathode process. A photograph shows the deposited metal on the surface of the current collector. (c) Current vs. time curve of UO₂ fluidised cathode in LiCl–KCl eutectic at 450 °C, argon flow rate: 600 cm³ min⁻¹, voltage: –2.2 V vs. Ag/AgCl. (b) and (c) have been reprinted from Abdulaziz *et al.*²³⁹

This results in a lower cell voltage and reduces the overall energy consumption; several studies have reported these promising features for liquid metal cathodes.^{62,243,244} Advantageous electrochemical properties have also been realised for the single-step reduction of Ti³⁺ in NaCl–KCl melt on a liquid tin cathode where the multi-step reduction on tungsten electrodes is avoided, incurring a lower energy penalty.²⁴⁵ In the nuclear industry, liquid cathodes have been widely used in waste disposal and spent-fuel processing. They have shown merit in health and safety concerns because they allow (depending on solubility) the target metal ions to dissolve within the liquid metal cathode reducing the radiation dose to workers and reducing the overall quantity of radioactive waste.

Gibilaro *et al.* investigated the use and application of liquid metal cathodes in molten fluoride systems.²⁴⁶ The work demonstrated successful studies on each of the following liquid metal cathodes: bismuth, gallium, lead, antimony and tin in two different melt systems LiF–NaF (750 °C) and LiF–CaF₂ (850 °C). In this technique, the target metal ions are anodically dissolved within the melt before being selectively recovered on the surface of the metal cathode; any fission product isotopes are found within the molten salt after it solidifies. Using transient electrochemical techniques such as linear sweep voltammetry, a hierarchy of reactivities

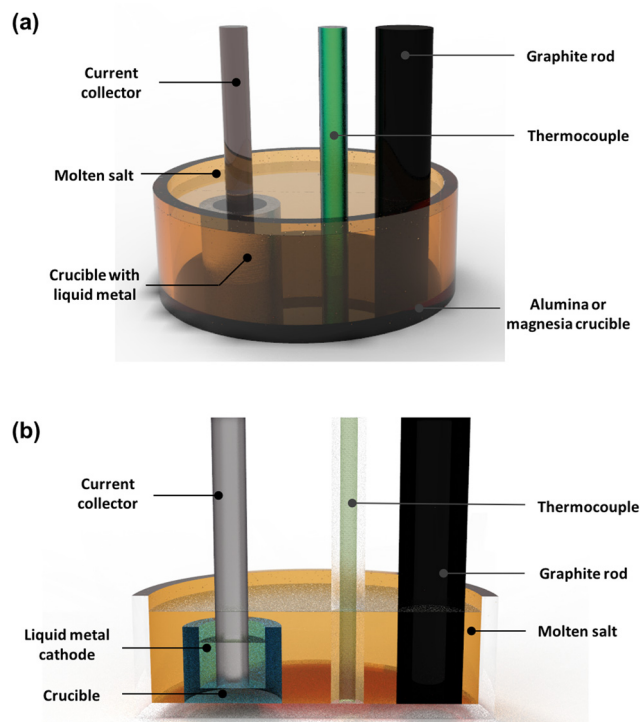


Fig. 10 A simplified arrangement of the liquid metal cathode electrochemical cell. (a) The current collector in this electrochemical cell sits in a liquid metal which is itself mixed with a molten salt electrolyte. This process works on the principle that the density of the liquid metal is greater than the surrounding electrolyte. (b) is a cross-sectional area representing these two different electrolytes (the molten salt medium and liquid metal cathode).

(gallium being the most reactive and bismuth the least) was found and subsequently, the rate of extraction was revealed from the data. Research programmes in Japan,^{59,60,243} US,^{58,63,247} and Korea,²⁴⁸ have made similar developments, exploring the electrochemical reduction of alkaline earths, lanthanides and some minor actinides in LiCl–KCl using a liquid Cd cathode. In 2007, the Idaho National Laboratory in the US reported engineering-scale (12–13 kg of depleted uranium processed) liquid Cd experiments in molten LiCl–KCl at 773 K.²⁴⁹

Notwithstanding the progress of liquid metal cathode work, appropriate selection and choice of design parameters must be made including: using a metal with low vapour pressure to avoid metal distillation, choosing a viable operating temperature range well below the boiling point of the metal, and ensuring that the density of the liquid metal is greater than the molten salt. Further work must also be undertaken on selecting an optimal material for crucible design. Vaden *et al.* chose a BeO crucible and upon product recovery found it to be damaged. Selecting a lower-cost material such as Al₂O₃ could work equally as well.²⁴⁹ On the other hand, many studies have used stainless steel baskets in the electrochemical reduction of uranium salts; XRD analysis in some cases has revealed iron impurities due to the homogeneous liquid alloy formation.¹⁰⁰ For metals to remain good candidates in scaling-up processes they must be non-toxic and inexpensive.



8 Summary and future outlook

The energy supply industry is changing rapidly, not least accelerated by global net-zero targets. Nuclear energy is regarded as one of the few low carbon technologies that can meet future global energy demands, but concerns surrounding its safe use and spent fuel management remain a major challenge. Effective methods for spent fuel processing and metal recovery must be available so as to minimise nuclear waste. Molten salt technology has shown promising capabilities in the extractive metallurgy of uranium and plutonium from metal oxides, MOX fuels and in the production of high-grade alloys. This review has shown that singular, binary and ternary molten salts, in specified molar ratios, are amenable to the electro-deoxidation processes of metal oxide nuclear fuels, although most advances cover only the molten chloride and fluoride systems. This may relate to the efficacy of the final product morphology, ease of handling, optical transparency, solubility of oxide ions, stability at both elevated temperatures and pressure and ultimately the reduced hygroscopic nature. Despite these advantages, it is difficult to separate the metal deposit from the molten salt and in most, if not all processes, some residual salt remains, skewing the reported recovery efficiency. Whilst it might still be true that molten chlorides (LiCl–KCl and NaCl–KCl) and fluorides (LiF–CaF₂) are optimal for electrochemical processes, there remains a void in experimental work on the lesser well-known molten salts which may also present favourable results depending on their interaction with transuranic and refractory elements and their physico-chemical properties.

The main body of this review comprises a thorough discourse of the most significant (defined in this context as the most abundant elements present in spent nuclear fuel) actinides and lanthanides (thorium, uranium, neptunium, plutonium, americium, curium and europium) concerning spent nuclear fuel reprocessing. Many experimental studies have been considered, some of which date back to the 1950s. Uranium and plutonium are two of the most widely studied elements and thus occupy a large section of this work. Both of these were studied by the US using precipitation, solvent extraction and electrorefining techniques for purposes of producing materials for nuclear weapons prior to civilian applications. Neptunium, americium and curium were present as by-products of nuclear decay or through neutron activation, and the extraction of these synthetic elements was necessary because they were impurities and were required to be removed from the uranium and plutonium either for further use in a nuclear reactor or for waste management purposes. Europium is the only lanthanide element studied in this work and the only viable means of extraction is *via* co-reduction with a reactive metal. Thorium is the last of the studied elements and perhaps the most promising for future applications; it is three to four times more abundant in the earth's crust than uranium and possesses a greater energy density compared to uranium as thorium reactor fuel theoretically can undergo a higher burn up. If the use of thorium in nuclear reactors, particularly molten salt reactors, is to come to fruition, investments will need to be directed towards understanding the operation and maintenance of such plant systems.

Much of the past work has sought to resolve the reduction pathways exhibited by the actinides and lanthanides and there has, to a large extent, been a wide range of success reported in a variety of operating conditions, cell designs and different molten salts. More studies have focused on chlorides in solution than on oxides. The selection of different electrode materials, a wide range of initial precursors (difference in oxidation states) and a variety of reference electrodes all contribute to many data inconsistencies in the literature. Whilst the proof-of-concept has demonstrated broad applicability, there needs to be a greater effort made towards reprocessing spent nuclear materials at the pilot-scale. Electroreduction, electrorefining and electrowinning are all processes that have yet to reach a mature stage. Improvements to their process operation and control are required. An understanding of criticality in the design process is also required. Some of the early indications as to whether the results associated with the transition from lab-scale to pilot-scale can be determined by use of computational fluid dynamic (CFD) models. Within these models, factors such as cell design, size, shape and arrangement of electrodes, operating current, liquid density and laminar viscosity all require consideration.

A variety of one, two and three-dimensional models for electrorefining have been reported in this review article and date back to the early 1990s. Each of these codes presents advantages and disadvantages in relation to how accurately a particular electrorefiner is modelled. Complete understanding of the operation of an electrorefiner is yet to be achieved; accurately predicting the fluid flow and appreciating the change of elemental oxidation states with time and current density distribution at the electrode/molten salt interface remain areas for future development.

A range of precursor types, cell designs and electrode configurations have been discussed. Initial electro-deoxidation studies of titanium dioxide pellets found that parameters such as porosity, pellet thickness and density are likely to affect the final product morphology and oxygen content. However, there are still lessons to be learnt from this process and others that focus on the production of refractory metals; in particular, whether the product morphology from different precursors is of relevance for industrial application. In light of this, more must be done to close the gap between the electrochemical and materials-based understanding of these processes. One of the key innovations for this technology remains in the use of the fluidised cathode process. It ensures both a constant and high surface area of contact and improves current efficiency. Continued study of its applicability to a broader range of materials and optimisation of process variables and conditions is required to assess the commercial promise for this process.

Molten salt electrochemistry provides a viable route to metal production, incurring a smaller energy penalty compared with harsh chemical reducing agents, and is anticipated to save on downstream costs in the production of powders and homogeneous alloys. The transition from laboratory to pilot-scale remains a challenge for this technology partly due to the chemical corrosion of electrochemical reactors at high temperatures, possible



irradiation hazards with respect to materials under a neutron flux and parasitic reactions resulting in the formation of alloys. Consideration of different anode/cathode materials to improve selectivity, optimising the reduction potential and improving the current efficiency are some of the strategies that can be employed. For a completely successful scale-up, there must also be progress made in economic feasibility which, as of yet, remains understudied. Above all, the future growth of this field relies on sustaining strong partnerships between the nuclear industry and universities.

List of abbreviations

3PI	Three-phase interline
AGR	Advanced gas-cooled reactor
ANL	Argonne National Laboratory
BWR	Boiling water reactor
CFD	Computational fluid dynamics
CSAF	Cross-sectional area factor
CV	Cyclic voltammetry
EBR	Experimental breeder reactor
ER	Electrorefining
FFC	Fray–Farthing–Chen
FP	Fission products
HEP	High-enriched plutonium
HEU	High-enriched uranium
IAEA	International Atomic Energy Agency
ILW	Intermediate level waste
LAMEX	Los Alamos Molten Plutonium Electrorefining
LEU	Low-enriched uranium
LNFP	Less noble fission products
MA	Minor actinides
MCE	Metallic cavity electrode
MOX	Mixed oxide
NM	Noble metal
NFP	Noble fission products
NM	Noble metals
P&T	Partition and transmutation
PUREX	Plutonium uranium reduction extraction
SA	Surface area
S.C.L.E.	Standard chlorine electrode
SWV	Square wave voltammetry
TBP	Tri- <i>n</i> -butylphosphate
THORP	Thermal oxide reprocessing plant
TPB	Triple-phase boundary
TRU	Transuranium (elements)
RE	Rare earth (elements)
XRD	X-ray diffraction

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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References

- World Nuclear Association, Nuclear Power in the World Today, <https://world-nuclear.org/information-library/current-and-future-generation/nuclear-power-in-the-world-today.aspx>, accessed 06/12/2022.
- D. J. Hill, *Nat. Mater.*, 2008, 7, 680–682.
- S. G. Thompson and G. T. Seaborg, *Modern Alchemy*, World Scientific, 1994, vol. 2, pp. 363–371.
- J. G. Reynolds, J. S. Page, G. A. Cooke and J. Pestovich, *J. Radioanal. Nucl. Chem.*, 2015, 304, 1253–1259.
- D. L. Clark, S. S. Hecker, G. D. Jarvinen and M. P. Neu, *The Chemistry of the Actinide and Transactinide Elements*, ed. L. R. Morss, N. M. Edelstein and J. Fuger, Springer, Netherlands, Dordrecht, 2008, pp. 813–1264.
- S. Thompson and G. Seaborg, *US Pat.*, US2785951A, 1957.
- B. C. Reed, *The History and Science of the Manhattan Project*, Springer, Berlin, 2014.
- B. C. Reed, *The Physics of the Manhattan Project*, Springer Nature, Switzerland, 4th edn, p. 2021.
- H. Anderson and B. Larned, *US Pat.*, US2924506A, 1960.
- E. R. Irish and W. H. Reas, *The Purex Process. A Solvent Extraction Reprocessing Method for Irradiated Uranium*, HW-49483 A, Washington DC, 1957.
- R. S. Herbst, P. Baron and M. Nilsson, *Advanced Separation Techniques for Nuclear Fuel Reprocessing and Radioactive Waste Treatment*, Woodhead Publishing, 2011, pp. 141–175.
- P. Baron, S. M. Cornet, E. D. Collins, G. DeAngelis, G. Del Cul, Y. Fedorov, J. P. Glatz, V. Ignatiev, T. Inoue, A. Khaperskaya, I. T. Kim, M. Kormilitsyn, T. Koyama, J. D. Law, H. S. Lee, K. Minato, Y. Morita, J. Uhlir, D. Warin and R. J. Taylor, *Prog. Nucl. Energy*, 2019, 117, 103091.
- NDA, *The Magnox Operating Programme (MOP9)*, available at: https://assets.publishing.service.gov.uk/government/uploads/system/uploads/attachment_data/file/457808/The_Magnox_Operating_Programme_MOP9.pdf, date accessed: 19/01/2022, Magnox Ltd, Sellafield Ltd, Dounreay Ltd, 2012.
- OECD, *Nuclear Energy Data*, available at: https://www.oecd-nea.org/jcms/pl_15082, date accessed: 19/01/2022, Boulogne-Billancourt, France, 2018.



- 15 P. Podvig and The International Panel on Fissile Materials, *Consolidating Fissile Materials in Russia's Nuclear Complex*, available at: <https://fissilematerials.org/library/rr07.pdf>, Princeton, 2009.
- 16 P. D. Wilson, *The nuclear fuel cycle from ore to wastes*, Oxford University Press, Oxford, 1996.
- 17 K. L. Murty and I. Charit, *An introduction to nuclear materials: fundamentals and applications*, John Wiley & Sons, Weinheim, 2013.
- 18 F. Kawakami, M. Tokiwai and Y. Fujii, *Prog. Nucl. Energy*, 2011, **53**, 974–979.
- 19 B. Zohuri, in *Nuclear Reactor Technology Development and Utilization*, ed. S. U.-D. Khan and A. Nakhobov, Woodhead Publishing, 2020, pp. 213–246.
- 20 OECD, *Technology Roadmap Update for Generation IV Nuclear Energy Systems*, 2014, available at: <https://www.gen-4.org/gif/upload/docs/application/pdf/2014-03/gif-tru2014.pdf>, date accessed: 19/01/2022.
- 21 J. Uhlir, *J. Nucl. Mater.*, 2007, **360**, 6–11.
- 22 C. Le Brun, *J. Nucl. Mater.*, 2007, **360**, 1–5.
- 23 G. F. Hewitt and J. G. Collier, *Introduction to nuclear power*, CRC Press, 2nd edn, 2000.
- 24 E.-Y. Choi and S. M. Jeong, *Prog. Nat. Sci.: Mater. Int.*, 2015, **25**, 572–582.
- 25 U. Basak, S. Bourq and A. Constantin *et al.*, *IAEA Tecdoc Series – Status and Trends in Pyroprocessing of Spent Nuclear Fuels*, Vienna, 2015, available at: <https://wwwpub.iaea.org/MTCD/Publications/PDF/TE-1967web.pdf>, date accessed: 07/02/2023.
- 26 M. Iizuka, K. Uozumi, T. Inoue, T. Iwai, O. Shirai and Y. Arai, *J. Nucl. Mater.*, 2001, **299**, 32–42.
- 27 E. González-Roubaud, D. Pérez-Osorio and C. Prieto, *Renewable Sustainable Energy Rev.*, 2017, **80**, 133–148.
- 28 R. W. Bradshaw and D. E. Meeker, *Sol. Energy Mater.*, 1990, **21**, 51–60.
- 29 T. Bauer, N. Pflieger, D. Laing, W.-D. Steinmann, M. Eck and S. Kaesche, High-Temperature Molten Salts for Solar Power Application, in *Molten Salts Chemistry: From Lab to Applications*, Elsevier, 2013, pp. 415–438.
- 30 Q. Guo Zhao, S. Yu Miao, H. Guo and Y. Ting Wu, *Appl. Therm. Eng.*, 2020, **171**, 115084.
- 31 B. Zhang, H. Xie, B. Lu, X. Chen, P. Xing, J. Qu, Q. Song and H. Yin, *ACS Sustainable Chem. Eng.*, 2019, **7**, 13391–13399.
- 32 A. M. Bonanos, M. C. Georgiou, K. G. Stokos and C. N. Papanicolas, *Appl. Therm. Eng.*, 2019, **154**, 294–301.
- 33 B. Mishra and D. L. Olson, *J. Phys. Chem. Solids*, 2005, **66**, 396–401.
- 34 G. Chen, D. Fray and T. Farthing, *Nature*, 2000, **407**, 361–364.
- 35 B. Claux, J. Serp and J. Fouletier, *Electrochim. Acta*, 2011, **56**, 2771–2780.
- 36 D. Hu, A. Dolganov, M. Ma, B. Bhattacharya, M. T. Bishop and G. Z. Chen, *JOM*, 2018, **70**, 129–137.
- 37 D. J. Fray, *JOM*, 2001, **53**, 27–31.
- 38 A. V. Bychkov and O. V. Skiba, in *Chemical Separation Technologies and Related Methods of Nuclear Waste Management*, ed. G. R. Choppin and M. K. Khankhasayev, Springer, Netherlands, Dordrecht, 1999, pp. 71–98.
- 39 S. Delpech, in *Molten Salts Chemistry From Lab to Applications*, ed. F. Lantelme and H. B. T. Groult, Elsevier, Oxford, 2013, pp. 497–520.
- 40 G. J. Janz, R. P. T. Tomkins, C. B. Allen, J. R. Downey Jr, G. L. Garner, U. Krebs and S. K. Singer, *J. Phys. Chem. Ref. Data*, 1975, **4**, 871–1178.
- 41 E. A. Ukshe, *Russ. Chem. Rev.*, 1965, **34**, 141.
- 42 G. J. Janz, G. L. Gardner, U. Krebs and R. P. T. Tomkins, *J. Phys. Chem. Ref. Data*, 1974, **3**, 1–115.
- 43 D. J. Fray and G. Z. Chen, *Mater. Sci. Technol.*, 2004, **20**, 295–300.
- 44 K. Ono and R. O. Suzuki, *JOM*, 2002, **54**, 59–61.
- 45 W. Xiao and D. Wang, *Chem. Soc. Rev.*, 2014, **43**, 3215–3228.
- 46 Y. Sakamura, *J. Nucl. Mater.*, 2011, **412**, 177–183.
- 47 T. Koyama, in *Advanced Separation Techniques for Nuclear Fuel Reprocessing and Radioactive Waste Treatment*, ed. K. L. Nash and G. J. Lumetta, Woodhead Publishing, 2011, pp. 269–310.
- 48 S. M. Jeong, H. S. Shin, S. H. Cho, J. M. Hur and H. S. Lee, *Electrochim. Acta*, 2009, **54**, 6335–6340.
- 49 E.-Y. Choi, J. W. Lee, J. J. Park, J.-M. Hur, J.-K. Kim, K. Y. Jung and S. M. Jeong, *Chem. Eng. J.*, 2012, **207**, 514–520.
- 50 E.-Y. Choi, J.-K. Kim, H.-S. Im, I.-K. Choi, S.-H. Na, J. W. Lee, S. M. Jeong and J.-M. Hur, *J. Nucl. Mater.*, 2013, **437**, 178–187.
- 51 J.-M. Hur, S.-S. Hong and H. Lee, *J. Radioanal. Nucl. Chem.*, 2013, **295**, 851–854.
- 52 S. M. Jeong, S.-B. Park, S.-S. Hong, C.-S. Seo and S.-W. Park, *J. Radioanal. Nucl. Chem.*, 2006, **268**, 349–356.
- 53 A. M. Abdelkader, K. T. Kilby, A. Cox and D. J. Fray, *Chem. Rev.*, 2013, **113**, 2863–2886.
- 54 J. L. Willit, W. E. Miller and J. E. Battles, *J. Nucl. Mater.*, 1992, **195**, 229–249.
- 55 Y. Sakamura and T. Omori, *Nucl. Technol.*, 2010, **171**, 266–275.
- 56 K.-C. Song, H.-S. Lee, J.-M. Hur, J.-G. Kim, D.-H. Ahn and Y.-Z. Cho, *Nucl. Eng. Technol.*, 2010, **42**, 131–144.
- 57 Z. Tomczuk, J. P. Ackerman, R. D. Wolson and W. E. Miller, *J. Electrochem. Soc.*, 1992, **139**, 3523.
- 58 J. J. Laidler, J. E. Battles, W. E. Miller, J. P. Ackerman and E. L. Carls, *Prog. Nucl. Energy*, 1997, **31**, 131–140.
- 59 T. Koyama, M. Iizuka, Y. Shoji, R. Fujita, H. Tanaka, T. Kobayashi and M. Tokiwai, *J. Nucl. Sci. Technol.*, 1997, **34**, 384–393.
- 60 M. Iizuka, T. Koyama, N. Kondo, R. Fujita and H. Tanaka, *J. Nucl. Mater.*, 1997, **247**, 183–190.
- 61 T. Koyama, M. Iizuka, N. Kondo, R. Fujita and H. Tanaka, *J. Nucl. Mater.*, 1997, **247**, 227–231.
- 62 Y. Sakamura, T. Hijikata, K. Kinoshita, T. Inoue, T. S. Storvick, C. L. Krueger, L. F. Grantham, S. P. Fusselman, D. L. Grimmer and J. J. Roy, *J. Nucl. Sci. Technol.*, 1998, **35**, 49–59.
- 63 J. J. Roy, *J. Electrochem. Soc.*, 1996, **143**, 2487.



- 64 D. Inman, G. J. Hills, L. Young and J. Bockris, *Trans. Faraday Soc.*, 1959, **55**, 1904–1914.
- 65 S. A. Kuznetsov, H. Hayashi, K. Minato and M. Gaune-Escard, *J. Electrochem. Soc.*, 2005, **152**, C203.
- 66 K. Dring, R. Dashwood and D. Inman, *J. Electrochem. Soc.*, 2005, **152**, D184.
- 67 R. Littlewood, *J. Electrochem. Soc.*, 1962, **109**, 525.
- 68 L. D. Brown, R. Abdulaziz, S. Simons, D. Inman, D. J. L. Brett and P. R. Shearing, *J. Appl. Electrochem.*, 2013, **43**, 1235–1241.
- 69 J. J. Roy, L. F. Grantham, D. L. Grimmett, S. P. Fusselman, C. L. Krueger, T. S. Storvick, T. Inoue, Y. Sakamura and N. Takahashi, *J. Electrochem. Soc.*, 1996, **143**, 2487.
- 70 *Encyclopedia of Electrochemistry of the Elements, Fused Salt Systems*, ed. J. A. Plambeck and A. J. Bard, Marcel Dekker, Inc., 1976, vol. 10.
- 71 H. Lee, G.-I. Park, J.-W. Lee, K.-H. Kang, J.-M. Hur, J.-G. Kim, S. Paek, I.-T. Kim and I.-J. Cho, *Sci. Technol. Nuclear Install.*, 2013, **2013**, 1–11.
- 72 R. G. Lewin and M. T. Harrison, *Woodhead Publishing Series in Energy*, Woodhead Publishing, Oxford, 2015, pp. 373–413.
- 73 O. Shirai, H. Yamana and Y. Arai, *J. Alloys Compd.*, 2006, **408**, 1267–1273.
- 74 Y. Sakamura, T. Hijikata, K. Kinoshita, T. Inoue, T. S. Storvick, C. L. Krueger, J. J. Roy, D. L. Grimmett, S. P. Fusselman and R. L. Gay, *J. Alloys Compd.*, 1998, **271**, 592–596.
- 75 Nuclear Energy Agency, Spent Nuclear Fuel Assay Data for Isotopic Validation, State-of-the-art Report, 2011.
- 76 I. Grenthe, J. Drożdżynski, T. Fujino, E. C. Buck, T. E. Albrecht-Schmitt and S. F. Wolf, in *The Chemistry of the Actinide and Transactinide Elements*, ed. L. R. Morss, N. M. Edelstein and J. Fuger, Springer, Netherlands, Dordrecht, 2006, pp. 253–698.
- 77 M. Genet, *Radiochim. Acta*, 1995, **70–71**, 3–12.
- 78 O. Hahn and F. Strassmann, *Naturwissenschaften*, 1939, **27**, 11–15.
- 79 H. L. Anderson, E. T. Booth, J. R. Dunning, E. Fermi, G. N. Glasoe and F. G. Slack, *Phys. Rev.*, 1939, **55**, 511–512.
- 80 E. Péligot, *J. Prakt. Chem.*, 1841, **24**, 442–451.
- 81 D. Inman, G. J. Hills, L. Young and J. O. Bockris, *Ann. N. Y. Acad. Sci.*, 1960, **79**, 803–829.
- 82 S. A. Kuznetsov, H. Hayashi, K. Minato and M. Gaune-Escard, *Electrochim. Acta*, 2006, **51**, 2463–2470.
- 83 L. Martinot and F. Caligara, *At. Energy Rev.*, 1973, **11**, 3–61.
- 84 S. A. Kuznetsov, L. Rycerz and M. Gaune-Escard, *J. Nucl. Mater.*, 2005, **344**, 152–157.
- 85 O. Shirai, K. Uozumi, T. Iwai and Y. Arai, *Anal. Sci.*, 2001, **17**, i959–i962.
- 86 P. Masset, D. Bottomley, R. Konings, R. Malmbeck, A. Rodrigues, J. Serp and J.-P. Glatz, *J. Electrochem. Soc.*, 2005, **152**, A1109.
- 87 S. D. Herrmann, S. X. Li, M. F. Simpson and S. Phongikaroon, *Sep. Sci. Technol.*, 2006, **41**, 1965–1983.
- 88 E.-Y. Choi, J.-M. Hur, I.-K. Choi, S. G. Kwon, D.-S. Kang, S. S. Hong, H.-S. Shin, M. A. Yoo and S. M. Jeong, *J. Nucl. Mater.*, 2011, **418**, 87–92.
- 89 C. S. Seo, S. Bin Park, B. H. Park, K. J. Jung, S. W. Park and S. H. Kim, *J. Nucl. Sci. Technol.*, 2006, **43**, 587–595.
- 90 G. Z. Chen, E. Gordo and D. J. Fray, *Metall. Mater. Trans. B*, 2004, **35**, 223–233.
- 91 Y. Deng, D. Wang, W. Xiao, X. Jin, X. Hu and G. Z. Chen, *J. Phys. Chem. B*, 2005, **109**, 14043–14051.
- 92 P. Kar and J. W. Evans, *Electrochim. Acta*, 2008, **54**, 835–843.
- 93 Y. Sakamura, M. Kurata and T. Inoue, *J. Electrochem. Soc.*, 2006, **153**, D31.
- 94 C. Nourry, P. Souček, L. Massot, R. Malmbeck, P. Chamelot and J.-P. Glatz, *J. Nucl. Mater.*, 2012, **430**, 58–63.
- 95 C. Hamel, P. Chamelot, A. Laplace, E. Walle, O. Dugne and P. Taxil, *Electrochim. Acta*, 2007, **52**, 3995–4003.
- 96 G. Mamantov and D. L. Manning, *Anal. Chem.*, 1966, **38**, 1494–1498.
- 97 M. Kolodney, *Production of plutonium by electrolysis*, Report LA-148, Los Alamos Scientific Laboratory, U.S. Atomic Energy Commission, 1944.
- 98 R. D. Baker and W. J. Maraman, in *Extractive and Physical Metallurgy of Plutonium and Its Alloys*, ed. W. D. Wilkinson, Interscience Publishers, New York, 1960, pp. 43–59.
- 99 B. Blumenthal and M. B. Brodsky, in *Plutonium*, ed. E. Grison, W. B. H. Lord and R. D. Fowler, London, 1960.
- 100 S. M. Jeong, H.-S. Shin, S.-S. Hong, J.-M. Hur, J. B. Do and H. S. Lee, *Electrochim. Acta*, 2010, **55**, 1749–1755.
- 101 L. J. Mullins, J. A. Leary and C. W. Bjorklund, *Large scale preparation of high purity plutonium metal by electrorefining. Preliminary Report*, Los Alamos Scientific Lab., N. Mex., LAMS-2441, 1960.
- 102 J. A. Leary and L. J. Mullins, *Preparation of Ultra-High Purity Plutonium*, Los Alamos Scientific Lab., Univ. of California, N. Mex., 1965.
- 103 L. J. Mullins and A. N. Morgan, *Large scale electrorefining of plutonium from plutonium-iron alloys*, Los Alamos Scientific Laboratory of the University of California, New Mexico, 1964, vol. 3029.
- 104 T. Paget, J. Mcneese, K. Fife, M. Jackson and R. Watson, in *Plutonium Handbook*, ed. D. Clark, D. A. Geeson and R. J. Hanrahan Jr, La Grange Park, IL, 2nd edn, 2019, pp. 201–286.
- 105 T. J. Paget, R. Watson and S. Slade, *Plutonium Futures*, Dijon, France, 2008.
- 106 T. J. Paget, S. M. Slade and R. F. Watson, *42nd IUPAC Congress*, Glasgow, Scotland, 2009.
- 107 D. F. Mclaughlin, C. E. Sessions and J. E. Marra, *Nucl. Technol.*, 1992, **99**, 242–251.
- 108 T. S. Soine, *Batch electrorefining of plutonium metal. Progress Report*, General Electric Co., Richland, Wash. Hanford Atomic Products Operation, 1965.
- 109 M. H. Curtis, *Continuous Electrowinning of Plutonium from Chloride Melts*, General Electric Co. Hanford Atomic Products Operation, Richland, Wash., p. 1963.
- 110 M. B. Brodsky and B. G. F. Carleson, *J. Inorg. Nucl. Chem.*, 1962, **24**, 1675–1681.
- 111 L. J. Mullins and A. N. Morgan, *Review of operating experience at the Los Alamos Plutonium Electrorefining Facility, 1963–1977*, Los Alamos National Lab., NM (USA), 1981.



- 112 L. J. Mullins, A. N. Morgan, S. A. Apgar III and D. C. Christensen, *Six-kilogram-scale electrorefining of plutonium metal*, Los Alamos National Lab., 1982.
- 113 D. S. Poa, Z. Tomczuk and R. K. Steunenbergh, *J. Electrochem. Soc.*, 1988, **135**, 1161.
- 114 O. Shirai, T. Iwai, Y. Suzuki, Y. Sakamura and H. Tanaka, *J. Alloys Compd.*, 1998, **271–273**, 685–688.
- 115 D. A. Nissen, *J. Inorg. Nucl. Chem.*, 1966, **28**, 1740–1743.
- 116 J. Serp, R. J. M. Konings, R. Malmbeck, J. Rebizant, C. Scheppeler and J.-P. Glatz, *J. Electroanal. Chem.*, 2004, **561**, 143–148.
- 117 L. Martinot and G. Duyckaerts, *Anal. Lett.*, 1971, **4**, 1–11.
- 118 G. M. Campbell and J. A. Leary, *Thermodynamic Properties of Pu Compounds from EMF Measurements: I. Pu Vs. Ag in LiCl-KCl Eutectic*, Technical Report LA-3399, United States Atomic Energy Commission, 1966.
- 119 L. Martinot, *J. Inorg. Nucl. Chem.*, 1975, **37**, 2525–2528.
- 120 P. Masset, R. J. M. Konings, R. Malmbeck, J. Serp and J. P. Glatz, *J. Nucl. Mater.*, 2005, **344**, 173–179.
- 121 Y. Sakamura, O. Shirai, T. Iwai and Y. Suzuki, *J. Alloys Compd.*, 2001, **321**, 76–83.
- 122 O. Shirai, M. Iizuka, T. Iwai and Y. Arai, *Anal. Sci.*, 2001, **17**, 51–57.
- 123 L. E. McCurry and G.-M. M. Moy, *Trans Am. Nuc. Soc.*, Los Alamos National Lab., Los Angeles, CA, US, 1987, vol. 55, p. 248.
- 124 R. Abdulaziz, L. D. Brown, D. Inman, S. Simons, P. R. Shearing and D. J. L. Brett, *Int. J. Electrochem. Sci.*, 2016, **11**, 10417–10435.
- 125 D. Lambertin, S. Ched homme, G. Bourges, L. Pescayre, S. Sanchez and G. Picard, *ECS Proc. Vol.*, 2004, **24**, 700.
- 126 A. H. Jones, R. Watson, T. Paget, R. Campbell-Kelly, T. Caldwell and D. J. Fray, *J. Nuclear Fuel Cycle Waste Technology*, 2015, **13**, 1–5.
- 127 M. Iizuka, Y. Sakamura and T. Inoue, *J. Nucl. Mater.*, 2006, **359**, 102–113.
- 128 M. Kurata, T. Inoue, J. Serp, M. Ougier and J.-P. Glatz, *J. Nucl. Mater.*, 2004, **328**, 97–102.
- 129 D. Rappleye, J. McNeese, R. Torres, K. Holliday and J. R. Jeffries, *J. Nucl. Mater.*, 2021, **552**, 152968.
- 130 J. G. Reavis, *Experimental studies of actinides in molten salts*, Los Alamos National Lab., LA-10340, New Mexico, 1985.
- 131 E. McMillan and P. H. Abelson, *Phys. Rev.*, 1940, **57**, 1185–1186.
- 132 G. T. Seaborg, E. M. Mcmillan, J. W. Kennedy and A. C. Wahl, *Phys. Rev.*, 1946, **69**, 366–367.
- 133 G. A. Burney and R. M. Harbour, *Radiochemistry of Neptunium: Nuclear Science Series*, NAS-NS-3060, Savannah River Laboratory, E. I. du Pont de Nemours & Co., South Carolina, 1974.
- 134 P. Masset, C. Apostolidis, R. J. M. Konings, R. Malmbeck, J. Rebizant, J. Serp and J.-P. Glatz, *J. Electroanal. Chem.*, 2007, **603**, 166–174.
- 135 O. Shirai, M. Iizuka, T. Iwai and Y. Arai, *J. Appl. Electrochem.*, 2001, **31**, 1055–1060.
- 136 L. Martinot and G. Duyckaerts, *Inorg. Nuclear Chem. Lett.*, 1969, **5**, 909–919.
- 137 C. L. Krueger, *J. Electrochem. Soc.*, 1991, **138**, 1186.
- 138 S. P. Fusselman, J. J. Roy, D. L. Grimmett, L. F. Grantham, C. L. Krueger, C. R. Nabelek, T. S. Storvick, T. Inoue, T. Hijikata, K. Kinoshita, Y. Sakamura, K. Uozumi, T. Kawai and N. Takahashi, *J. Electrochem. Soc.*, 1999, **146**, 2573–2580.
- 139 A. Cassol, L. Magon, G. Tomat and R. Portanova, *Inorg. Chem.*, 1972, **11**, 515–519.
- 140 R. M. Rush and J. S. Johnson, *J. Phys. Chem.*, 1963, **67**, 821–825.
- 141 A. Cassol, L. Magon, R. Portanova and E. Tondello, *Radiochim. Acta*, 1972, **17**, 28–32.
- 142 R. Lysy and G. Duyckaerts, *Anal. Chim. Acta*, 1978, **96**, 125–132.
- 143 I. B. Polovov, C. A. A. Sharrad, I. May, B. D. D. Vasin, V. A. Volkovich and T. R. R. Griffiths, *ECS Trans.*, 2007, **3**, 503–511.
- 144 L. Martinot, *J. Less Common Metals*, 1991, **170**, 121–126.
- 145 G. T. Seaborg, R. A. James and L. O. Morgan, *Modern Alchemy*, 1994, pp. 24–52.
- 146 B. Still, *Nat. Chem.*, 2017, **9**, 296.
- 147 J. Serp, P. Chamelot, S. Fourcaudot, R. J. M. Konings, R. Malmbeck, C. Pernel, J. C. Poignet, J. Rebizant and J. P. Glatz, *Electrochim. Acta*, 2006, **51**, 4024–4032.
- 148 D. Lambertin, J. Lacquement, S. Sanchez and G. S. Picard, *Plasmas Ions*, 2000, **3**, 65–72.
- 149 A. Laplace, J. Lacquement, C. Maillard and L. Donnet, *Proc. Atalante 2004: Advances for Future Nuclear Fuel Cycles*, Nimes, France, 2004.
- 150 H. Boussier, J. P. Glatz, R. J. M. Konings, R. Malmbeck, C. Pernel, J. Rebizant, C. Scheppeler and J. Serp, PYRO Project Status, 2003, FIKW-CT-2000-00049.
- 151 G. T. Seaborg, R. A. James and A. Ghiorso, *The New Element Curium (Atomic Number 96)*, AECD-2182, United States, 1948.
- 152 H. J. Groh, R. T. Huntoon, C. S. Schlea, J. A. Smith and F. H. Springer, *Nuclear Appl.*, 1965, **1**, 327–336.
- 153 G. J. Lumetta, M. C. Thompson, R. A. Penneman and P. G. Eller, in *The Chemistry of the Actinide and Transactinide Elements*, ed. L. R. Morss, N. M. Edelstein and J. Fuger, Springer, Netherlands, Dordrecht, 2008, pp. 1397–1443.
- 154 Y. V. Lobanov, G. V. Buklanov, F. S. Abdullin, A. N. Polyakov, I. V. Shirokovsky, Y. S. Tsyganov and V. K. Utyonkov, *Nucl. Instrum. Methods Phys. Res., Sect. A*, 1997, **397**, 26–29.
- 155 R. A. Penneman and T. K. Keenan, *The Radiochemistry of Americium and Curium*, NAS-NS-3006, Washington DC, 1960.
- 156 A. Osipenko, A. Maershin, V. Smolenski, A. Novoselova, M. Kormilitsyn and A. Bychkov, *J. Electroanal. Chem.*, 2011, **651**, 67–71.
- 157 T. Koyama, T. R. Johnson and D. F. Fischer, *J. Alloys Compd.*, 1992, **189**, 37–44.
- 158 V. P. Kolesnikov, G. N. Kazantsev and O. V. Skiba, *Radiochim. Acta*, 1977, **19**, 545–548.
- 159 A. Osipenko, A. Maershin, V. Smolenski, A. Novoselova, M. Kormilitsyn and A. Bychkov, *J. Nucl. Mater.*, 2010, **396**, 102–106.



- 160 A. J. Bard and L. R. Faulkner, *Electrochem. Methods*, 2001, **2**, 580–632.
- 161 A. Osipenko, A. Mayershin, V. Smolenski, A. Novoselova and M. Kormilitsyn, in *Recent Trends in Electrochemical Science and Technology*, ed. U. Kumar, Sur, IntechOpen, 2012.
- 162 T. T. Tsong, *Prog. Surf. Sci.*, 2001, **67**, 235–248.
- 163 A. Ugale, T. N. Kalyani and S. J. Dhoble, in *Lanthanide-Based Multifunctional Materials*, ed. P. Martín-Ramos and M. B. T.-L.-B. M. M. Ramos Silva, Elsevier, 2018, pp. 59–97.
- 164 H. N. McCoy, *J. Am. Chem. Soc.*, 1936, **58**, 2279–2281.
- 165 T.-J. Kim, Y.-J. Jung, S.-H. Kim, S.-W. Paek and D.-H. Ahn, *Bull. Korean Chem. Soc.*, 2011, **32**, 863–866.
- 166 S. Bratsch and H. B. Silber, *Polyhedron*, 1982, **1**, 219–223.
- 167 K. E. Johnson and J. R. Mackenzie, *J. Electrochem. Soc.*, 1969, **116**, 1697.
- 168 J. P. Schobrechts, B. P. Gilbert and G. Duyckaerts, *J. Electroanal. Chem. Interfacial Electrochem.*, 1983, **145**, 139–146.
- 169 S. A. Kuznetsov and M. Gaune-Escard, *Electrochim. Acta*, 2001, **46**, 1101–1111.
- 170 S. A. Kuznetsov and M. Gaune-Escard, *J. Electroanal. Chem.*, 2006, **595**, 11–22.
- 171 M. Gibilaro, L. Massot, P. Chamelot, L. Cassayre and P. Taxil, *Electrochim. Acta*, 2009, **55**, 281–287.
- 172 C. Nourry, L. Massot, P. Chamelot and P. Taxil, *J. Appl. Electrochem.*, 2009, **39**, 927–933.
- 173 S. E. Bae, Y. J. Park, S. K. Min, Y. H. Cho and K. Song, *Electrochim. Acta*, 2010, **55**, 3022–3025.
- 174 M. Gibilaro, L. Massot, P. Chamelot and P. Taxil, *J. Nucl. Mater.*, 2008, **382**, 39–45.
- 175 P. Taxil, P. Chamelot, L. Massot and C. Hamel, *J. Min. Metall., Sect. B*, 2003, **39**, 177–200.
- 176 C. A. Schroll, S. Chatterjee, T. G. Levitskaia, W. R. Heineman and S. A. Bryan, *Anal. Chem.*, 2013, **85**, 9924–9931.
- 177 W. Huang, L. Tian, C. She, F. Jiang, H. Zheng, W. Li, G. Wu, D. Long and Q. Li, *Electrochim. Acta*, 2014, **147**, 114–120.
- 178 M. R. Bermejo, F. de la Rosa, E. Barrado and Y. Castrillejo, *J. Electroanal. Chem.*, 2007, **603**, 81–95.
- 179 L. Massot, P. Chamelot, L. Cassayre and P. Taxil, *Electrochim. Acta*, 2009, **54**, 6361–6366.
- 180 C. Caravaca, G. Córdoba and M. J. Tomás, in 8th Information Exchange Meeting on P&T, Las Vegas, 2004.
- 181 S. A. Kuznetsov, L. Rycerz and M. Gaune-Escard, *J. New Mater. Electrochem. Syst.*, 2006, **9**, 313.
- 182 M. S. Wickleder, B. Fourest and P. K. Dorhout, in *The Chemistry of the Actinide and Transactinide Elements*, ed. L. R. Morss, N. M. Edelstein and J. Fuger, Springer, Netherlands, Dordrecht, 2008, pp. 52–160.
- 183 I. Baghni, Y.-S. Wu, J.-Q. Li, C. Du and W. Zhang, *Trans. Nonferrous Met. Soc. China*, 2003, **13**, 1253–1259.
- 184 T. E. Leontis, *JOM*, 1952, **4**, 633–642.
- 185 H. O. Poppleton and F. E. Block, *A Process for the Preparation of Thorium by Sodium Reduction of Thorium Tetrachloride*, US Department of the Interior, Bureau of Mines, 1964, vol. 6527.
- 186 L. Martinot, *J. Less-Common Met.*, 1989, **147**, 73–77.
- 187 L. Martinot, *Handbook on the Physics and Chemistry of the Actinides*, Elsevier Science Publishers, 1991.
- 188 L. Cassayre, J. Serp, P. Soucek, R. Malmbeck, J. Rebizant and J. P. Glatz, *Electrochim. Acta*, 2007, **52**, 7432–7437.
- 189 T. M. Klapötke and A. Schulz, *Polyhedron*, 1997, **16**, 989–991.
- 190 G. Ionova, C. Madic and R. Guillaumont, *Polyhedron*, 1998, **17**, 1991–1995.
- 191 R. Srinivasan and S. N. Flengas, *Can. J. Chem.*, 1964, **42**, 1315–1322.
- 192 L. Martinot, J. Bohet, G. Duyckaerts and W. Muller, *Inorg. Nuclear Chem. Lett.*, 1977, **13**, 315–319.
- 193 L. H. Meyer, *J. Electrochem. Soc.*, 1960, **107**, 43.
- 194 F. R. Clayton, G. Mamantov and D. L. Manning, *J. Electrochem. Soc.*, 1974, **121**, 86.
- 195 R. P. Elliott, *Constitution of binary alloys: first supplement*, McGraw-Hill, New York, 1965.
- 196 L. Martinot and J. Fuger, *J. Less-Common Met.*, 1986, **120**, 255–266.
- 197 I. Johnson, *J. Nucl. Mater.*, 1988, **154**, 169–180.
- 198 T. Kobayashi and M. Tokiwai, *J. Alloys Compd.*, 1993, **197**, 7–16.
- 199 T. Kobayashi, R. Fujita, M. Fujie and T. Koyama, *J. Nucl. Sci. Technol.*, 1995, **32**, 653–663.
- 200 T. Kobayashi, R. Fujita, H. Nakamura and T. Koyama, *J. Nucl. Sci. Technol.*, 1997, **34**, 50–57.
- 201 J. H. Lee, Y. H. Kang, S. C. Hwang, H. S. Lee, E. H. Kim and S. W. Park, *Nucl. Technol.*, 2008, **162**, 107–116.
- 202 S. H. Kim, S. Bin Park, S. J. Lee, J. G. Kim, H. S. Lee and J. H. Lee, *Nucl. Eng. Des.*, 2013, **257**, 12–20.
- 203 S. Ghosh, B. Prabhakara Reddy, K. Nagarajan and P. R. Vasudeva Rao, *Nucl. Technol.*, 2010, **170**, 430–443.
- 204 S. Ghosh, S. Vandarkuzhali, N. Gogoi, P. Venkatesh, G. Seenivasan, B. P. Reddy and K. Nagarajan, *Electrochim. Acta*, 2011, **56**, 8204–8218.
- 205 R. O. Hoover, S. Phongikaroon, M. F. Simpson, S. X. Li and T.-S. Yoo, *Nucl. Technol.*, 2010, **171**, 276–284.
- 206 R. K. Ahluwalia and T. Q. Hua, *Nucl. Technol.*, 2002, **140**, 41–50.
- 207 R. K. Ahluwalia, T. Q. Hua and H. K. Geyer, *Nucl. Technol.*, 2001, **133**, 103–118.
- 208 R. Ahluwalia, T. Q. Hua and H. K. Geyer, *Nucl. Technol.*, 1999, **126**, 289–302.
- 209 B. G. Park and I. S. Hwang, *Proc. of the Korean Nuclear Society autumn meeting*, Seoul, Republic of Korea, 1999, p. 293.
- 210 J. Zhang, *Prog. Nucl. Energy*, 2014, **70**, 279–286.
- 211 R. M. Cumberland and M.-S. Yim, *Ann. Nucl. Energy*, 2014, **71**, 52–59.
- 212 R. M. Cumberland and M.-S. Yim, *J. Electrochem. Soc.*, 2014, **161**, D147.
- 213 J. O. Bockris, B. E. Conway, E. Yeager and R. E. White, *Comprehensive treatise of electrochemistry, Electrochemical Processing*, Plenum Press, New York, 1st edn, 1981, vol. 2.
- 214 D. S. Maha Vishnu, N. Sanil, L. Shakila, R. Sudha, K. S. Mohandas and K. Nagarajan, *Electrochim. Acta*, 2015, **159**, 124–130.
- 215 N. Xin-miao, D. Ling-yang, B. Chen-guang and C. Deng-fu, *Trans. Nonferrous Met. Soc. China*, 2006, **16**, s723–s727.



- 216 H. S. Shin, J. M. Hur, S. M. Jeong and K. Y. Jung, *J. Ind. Eng. Chem.*, 2012, **18**, 438–442.
- 217 R. Barnett, K. T. Kilby and D. J. Fray, *Metall. Mater. Trans. B*, 2009, **40**, 150–157.
- 218 T. Wu, X. Jin, W. Xiao, X. Hu, D. Wang and G. Z. Chen, *Chem. Mater.*, 2007, **19**, 153–160.
- 219 C. Schwandt and D. J. Fray, *Electrochim. Acta*, 2005, **51**, 66–76.
- 220 G. Qiu, M. Ma, D. Wang, X. Jin, X. Hu and G. Z. Chen, *J. Electrochem. Soc.*, 2005, **152**, E328.
- 221 D. Wang, G. Qiu, X. Jin, X. Hu and G. Z. Chen, *Angew. Chem., Int. Ed.*, 2006, **45**, 2384–2388.
- 222 G. Qiu, K. Jiang, M. Ma, D. Wang, X. Jin and G. Z. Chen, *Zeitschrift für Naturforschung A*, 2007, **62**, 292–302.
- 223 G. Li, D. Wang, X. Jin and G. Z. Chen, *Electrochem. Commun.*, 2007, **9**, 1951–1957.
- 224 G. Qiu, X. Feng, M. Liu, W. Tan and F. Liu, *Electrochim. Acta*, 2008, **53**, 4074–4081.
- 225 K. Rao, D. Brett, D. Inman and R. Dashwood, *Molten Salt Discussion Group*, London, 2008.
- 226 L. D. Brown, PhD thesis, University College London, 2015.
- 227 J. Peng, Y. Deng, D. Wang, X. Jin and G. Z. Chen, *J. Electroanal. Chem.*, 2009, **627**, 28–40.
- 228 T. Berent, I. Fells and R. Mason, *Nature*, 1969, **223**, 1054–1055.
- 229 C. Oloman and A. P. Watkinson, *J. Appl. Electrochem.*, 1979, **9**, 117–123.
- 230 Y. Xiong and H. T. Karlsson, *Adv. Environ. Res.*, 2002, **7**, 139–145.
- 231 S. Germain and F. Goodridge, *Electrochim. Acta*, 1976, **21**, 545–550.
- 232 F. Goodridge, C. J. H. King and A. R. Wright, *Electrochim. Acta*, 1977, **22**, 1087–1091.
- 233 M. Kalaruban, P. Loganathan, W. G. Shim, J. Kandasamy, G. Naidu, T. V. Nguyen and S. Vigneswaran, *Sep. Purif. Technol.*, 2016, **158**, 62–70.
- 234 M. J. Nelson, G. Nakhla and J. Zhu, *Engineering*, 2017, **3**, 330–342.
- 235 D. R. Chadeesingh and A. N. Hayhurst, *Fuel*, 2014, **127**, 169–177.
- 236 A. N. Hayhurst, *Combust. Flame*, 1991, **85**, 155–168.
- 237 D. Hu, X. Zeng, F. Wang, M. Haruna Adamu and G. Xu, *Fuel*, 2021, **290**, 120038.
- 238 P. Chaiwang, H. Chitcharoenyoo, P. Piumsomboon and B. Chalermssinsuwan, *Particuology*, 2022, **67**, 47–56.
- 239 R. Abdulaziz, L. Brown, D. Inman, C. A. Sharrad, A. Jones, P. R. Shearing and D. J. L. Brett, *J. Electrochem. Soc.*, 2017, **164**, H5280–H5285.
- 240 R. Abdulaziz, L. D. Brown, D. Inman, P. R. Shearing and D. J. L. Brett, *Electrochim. Acta*, 2017, **226**, 18–28.
- 241 M. Mirza, R. Abdulaziz, W. C. Maskell, C. Tan, P. R. Shearing and D. J. L. Brett, *Electrochim. Acta*, 2021, **391**, 138846.
- 242 S. Jiao, H. Jiao, W. Song, M. Wang and J. Tu, *Int. J. Miner., Metall. Mater.*, 2020, **27**, 1588–1598.
- 243 T. Kato, T. Inoue, T. Iwai and Y. Arai, *J. Nucl. Mater.*, 2006, **357**, 105–114.
- 244 K. Kinoshita, T. Inoue, S. P. Fusselman, D. L. Grimmer, J. J. Roy, R. L. Gay, C. L. Krueger, C. R. Nabelek and T. S. Storvick, *J. Nucl. Sci. Technol.*, 1999, **36**, 189–197.
- 245 H. Jiao, J. Wang, L. Zhang, K. Zhang and S. Jiao, *RSC Adv.*, 2015, **5**, 62235–62240.
- 246 M. Gibilaro, S. Bolmont, L. Massot, L. Latapie and P. Chamelot, *J. Electroanal. Chem.*, 2014, **726**, 84–90.
- 247 J. P. Ackerman and J. L. Settle, *J. Alloys Compd.*, 1991, **177**, 129–141.
- 248 Y. S. Hwang, M. S. Jeong and S. W. Park, *Prog. Nucl. Energy*, 2007, **49**, 463–472.
- 249 D. Vaden, S. X. Li, B. R. Westphal, K. B. Davies, T. A. Johnson and D. M. Pace, *Nucl. Technol.*, 2008, **162**, 124–128.

