One-pot synthesis of Mo₂C&MoS₂ loaded on N/S co-doped carbon materials as the electrocatalyts for hydrogen evolution reation

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Abstract:

Biomass is a potential precursor for preparing functional carbon materials, as the organic component containing nitrogen (N), sulfur (S) would create the atomic dopants to adjust electronic properties of the carbon material. The heteroatom-doped carbon materials loaded with non-precious metal compound is considered to be a promising alternative for hydrogen evolution reaction (HER). Herein, Mo₂C and MoS₂ nanoparticles with a size of 5~20 nm uniformly loaded on the N/S-co-doped carbon sheets (NSC) are prepared via direct pyrolysis of ginkgo leaves with ammonium

molybdate tetrahydrate (AMT). Carbon black (CB) is used to replace ginkgo leaves to obtain Mo₂C@NC₃ to further study the influence of N/S-containing organic components in ginkgo leaves. By comparing with Mo₂C@NC₃, the XPS reflect the characteristic peak of Mo is positively shifted while the characteristic peak of N 1s is obviously negatively shifted, indicating a stronger electron interactions of Mo₂C&MoS₂@NSC₃. Benefited from the intense electronics transfer, the efficiency electrochemical active areas of Mo₂C&MoS₂@NSC₃ is larger than Mo₂C@NC₃ (10.72 *vs.* 4.97 mF/cm⁻²), thus the Mo₂C&MoS₂@NSC₃ gives a smaller overpotential of 209 mV to obtain the current density of 10 mA cm⁻², together with a low Tafel slope value of 85.5 mV dec⁻¹.

Keywords: Mo₂C&MoS₂, N/S co-doped carbon, electrocatalysts, hydrogen evolution reaction, ginkgo leaves

1 Introduction

The huge energy demand and environmental pollution of human society enforce the exploration for alternatives of traditional energy. Hydrogen gas (H₂) is regarded as an efficient and pollution-free energy carrier, thus the methods for generating H₂ have been world-wide investigated [1-3]. Steam reforming of fossil resources is a commonly industrial method to produce H₂ but will result in the consuming of fossil fuel and emit carbon dioxide (CO₂) [4]. By contrast, electrolysis water splitting is an appealing strategy to obtain H₂ for its easy availability of raw materials and emission-free [5, 6]. Hydrogen evolution reaction (HER) is the cathode reaction during the water splitting. Among various electrocatalysts, platinum-group based catalysts (Pt, Pd, et al.) have the best activity to electrocatalytic HER due to its optimal hydrogen bonding energy [7, 8]. The high-cost and scarce reserves seriously limit the scalable application of noble metals in electrocatalytic H₂ production,

extensive efforts have been devoted to search the substitutes with earth-abundant sources [9-11]. According to the volcano plots, transition metals such as nickel (Ni), cobalt (Co) and molybdenum (Mo) are the idea alternatives for H₂ production [12-18]. The carbides of transition metal possess the Pt-like band structure which endow them to adsorb and activate hydrogen [19]. Molybdenum carbides (Mo₂C) have already been demonstrated similar chemistry peculiarity to Pt-group catalysts in catalyzing hydro-processing reaction [20, 21]. However, the poor conductivity and relative strong of hydrogen adsorption energy restrict the HER performance of Mo₂C [19, 22, 23]. The molybdenum sulfide (MoS₂) possesses moderate hydrogen bonding also suffers the low conductivity which serious restrict its catalytic activity for HER [24]. More than that, the typical lamellar crystal structure results in the violent aggregation which make against the active sites exposure [25, 26]. It is more sufficient to enhance the catalytic activity of Mo₂C and MoS₂ by loading it on a high conductivity matrix [27, 28].

Carbon-based matrix such as carbon nanotube, graphene and carbon fiber have been attracted many attentions for it satisfactory effect on promoting the electrons transfer [29-32]. Embedding the metal compounds on carbon support will benefit to downsize the size of metal particles and promote its uniform dispersion, thus benefit to expose more active sites [19]. The coupling of Mo₂C, MoS₂ and carbon has also been shown to be beneficial for optimizing the Mo-H bond strength [33, 34]. Especially, the carbon matrix doped with heteroatoms have been proven to further boost the HER performance of Mo₂C and MoS₂ by enhancing the catalytic activity of carbon atoms adjacent to doped atoms [19, 35]. While the typical techniques for manufacturing heteroatoms doped carbon are to introduce the exogenous heteroatoms sources (melamine [36], thiourea [37], etc.) into the

carbon matrix. The Mo₂C&MoS₂ loaded carbon matrix is generally prepared via hydrothermal followed post-calcination method which is relative complex and with large energy consumption [33, 38]. Compared with such normal carbon materials, biomass is an abundant renewable source consisted of several organic contents containing the nitrogen (N), sulfur (S) and phosphorus (P) element, which can be introduced into the carbon skeleton as dopants during the biomass conversion [39, 40]. The plenty of surface functional group endow its with huge space to be functionalized and applicated in the electrocatalysis such as HER [41], oxygen evolution reaction (OER) [42-44], carbon dioxide reduction reaction (CO_2RR) [45, 46]. Furthermore, the nonmetal element may combine with metal to form metal compounds to enhance the catalytic activity which is taking full use of the pristine properties of biomass [16, 17].

Ginkgo trees are widely planted in the China as a landscape tree. The ginkgo leaves are rich in N and S element which make it is possible to be converted to N/S doped carbon via simple thermal treatment. Herein, as shown in scheme 1, the ginkgo leaves are chosen as the biomass precursor and mixed with ammonium molybdate tetrahydrate (AMT) by simply grounding. The AMT decomposed into ammonia (NH₃) and molybdenum trioxide (MoO₃) during the pyrolysis. The escape of NH₃ produce the addition N atoms. While the MoO₃ combined with the C and S to form the Mo₂C and MoS₂. The Mo₂C and MoS₂ nanoparticles with a size of 5~30 nm homogeneous dispersed on the N/S co-doped carbon sheets (Mo₂C&MoS₂@NSC). As expect, the Mo₂C and MoS₂ serves as the efficient active sites and significantly enhance the HER performance for ginkgo leaves derived carbon materials. More than that, the activity of Mo₂C&MoS₂@NSC is better than Mo₂C@NC₃ derived from carbon black and AMT, which have reflected the advantage of ginkgo

leaves as the precursor, that the heteroatom originated from ginkgo leaves play a critical role in electrocatalytic HER.



Scheme 1 Schematic illustration of the fabrication process of Mo₂C&MoS₂@NSC_x.

2 Experiment

2.1 Materials

Ammonium molybdate tetrahydrate ((NH₄)₆Mo₇O₂₄·4H₂O, 99.99%, AMT), sodium molybdate (Na₂MoO₄·2H₂O), thiourea (CH₄N₂S), sulfuric acid (H₂SO₄, A.R.), and carbon black (CB) were purchased from Chron Chemicals. The ginkgo leaves used in this work were collected from the street in Nanjing, China. The collected leaves were firstly washed with ultrapure water for at least three times and dried in an oven at 80 °C for all night. The dried ginkgo leaves were then smashed by crusher. The obtained powders of ginkgo leaf were with a size of ~300 mesh.

2.2 Synthesis of Mo₂C&MoS₂@NSC_x

0.6 g Ginkgo leaf powders were mixed with AMT (0.6, 0.3, 0.2, 0.15 g) and then grounded for 20 mins. The mixtures were transferred into the tube furnace and calcinated at 900 °C for 5 h in N₂ flow. The as-prepared black samples were denoted as $Mo_2C\&MoS_2$ loaded on N/S co-doped carbon material ($Mo_2C\&MoS_2@NSC_x$, x=1, 2, 3, 4, x represents the weight ratio of Ginkgo leaf powders

and AMT). Solid residue from direct pyrolysis of ginkgo leaves at 900 $^{\circ}$ C for 5 h under N₂ protection is called as the N/S co-doped on carbon material (NSC).

2.3 Syntheses of Mo₂C@NC₃

0.6 g CB powders were mixed with 0.2 g AMT and then grounding for 20 mins. The mixtures were pyrolysis at 900 $^{\circ}$ C for 5 h under N₂ protection. The as-obtained products were named Mo₂C@NC.

2.4 Syntheses of MoS₂@NSC₃

To avoid the formation of Mo₂C during the pyrolysis, the MoS₂@NSC was prepared via a typical hydrothermal method as reported in Ref. [47] with slightly modification. 0.25 g CH₄N₂S, 0.2 g AMT and 0.6 g CB powders were mixed with 27 mL ultrapure water and ultrasonic mixing for 30 min. The mixture was then transferred into the 50 mL Teflon-lined autoclave and treated at 200 °C for 18 h. The black precipitates were washed with ultrapure water, ethanol in succession and then dried at 80 °C in an oven for whole night. Then, the obtained composites were calcination at 900 °C for 5h with the protection of N₂ flow. The as-prepared sample was named MoS₂@NSC₃.

2.4 Materials characterization

The elemental analysis was carried out via Germany THERMO FISHER SCIENTIFIC. The morphology of the as-prepared catalyst was tested by the scanning electron microscopy (SEM, FEI quanta 400FEG). The transmission electron microscopy (TEM, FEI Tecnai 20 electron microscope) and the high-angle annular dark field-scanning transmission electron microscope (HAADF-STEM, FEI TALOSF200S) was employed to further investigate the detailed morphology and analysis the

crystal lattice of hybrid catalysts. The energy-dispersive X-ray spectroscopy-mapping was along with HAADF-STEM to characterize the element distribution. The X-ray powder diffraction (XRD, Bruker D8) equipment was taken to further prove the structural compositions of the as-prepared materials. The surface element composition and chemical state of all catalyst was studied through K-Alpha X-ray photoelectron spectroscopy (XPS, Thermo ESCALAB 250XI). Calibrating with the binding energy using C1s at 284.6 eV. The Raman spectra was conducted using iHR550 Raman microscope (HOR-IBA scientific) at a range of 400~4000 cm⁻¹. The N₂ adsorption experiment on the Quanta 250F and FEI by Brunauer-Emmett Teller (BET) equation at 77.5 K was applied to measure the specific surface area. The ICP Optical emission spectrometer Varian 720-ES (ICP-OES) was employed to analysis the total Mo content of the as-prepared catalysts.

2.5 Electrochemical measurement

The electrochemical tests for HER were conducted on a conventional three-electrode system on CHI 760E electrochemical workstation at room temperature. The graphite rod was selected as the counter electrode. The mercury/mercurous sulfate electrode (MCE, Hg/Hg₂SO₄) was used as the reference electrode in 0.5 M H₂SO₄. The measured potentials were converted potentials vs. reversible hydrogen electrode (RHE) via Eq.(1). All the HER tests were conducted in the saturated N₂ solution via introducing N₂ into the electrolyte solution for at least 30 minutes.

$$E_{vs.RHE} = EvsHg/Hg_2SO_4 + E^{\theta}Hg/Hg_2SO_4 + 0.059 pH$$
(1)

The glass carbon electrode (GCE, 0.196 cm⁻²) was served as the working electrode via modified with the catalysts ink. The catalysts ink was obtained by mixed 10 mg of the as-prepared catalysts

with 1000 µL of ethanol/deionized water ($V_{C_2H_5OH}$: V_{H_2O} = 1 : 3) and sonicated for 30 minutes. The well mixed catalysts ink with 10 µL was dropped onto the GC and dried naturally. Then the 10 µL Nafion (5 wt.%) was dropped onto the GCE and dried in the room temperature. The catalysts ink of commercial Pt/C was prepared through the same procedure. For the HER tests, the linear sweep voltammetry (LSV) plots were recorded with 1 mV s⁻¹. And the cyclic voltammetry (CV) tests were performed with the scan rates of 10, 30, 50, 70, 90 mV s⁻¹, respectively.

3 Results and discussion

3.1 Physical-chemistry properties

Before the experiment, the element distributions of raw ginkgo leaves were measured through elemental analysis. The weight ratio of every element was shown in Table S1. The weight ratio of nitrogen (N) and sulfur (S) is 1.070 wt.% and 0.613 wt.% respectively, which can serve as the heteroatom dopants on the carbon frameworks. The scanning electron microscopy (SEM) reveals the morphology of raw ginkgo leaf powders is stack microparticles as shown in Figure S1A. After direct pyrolysis at 900 °C for 5 h, the microparticles of ginkgo leaves is broken into the smooth sheets (Figure S1B) which induced by the decomposition of organic components. After the AMT is introduced into the pyrolysis process, the carbon sheets are partially destroyed by reacting with AMT. The calcined products are labeled as $Mo_2C\&MoS_2@NSC_x$ (x=1, 2, 3, 4, x represents the weight ratio of Ginkgo leaf powders and AMT). There are many fragments stacked on the spacious carbon flakes as shown in Figure S2. The shattered carbon flakes will benefit for the increase of the specific surface area of hybrid catalysts. However, the excessive AMT will seriously consume the carbon sheets, thus lead to the aggregation of fragments which will cause the reduce of specific

surface area. There may be a best ratio of ginkgo leaves and AMT to obtain a relative higher specific surface area.

Through the N₂ adsorption/desorption tests, the specific surface area and pore structure of $Mo_2C\&MoS_2@NSC_x$ was investigated. As shown in in Figure 1 and Table S2, the specific surface areas of $Mo_2C\&MoS_2@NSC_x$ (x=2, 3, 4) are 189.08, 226.04 and 168.05 m² g⁻¹ respectively (Figure 1A) with a large number of nanopores mostly with a size less than 50 nm (Figure 1B). The specific surface area reached to the highest value when the mass ratio of ginkgo leaves and AMT was 3. The AMT promote the crush of carbon flakes derived from ginkgo leaves and thus increase the specific surface areas of hybrid catalysts. As the addition of AMT is equally to the ginkgo leaf powders (x=1), the specific surface areas are sharply reduced (6.12 m² g⁻¹). The larger specific surface area and pores structure of $Mo_2C\&MoS_2@NSC_3$ provide more active sites, which will boost catalytic activity. The relatively higher pore volume (Table S2) facilitates the mass transport of reactants to the active sites.



Figure 1 (A) N₂ adsorption/desorption isotherms and (B) Pore size distribution curves of

Mo₂C&MoS₂@NSC_x (x=1, 2, 3, 4).

The Raman spectra was used to estimate the graphitization of carbon materials which have further proved the excessive consumption of carbon sheets at a higher addition amount of AMT. As shown in Figure 2A, the two distinct peaks are named D band and G band, which the D band represents the disordered carbon (at 1340 cm⁻¹) and the G band indicates the graphite carbon (at 1585 cm⁻¹). Notably, there are not obvious D band and G band exhibited in the Raman spectra of $Mo_2C\&MoS_2@NSC_1$ because the carbon skeleton is nearly depletion in the conversion process. The value of I_D/I_G for Mo₂C&MoS₂@NSC_x (x=2, 3, 4) is 0.91, 0.98 and 0.95 respectively. The relative higher value for the Mo₂C&MoS₂@NSC₃ means more structural defects which may enhance the electron exchange and electrochemical activity [10]. The X-ray diffractions (XRD) was employed to confirm the crystal structure of Mo₂C&MoS₂@NSC₃. Besides, as the Mo₂C&MoS₂@NSC₃ own the highest specific surface area and I_D/I_G value, the Mo₂C@NC₃ was designed using CB powders to be the control sample and its crystal structure was also investigated. As shown in Figure 2B and Figure S3A, the intensive diffraction at 34.4°, 37.9°, 39.3°, 52.1°, 61.6°, 69.5°, 74.6° and 75.5° corresponding to (1 0 0), (0 0 2), (1 0 1), (1 0 2), (1 1 0), (1 0 3), (1 1 2) and (2 0 1) planes in wellcrystallized hexagonal β -Mo₂C (JCPDS No. 35-0787) respectively [18]. For Mo₂C&MoS₂@NSC_x, an additional diffractions peak at 14.2° can be attributed to the (0 0 2) planes in hexagonal MoS₂ (JCPDS No. 37-1492) [48]. With the decrease of AMT, the intensity of MoS₂ peak increases. The peak intensity of Mo₂C&MoS₂@NSC₃ is the strongest, which may be ascribed to the saturation of the combination of Mo and S with the combination of Mo and C, while Mo₂C&MoS₂@NSC₄ exhibit relatively low peaks intensity than Mo₂C&MoS₂@NSC₃ because the insufficient Mo cannot combine more S. Herein, The MoS₂ is originated from the *in-situ* vulcanization of Mo nanoparticles.

The coexist of Mo₂C and MoS₂ will enrich the catalytic sites and accelerate the electron interactions of hybrid catalysts [33, 48]. The ratio of Mo₂C and MoS₂ of Mo₂C&MoS₂@NSC₃ was determined of ~ 34.7:1 via RIR (Reference Intensity Ratio) method (Figure S4), the detailed methods and analyses are shown in supporting information. Furthermore, the total Mo content of Mo₂C&MoS₂@NSC₃ is 453.67 mg/g via the ICP-OES tests. Therefore, the content of Mo₂C and MoS₂ of Mo₂C&MoS₂@NSC₃ is 440.96 mg/g and 12.70 mg/g, respectively. In addition, the chemical stats of Mo-based compounds and N/S elements dispersed on the surface played crucial role in the electrocatalytic activity. The surface compositions of the Mo₂C&MoS₂@NSC₃ and Mo₂C@NC was further analysis via X-ray photoelectron spectroscopy (XPS).



Figure 2 (A) Raman and (B) XRD patterns of Mo₂C&MoS₂@NSC_x (x=1, 2, 3, 4).

The full spectrum (Figure 3A) indicates that the Mo₂C&MoS₂@NSC₃ consisted of C, N, O, S, Mo elements. Fitted Mo 3d spectrum (Figure 3B) indicates the existence of Mo⁴⁺ and Mo⁶⁺ species, the Mo⁴⁺ is originated from the MoS₂ and oxide Mo₂C. The Mo⁴⁺ and Mo⁶⁺ is attributed to the molybdenum trioxide resulting from surface oxidation of Mo₂C. The peaks at 228.86 and 232.95 eV indicate the Mo⁴⁺ 3d_{5/2} and Mo⁴⁺ 3d_{3/2} states of Mo, respectively [49-51], which are shifted toward higher BE after the formation of the MoS₂. This may be associated with the higher electronegativity of S compared to C. The fine spectrum of the Mo 3d was deconvolved to further analysis the surface chemical states of Mo atoms. As shown in Figure S5, there are also two peaks located at 229.25 and 231.75 eV in the fitted Mo 3d fine spectrum of Mo₂C&MoS₂@NSC₃ compared with Mo₂C@NC₃. These two peaks indicated the Mo⁴⁺ 3d_{5/2} and Mo⁵⁺ 3d_{5/2} [52], which is also resulted from the surface oxidation of Mo₂C. The full width at half maximum of C oxidation states is decreased from 1.42 to 1.21 eV after growing MoS₂ on ginkgo leaves derived carbon matrix, which indicates a higher spin coupling effect between Mo and C atoms in Mo₂C&MoS₂@NSC₃ catalysts (Figure S6). While for S 2p fine spectrum (Figure 3C), the two peaks at 161.64 and 162.92 eV corresponding to the S 2p_{3/2} and S 2p_{1/2} states of S, respectively [48]. It confirms that the S was transformed into the dopants in carbon matrix and simultaneously combined with Mo. For Mo₂C@NC₃, there is no characteristic peak of S. Besides, there are also no obvious N 1 peaks because the N atoms only came from the AMT which caused the extremely low N dopants (only 1.60% in Table S3) in $Mo_2C@NC_3$. The content of N in $Mo_2C&MoS_2@NSC_3$ is as high as 14.44%. By comparing the N 1 spectrum of the $Mo_2C\&MoS_2@NSC_3$ and $Mo_2C@NC_3$, the four peaks of fitted N 1 spectrum of Mo₂C&MoS₂@NSC₃ at 396.51 eV, 398.36 eV, 399.48 eV and 401.26 eV in Figure 3D indicating the existence of N-Mo, pyridinic N, pyrrolic N and graphite N, respectively [49]. The states of N element in Mo₂C@NC₃ sample are similar to that in Mo₂C&MoS₂@NSC₃ (Figure S7), but the BEs of N elements in Mo₂C&MoS₂@NSC₃ are obviously negatively shift due to the presence of S, which influence the electronic coupling effect between C and Mo. As shown in Figure 3B, the BEs for Mo⁴⁺ and Mo⁶⁺ of Mo₂C&MoS₂@NSC₃ is positively shifted compared with Mo₂C@NC₃, which suggest that the electron transfer from Mo to the S atoms due to its high

electronegativity. The electronegativity of N atoms is higher than S atoms, the electron enrich in the surface of NSC will be captured by the N atoms, thus cause the negatively shift of BEs of N elements. Such results confirm the strong electron interaction of the Mo₂C&MoS₂@NSC₃, which can enhance the HER activity. Therefore, the XPS analysis proves the possible N- and S-doping in carbon matrix and the successful synthesis of Mo₂C and MoS₂. The N/S dopants with higher electronegativity into the carbon lattice will lead to the destroy of electric neutrality and the charge density of carbon atoms, thus cause the defects of disordered carbon.



Figure 3 (A) Full XPS spectra of Mo₂C@NC₃ and Mo₂C&MoS₂@NSC₃. (B) Mo 4d and (C) S 2p

XPS scan spectra for Mo₂C@NC₃ and Mo₂C&MoS₂@NSC₃. (D) N 1s XPS spectra for

 $Mo_2 C\&MoS_2 @NSC_3.$

Transmission electron microcopy (TEM) and high-resolution transmission electron microcopy (HRTEM) were carried out to in-depth analysis the structure of Mo₂C&MoS₂@NSC₃. As shown in Figure 4A-C, it can be seen that the Mo nanoparticles disperse on the surface of carbon flakes homogenously with a size range from 5 to 20 nm. From the HRTEM images (Figure 4D) of Mo₂C&MoS₂@NSC₃, it shows that there are two different crystal structures inside. The lattice plane distance of 0.15 nm corresponds to the (1 1 0) plane of Mo₂C, and 0.16 nm accords with the (1 1 0) plane of MoS₂, which was well corresponding to the XRD results. In addition, the element distribution of Mo, C, S, N, O was revealed by HAADF-STEM mapping images (Figure 4E). It is worth noting that the S content is highly overlapped with the region of Mo, which is consistent with the formation of MoS₂. In addition, a small amount of S is doped into the carbon sheet. The successful doping of S and N would change the electrical neutrality of carbon and thus promote the charge transfer capability [40, 53]. Furthermore, the Mo₂C and MoS₂ have more moderate hydrogen boding energy which would facilitate the adsorption and desorption of intermediate media, thus resulted in more fast kinetic process [48].



Figure 4 (A, B, C) TEM image, (D) HRTEM images of Mo₂C&MoS₂@NSC₃. (E) STEM image and the element mapping image of N, C, S and Mo elements of Mo₂C&MoS₂@NSC₃.

3.2 HER performance in acid electrolyte

The HER performance of the as-prepared samples and benchmark 20% commercial Pt/C catalyst in 0.5 M H₂SO₄ was firstly explored. Figure 5A displayed the linear polarization curves (LSV) plots with a scan rate of 1 mV S⁻¹. As one can see, the commercial Pt/C exhibits anticipated electrocatalytic performance with a near zero onset potential, together with 35 mV at standard current of 10 mA cm⁻². The overpotential of Mo₂C&MoS₂@NSC_x (x=1, 2, 3, 4) to drive the current density to 10 mA cm⁻² was 331, 226, 209, and 220 mV (η_{10}), respectively. As expected, the Mo₂C&MoS₂@NSC₃ has the best electrocatalytic activity among the Mo₂C&MoS₂@NSC_x, and is much better than that of Mo₂C@NC₃ (713 mV) and NSC (> 1100 mV). The introduction of Mo₂C

carbon for catalytic HER. Meanwhile, the addition amount of AMT seriously affects the HER performance of the as-synthesized electrocatalysts. Especially, the excessive addition of AMT will result in the depletion of carbon skeleton and cause the decrease of catalytic activity. The $MoS_2@NSC_3$ is also prepared and its crystal structure is identified via XRD (Figure S3B). The overpotential for Mo₂S@NSC₃ to afford 10 mA cm⁻² current density is 498 mV, which is better than Mo₂C@NC₃, but worse than Mo₂C&MoS₂@NSC₃. In Figure 5B and S8, the corresponding Tafel slopes for Mo₂C&MoS₂@NSC_x (x=1, 2, 3, 4) MoS₂@NSC₃ and Mo₂C@NC₃ are 250.8 mV dec⁻¹, 110.5 mV dec⁻¹, 85.5 mV dec⁻¹, 111.6 mV dec⁻¹, 129.7 mV dec⁻¹ and 208.6 mV dec⁻¹, respectively. The Tafel slope values of Mo₂C&MoS₂@NSC₃ indicate its likely occur through the Volmer-Heyrovsky mechanism. Moreover, the Mo₂C&MoS₂@NSC₃ has a C_{dl} value of 10.72 mF cm⁻², which is far higher than Mo₂C&MoS₂@NSC₁ (6.04 mF cm⁻²), Mo₂C&MoS₂@NSC₂ (6.39 mF cm⁻²) ²), Mo₂C&MoS₂@NSC₄ (9.91 mF cm⁻²), MoS₂@NSC₃ (5.62 mF cm⁻²) and Mo₂C@NC₃ (4.97 mF cm⁻²) (Figure S9 and 5C), indicating that Mo₂C&MoS₂@NSC₃ exhibits a larger active surface area. The HER performance of the as-prepared Mo₂C&MoS₂@NSC₃ is compared with the reported Mo₂C or MoS₂ based electrocatalysts as shown in Table S4. It is observed that the HER activity of the asobtained electrocatalysts can contend with other Mo-based electrocatalysts. Besides, the remarkable stability of Mo₂C&MoS₂@NSC₃ within 15 h of continuous operation in a 0.5 M H₂SO₄ solution, was proved by a chronoamperometric test.



Figure 5 (A) LSV curves of NSC, Mo₂C@NC₃ and Mo₂C&MoS₂@NSCx (x=1, 2, 3, 4) in 0.5 M
H₂SO₄ electrolyte. (B) Corresponding Tafel curves derived from (A). (C) C_{dl} stands for ECSA of the five electrocatalysts. (H) Durability test of Mo₂C&MoS₂@NSC₃.

As manifested in Figure 5D, there is a negligible current decay (< 10%) and almost invariable catalytic activity after 2000 cycles (Figure 5D inset), which indicates the superior stability of Mo₂C&MoS₂@NSC₃. After stability tests, the catalysts were recycled to employ the XRD, XPS, SEM and TEM testing to further prove the stability of the Mo₂C&MoS₂@NSC₃. The crystal structure of Mo₂C&MoS₂@NSC₃ was firstly identified via XRD (Figure S10). All the diffraction peaks are matched well with original Mo₂C&MoS₂@NSC₃. The same BEs of Mo 3d, N 1s, S 2p, (Figure S11 A-D) are found for the recycled catalysts after HER tests, which demonstrate that there

is nearly no change in the surface valence sate of various species of Mo₂C&MoS₂@NSC₃. In addition, representative SEM image of the surface after i-t tests are shown in Figure S12 A, which suggests that the morphology of the accumulated fragments on carbon matrix have not changed during the stability tests. As shown in the TEM image (Figure S12 B), the Mo₂C and MoS₂ nanoparticles are homogeneously dispersed on the carbon support. Such observations indicate no evidence of any phase transformation during the stability tests, suggest the robust stability of the asprepared catalysts.

Utilizing the microstructure of ginkgo leaves and its characteristics of being rich in N and S elements, Mo₂C&MoS₂@NSC₃ catalyst was synthesized by mixing and calcining ginkgo leaves with AMT in one step. There are three possible explanations for the high catalysis characteristics our synthesized Mo₂C&MoS₂@NSC₃ catalyst: (1) ginkgo leaves not only serve as the N/S doped carbon supports, but also served as the reactants to combine with Mo to *in-situ* form Mo₂C and MoS₂, which have been proved to favor the electron interactions. Meanwhile, the synergistic effect of Mo₂C&MoS₂ with NSC make Mo₂C&MoS₂@NSC₃ show an excellent HER performance, (2) the carbon structural defects which are beneficial for electrochemical reactions, (3) the larger specific surface area, pores structure and high pore volume of Mo₂C&MoS₂@NSC₃ are conducive to the uniform dispersion of active components, and play a positive role in reducing the size of active centers and preventing sintering agglomeration of active components, thus leading to high HER activity. Our study shed light up a universal and effective way to fabricate dual phase of Mo-based carbon catalysts derived from ginkgo leaves as the electrocatalysts with high-active, long-term stability and cost-effective for HER.

4 Conclusions

In summary, an environment-friendly one-pot protocol to prepare Mo₂C&MoS₂ loaded on N/S codoped carbon nanosheets was designed by using the common biomass waste-ginkgo leaves. Benefited from the N and S contents of ginkgo leaves, the dopants of N and S atoms was *in-situ* introduced into the carbon skeleton without additive agents. The Mo₂C and MoS₂ was resulted from the reactions between Mo and C (or S) in the pyrolysis process. The Mo₂C and MoS₂ as the active sites and N/S co-doped carbon matrix endows the hybrids catalyst with superior activity to catalytic H₂ evolution. During the HER tests in 0.5 M H₂SO₄, the Mo₂C&MoS₂@NSC₃ only needs 209 mV to drive the cathodic current density to 10 mA cm⁻², as well as a relatively low Tafel slope of 85.5 mV dec⁻¹ and superb durability. The hybrid electrocatalysts was demonstrated to be potential to serve as the HER catalysts. In addition, it also offers an efficient way to utilize the waste biomass which was rich in N and S contents.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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