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A new three-dimensional subsulfide $\text{Ir}_2\text{In}_8\text{S}$ with Dirac semimetal behavior

Jason F. Khoury¹, Alexander J. E. Rettie^{2,3}, Mojammel A. Khan², Nirmal J. Ghimire², Iñigo Robredo^{4,5}, Jonathan E. Pfluger⁶, Koushik Pal⁶, Chris Wolverton⁶, Aitor Bergara^{4,5,7}, J. S. Jiang², Leslie M. Schoop⁸, Maia G. Vergniory^{4,9}, J. F. Mitchell², Duck Young Chung², and Mercuri G. Kanatzidis^{1,2*}

¹*Department of Chemistry, Northwestern University, Evanston, Illinois 60208, United States*

²*Materials Science Division, Argonne National Laboratory, Argonne, Illinois 60439, United States*

³*Department of Chemical Engineering, University College London, London WC1E 7JE, United Kingdom*

⁴*Donostia International Physics Center, Paseo Manuel de Lardizabal 4, 20018 Donostia-San Sebastian, Spain*

⁵*Condensed Matter Physics Department, University of the Basque Country UPV/EHU, 48080 Bilbao, Spain*

⁶*Department of Materials Science and Engineering, Northwestern University, Evanston, Illinois 60208, United States*

⁷*Centro de Física de Materiales, Centro Mixto CSIC-UPV/EHU, 20018 Donostia, Spain*

⁸*Department of Chemistry, Princeton University, Princeton, New Jersey 08540, United States*

⁹*Ikerbasque, Basque Foundation for Science, E-48011 Bilbao, Spain*

Abstract

Dirac and Weyl semimetals host exotic quasiparticles with unconventional transport properties, such as high magnetoresistance and carrier mobility. Recent years have witnessed a huge number of newly predicted topological semimetals from existing databases; however, experimental verification often lags behind such predictions. Common reasons are synthetic difficulties or the stability of predicted phases. Here, we report the synthesis of the Type-II Dirac semimetal $\text{Ir}_2\text{In}_8\text{S}$, an air-stable compound with a new structure type. This material has two Dirac crossings in its electronic structure along the Γ -Z direction of the Brillouin zone. We further show that $\text{Ir}_2\text{In}_8\text{S}$ has a high electron carrier mobility of $\sim 10,000 \text{ cm}^2/\text{Vs}$ at 1.8 K, and a large, non-saturating transverse magnetoresistance of $\sim 6000\%$ at 3.34 K in a 14 T applied field. Shubnikov de-Haas oscillations reveal several small Fermi pockets and the possibility of a nontrivial Berry phase. With its facile crystal growth, novel structure type, and striking electronic structure, $\text{Ir}_2\text{In}_8\text{S}$ introduces a new material system to study topological semimetals and enable advances in the field of topological materials.

Introduction

The field of topological semimetals (TSMs) has been blossoming since the discovery of several three-dimensional (3D) materials with exotic physical properties, such as Cd_3As_2 and Na_3Bi .¹⁻⁷ As 3D analogues of graphene, these materials have linearly dispersing bands that cross at the Fermi level to create four-fold degenerate Dirac points in momentum space and are referred to as Dirac semimetals (DSMs). These Dirac points harbor massless Dirac fermions that engender remarkable transport properties, including high carrier mobility and extremely large magnetoresistance.⁸⁻¹¹ If a material with symmetry protected band crossings lacks inversion or time reversal symmetry (TRS), the massless fermions are only two-fold degenerate instead of four, referred to as a Weyl semimetal (WSM).^{6, 8-9, 12} These Weyl fermions have different surface states and transport behavior compared to DSMs due to their inherent chirality or “handedness”.¹³ The interest in TSMs is not limited to just their transport properties; it also extends to other areas such as optical switches or infrared sensors.¹⁴ In addition, it has been predicted that topological superconductivity (TSC) could arise in type-II DSMs with tilted Dirac cones.¹⁵⁻²¹ Understanding the physical phenomena linked to DSMs and advancing the knowledge of how Dirac and Weyl fermions behave calls for new and diverse examples of stable material systems in which they reside.

To date, there exist a few dozen experimentally verified topological semimetals, while many more have been predicted but not yet synthesized.⁶⁻⁷ There are many factors that slow down or impede experimental verification of predictions: First, the predicted materials might not be synthesizable in single crystalline form. Second, they might be extremely air sensitive, which will drastically complicate measurements. Third, a wide combination of different experimental techniques is required to definitively determine the topological nature of a system. Lastly,

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3 prediction based on DFT can be incorrect, either by underestimating band gaps, omitting magnetic
4 order, or by predicting unreliable materials.²² The currently available candidate materials are
5 limited in their structural diversity, with most corresponding to the same handful of structure types.
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7 In particular, the class of square net materials has garnered considerable attention as a source for
8 TSM candidates because of their wide range of tunable and synthesizable analogues.²³ Still, these
9 materials all show related electronic structures as a result of their similar crystal structures.
10
11 Synthesizing air-stable TSMs with novel structure types as well as facile growth conditions is
12 important in further understanding their properties, as there are several different types of fermion
13 degeneracies predicted to exist that have not yet been verified experimentally.²⁴⁻²⁵ Considering
14 recent advances that catalogued all known materials based on their topological properties, new
15 structure types and compounds have become crucial for advancing the field.²⁶⁻³⁰ Indeed, more
16 unique candidate materials and diverse structure types must be identified to provide new insight
17 in understanding topological behavior. In this work, we describe the synthesis, crystal structure,
18 and electronic transport characterization of Ir₂In₈S, a new, air-stable subsulfide compound that
19 represents a Dirac semimetal candidate material.
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37 **Results and Discussion**

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40 Ir₂In₈S crystallizes in a new structure type in the *P4₂/mnm* space group (Figure 1). Open
41 Quantum Materials Database (OQMD) calculations show that Ir₂In₈S is a ternary phase that resides
42 on the convex hull, with a formation enthalpy of -0.266 eV/atom (Figure S6). Ir₂In₈S is a
43 subsulfide, an unusual type of compound that lies between the classes of intermetallics and salt-
44 like chalcogenides. Subsulfides contain metal-metal (Ir-In) along with metal-chalcogenide (In-S)
45 interactions, causing the metal atoms to have very low formal oxidation states. Subchalcogenide
46 materials are not a well-known class of compounds but they could be of great interest because of
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3 their potential for topological, superconducting and charge density wave (CDW) behavior, as their
4 lower dimensional metallic substructures and exotic bonding motifs facilitate these interactions.³¹⁻
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8 ³³ However, despite their promise for remarkable physical properties, it is a great challenge to
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10 synthesize and characterize new subchalcogenide materials, as they are notoriously difficult to
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12 form due to phase separation into constituent binary intermetallic and chalcogenide materials.
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14 Here, we overcome this difficulty by utilizing an indium metal flux to synthesize Ir₂In₈S, a method
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16 rarely used for these types of materials.³⁴⁻³⁵ Through this method, we can grow large single crystals
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18 on the scale of several millimeters that are stable in air, water, and dilute acids, making them ideal
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20 candidates for electrical transport measurements.
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24 The structure of Ir₂In₈S consists of a dense three-dimensional framework assembled by
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26 eight-coordinate IrIn₈ polyhedra with chalcogenide atoms found in the channels along the *c* axis
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28 (crystallographic information in Tables S1-S3). The IrIn₈ polyhedra are distorted bi-capped
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30 trigonal prisms that are corner sharing along the *a* and *b* axes but alternate between corner and
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32 edge sharing along the *c* axis, resulting in the tetragonal nature of this compound. The sulfur atoms
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34 in the channels adopt an approximately square planar coordination with the surrounding indium
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36 atoms, forming long In-S bonds of approximately 2.8487(16) Å. The indium coordination
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38 environment is two Ir-In bonds (2.7397(14) Å) along with a short In-In bond of approximately
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40 2.689(3) Å connecting the edge sharing polyhedra along the *c* axis, and this bond length is shorter
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42 than the In-In dimer in InS (2.7624(4) Å).
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47 The chemical origin of subchalcogenide materials comes from an intermetallic network
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49 that has been partially oxidized by chalcogenide atoms, causing regions of metallic behavior and
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51 regions of ionic or salt-like behavior. In the case of Ir₂In₈S, the electronegative sulfide anions act
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53 as electron acceptors, taking electron density from the more electropositive indium atoms. In a
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3 sense, the indium is the bridging atom in the framework of the structure, as it interacts with both
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5 iridium and sulfur, but those two atoms do not directly bond with one another due to their large
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7 electronegativities. The fact that the chalcogenide ions in the channels act as ionic species is
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9 actually unusual for most subsulfides and subselenides, as they have shown in other structures that
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11 they can form polar covalent bonds with both chalcogenide atoms and transition metals.³⁶⁻⁴⁰ In a
12
13 sense, the bonding in Ir₂In₈S has more in common with subhalide materials, as the sulfur atoms
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15 act as counter ions that oxidize the intermetallic framework.⁴¹⁻⁴³
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19 To study the presence of topological behavior in Ir₂In₈S, we employed the CheckTopMat
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21 application from the Bilbao Crystallographic Server (BCS) and computed the irreducible
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23 representations (irreps) of the occupied bands at the high symmetry points Γ and Z . This procedure
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25 checks whether the band structure is topological at the Fermi level. The calculations indicate that
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27 Ir₂In₈S is an enforced semimetal (ES), which means that the semimetallic behavior is protected by
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29 symmetry.⁴⁴ Two band crossings are observed along the line A located only 0.025 and 0.040 eV
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31 above the Fermi level, see Figure 2A-B. Following inversion or time reversal symmetry, there are
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33 two more crossings along the Γ - Z line. The electronic structure is similar to Na₃Bi but with 4 total
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35 crossings instead of 2.⁴
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40 We also computed the irreducible representations above the Fermi level along the high
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42 symmetry line A connecting Γ and Z using Quantum Espresso to obtain information about the
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44 symmetry element which protects the crossing (see Figure 2C).⁴⁵⁻⁴⁷ Based on our group theoretical
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46 analysis, we identify this to be the 4₂ screw axis, which differentiates the representations of the
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48 bands that cross and prevents their hybridization. Any perturbation that breaks this symmetry will
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50 enable hybridization of the bands and could open a gap. The general band dispersion of an
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52 anisotropic, tilted Dirac crossing is:
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$$E(\vec{k}) = \vec{w}_0 \cdot \vec{k} \pm \sqrt{(v_F^x k_x)^2 + (v_F^y k_y)^2 + (v_F^z k_z)^2} \quad (1)$$

Where v_F^i is the i^{th} component of the Fermi velocity and w_0 is the tilt vector of the cone.⁴⁸ We performed ab-initio calculations and computed the band dispersion in the y and z directions centered at the crossings. A depiction of the lower tilted Dirac cone can be seen in Figure 2D, and fits and their values are displayed in Figure S5 and Table S4. Information about the Fermi velocity was extracted by fitting the bands to the dispersion given in Equation (1).

Inversion (and time reversal) symmetry dictates that band dispersion along $k_x > 0$ must be the same as along the $k_x < 0$ direction (the same holds for k_y). Since the crossings are located along the Γ - Z line ($k_x = k_y = 0$), the following must be true: $\omega_0^x = \omega_0^y = 0$. The 4-fold rotation axis along z demands that $v_F^x = v_F^y$. Comparing the Fermi velocity values to the ones in Na₃Bi, we see that $v_F^{x,y}$ are 10 times smaller but v_F^z is 10 times larger (Figure S5 and Table S4). The two Dirac points above the Fermi level in Ir₂In₈S may contribute to some of the notable electronic properties found in Ir₂In₈S, as discussed below, such as the high carrier mobility and Shubnikov de-Haas (SdH) oscillations.

Ir₂In₈S displays metallic resistivity (Figure 3A) and a negative Hall resistivity, suggesting that electrons are the majority carrier type (Figure S2). The residual resistivity ratio (RRR, $\rho_{300} \text{ K} / \rho_{1.8 \text{ K}}$) for the crystals used for the Hall and quantum oscillations measurements seen below were 88 and 40, respectively, indicating high quality crystals with few intrinsic defects. In Ir₂In₈S, the Hall resistivity as a function of magnetic field becomes noticeably nonlinear below 50 K, suggesting an incorporation of hole carriers. To account for this, two-band fits of the data were taken into consideration when determining the Hall coefficients below 50 K (Table S5). The n-type carrier concentration of Ir₂In₈S is approximately 10^{20} cm^{-3} , and this value is relatively constant

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3 as a function of temperature. The electron and hole carrier mobilities of Ir₂In₈S are high, with
4 values of ~10,000 cm² V⁻¹ s⁻¹ at 1.8 K (Figure 3B). The low temperature carrier mobility of our
5 sample of Ir₂In₈S is comparable to that of the Weyl semimetal WTe₂ (~10⁴ cm²V⁻¹s⁻¹), but
6 significantly less than that of the monpnictides (e.g., TaP $\mu \sim 10^5$ cm²V⁻¹s⁻¹).⁴⁹⁻⁵¹ The electron and
7 hole carrier concentrations and mobilities are almost identical when fitted with the two-band
8 model, suggesting that Ir₂In₈S is a nearly fully compensated semimetal.⁵⁰ The large carrier mobility
9 values are likely a result of the invariance to backscattered electrons from the symmetry protected
10 band crossings near the Fermi level, as the relaxation time τ is related to both mobility and drift
11 velocity v_D .
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24 The magnetoresistance (MR) data in Figure 3C-D corroborate the high electron mobility
25 of this structure, as the material has large, non-saturating MR% of 6038% at 14 T and 3.34 K in a
26 transverse configuration (I || [001], B || [110]). Since magnetoresistance follows $MR = 1 + (\mu B)^2$,
27 where μ is mobility and B is applied magnetic field), the fact that Ir₂In₈S has a high MR% is in
28 line with its high carrier mobility. The MR% differs greatly in sinusoidal fashion as the applied
29 magnetic field direction changes, reaching a minimum of 225% when both current and field are
30 along the c axis in a longitudinal configuration (Figure 3D), suggesting that the overall Fermi
31 surface is anisotropic. Ir₂In₈S has a lower MR% than the reported values of other fully
32 compensated semimetals such as WTe₂, LaSb, and TaAs₂.^{50, 52-53} This modest MR% may reflect a
33 compromised crystal quality in comparison to crystals of the latter compounds with RRR in the
34 10²-10³ range. We note, however, that a 3 x 10⁵ MR% has been reported for compensated TaP
35 with RRR ~ 65.⁴⁹
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52 Shubnikov de-Haas oscillations are shown in Figure 4. The full battery of angle and
53 temperature dependent data revealed two small, predominant frequencies at 86 (F_{d1}) and 155 T
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($F_{\alpha 2}$), as well as a third notable pocket at 241 T (F_{β}). The strongest frequency, F_{α} , has one harmonic ($F_{2\alpha 1}$) at 171 T and $F_{\alpha 2}$ has one harmonic ($F_{2\alpha 2}$) at 303 T. α_1 and α_2 do not vary strongly as a function of angle, suggesting that these pockets are isotropic in nature despite the calculated Fermi surface predicting a more ellipsoidal topology for both (Figures S3 and S4). For $F_{\alpha 1}$, we can use the Lifshitz-Onsager relation to determine the extremal cross-sectional area (S_F) of the Fermi surface, Equation (2)

$$F = \frac{\hbar}{2\pi e} S_F \quad (2)$$

where F is the frequency of the pocket and h is Planck's constant.⁵⁴ Ir₂In₈S has an S_F of 0.00821 Å⁻², which occupies 2.04% of the Brillouin zone area along the $k_x - k_y$ plane, and a Fermi momentum $k_F = \sqrt{\frac{S_F}{\pi}} = 0.051$ Å⁻¹. $F_{\alpha 1}$ and $F_{\alpha 2}$ survive up to 20 K, while F_{β} is still visible at 15 K.

We can fit the decay amplitude of each frequency to determine the effective mass of the pocket by the Lifshitz-Kosevich (LK) formula:

$$A(T, B) \sim e^{\frac{-2\pi^2 k_B T_D}{\hbar \omega_C} - \frac{2\pi^2 k_B T}{\hbar \omega_C}} \frac{1}{\sinh \frac{2\pi^2 k_B T}{\hbar \omega_C}} \quad (3)$$

where k_B is Boltzmann's constant, T_D is the Dingle temperature, B is applied field, T is temperature, and ω_C is the cyclotron frequency ($\omega_C = eB/m^*$, with m^* as the carrier effective mass). The value of B for fitting was the average of the minimum and maximum value in the Fast Fourier Transform data. The LK fit shown in Figure 4C reveals an m^* of 0.234(6) m_e for $F_{\alpha 1}$, with m_e being the mass of a free electron in vacuum. $F_{\alpha 2}$ has a nearly identical fitted effective mass to $F_{\alpha 1}$ ($m^* = 0.234(4)m_e$), and F_{β} , F_{δ} , and F_{γ} have notably heavier fitted effective masses (0.367(6) m_e , 0.42(2) m_e , and 0.34(1) m_e , respectively). The low effective masses of the two primary alpha frequencies are coincident with the high electron carrier mobility and persistence of SdH

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3 oscillations up to 20 K. The prominent α_1 and α_2 pockets have cross sectional areas of 0.00821
4 and 0.0148 \AA^{-2} , respectively, which are consistent with the calculated cross sectional areas of
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6 0.00815 and 0.0120 \AA^{-2} from the Fermi surface calculations (see Figure S4).
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10 To explore the possibility of topological behavior, the Landau level (LL) fan diagram can
11 be plotted to approximate the value of the Berry phase by indexing the peaks and valleys of the
12 $F_{\alpha l}$ frequency after applying a band pass filter to the data. The Onsager phase (γ) is determined by
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14 the equation $\gamma = \frac{1}{2} - \varphi_B$, where φ_B is the Berry phase.⁵⁴ The intercept of the LL fan diagram can be
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16 used to determine γ , which can then extract the value of φ_B . For $\text{Ir}_2\text{In}_8\text{S}$, we obtain an intercept of
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18 0.03, which might suggest a topologically nontrivial Berry phase value of $\sim\pi$. We note that the
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20 Landau fan diagram derived from our data only reaches a minimum Landau level of 6, which is
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22 far from the quantum limit and would require much higher applied magnetic fields to definitively
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24 verify this claim.
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31 **Conclusion**

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33 In conclusion, $\text{Ir}_2\text{In}_8\text{S}$ is a novel Dirac semimetal candidate that crystallizes in a new
34 structure type. $\text{Ir}_2\text{In}_8\text{S}$ is a subsulfide: A rare, metal-rich compound that contains both metal-metal
35 and metal-chalcogenide bonding interactions. The symmetry analysis of its electronic band
36 structure indicates that it has two symmetry-protected crossings next to each other along the Γ - Z
37 direction. Analysis of the SdH oscillations show several small Fermi surface pockets, a small
38 effective mass, and the possibility of a nontrivial Berry phase in the electronic structure. These
39 experimental data, along with the theoretical analysis of its band structure, point to the topological
40 character of this material. $\text{Ir}_2\text{In}_8\text{S}$ is therefore a notable new entry in the field of topological
41 semimetals, as it is not found in any database and offers a new system for future studies on its
42 quantum properties.
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Experimental Section

Synthesis. For Ir₂In₈S, Ir (1 mmol, 0.1922 g), S (0.5 mmol, 0.0163 g), and In (20 mmol, 2.2964 g) were loaded into alumina crucibles. A 100-mesh stainless steel filter was placed on top of the alumina crucibles, and a small piece of alumina tubing was placed on top of the filter as a counterweight. The crucible, filter, and tubing were loaded into 18 mm fused silica tubes and sealed under vacuum at $\sim 3 \times 10^{-3}$ mbar, heated to 1000 °C in 12 h, held there for 24 h, and cooled to 650 °C in 24-48 h. After being held at 650 °C for a few hours, the tubes were removed from the furnace and immediately centrifuged to remove excess indium flux. Residual indium on the surface of the crystals was removed by etching in dilute (10% by weight) HCl for 2-3 hours, and then the crystals were filtered and washed with water and acetone. The crystals are stable in air and stable in dilute HCl for at least 2-3 days. Yields for these reactions range from 50-75% by weight depending on the limiting reagent.

Single Crystal X-ray Diffraction. Crystals of Ir₂In₈S were adhered to a glass fiber with super glue. Single crystal x-ray diffraction was performed at 200 K on a Bruker-APEX II CCD diffractometer with MoK α radiation ($\lambda = 0.71073$ Å). The data were integrated using SAINT-v8.38A and the multi-scan absorption correction was applied using SADABS.⁵⁵ The structures were solved with intrinsic phasing in the SHELXT software package, and refined with SHELXL by the Least Squares method.⁵⁶ The crystallographic information for the structures can be found in Tables S1-S3. Additional crystallographic information can be found in the supporting information.

Density Functional Theory (DFT) Calculations. Band structure calculations were performed using Density Functional Theory (DFT) as implemented in the Vienna Ab initio Simulation Package (VASP).⁵⁷⁻⁶⁰ The interaction between ion cores and valence electrons was

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3 treated by the projector augmented-wave method (PAW), the generalized gradient approximation
4 (GGA) for the exchange-correlation potential with the Perdew-Burke-Ernkzerhof for solids
5 parametrization and spin-orbit coupling (SOC) was taken into account by the second variation
6 method.⁶¹⁻⁶³ A Monkhorst-Pack k-point grid of $(6 \times 6 \times 6)$ for reciprocal space integration and 500
7 eV energy cutoff of the plane-wave expansion have been used. Atomic positions were fixed and
8 therefore the symmetries of the space groups were preserved without disorder. The Open Quantum
9 Materials Database (OQMD) was used to perform stability calculations on $\text{Ir}_2\text{In}_8\text{S}$.

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20 **Charge Transport.** Temperature-variable resistivity and Hall effect measurements on
21 single crystals (approximate dimensions: $1 \times 1 \times 0.3 \text{ mm}^3$) were conducted on a Quantum Design
22 Dynacool Physical Property Measurement System (PPMS) between 1.8 and 300 K. Resistivity
23 was measured in a 4-point collinear geometry (with current in an arbitrary direction along the ab
24 plane) and the Hall effect measured using two Hall voltage contacts placed perpendicular to the
25 axis of current flow. The magnetic field was applied perpendicular to the axis of current flow from
26 -9 to +9 T. Temperature and field were cycled multiple times to confirm data reproducibility.
27 Electronic resistivity and Hall effect measurements were performed simultaneously on the same
28 sample. Stable, Ohmic contact was achieved by thermally evaporating Au metal pads, or sputtering
29 Pt metal pads on the sample and attaching Au wires with silver paste (Dupont 4929N). The residual
30 resistivity ratio (RRR, $\rho_{300 \text{ K}}/\rho_{1.8 \text{ K}}$) for this crystal was ~ 88 . Multiband fits for the low temperature
31 data used a simple two-band model that has been described elsewhere.⁶⁴

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49 To perform SdH Oscillations, single crystals were oriented such that current was along the
50 c axis and the perpendicular axis was the (110) plane. Samples were measured on a Quantum
51 Design PPMS with a 14 T magnet attachment between 3.34 K and 25 K. Resistivity was again
52 measured in a 4-point collinear geometry. The magnetic field was initially applied perpendicular
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3 to the axis of current flow from 5 to 14 T to measure magnetoresistance as a function of field,
4 revealing the SdH oscillations. On the PPMS rotator, the MR was measured from 0 to 100 degrees
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6 in 10 degree increments through the same range. The RRR ($\rho_{300\text{ K}}/\rho_{3.34\text{ K}}$) for this oriented crystal
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8 was ~ 40 . A band pass filter was applied using OriginLab 8 to isolate the SdH frequency used to
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10 create the Landau fan diagram.
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15 **Associated Content**

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18 The supporting information is available free of charge via the internet at the ACS Publications
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20 website.
21

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23 Experimental details, SEM/EDS, Hall resistivity and fitting information, FFT data,
24
25 additional Fermi surface and band calculations, and additional crystallographic
26
27 information (CIF)
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29 **Author Information**

30
31 Corresponding Author

32
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34 *E-mail: m-kanatzidis@northwestern.edu
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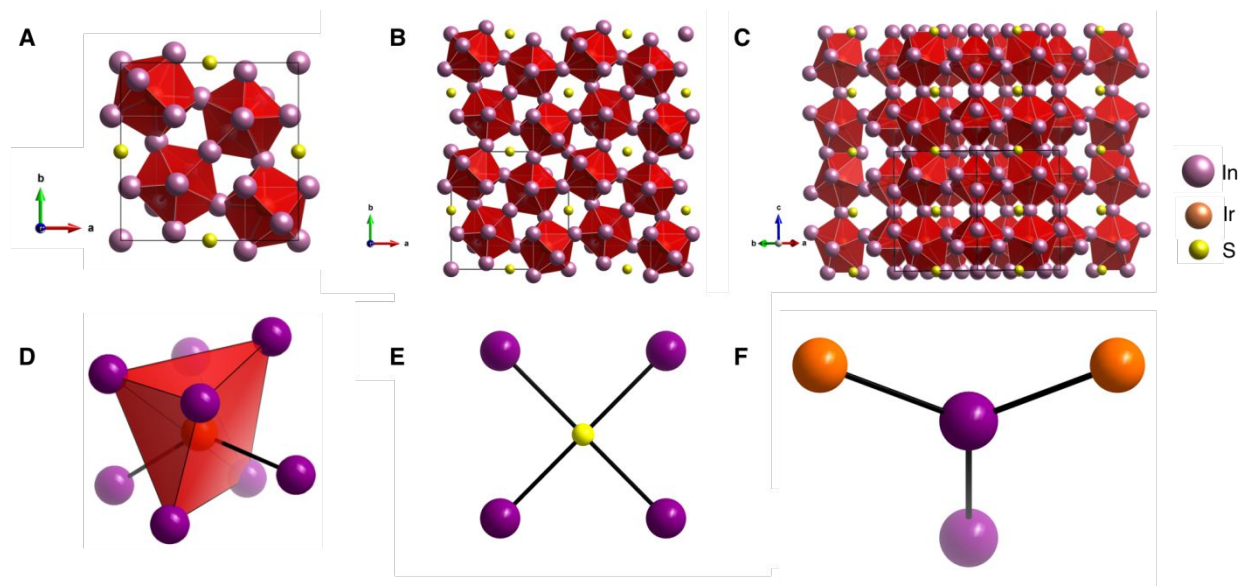


Figure 1: The structure of Ir₂In₈S. (A) Individual unit cell of Ir₂In₈S along the *c* axis. The space group is *P4*₂/*mmm*, and the lattice parameters are *a* = 9.8937(16) Å, *c* = 10.0991(17) Å, $\alpha = \beta = \gamma = 90^\circ$. (B, C) Extended structure along the *ab* plane (B) and the *c* axis (C). The connectivity along the *c* axis alternates between corner and edge sharing polyhedra. (D) Individual IrIn₈ coordination environment, showcasing the distorted bi-capped trigonal prism geometry. (E) Coordination environment of sulfur, adopting an approximately square planar geometry with In-S bonds (2.8487(16) Å). (F) Coordination environment of indium, showcasing the Ir-In bonds (2.7397(14) Å) and In-In bond (2.689(3) Å) that forms the edge sharing polyhedra along the *c* axis.

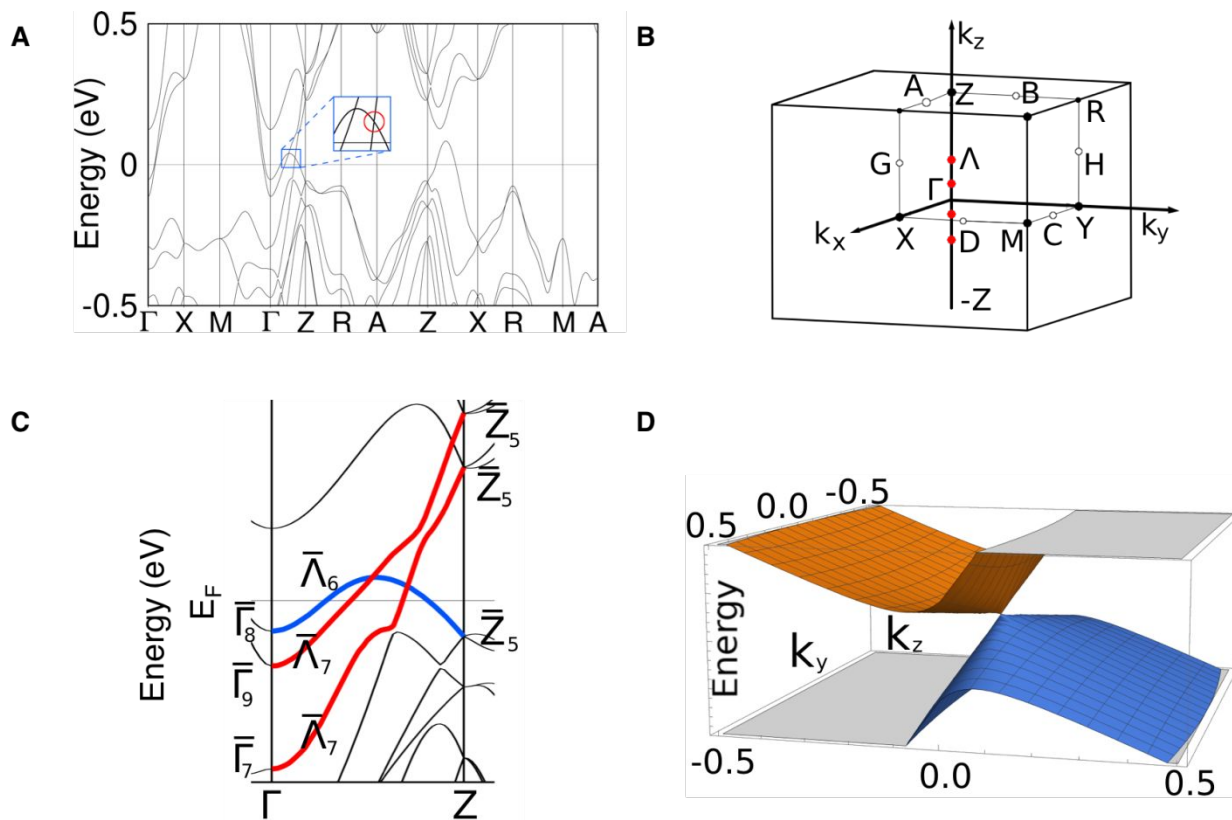


Figure 2: (A) $\text{Ir}_2\text{In}_8\text{S}$ band structure. All band structures have two Dirac crossings in the line Γ -Z. Highest fitted values correspond to the crossing inside the red circle in $\text{Ir}_2\text{In}_8\text{S}$ band structure. (B) Brillouin zone of the space group $P4_2/mnm$ (136). The two Dirac crossings near the Fermi level are depicted as red dots. (C) Irreducible representations (irreps) at high-symmetry points and the line connecting them. Blue color stands for Λ_6 irrep and red color for Λ_7 . The lighter color bands are the ones demanded by the group theoretical analysis but not relevant for the protection of the crossings. (D) Visual representation of the lower tilted Dirac cone that is circled in 2A.

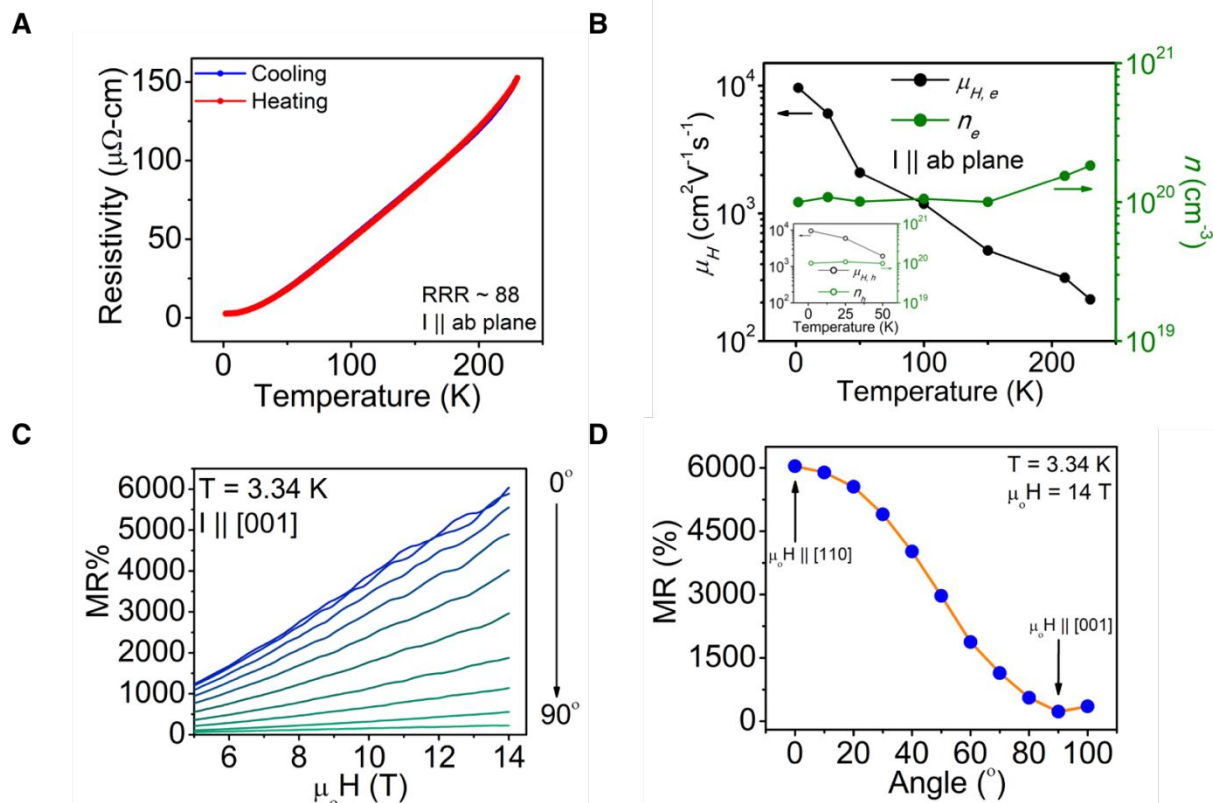


Figure 3: (A) Resistivity (ρ) and (B) Hall transport data for $\text{Ir}_2\text{In}_8\text{S}$. $\text{Ir}_2\text{In}_8\text{S}$ has an electron and hole (inset) mobility of $\sim 10,000$ $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ at 1.8 K and a carrier concentration of $\sim 10^{20}$ cm^{-3} . (C) Angle dependent magnetoresistance on an oriented crystal of $\text{Ir}_2\text{In}_8\text{S}$ at 3.34 K from 5 to 14 T. (D) Maximum MR% at 14 T as a function of angle; the approximately $\cos^2(\theta)$ dependence of the magnetoresistance can be seen by the orange line.

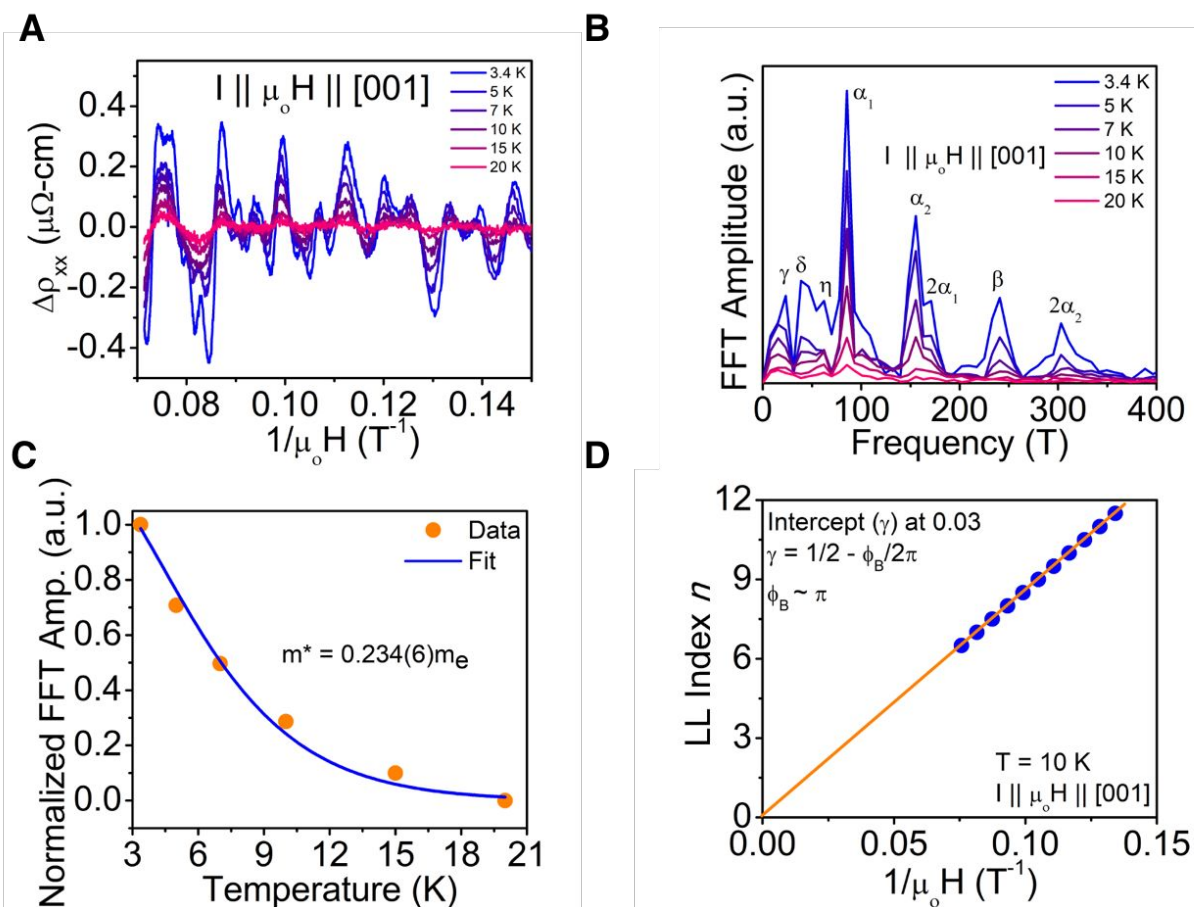


Figure 4: (A) Shubnikov de-Haas (SdH) oscillations for $\text{Ir}_2\text{In}_8\text{S}$ as a function of temperature. The oscillations remain clearly visible up to 15 K. (B) Fast Fourier Transform of the variable temperature SdH oscillations. Two predominant frequencies (F_{α_1} and F_{α_2}) can be seen at 86 and 155 T, with another F_{β} pocket at ~ 241 T. (C) Lifshitz-Kosevich (LK) fit of the decay amplitude for the F_{α_1} data, yielding effective mass $m^* = 0.234(6)m_e$. (D) Landau fan diagram from the 10 K SdH data for F_{α_1} .

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