

A Simultaneous X-ray Diffraction-Differential Scanning Calorimetry Study into the Phase Transitions of Mefenamic Acid

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Synopsis The phase transitions of mefenamic acid (MA) have been studied by synchrotron X-ray powder diffraction combined with differential scanning calorimetry. A direct transition from form I to forms II and III was noted, with the possibility of a new and as yet unidentified form observed during heating.

Abstract In this study, the polymorphic transitions of mefenamic acid (MA) were studied by synchrotron X-ray powder diffraction combined with differential scanning calorimetry (XRD-DSC). The initial material was found to be phase-pure form I. When this was heated, two endotherms were observed by DSC, at 162.72 and 219.55 °C. The former was found to correspond to a solid-solid enantiotropic transition from form I to a mixture of form II and III. The latter is the melting point of form II. As form I is heated, significantly greater unit cell expansion is seen in the a direction than in b and c, which can be explained because there are stronger intermolecular interactions in the bc plane. Refinements of the MA diffraction models against the patterns collected during heating revealed that at 175 °C there exists a mixture of forms I, II and III, while only forms II and III remain at 205 °C. However, reflections are observed at both temperatures which cannot be fitted with the known forms of MA. It is hypothesised that a new form of MA is formed upon heating. The stability of MA after the enantiotropic transition temperature is form II > III > I, which differs from the previously reported II > I > III.

Keywords: Synchrotron X-ray diffraction; differential scanning calorimetry; hyphenated techniques; mefenamic acid; polymorphic transition

1. Introduction

Most drugs are administered orally as solid materials. These must then dissolve in an aqueous environment (e.g. in the stomach and small intestine) to make an effective medicine – this is required for the drug to be absorbed by the body (Aulton & Taylor, 2013). However, up to 70% of new drug candidates have poor aqueous solubility and do not dissolve easily (Gaisford, 2013; Kawabata *et al.*, 2011; Morissette *et al.*, 2004). Accessing and stabilising more soluble solid forms of a drug can alleviate these problems, but systematic approaches to doing this remain highly elusive.

Mefenamic acid (MA, 2-[(2,3-dimethylphenyl)amino]benzoic acid) belongs to the non-steroidal anti-inflammatory class of drug (NSAIDs). It potently inhibits prostaglandin synthetase, primarily acting as an analgesic and anti-inflammatory agent (Cesur & Gokbel, 2008; Kato *et al.*, 2006). MA belongs to Class II of the Biopharmaceutical Classification System, having low solubility and high permeability, which hampers its use in the clinic (SeethaLekshmi & Guru Row, 2012). In addition, its high hygroscopicity and tendency to adhere to surfaces cause great problems during granulation and tableting (Otsuka, 2004). Therefore, extensive efforts have been made to alter MA's solubility and physical properties, for instance through crystal modification. To date, three polymorphs of MA have been reported: the crystal structures are given in Figure 1, and unit cell parameters are listed in Table 1. In all three systems the MA molecules pack as dimers and the dominant intermolecular forces in the structures are hydrogen bonds between the hydrogen on O2 of one molecule and O1 on the adjacent molecule ($O1 \cdots H-O2$; $\sim 1.68 \text{ \AA}$). In forms I and III, the dimer is further stabilised through $\pi \cdots H-C3$ interactions, while in form II, $\pi \cdots H-C4$, $\pi \cdots H-C5$ and $\pi \cdots \pi$ interactions support the dimer units. The phenyl ring carboxylic group and bridging amino group are coplanar, stabilised by an intramolecular $O1 \cdots H-N1$ hydrogen bond (Abbas *et al.*, 2017).

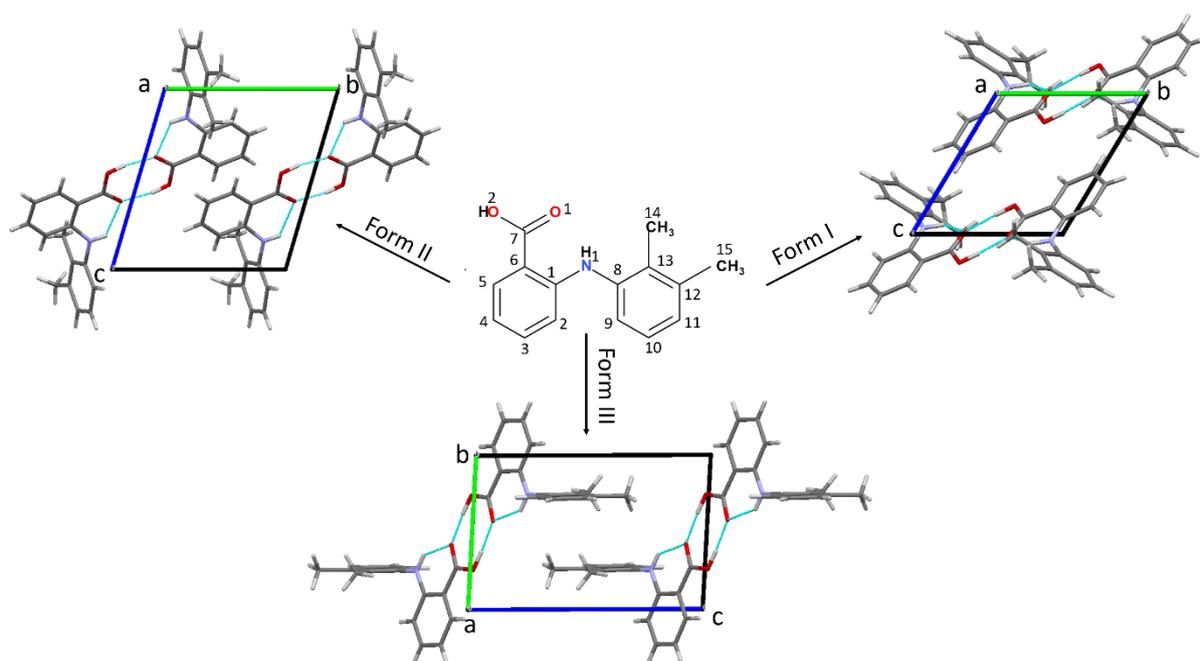


Figure 1

The molecular packing of MA forms I, II and III, viewed down the a-axis. Disorder present in the structure of form II is omitted for clarity.

Table 1

Crystallographic data for MA forms I, II and III.

Form	I^a	II	III
CSD ID	XYANAC	XYANAC02	XYANAC03
Crystal system	Triclinic	Triclinic	Triclinic
Space group	P-1	P-1	P-1
Temperature (K)	295	298	298
a/ Å	14.556	7.7584(5)	7.723(2)
b/ Å	6.811	9.2772(6)	7.934(10)
c/ Å	7.657	9.3991(4)	11.232(10)
α/ °	119.57	106.308(5)	83.590(10)
β/ °	103.93	91.847(4)	80.940(10)
γ/ °	91.3	101.865(5)	67.510(10)
Cell volume (Å³)	631.766	632.52	626.96
Density (g/cm³)	1.268	1.267	1.278
Z	2	2	2
Z'	1	1	1

^a SUs are not given in the CSD for form I.

MA form I is the commercially available form, and in the lab can be obtained by cooling a supersaturated acetone solution (Kato *et al.*, 2006). Form II is prepared by rapid cooling a supersaturated dimethyl formamide (DMF) solution (Kato *et al.*, 2006; SeethaLekshmi & Guru Row, 2012) or by slow evaporation of a chloroform solution in low humidity conditions (SeethaLekshmi & Guru Row, 2012). MA form III was obtained fortuitously during co-crystallisation, and can be prepared by slow evaporation of a MA DMF/methanol solution with adenine present as an additive (SeethaLekshmi & Guru Row, 2012). MA form II is reported to exhibit a higher solubility than form I in several solvents; however, being metastable it will slowly convert to form I at room temperature (Kato *et al.*, 2006). SeethaLekshmi and Guru Row (2012) determined that the relative stabilities of MA at ambient conditions lie in the order form I > II > III, which changes to form II > I > III at 175 °C and above.

In this work a novel method is employed to analyse the phase transitions of MA, combining high energy synchrotron X-ray powder diffraction with differential scanning calorimetry (XRD-DSC) (Figure 2). By using this technique, structural changes can be quantified via the XRD patterns, and phase transitions during heating and cooling observed by DSC. The technique allows for real time characterization, as both XRD and DSC data are obtained simultaneously on the same sample (Clout *et al.*, 2016). Clout *et al.* (2016) demonstrated that XRD-DSC experiments can be performed after simply modifying a laboratory DSC instrument by drilling holes through the furnace to allow a synchrotron X-ray beam to pass unhindered through the sample pan (Figure 2). The XRD-DSC approach has been applied to study phase transitions in a range of systems including carbamazepine (Clout *et al.*, 2018), paracetamol/lactose blends (Telford *et al.*, 2016), and spray-dried amorphous solid dispersions of olanzapine (Askin *et al.*, 2019). In this work, we make a further advance to the technique by dint of incorporating a controlled cooling facility into the experimental set-up.

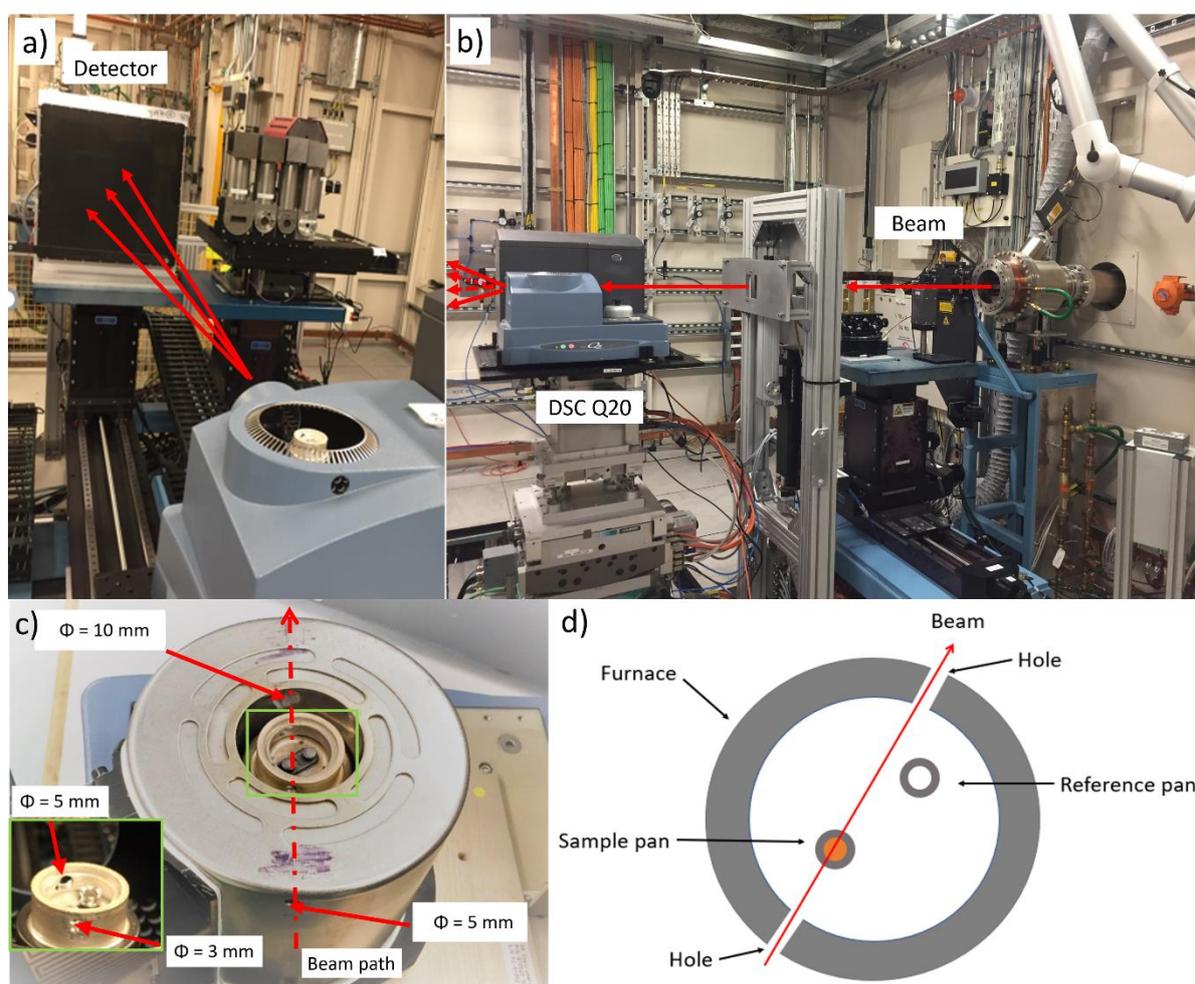


Figure 2

Images showing the XRD-DSC experimental approach. a) and b) The experimental set up for XRD-DSC experiments on Beamline I12 at the Diamond Light Source. The red lines indicate the path of the

X-ray beam; c) A digital photograph of the modified Q20 DSC (inset: furnace; main image: cooling unit); d) A schematic of the modified DSC furnace for XRD-DSC experiments

2. Materials and methods

MA (lot L102034, 98%) was obtained from Alfa Aesar and used as supplied. Standard laboratory XRD experiments were undertaken on a MiniFlex 600 diffractometer (Rigaku) supplied with Cu-K α radiation ($\lambda = 0.15418$ nm, 40 kV, 15 mA). MA samples were loaded in low volume glass holders and scanned from 5 to 50° in 0.02° steps at 2°/min. Nuclear magnetic resonance (NMR) experiments were performed on a Bruker Avance 400 MHz spectrometer, with DMSO-d₆ as the solvent.

Thermal analysis was carried out using a DSC (Q2000 calorimeter, TA Instruments). Nitrogen was used as the purge gas, with a flow rate of 50 mL/min. Three DSC methods were employed: in the first method, the sample was heated from 20 to 240 °C at 10 °C/min. In the second method the sample was heated from 20 to 180 °C at 10 °C/min, cooled to 0 °C and then reheated to 240 °C at the same heating rate. In the third method, MA was heated from 20 to 180 °C at 100 °C/min. Thermogravimetric analysis (TGA; Discovery TGA, TA Instruments) was also performed. Samples were heated from room temperature to 300 °C at 10 °C/min under a nitrogen purge at 25 ml/min. All temperatures are reported as onset temperatures. These are preferred because they usually stay unchanged with variations in heating rate or sample preparation method, which might cause broadening of the transition peaks resulting in the shift of peak temperature. The temperatures given for DSC data are thus the temperatures where polymorphic transitions begin, rather than when they are complete.

Simultaneous XRD-DSC experiments were carried out on the Joint Engineering, Environment and Processing Beamline I12 (JEEP) at the Diamond Light Source (Drakopoulos *et al.*, 2015). A Q20 DSC (TA Instruments) fitted with a refrigerated cooling system (RCS) attachment (TA Instruments) was mounted onto the sample stage in the experimental hutch, as depicted in Figure 2a and b. The DSC and RCS had both previously been modified to permit passage of the synchrotron beam. Holes were drilled in the cooler and furnace, with the entry holes being 5 mm (RCS) and 3 mm (furnace) in diameter and the exit holes 10 mm (RCS) and 5 mm (furnace) (see Figure 2c). The wavelength of the X-ray beam was monochromated to 0.234 Å (53.2 keV), and the beam diameter was 0.5 mm. The beam is almost parallel at the sample position. A Thales Pixium RF4343 detector was located 1.9 m behind the sample. The wavelength and sample-detector distance were calibrated with cerium dioxide (CeO₂) prior to experiments beginning, according to a previously reported calibration procedure (Hart *et al.*, 2013). CeO₂ was measured at two distances (L and L+100 mm). Since all the d-spacings of CeO₂ are known, the peak positions on the detector plate can be used to calibrate the wavelength and sample-detector distance. This calibration was undertaken using the DAWN Science Workbench (Basham *et al.*, 2015; Filik *et al.*, 2017). The DSC was calibrated with an indium standard.

For XRD-DSC experiments, MA samples of ca. 20 mg were loaded into crimped Tzero aluminium pans with pin-holed hermetic lids. The samples were loaded into the DSC furnace and heated from 20

°C to 240 °C at 10 °C/min, with nitrogen used as a purge gas. DSC data were recorded using the TA Advantage software and analysed in the TA Universal Analysis software. The Diamond Generic Data Acquisition (GDA) software was employed to collect diffraction patterns for 5 s, with a 1 s pause between each scan. This permits a diffraction pattern to be collected for every 1 °C increase in sample temperature.

XRD data were analysed as follows. The DAWN Science Workbench was used to mask and convert the 2D Pixium datasets into 1D diffraction patterns by azimuthal integration (Basham *et al.*, 2015; Filik *et al.*, 2017). The raw XRD data were then plotted as a function of temperature in contour plots using OriginPro 2017. TOPAS-Academic V5 was employed to analyse selected patterns with the Rietveld method (Rietveld, 1967; Rietveld, 1969) implemented within the software, in order to obtain realistic values for the unit cell parameters at elevated temperatures (Coelho *et al.*, 2012). Since synchrotron X-ray data were collected as 2D images, no zero point needed to be refined. Linear interpolation and subtraction were used for background refinement, and peak shapes were refined with pseudo-Voigt functions. For lattice parameters, least-squares fitting was employed and the intensities of calculated reflections iterated to best fit the observed data (McCusker *et al.*, 1999). If an XRD pattern contained more than one phase, the peak shapes for each phase were set to be the same and the phase fractions were refined.

3. Results

3.1. Physical properties of MA

MA was first characterised by XRD, TGA and DSC. The XRD profile confirms the material supplied to be form I (Supporting Information, Figure S1). According to the TGA profile (Figure 3), thermal degradation began at ca. 180°C. The DSC data for MA form I are also shown in Figure 3. Two endothermic peaks are visible with onsets at 162.72 and 219.55 °C, respectively corresponding to the polymorphic transition from form I to form II and the melting of form II. The DSC and TGA results together suggest that degradation occurs on melting, which has not been noted for MA previously.

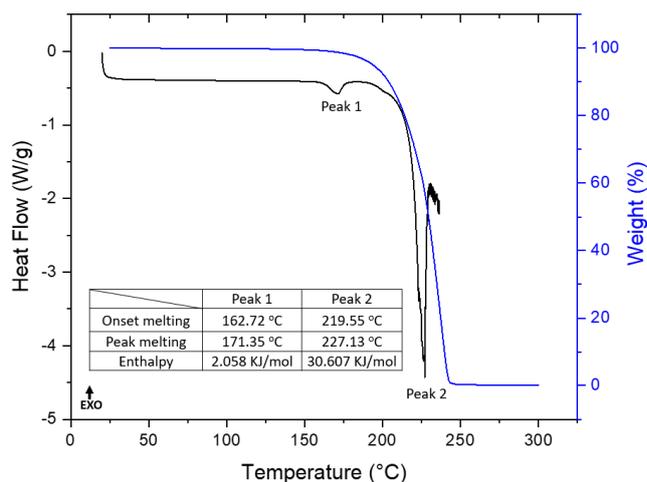


Figure 3

Overlay of the DSC and TGA results obtained on as-supplied MA (the enthalpy of the second endotherm in DSC is not accurate due to the degradation of MA).

Using a DSC heat-cool-heat cycle, MA was heated to a temperature just above the completion of the first endotherm, cooled down to 0 °C, and then reheated to 240 °C (see Figure 4). The results demonstrated that no recrystallisation happened during the reheating process (the peak at 15 °C is a baseline shift artefact), and the data obtained during the second heat present only an endothermic peak at 220 °C. This is the melting point of form II according to the literature (SeethaLekshmi & Guru Row, 2012). These observations indicate that after the first endothermic event, form I has been converted into form II. When increasing the heating rate to 100 °C/min (Figure S2), the DSC data show that the first (smaller) endothermic peak shifts to higher temperature, confirming it arises as a result of kinetic rather than thermodynamic events. Hence, the first endothermic transition can be confirmed as the enantiotropic transition from MA form I to form II.

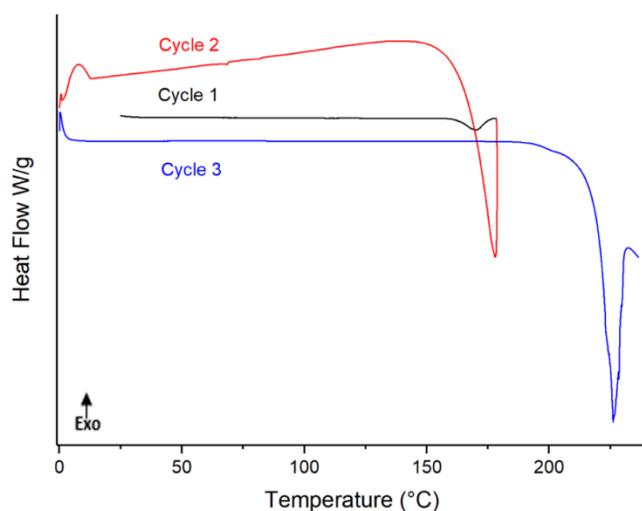


Figure 4

DSC data collected during heat-cool-heat cycles with MA (cycle 1 and cycle 3: heating, cycle 2: cooling).

3.2. XRD-DSC

XRD-DSC data for MA can be seen in Figure 5. Data collection began at 25 °C and no changes in the position of the Bragg reflections are seen until around 160 °C, at which point a second crystalline phase appears. Both the initial and the second crystalline phase coexist from 160 °C to 179 °C, when the MA is completely converted to the latter. Slight changes in the diffraction pattern are observed between 200 to 220 °C, which may be because MA begins to melt in this temperature range. The DSC trace shows a small endothermic event with an onset at 163.47 °C coinciding with the polymorphic transition. Following this, there is a much larger endothermic peak at 220 °C. The latter results from the melting of form II, as is evident from the total lack of diffraction intensity above this temperature.

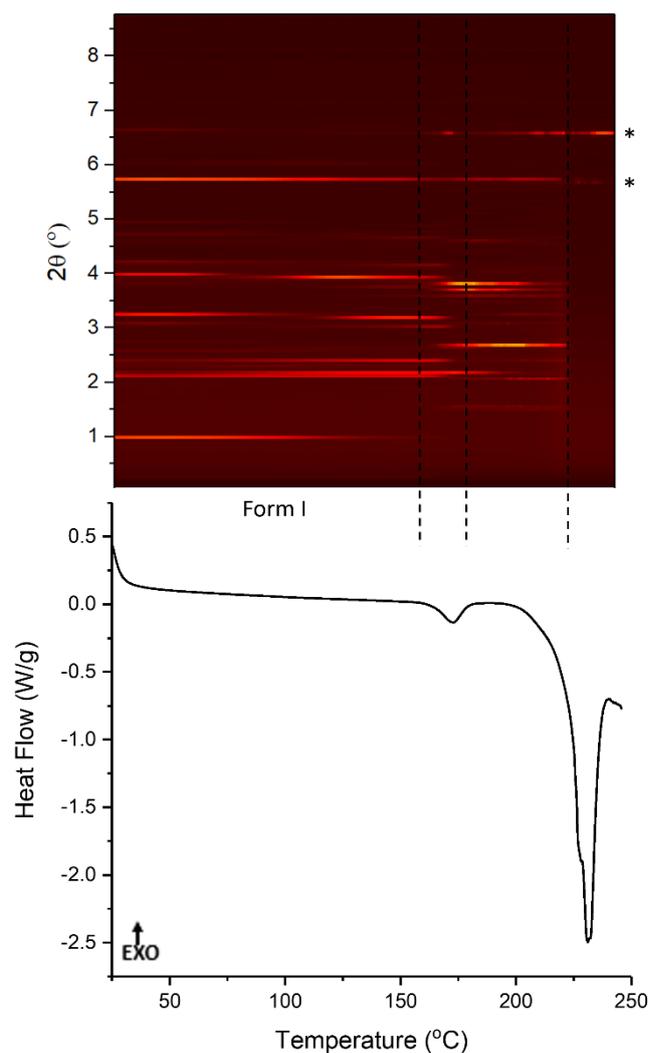
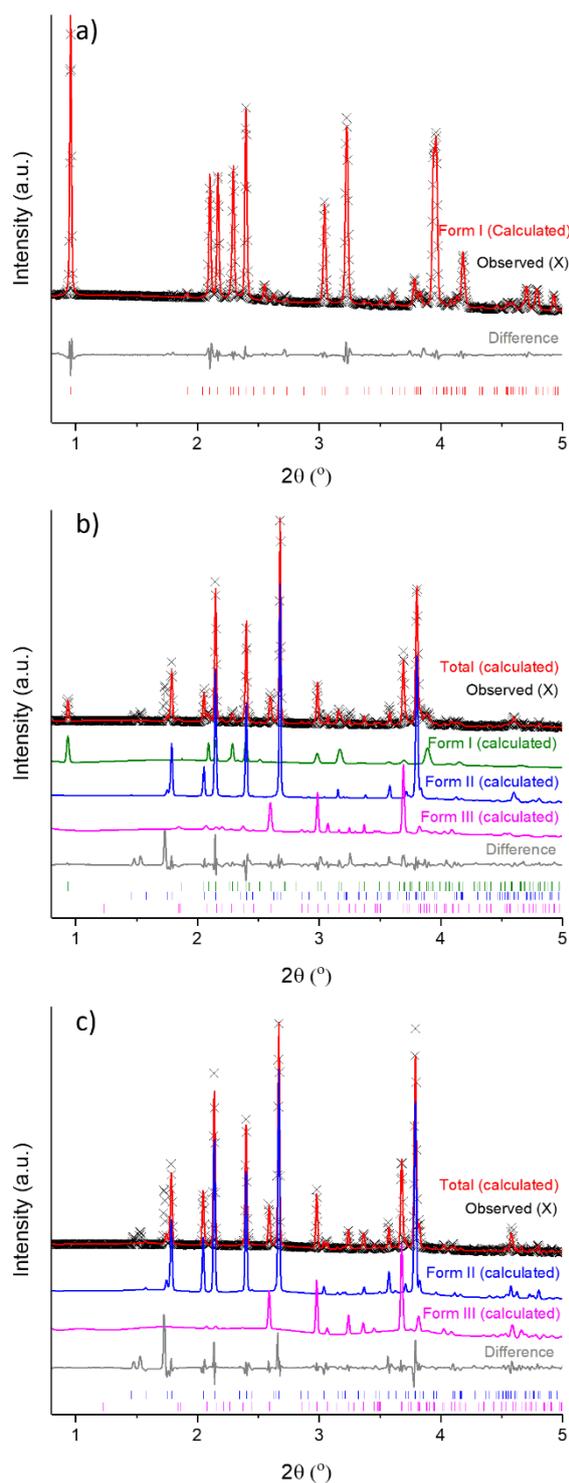


Figure 5

XRD-DSC data for MA (reflections marked * are a result of the aluminium pan).

Rietveld refinement was carried out using the structure models XYANAC (form I) (McConnell *et al.*, 1976), XYANAC02 (form II) (Lee *et al.*, 2006) and XYANAC03 (form III) (SeethaLekshmi & Guru Row, 2012) from the Cambridge Structural Database (CSD). The refinement against the initial material at 25 °C confirms that the sample is phase-pure form I (Figure 6a, Table 2). Rietveld refinement of the pattern recorded at 175 °C, after the profile change, shows that the diffraction pattern contains features corresponding to each of forms I, II and III (Figure 6b, Table 2). At 205 °C, the sample appears to contain form II as well as some reflections from form III (Figure 6c, Table 2). The R_{wp} (weight profile R-factor) values of the last two refinements are relatively high, because some of the observed peaks cannot be fitted.

**Figure 6**

Rietveld refinement against diffraction patterns recorded for MA during heating at 10 °C/min. Data are shown at a) 25 °C (tick marks show the position of allowed reflections of MA form I); b) 175 °C (tick marks show the position of allowed reflections of MA form I (upper), form II (middle) and form III (lower)); c) 205 °C (tick marks show the position of allowed reflections of MA form II (upper) and form III (lower)).

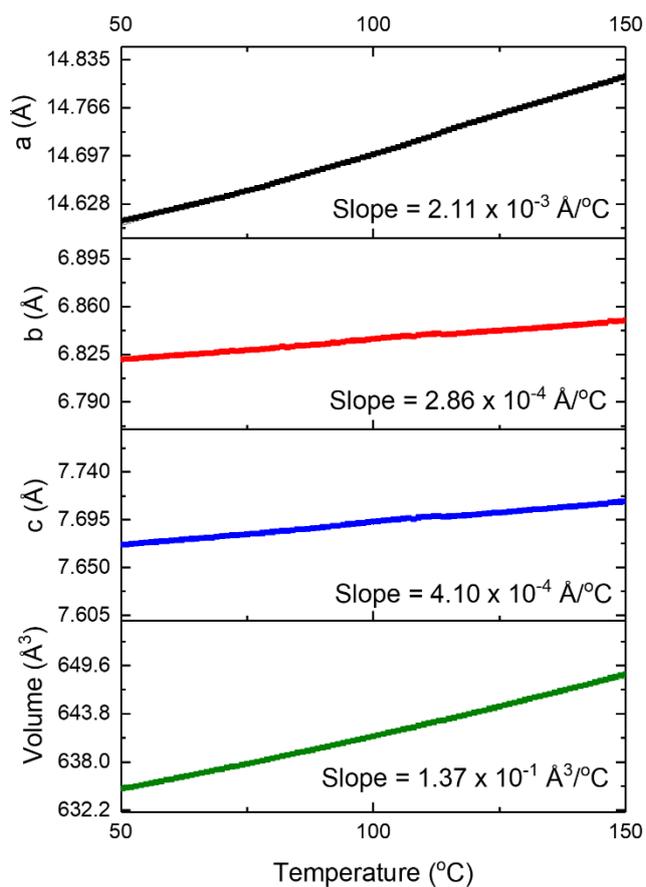
Table 2

Refinement parameters for MA form I, form II and form III. The starting models were taken from the CSD (Form I: XYANAC, Form II: XYANAC02, Form III: XYANAC03).

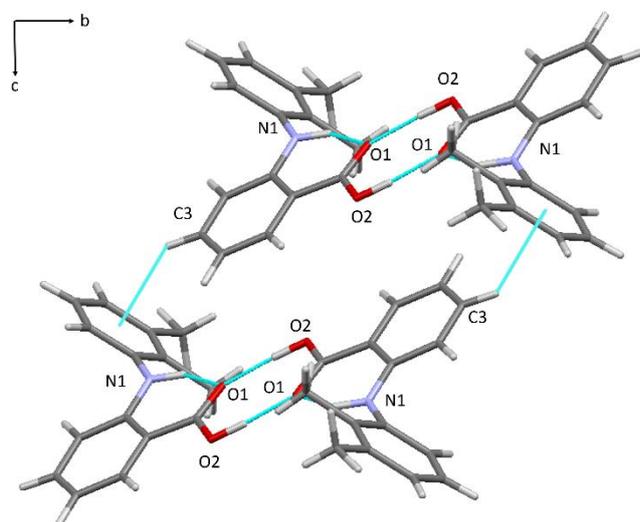
T/ °C	25	175			205	
Space group	Form I	Form I	Form II	Form III	Form II	Form III
	P-1	P-1	P-1	P-1	P-1	P-1
a/ Å	14.626(1)	14.802(10)	7.8857(19)	7.9059(51)	7.9115(11)	7.8667(32)
b/ Å	6.8245(3)	6.8607(5)	9.1451(25)	7.8927(24)	9.1585(11)	7.9037(15)
c/ Å	7.6766(4)	7.6694(5)	9.6664(35)	11.098(5)	9.6889(16)	11.131(3)
α / °	119.77(0)	120.44(4)	106.83(2)	82.760(40)	106.79(1)	83.218(22)
β / °	104.07(0)	103.28(6)	92.383(18)	79.374(46)	92.599(9)	79.875(21)
γ / °	91.007(5)	91.200(2)	102.69(3)	67.148(37)	102.70(1)	67.417(21)
R _{wp}	5.0251		13.307		13.633	
Phase fraction ^a	99.0% ^b	12.2% ^c	64.0% ^c	23.7% ^c	72.1% ^c	25.9% ^c

^aThe representative error of the phase fractions cannot be calculated because of the graininess of the sample. ^bThe purity of MA is 99%, which results in the phase fraction of MA form I at 25 °C being less than 100%. ^cThe total phase fraction does not reach 100% at elevated temperatures because the refinement cannot perfectly fit the observed patterns.

A number of the MA form I reflections shift to lower 2θ angles as the material is heated from 25 to 150 °C (Figure 5). This is caused by unit cell expansion (Figure 7). The quality of the data collected permitted batch refinements to be performed against all the patterns collected. The unit cell parameters could be extracted from these and plotted as a function of temperature (Figure 7). It is clear that the a dimension expands significantly more than either the b or c dimensions; the per degree increase in temperature is 7- and 5-fold greater than that in b and c respectively. The dominant intermolecular forces in the structure of form I are hydrogen bonds between the hydrogen on O2 and O1 of an adjacent molecule (O1...H—O2; Figure 8). These comprise the only intermolecular hydrogen bonding throughout the structure, with an H...O distance of 1.68 Å (Abbas *et al.*, 2017). This holds the MA molecules together in dimers, and adjacent dimers are stabilised through π ...H—C3 interactions (2.77 Å). When viewed in the bc plane, it is clear that these interactions stabilise the structure in both the b and c directions. While the intermolecular H-bond offers some support in the a direction, this is relatively minimal. As a result, the thermal unit cell expansion is more distinct in the a dimension than in b and c.

**Figure 7**

Lattice parameters and unit cell volume as a function of temperature for MA form I.

**Figure 8**

Molecular interactions in the *bc* plane of MA form I, showing O \cdots H—O intermolecular bonding, N \cdots H—O intramolecular bonding and $\pi\cdots$ H—C interactions.

The integrated total diffraction intensity for each pattern as a function of temperature is plotted in Figure 9. At about 150 °C the amount of form I begins to decrease, and rapid growth of form II and III is seen. This suggests that form I transforms to forms II and III during the first endothermic peak. The rate of change is almost linear from 150 °C until around 175 °C, at which point there is no more form I present. The trend line of the most linear part of two curves was analysed (from 169 to 174 °C) and a decline in form I of -5.41 °C^{-1} , and growth of form II/III at 6.43 °C^{-1} was seen. The similarity between these two numbers indicates the conversion is a direct transition between the polymorphs. The integrated data were converted to phase fractions and plotted as a function of temperature (Figure 9, inset). The two curves cross at around half of the maximum quantity of either of the two phases, indicating that the crystal structure does not transit via a complete melt of the sample, but rather there is a solid-solid phase transition. Overall, the data suggest that the small endothermic peak at 163.47 °C is likely to be a solid-solid enantiotropic transition from MA form I to forms II and III. After the transition point, the stability of the MA polymorphs lies in the order form II > III > I. This differs to that previously reported series, which found II > I > III (SeethaLekshmi & Guru Row, 2012). The reason behind this difference is not clear, but SeethaLekshmi found that the difference in energy between form I and III is small at elevated temperatures; hence, it might be that the small changes in sample environment and experimental conditions between our two studies are enough to cause the stability order to alter.

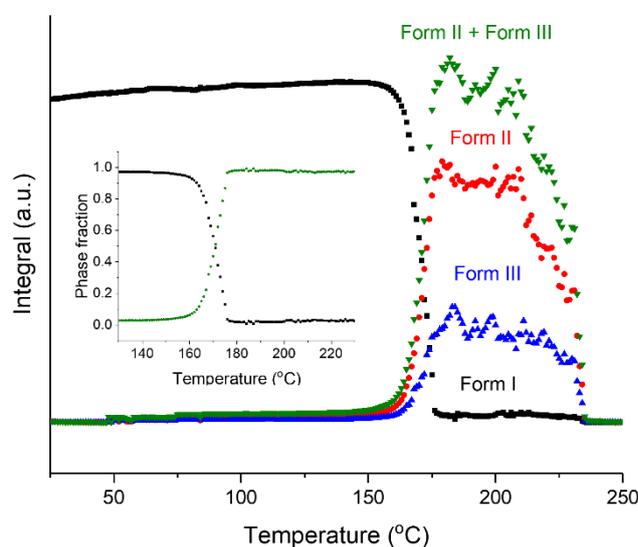


Figure 9

Plot of integrated total diffraction intensity for MA form I, II and III as a function of temperature, with inset plot of phase fraction as a function of temperature.

MA has three reported polymorphic forms, but the patterns collected at 175 and 205 °C cannot be perfectly fitted with the known polymorphs. Looking at the data from 205 °C (Figure 6c), form II accounts for most reflections in the observed pattern, but there are unmatched peaks at 1.47, 1.53, 1.72, 2.58, 2.97 and 3.67°. The form III fit is rather poor, but form III does match some observed reflections which cannot be attributed to form II (2.58, 2.97 and 3.67°). Additional reflections at 1.47, 1.53 and 1.72° cannot be matched with any of form I, II or III, however. We hypothesised that these could correspond either to degradation products, or to a new polymorph of MA. To investigate the former, MA samples were heated to 175 or 205 °C following the same heating protocol as for XRD-DSC, cooled to room temperature, and analysed by ¹H NMR (Figure S3). The resonances observed in the NMR spectra before and after heating are identical. Thus, although degradation does occur upon heating (see Figure 3), the products of this process must be volatile and cannot account for the additional reflections. Since degradation is ruled out, the additional reflections must arise from a new polymorph of MA. In an attempt identify this, predicted crystal structures were explored (Case *et al.*, 2018). None of the low-energy structures provide a good match to the unassigned reflections (Figure S4). It thus appears that an as-yet unknown polymorph of MA was generated during heating. This is highly unstable: when MA is heated to 175 or 205 °C in the DSC, cooled to room temperature, and analysed by XRD, the patterns obtained match perfectly with form II (Figure S5). The new polymorph therefore quickly converts to form II when the temperature is reduced.

4. Conclusions

Combined synchrotron X-ray powder diffraction - differential scanning calorimetry (XRD-DSC) experiments undertaken in this work permitted new insight into the polymorphic transitions of mefenamic acid (MA) to be obtained. When commercially sourced MA form I was heated, two endotherms were observed, at 162.72 and 219.55 °C. The former corresponds to an enantiotropic transition between forms I and II/III, while the latter is the melting of form II. As form I is heated, the unit cell expands, and we observe significantly greater unit cell growth in a than in b and c. This is a result of the bc plane being supported by H-bonding. The conversion between form I and II/III appears to occur via a solid-solid pathway, with no wholesale melting observed. At temperatures above the enantiotropic transition point, the diffraction patterns contain reflections which cannot be matched with any of the known forms of MA, suggesting the existence of an as-yet unknown polymorph.

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Supporting Information

Table S1 Crystallographic data for computationally predicted MA polymorphs, taken from Case *et al.* 2018.

Structure ID	M318	M497	M664	M1069	M1090	M3912
Space group	C2/c	P-1	Pbca	Pbca	Pbca	P-1
a/ Å	24.6	8.1	23.0	23.1	23.2	14.3
b/ Å	6.9	8.5	16.1	15.6	15.6	28.2
c/ Å	16.3	9.9	7.0	7.1	7.1	4.3
α /°	90	70.1	90	90	90	152.7
β /°	67.6	96.1	90	90	90	54.9
γ /°	90	82.7	90	90	90	125.5
Cell volume (Å ³)	2558	627.48	2592.1	2558.56	2569.63	639.342

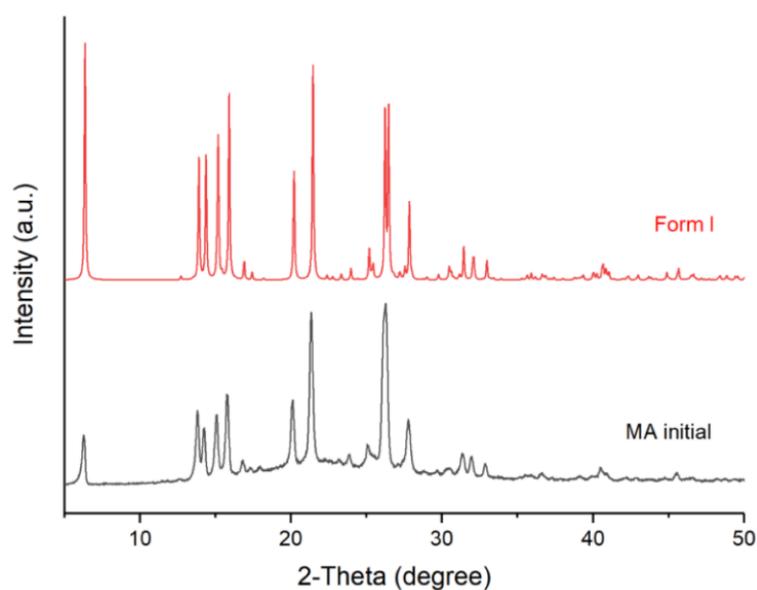
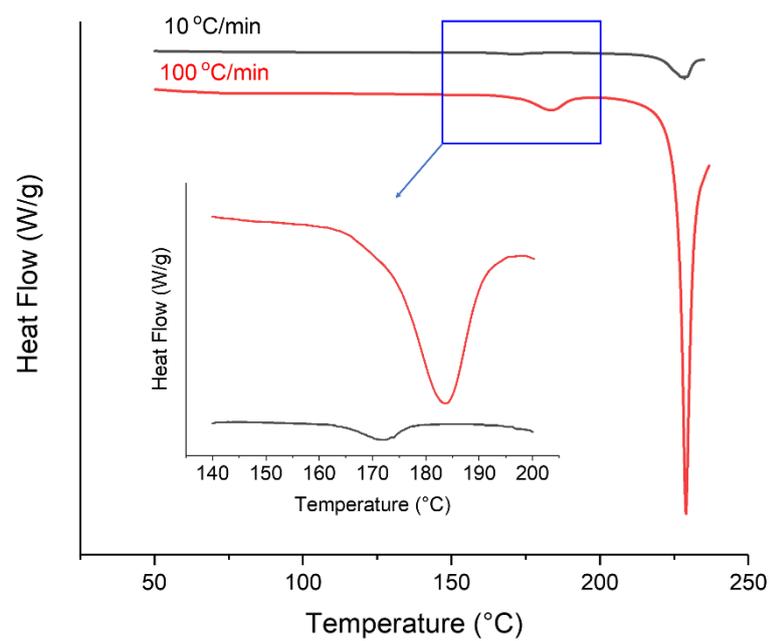
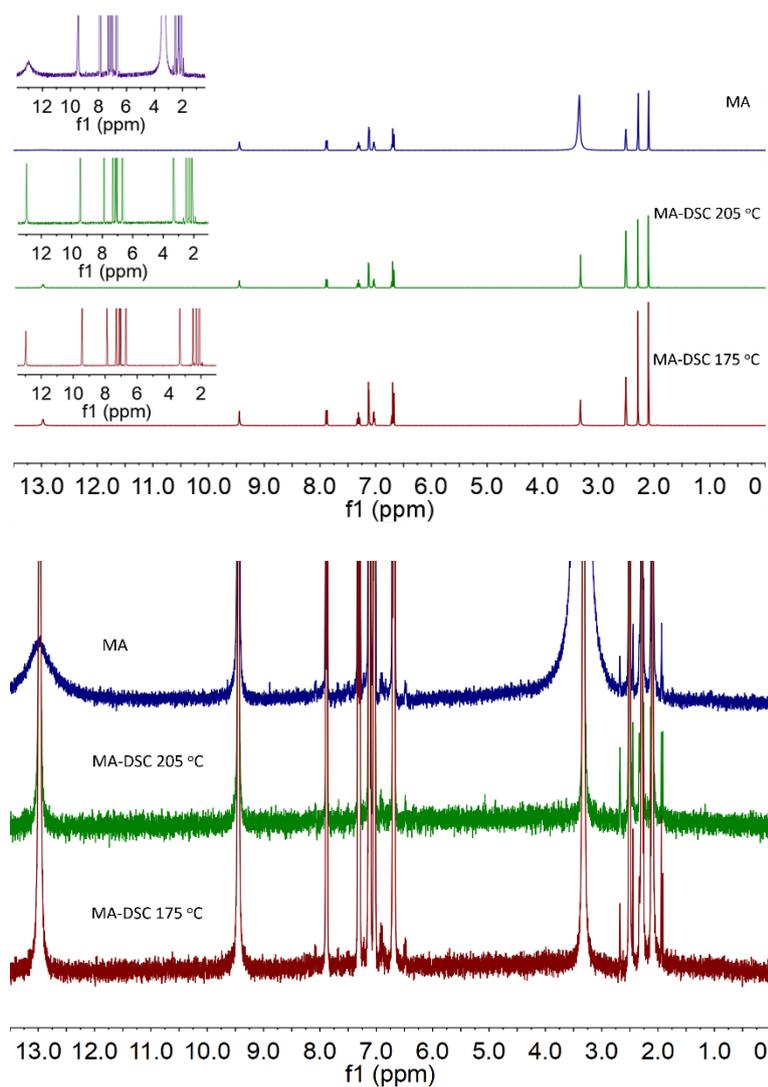


Figure S1

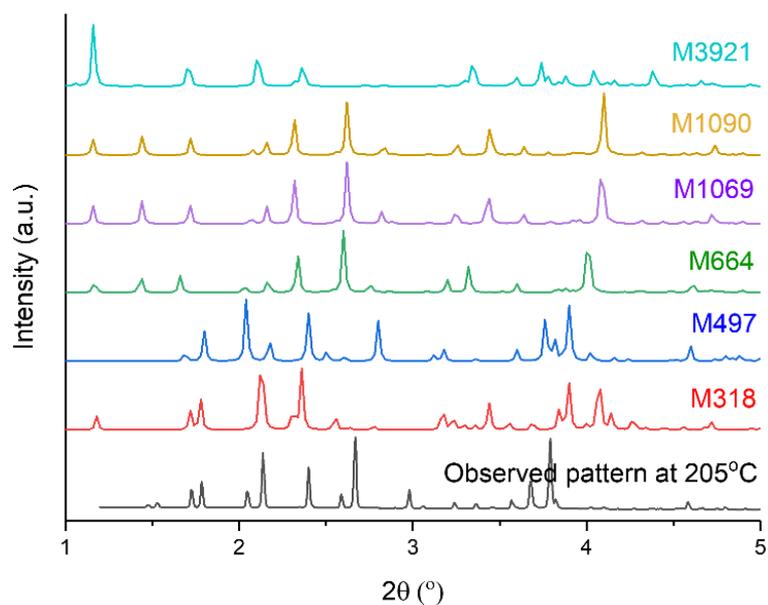
The XRD patterns of as-supplied MA and that of MA form I (XYANAC) calculated from the CSD (wavelength: 1.5418 Å).

**Figure S2**

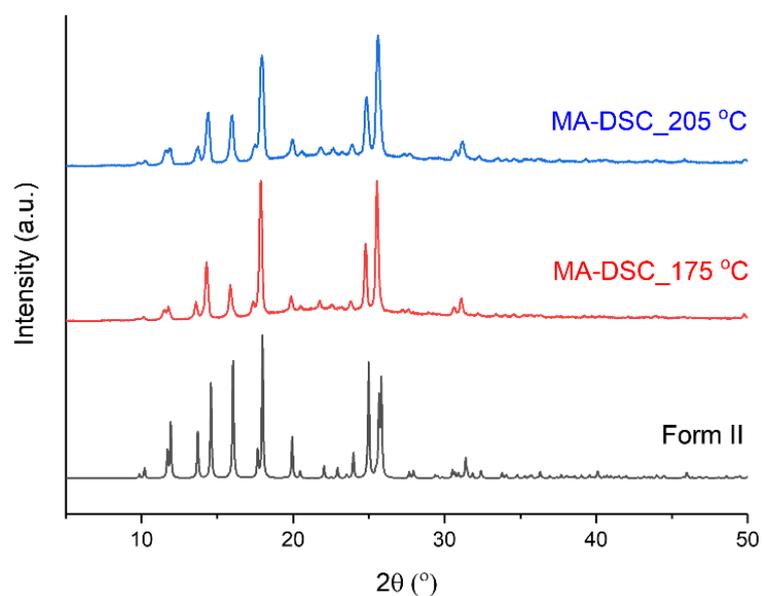
DSC profiles of MA heated from 20 to 230 °C at 10 and 100 °C/min. Exo up.

**Figure S3**

NMR spectra of MA heated to 175 and 205 °C and that of as-supplied MA. The peak at 13 ppm corresponds to the MA COOH group; this is hard to see in the as-supplied MA sample, but can be observed upon enlarging the spectrum.

**Figure S4**

The XRD pattern of MA observed at 205 °C, together with computationally predicted patterns taken from Case *et al.* 2018 (wavelength: 0.234 Å). Distinct reflections at 1.47 and 1.53° cannot be matched with any of form I, II or III; it is clear that none of the predicted patterns fits these either.

**Figure S5**

XRD patterns of MA form II (XYANAC02) calculated from the CSD (wavelength: 1.5418 Å), together with that of MA heated in DSC to 175 and 205 °C.