

Magnetically Coated Bioabsorbable Stents for Renormalization of Arterial Vessel Walls after Stent Implantation

J.S. Lee¹, P. Han², H. Song¹, D. Kim^{1,6}, H. Lee², M. Labowsky⁷, S. Ylä-Herttuala⁸, J. Hytönen⁸, M. Gülcher⁹, A.J. Sinusas^{4,5}, J. Martin⁵, A. Mathur¹⁰, T.M. Fahmy^{1,2,3,*}.

Departments of ¹Biomedical Engineering, ²Chemical and Environmental Engineering, ³Immunobiology, ⁴Internal Medicine, and ⁵Cardiovascular Medicine, Yale University, New Haven, CT 06511, USA.

⁶Department of Pharmaceutical Sciences, Irma Lerma Rangel College of Pharmacy, Texas A&M Health Science Center, College Station, TX 77843, USA.

⁷Ansama Research LLC, Wayne, NJ 07470, USA.

⁸A.I. Virtanen Institute for Molecular Sciences, University of Eastern Finland, Kuopio, 70210, Finland.

⁹QualiMed Innovative Medizinprodukte GmbH, Winsen, 21423, Germany.

¹⁰Department of Clinical Pharmacology, William Harvey Research Institute, Queen Mary University of London, London E1 4NS, UK.

Abstract

The insertion of a stent in diseased arteries is a common endovascular procedure that can be compromised by the development of short and long-term inflammatory responses leading to restenosis and thrombosis respectively. While treatment with drugs, either systemic or localized, have decreased the incidence of restenosis and thrombosis these complications persist and are associated with a high mortality in those that present with stent thrombosis. We reasoned that if stents could be made to undergo accelerated endothelialization in the deployed region, then such an approach would further decrease the occurrence of stent thrombosis and restenosis thereby improving clinical outcomes. Towards that objective, the first step necessitated efficient capture of progenitor stem cells, which eventually would become the new endothelium. To achieve this objective, we engineered intrinsic ferromagnetism within non-magnetizable, biodegradable Magnesium (Mg) bare metal stents. Mg stents were coated with biodegradable polylactide (PLA) polymer embedding magnetizable iron-Platinum (FePt) alloy nanoparticles, Nano-Magnetic[®] particles-ⁿMags, which increased the surface area and hence magnetization of the stent. ⁿMags uniformly distributed on stents enabled capture, under flow, up to 50 ml/min, of systemically injected iron oxide (IO) labeled progenitor stem cells. Critical parameters enhancing capture efficiency were optimized and we demonstrated the generality of the approach by showing that ⁿMag-coated stents can capture different cell types. Our work is a potential paradigm shift in engineering stents because implants are rendered as tissue in the body and this, “natural stealthiness” reduce or eliminate issues associated with pro-inflammatory immune responses post-implantation.

Keywords: ferromagnetic, paramagnetic, drug-eluting stent, nanoparticles, cell therapy, cardiovascular disease, Iron-Platinum, magnetization.

Introduction

In 2016, more than 0.75 million stents were implanted worldwide to remedy atherosclerotic cardiovascular diseases and over 0.33 million in the United States alone¹. In-stent restenosis, however, is a frequent unwanted side effect originating with an inflammatory immune response that attempts to restore immune competence and limit damage in the affected area post foreign material deployment². In addition, because of “tissue scarring”, smooth muscle cells proliferate abnormally resulting in narrowing of the vascular lumen around the inserted stent. Conventional treatment for these complications entail the use of anti-mitotic agents coating the stent to prevent restenosis and anti-platelet medications such as Clopidogrel (Plavix), Prasugrel (Effient), Ticagrelor (Brilinta), Ticlopidine (Ticlid) to prevent thrombosis. Use of these medications can result in unwanted side effects, and are rendered less than optimally effective due to patient non-compliance to the drug regimen.^{3, 4} For those reasons, drug eluting stents have raised hopes since they provision a drug locally to the affected region, minimizing side effects, and in a sustained manner, facilitating automatic patient compliance.⁵ However, the main complication have been the long-term effects such as late thrombosis and delayed endothelialization which, collectively, may result in myocardial infarctions.⁶ For example, it has been reported that stent thrombosis develops in a significant fraction of patients implanted with Drug Eluting Stents⁷ leading to delayed arterial healing.^{8, 9} Thus, to achieve long-term protection, a new methodology is needed that inhibits restenosis in the short-term and abrogates long term complications like thrombosis. A methodology that renormalizes the vasculature after stent implantation would serve this objective and such an approach would need to begin with a means to achieve rapid and efficient seeding of the affected area with endothelial progenitor stem cells.

Stent endothelialization here refers to capture, growth or proliferation of a fresh layer of endothelial cells on the stent without abrogating its mechanical function.¹⁰ This can be achieved using administered endothelial stem cell progenitors.¹¹ This hypothesis is supported by prevailing thoughts that delivery and localization of progenitor stem cells to the stent implant site may quickly regenerate the endothelium and prevent narrowing of the vascular lumen.¹² Early efforts aimed at rapid endothelialization involved precoating stents with cultured cells or capturing with ligands or antibodies matured endothelial cells intravenously inoculated allowing for the attachment of those cells and their subsequent differentiation and proliferation.^{13, 14}

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3 Pioneering work on endothelialization was performed as early as the late 1970s, when Herring *et*
4 *al.* first reported on autologous endothelial cell seeding.¹⁵ Endothelial cell seeding was shown to
5 be effective in reducing graft thrombogenicity and intimal hyperplasia in large animal models¹⁶
6 and promising results were reported by Ortenwall in human trials.¹⁷ These techniques, however,
7 are still facing inefficiency in cell retention and non-specific capture of inflammatory cells
8 within the target site.¹⁸

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10 A potential strategy for achieving complete coverage of the implanted stent with a
11 functional endothelial layer is to provide rapid, uniform, and specific cell recruitment using
12 external forces guided, for example, by force fields such as magnetic fields. This technology
13 relies on rendering cells magnetically responsive, with methods such as labeling with Iron Oxide
14 nano or microparticles. The feasibility of this approach has been demonstrated by several groups
15 showing IO labeled cells can be targeted to paramagnetic medical devices using locally/remotely
16 applied magnetic fields. Paramagnetic stents (i.e; instantly magnetized in the presence of a
17 strong magnetic flux, and demagnetized instantly in its absence), are limited in their application
18 by the insufficient magnetic force gradients experienced by deeper areas of the body, such as
19 coronary vessels where stenting may take place. Furthermore, the inconvenient aspect (a large
20 magnet needs to be available for the procedure) makes this a less attractive option for outpatient
21 or point-of-care clinical settings, which are often not equipped with sophisticated MRI or large
22 magnetic flux equipment.

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24 An optional methodology is to introduce an intrinsic magnetic moment within the stent,
25 i.e., ferromagnetic. Pislaru *et al.* tried this approach using stainless steel interspaced with
26 synthetic vascular grafts. Others, such as Uthamaraj *et al.*, used stainless steel stents rendered
27 ferromagnetic to capture IO-labeled endothelial cells.¹⁹ Stainless steel, however, is not
28 biodegradable and thus require long-term drug therapy with antibiotics or antiplatelet therapy. to
29 minimize complications. Further, non-bioabsorbable stents prevent lumen expansion, impair
30 vessel geometry and often lead to obstruction of blood vessel side branches.²⁰

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32 Magnesium bare metal stents, on the other hand, are bioabsorbable. In addition, Mg
33 introduces additional features which make it a very attractive choice for stent manufacture.
34 (Table 1)²¹ First, the use of polymeric coatings such as PolyLactide acid (PLA), or Polylactic-
35 co-glycolic (PLGA) results in release of acidic monomers during the degradation process

lowering the pH in and around the sten. An acidic microenvironment will impede cell survival and thus the development of a functional endothelial cell layer. Mg is known to impart an acid neutralizing effect, which is why several antacid and gastric reflux medications such as Mylanta and Milk of Magnesia have Mg as the active ingredient.^{22, 23} Mg will therefore improve cell viability in the presence of degradable polymer coatings.²⁴ Second, Mg has anti-inflammatory immune properties, leading to, for example reduction in IL-6 and IL-13.^{25, 26} Third, Mg alloys show excellent anti-platelet deposition and low thrombogenicity.^{27, 28} Applications in the field of bone regenerative medicine, for example, show that Mg screws can promote human bone marrow stem cell (hBMSC) differentiation and proliferation²⁹ and the same for Mg scaffolds in bone regeneration. Finally, there is evidence of enhanced antioxidant enzyme activity following Mg supplementation.³⁰ Table 1 below, summarizes these features and properties associated with the use of Mg. These properties together with the biodegradability aspect of the material and the engineering of an intrinsically magnetic moment thus offer an unprecedented stent technology that facilitates renormalization of vascularity after stent implantation.

Table 1. Summary of Features Associated with use of Mg as the stent base material

Property	Beneficial Impact	Ref.
Bioabsorbability	Minimizes host foreign body response, endothelial cell dysfunction and thus reduces or abrogates the need for long-term drug regimens post stent deployment.	7
Acid-neutralization	Improves cell viability in the presence of degrading polymeric (PLA or PLGA) coatings.	22-24
Anti-inflammation	Reduces release of pro-inflammatory cytokines such as IL-6 and IL-13. Reduced fibrosis and platelet deposition and thus lowering potential thrombogenicity	25-28
Anti-Oxidant	Reduced oxidative stress and lowering of free radicals	30

We developed a novel bioabsorbable ferromagnetic stent as a platform to recruit cells tagged with IOs. We note that IO labelling of cells is an already FDA approved, used for tracking cells by MRI.³¹ and commercial kits are available that tag human cells with IO for separation and concentration. Our technology combines the use of conventional bioabsorbable Mg metal stents with an ⁿMg-embedded PLA polymer coating (ⁿMg stent) yielding a uniform

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stent magnetic moment that was capable of attracting cells under flow. Because, Mg has been used to fabricate stents with minimal degree of thromboembolic events and inflammation,²⁸ we hypothesized that ⁿMag-Mg stents will capture cells and cover the stent with progenitor stem cells and that effect will enhance stent patency and reduce or eliminate complications over the long-term.

Results and Discussion

The magnetic force due to surface incorporated ⁿMag: Rationale for use of nanoparticles and choice of FePt as the magnetizable alloy (ⁿMags). The intrinsic magnetic moment of a stent would need to be sufficiently strong to capture cells flowing within distances that may exceed several diameters of the stent itself. We reasoned that a strong magnetic field can be created by coupling magnetic nanoparticles to the Mg metallic coronary stents (3 mm diameter, D and 18 mm length, L) because the magnetic flux density is directly proportional to the surface area of the magnet according to eq. 1, a higher magnetic force can be introduced by coating with nanoparticles.

$$F_{coated\ stent} = N_{nMags} \frac{B^2 A}{2\mu_0} \quad (1)$$

F = magnetic force imparted by stent, B = flux density, A = surface area, μ_0 = permeability of the medium, N_{nMags} = number of beads coating the stent, typically several thousand per stent.

For 7000 particles, for example, and assuming a flux density that is unchanged after coating the stent, the ratio of magnetic force of an uncoated to coated would be:

$$\frac{F_{coated}}{F_{uncoated}} = 7000 (4\pi r^2) \quad (2)$$

which is a significant increase in the magnetic field strength solely based on the incorporation of magnetic nanoparticles on the surface.

Diameter or radius of the particles is thus critical for magnetization force. On the nanoscale, Fe alone does not form ordered magnetizable crystals. Hence doping with Pt gives rise to a stable face centered tetragonal (fct) crystal with high magnetic anisotropy.³²⁻³⁴ The choice of FePt alloy was thus based on achieving a stable crystal structure alignment of the magnetic dipoles.^{32 35}

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5 **Synthesis and characterization of ⁿMags.** Monodisperse FePt nanoparticles (NPs) were
6 prepared by chemical reduction of platinum(II) acetylacetonate (Pt(acac)₂) and iron(III)
7 acetylacetonate (Fe(acac)₃) (Figure 1a).³⁶ We utilized transmission electron microscopy (TEM)
8 and high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM)
9 to show FePt particles of approximately 1-3 nm diameter (Figure 1d) and co-localization of the
10 Fe and Pt elements within individual nanoparticles (Figure 1b). The FePt particles have a
11 chemically disordered face-centered cubic (fcc) structure and can be transformed into an ordered
12 fct structure after proper thermal treatment (annealing) to yield stable ferromagnetic properties.
13 Heat treatment is known to lead to sintering and agglomeration of NPs thus broadening the size
14 distribution and increasing particle size (micrometers) (Figure 1d). To reduce thermally-induced
15 coalescence of FePt NPs, a silicon dioxide coating (SiO₂ has a high-melting-point, T_m: 1713 °C)
16 was used to protect FePt NPs prior to heat treatment.³⁷

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27 The compositionally ordered L10 phase of FePt NP was formed when SiO₂ coated FePt
28 NPs were annealed under forming gas (93% N₂+7% H₂) at 650 °C. The SiO₂ layer reduced
29 thermal sintering of FePt particles, yielding a mean size of 200 nm for coated FePt NP (Figure
30 1g). Particle sizes in this range are amenable to dispersion in organic solvents, which is critically
31 required for uniform coating of stents with the particles. After annealing, the silica layer was
32 completely removed by washing with HF solution (1 %). Within 1 min after start of the
33 annealing process, a phase transformation can be observed using X-ray diffraction. This phase
34 change imparts the ferromagnetic properties of FePt (Figure 1f).

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41 To assess the magnetic character of the particles, a superconducting quantum interference
42 device (SQUID) was utilized to derive in-plane magnetic hysteresis of the particles (Figures 1c
43 and 1e). SQUID testing showed a high uniaxial magnetocrystalline anisotropy which is
44 responsible for the measured coercivity of approximately 1.0 T post-annealing. Henceforth, the
45 magnetic FePt particles of size 300 nm and an average coercivity of 1.0 T were termed, “ⁿMags”,
46 to reflect their nanoscale ferromagnetic properties. To induce permanent ferromagnetization,
47 ⁿMags were placed in a 1.5 T magnetic field solenoid overnight. We note, that this solenoid is the
48 same as that typically used for human MRI tests. We also observed that magnetization can be
49 induced in less than an hour in the same magnet (data not shown), after which the characteristic,
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dispersible black particle powder agglomerated into clusters as a result of the magnetization of individual particles which then attracted each other (Figure 1h).

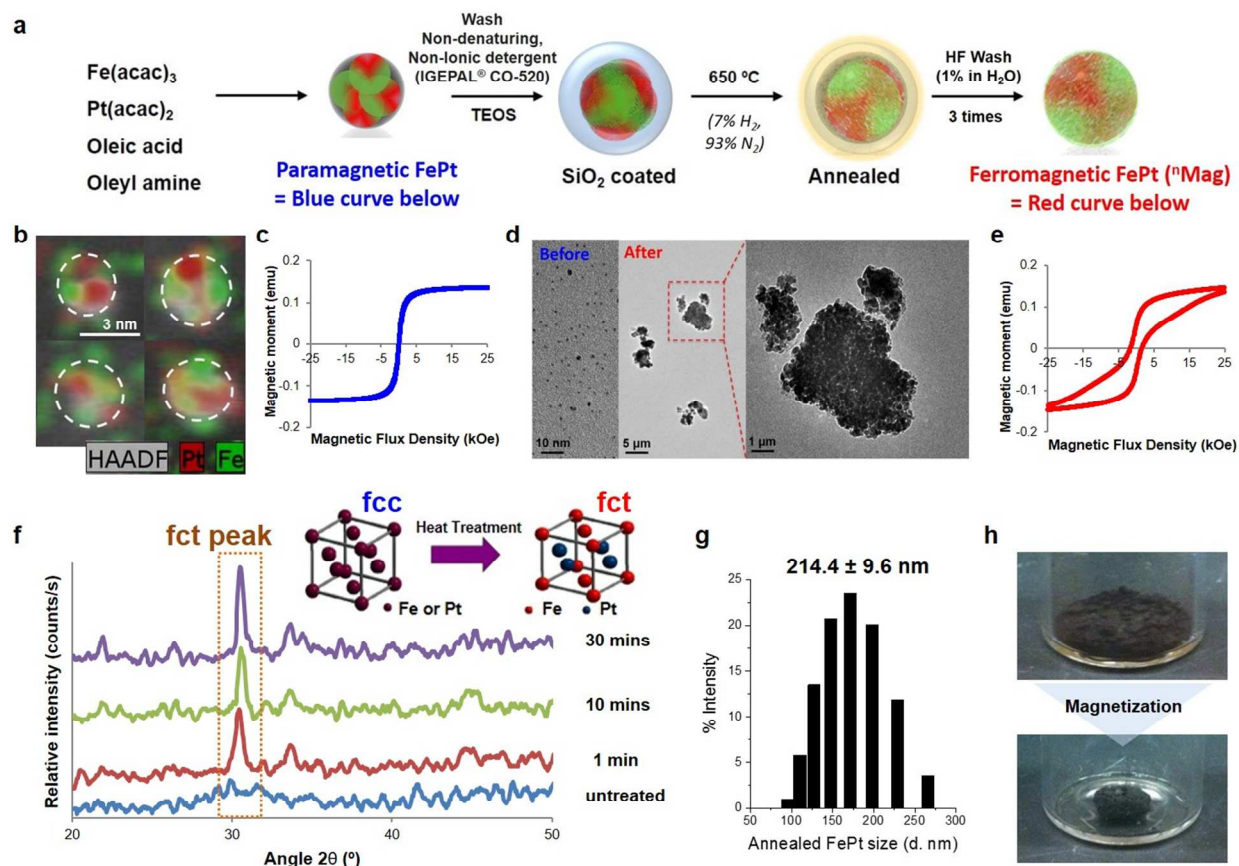


Figure 1. Synthesis and characterization of magnetizable FePt NPs ($^{\text{M}}\text{Mags}$). (a) Synthetic process involving reaction of the constituent elements to form FePt NPs, followed by chemical reduction, coating with SiO_2 , annealing at 650°C then silica coat removal by an HF wash. (b) Demonstration of the blending and localization of Fe and Pt in nanoparticles using High-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM). (c) Paramagnetism. A hysteresis in the magnetization loop of non-annealed NPs. (d) Transmission electron microscopy (TEM), showing FePt particles of approximately 1-3 nm diameter. (e) Ferromagnetism. A hysteresis in the magnetization loop demonstrates induction of permanent magnetic dipole moments (magnetic memory) in annealed NPs. The hysteresis is responsible for the high uniaxial magnetocrystalline anisotropy leading to coercivity. (f) Assessment of magnetic anisotropy using X-ray diffraction (XRD). XRD patterns show transition of FePt alloy NPs from

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3 chemically disordered, low magnetic anisotropy (fcc) to, ordered high magnetic anisotropy (fcc)
4 structure (peaks shown enclosed by red dotted square) as a function of the annealing time. (g)
5 Measurement of particle size after annealing using dynamic light scattering (DLS) (mean \pm std.).
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7 (h) Simple visual assessment of induction of magnetization. Particles magnetized in a 1.5 T coil
8 form an aggregate cluster.
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14 **Fabrication of the ^{64}Ni Mag coated Stent.** Bare metal Mg stents were made into magnetic
15 stents by spray coating with a polymer-solvent solution composed of the FDA approved
16 poly(lactic acid) (PLA) polymer as the base substrate mixed with ^{64}Ni Mags at a ratio of 30% by wt.
17 The stent and coating properties were measured using a scanning electron micrographs (SEM):
18 Mg strut thickness: 100 μm , PLA coating: 30 μm (Figure S1). PLA was chosen as the
19 embedding substrate for three reasons: A) controlled hydrolysis-induced biodegradation, B)
20 physio-chemical properties (PLA hydrophobicity closely matched that of the ^{64}Ni Mags, enabling
21 ^{64}Ni Mags uniform dispersion and retention within the coat.), C) minimal water absorption, enabling
22 stable coating for at least a couple of weeks after implantation (see Figure S2). The schematic in
23 Figure 2a depicts the process of electro spraying of the ^{64}Ni Mag-polymer-coated stent. Here ^{64}Ni Mags
24 dispersed in a PLA CHCl_3 solution were electro sprayed over the Mg stent while rotating the
25 stent axially (Details are in the method section.). The PLA coating allowed the addition of ^{64}Ni Mags
26 into the stent coating at a 30% of total coating mass. The stent coated with ^{64}Ni Mags was
27 magnetized in a 1.5 T magnetic field overnight, BRUKER MRI Solenoid. While we chose 24 h
28 to ensure complete dipole alignment, similar results were with magnetization were observed as
29 brief as 1 h post insertion of the device in the magnetic field (data not shown). Magnetization
30 was evident given the propensity of the device to attract steel objects (Figure 2b).
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44 Figure 2c depicts a rough approximation of the magnetic field lines around a coated
45 cylindrical object, such as a stent. This figure was generated by superimposing the individual
46 fields of some 7000 axially-aligned dipoles symmetrically located inside a cylindrical tube
47 (stent). Strikingly, this rough approximation predicts that the field lines will pull cells to the
48 luminal area of the stent. Note, this field differs from that in the interior of a solenoid in which
49 the field lines are parallel to the cylinder's surface and, thus, would not attract cells to the surface
50 and push the cells towards the ends. A more rigorous future calculation will map the trajectories
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of the magnetized stem cells as they flow through the stent. Therefore, fortuitously, coating with the nanoparticles introduces magnetic forces that “pull” cells toward the luminal side of the stent but also facilitate uniform distribution of those cells across the length of the stent.

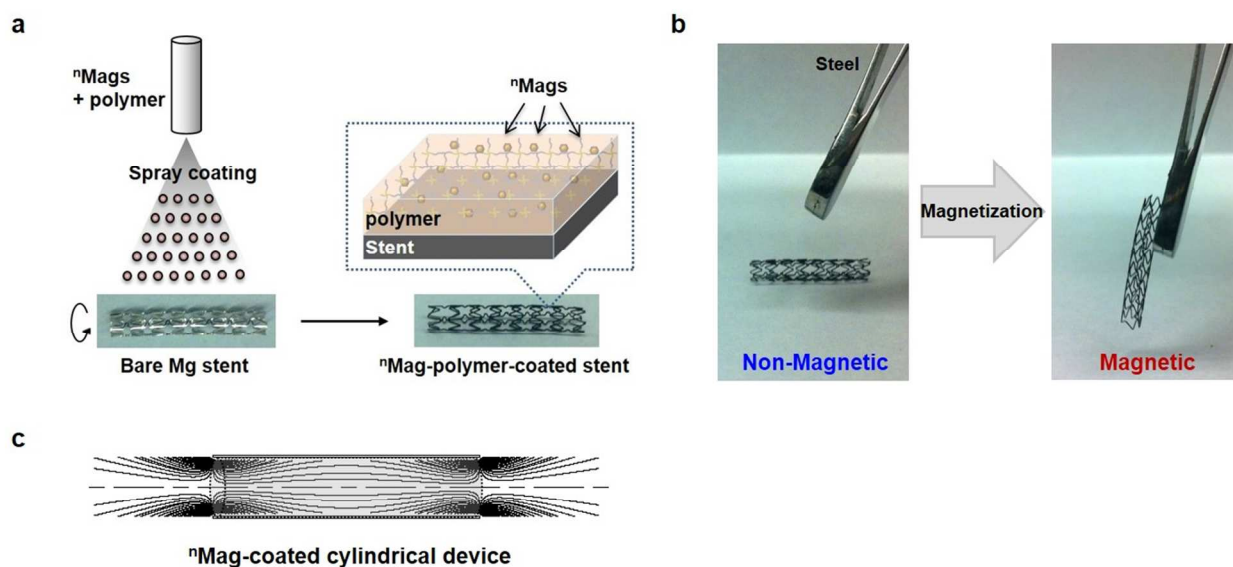


Figure 2. Schematic of the procedure for coating the stent with ^{125}Mg . (a) Electro spray deposition of PLA and ^{125}Mg on an axially rotating Mg stent. The process leads to embedding of the ^{125}Mg after coating. (b) Visual assessment of ^{125}Mg stent magnetization. No response was observed before magnetization. (c) Magnetic field lines inside a particle coated cylindrical device.

Cell Capture Efficiency: To quantitate capture efficiency, cells were labeled by either internalizing IO or surface receptor labeled with antibody coated-IO. A schematic of the bioreactor flow set-up is shown in Figure 3a. Under flow of 25 ml/min we quantitated capture efficiency over time as shown in Figure 3, where cells were injected into 3 cm away from the ^{125}Mg stent. A control stent, polymer-coated stent without ^{125}Mg was used to compare. The presence of a magnetic surface generated by ^{125}Mg substantially increased cell capture onto the stent (Figure 3b). The percentage of cell capture of ^{125}Mg stent was up to 83%, while it was 12% for polymer-coated stent (Figure 3c). It was clear that the magnetic field generated by the ^{125}Mg substantially increased cell capture onto the stent even at flows up to 50 mL/min, corresponding

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3 to normal physiological blood flow in the proximal coronary artery. This equates to a shear stress
4 of 840 dyne·s/cm² (typical shear stress in coronary arteries is < 400 dyne·s/cm²). For these
5 experiments, the fluid shear stress (t , dyne·s/cm²) in the channel was calculated according to the
6 equation:
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$$t = \frac{6Q\mu}{bh^2} \quad (3)$$

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12 where Q is the flow rate (50 mL/ min), μ is the viscosity of the perfusate (0.007 poise), b is the
13 width of the flow channel (0.3 cm), and h is the height of the flow channel (0.3 cm).
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17 To evaluate degradation, a fluorophore was co-encapsulated in the polymer coating
18 reporting on polymer degradation via fluorophore release from the degraded coating. First, we
19 observed that the presence of ⁶⁴Mags did not impact the integrity of the stent coating. This was
20 demonstrated by comparing the release kinetics of fluorophore from polymer coated stents with
21 and without the ⁶⁴Mags (Figure S2). Degradation of ⁶⁴Mag stent was evaluated in a convective
22 flow bioreactor and a slow degradation for up to 35 d was observed (Figure 3d). No obvious
23 recoiling of stent was found and the original diameter was maintained for up to 3 weeks.
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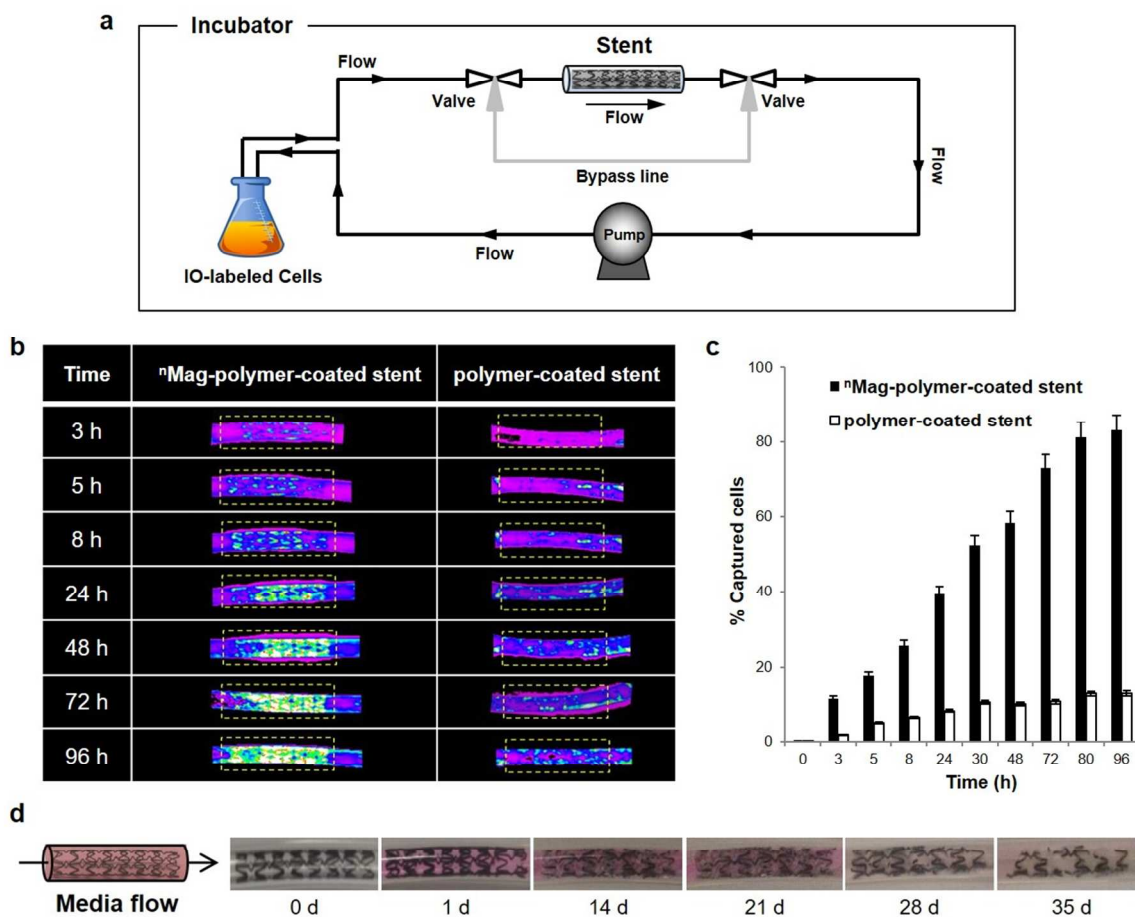
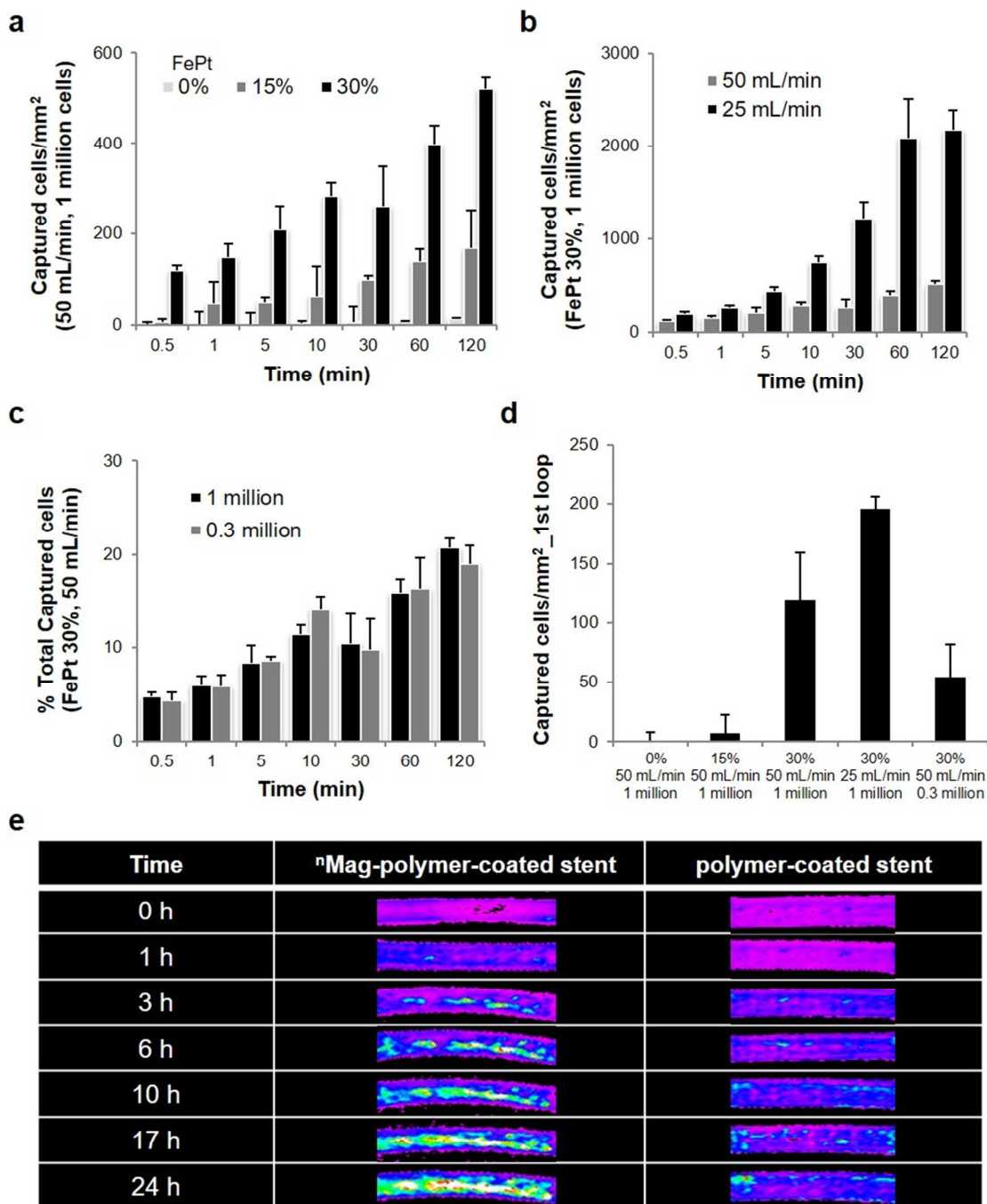


Figure 3. Efficiency of Cell Capture in a circulation flow bioreactor. (a) A flow-loop bioreactor set-up was used to mimic physiological shear flow conditions and efficiency of cell capture under those conditions. Macrophages (1×10^6) that have endocytosed IO particles in the media were used as a first demonstration of general magnetic cell capture. Macrophages were also fluorescently labeled for visualization and quantification of cell capture. (b) Cell capture by the ^{64}Mg stent over time compared to a control stent with polymer but no ^{64}Mg s. (c) Quantitative assessment of cell capture over time. (d) The mode of degradation of ^{64}Mg stent was monitored in a flow-loop bioreactor over one month.

Impact of other parameters on cell capture. Critical parameters that impact the capture efficiency include, concentration of ^{64}Mg s, flow rates (shear stress), and the initial cell number offered for capture. To ascertain the effect of these factors and to establish the generality of the platform in cell capture we repeated capture efficiency studies with Human Umbilical Vein

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3 Endothelial Cells (HUVECs) which internalized iron-oxide by endocytic uptake (Figure S4). The
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5 ^{60}Fe concentration and shear force were shown to be the most effectual variables on cell
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7 capture. At a fixed shear stress of $840 \text{ dyne}\cdot\text{s}/\text{cm}^2$ (50 mL/min), the presence of the ^{60}Fe on the
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9 stent (30 wt%) yielded a cell capture density of more than $500 \text{ cells}/\text{mm}^2$, a 37-fold increase in
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11 the number of retained cells as compared to a stent with no ^{60}Fe (0 wt%) (Figure 4a). Cells
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13 were also captured more efficiently (Figure 4b), when the flow rate was reduced by half. In
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15 contrast, the number of injected cells did not significantly influence the cell capture efficiency
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17 when the initial input number of cells was reduced by a 1/3 (from 1 to 0.3 million) (Figure 4c).
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19 Of the captured cells, about 10-20 % were attracted in the first pass through the circulation loop
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21 (Figure 4d). Understanding cell capture in the first pass is a critical parameter for translational
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23 application since in clinical settings cells are injected proximally to the stent and they then
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25 circulate through the body. Thus, first pass cell capture is crucial as it prognosticates the
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27 potential success of the device in capturing a sufficient number of cells that will proliferate and
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29 ultimately renormalize the impacted region.
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48 **Figure 4.** Capture of circulating HUVECs by ¹²⁵Mg-polymer-coated stent. The effects of (a)
49 ¹²⁵Mg seeding concentration, (b) flow rate, and (c) the number of cells infused into the stent flow
50 chamber. (d) The number of retained cells after the first pass (the most crucial pass in clinical
51 practice) (e) Visualization of HUVEC retention of HUVECs at 50 mL/min and 24 h using stents
52 loaded with 30 wt% ¹²⁵Mags and 0% ¹²⁵Mags.

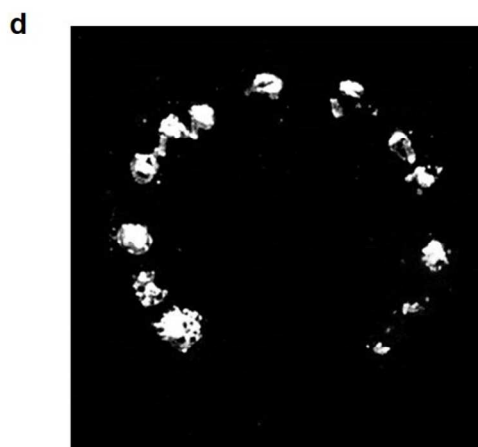
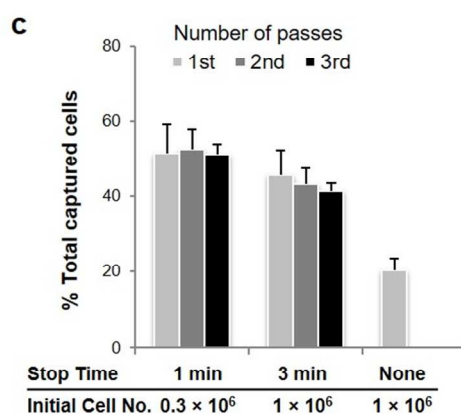
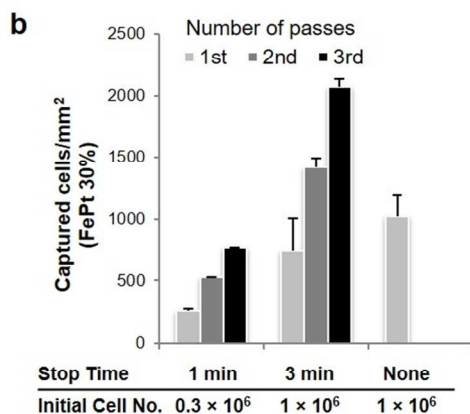
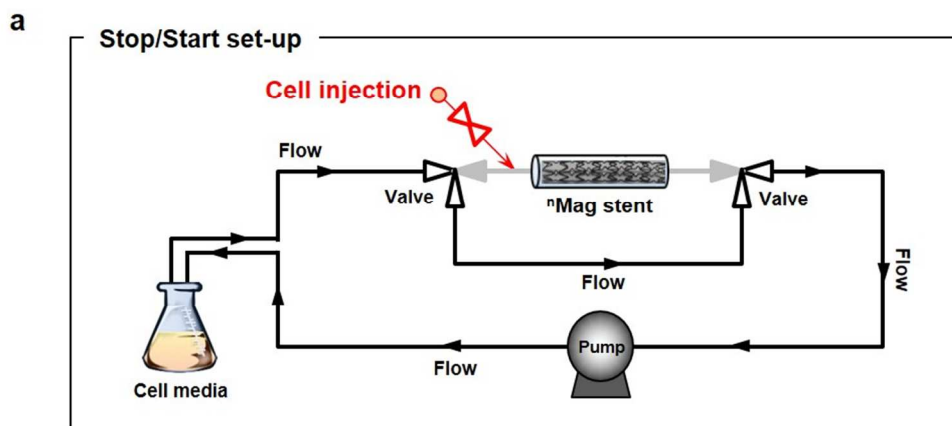
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5 **Clinically relevant flow conditions and their impact on capture of progenitor stem**
6 **cells.** To demonstrate the utility of the magnetic stent in stem cell capture for re-
7 endothelialization, we next labeled CD34+ human hematopoietic progenitor stem cells (hHSCs)
8 with the clinically approved CliniMACS (a dextran-coated superparamagnetic iron oxide agent)
9 immobilized with anti-CD34 antibodies using manufacturer protocol (supplementary
10 information).

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16 A clinically relevant protocol for HSC cell therapy utilizes what is termed as a “stasis
17 protocol” (Figure 5a). For stasis conditions, the flow bypassed while hHSCs were infused for 1
18 to 3 min before resuming flow (25 mL/min), and the cells were not recirculated (only first pass).
19 This procedure was repeated three times and compared to a normal continuous flow set up as has
20 been done previously. We note that the stasis method is of practical significance as it a clinical
21 procedure in current practice and was used to demonstrate the promise of cell therapy in previous
22 clinical trial work utilizing autologous stem cell therapy in acute myocardial infraction. The
23 procedure involved injections of cells through a balloon catheter (normal ‘over the wire’ PTCA
24 catheter) while the balloon was inflated (occluding flow) for a period of 3 min, delivering a total
25 of 3.3-5 mL of cell suspension. Next the myocardium was perfused by deflating the balloon for 3
26 min to restore flow in the coronary artery. The process was repeated until the total amount of
27 cells had been delivered. As opposed to the Stasis method, there has been little to no clinical
28 implementation of the continuous flow method in clinical settings since technology was not
29 available to capture cells under continuous shear flow in contrast to the stop and start, stasis
30 method. Given the challenge with implementing the continuous flow method, we compared both
31 using the magnetic stent technology in cell capture.

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44 The number of captured hHSCs and capture efficiency was significantly increased (at
45 least two-fold) using the stasis procedure (Figures 5b and 5c). This is likely due to the longer
46 time cells spend in proximity to the magnetic stent which facilitate higher efficiency capture. The
47 maximum number of cells retained on ¹²⁵I-Mag stent (30% of FePt NPs) was more than 2000
48 cells/mm² (700,000 cells per stent) considering that, like other cells, endothelial progenitor cells
49 can migrate from the outer basolateral surface through the stent struts to proliferate and thus
50 endothelialize the luminal surface as well as the basolateral region. Here, it is observed that cells
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occupy both the luminal and basolateral surface of the stent throughout the entire surface (Figure 5d). Previous work has reported that local delivery (without magnetic localization) of up to 10^6 autologous endothelial progenitor cells in balloon-injured rabbit carotid arteries can result in inhibition of neointimal hyperplasia.



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3 **Figure 5.** Capture of Human Hematopoietic Stem Cells (hHSCs) (a) Schematic of the “Stop-start
4 stasis/ flow” set-up, a clinically relevant cell administration procedure. Here the flow was
5 stopped and hHSCs were infused for 1 to 3 min before restarting flow (25 mL/min). The number
6 of hHSCs used was 0.3×10^6 for 1 min stasis and 1.0×10^6 for 3 min stasis set up, respectively.
7 Number of hHSCs captured on ^{60}Ni Mag stent (FePt 30%) after the 1st, 2nd, and 3rd cell passes
8 depended on the protocol conditions as described. For comparison, 1.0×10^6 cells were
9 continuously infused for 10 min without holding flow (25 mL/min). (b) Number of cells and (c)
10 capture efficiency (cumulative) are presented. Cross-sections of the ^{60}Ni Mag stent (d) before and (e)
11 after cell capture. The coverage of target stem cells along the length of the magnetic stent is
12 represented by fluorescence images (bright field: ^{60}Ni Mag stent struts, green: progenitor stem cells).
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23 **Evaluation of ^{60}Ni Mags biodistribution and nanotoxicology.** Ultimately, the use of this
24 novel device in vivo, necessitates a rigorous understanding of not only the safety and fate of the
25 device in vivo, but also of that of the individual components. Those components are primarily
26 the ^{60}Ni Mags and polymeric coating. For this, we evaluated the biodistribution, clearance, and the
27 acute hematological and organ toxicity of the ^{60}Ni Mag and PLA composite material in a healthy
28 mouse model. We note that the focus here was on the individual elements and not the entire
29 assembly, since any toxic side-effects or unsafe biodistribution profile would be a direct result of
30 these elements either having an adverse toxic side effect or abnormal biodistribution throughout
31 the body. These studies represent the worst possible scenario in that it assumes the entire device
32 disintegrates into its individual constituents and are circulating in the body. To simulate this
33 condition nanoparticles made from PLA and encapsulating ^{60}Ni Mags were used
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42 First, the ^{60}Ni Mag-PLA combination, like other particulate matter, was observed to
43 accumulate in the liver, pancreas, spleen, stomach, and intestines after polymer degradation
44 (Figure S3). The coating material showed peak accumulation in organs at day 2 and was mostly
45 cleared from animals 7 days post systemic administration. A complete blood count panel (CBCs)
46 and clinical chemistries for liver and kidney function were measured at day 1, 7, and 14 days
47 (Figure S4). No significant toxicities were observed in serum measurements of alkaline
48 phosphatase or serum alanine aminotransferase. Normal physiological ranges for mouse alkaline
49 phosphatase are approximately 62-209 IU/L and for alanine aminotransferase approximately 28-
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3 132 IU/L. No renal toxicity was observed, as blood urea nitrogen levels were within the normal
4 mouse reference range of 18-29 mg/dL. Complete blood counts demonstrated that normal
5 physiological ranges leukocyte counts, platelet counts, and hemoglobin content.
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9 To ascertain levels of acute cytokines that may be induced as a result of the material,
10 bone marrow-derived macrophages (BMMs) harvested from C57BL/6 mice were assessed for
11 secretion of TNF- α , IFN γ , and IL-4. IL-4 was measured as a proxy for potential allergic
12 responses, IFN γ and TNF- α for inflammatory responses (Figure S5). Generally, the coating
13 material was non-inflammatory and neither the particles nor polymer showed signs or instigated
14 an allergic response. An elevated secretion of IFN γ was noted only at a high ⁶⁴Mg-PLA dose of
15 1 mg/mL.
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21 In this work we demonstrated a novel, non-incremental and significant advance in
22 technology that may solve issues related to the long-term complications with stent implantation.
23 The key new feature is a strong intrinsic magnetic capability engineered in a metallic non-
24 ferromagnetic stent made from Magnesium, a bioabsorbable metal with well-known and
25 demonstrated benefits in stent manufacture, and applications in tissue engineering and
26 regenerative medicine. Yet, despite its promise as a bioabsorbable metal with attractive
27 mechanical properties for stent fabrication, early and long-term intravascular and angiographic
28 studies in human trials with Mg stents showed that the bare metal stents alone do not inhibit
29 development of restenosis in the first few months after implantation.³⁸ In human clinical trials, it
30 was found that the rate of stent degradation, which proved to be too fast, compromise the stent
31 radial strength leading to recoiling early after arterial deployment and ultimately playing a
32 significant role in the development restenosis and late term thrombosis. Such complications are
33 alleviated if endothelialization was made to happen early after implantation. Just using the bare
34 metal stent, endothelialization was observed as early as 1 month after implantation.³⁸ Given that
35 this work focuses on inducing rapid capture of injected cells then one would expect that the
36 degradation rate of the bare metal stent will play a smaller role since the tissue regenerative
37 process is expedited, thereby abrogating any advanced side-effects due to implant presence. As
38 such, our technology obviates new structural designs or the search for Mg alloys that may slow
39 down the degradation process. It may also obviate the need for drug regimens either
40 administered systemically or locally through drug eluting stents. With this technology, tissue
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3 regeneration would occur faster relative to any potential mechanical failure and while, we did not
4 encapsulate a drug in the polymer coating, the current design does accommodate for such an
5 addition, if deemed attractive for a particular stent application. In summary, this newer design
6 offers several critical features that make it an alternative to current gold standards: 1) A safe
7 toxicological profile of the magnetic nanoparticle and polymer components together with rapid
8 clearance of the Np from the body. 2) the elimination of the requirement for strong external
9 magnets, such as MRI solenoids in the operating room, as would be needed for a paramagnetic
10 stent. 3) The properties of the Mg metal itself; showing excellent mechanical and biological
11 properties that minimize inflammation, neutralize the acidic polymer products, and degradative
12 properties that were no doubt key in why this material was chosen for testing in humans.
13 Therefore, this study adds additional attractive features to the bare metal stent used in man and
14 these features promise to solve the issues that have limited durable and long-term stent
15 therapeutic outcomes.
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28 **Methods.**

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30 **Fabrication of the Magnetic nanoparticles (ⁿMag).** The bare metal stent was coated with
31 magnetic nanoparticles (FePt NPs) embedded in a biodegradable polymer (PLA). Synthesis of
32 FePt np involved simultaneous chemical reduction of Pt(acac)₂ and Fe(acac)₃ (See
33 supplementary information) and a previous report that was the basis of the current fabrication
34 process.³³ FePt NPs exhibited maximal coercivity when the composition was a molar ratio of 1:1
35 (Fe:Pt). The synthesized FePt NPs were coated with SiO₂ by base-catalyzed silica formation
36 using tetraethylorthosilicate (TEOS) to minimize thermal sintering of FePt particles.³⁴ Detailed
37 protocols are in the supplementary information. The SiO₂ coated FePt particles were annealed in
38 a tube furnace (three Zones Split Quartz Tube Furnace, MTI Co.) for 10 min at 650 °C under a
39 forming gas flow with the following composition (7% H₂/93% N₂). Next, the silica coating was
40 completely removed by washing the particles with HF (1%) in deionized (DI) water three times
41 followed by a DI water wash twice.
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53 **Coating of the Magnesium stent with Magnetic nanoparticles (ⁿMag-Mg Stent).** Bare metal
54 Mg stents with a strut thickness of 100 μm were coated with PLA-ⁿMag mixture (PLA:
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LACTEL[®], inherent viscosity 0.15 - 0.35 dL/g, Durect co.) 1 mg/mL in CHCl₃. The PLA to nMag ratio by wt was 30%. The mixture was electrospayed³⁹ over Mg stents in the ‘cone jet’ mode using ElectroNanospray (Nanocopoeia Inc.TM, rotation speed: 30 rpm, flowrate: 0.03 mL/min, injection volume: 250 μL) to create a 30 μm PLA coating. Magnetization was induced in the dried coated stents by inserting the coated stents in the bore of a Bruker MRI solenoid (1.5T) overnight.

A rough approximation of the magnetic field in the stent. Magnetic field (B^*) was generated by superimposing the individual fields of axially-aligned dipoles of ⁿMags that were symmetrically located inside a cylindrical tube (stent) with a length (L) to radius (R) ratio of 9 (18mm long x 4mm diameter):

$$B^*(r) = \sum_{j=1}^N \left[\frac{3r_j(m \cdot r_j)}{|r_j|^3} - \frac{m}{|r_j|^3} \right] \quad (4)$$

Where r_j is distance vector from the j th dipole to the point of interest (r) and the unit magnetic moments (m) of all the dipoles point in the axial direction.

Cell labeling for capture efficiency experiments. State all cells used here. were stained with carboxyfluorescein succinimide ester (CFSE) for tracking (supplementary information). Stained cells were then labeled with iron oxide) by one of two methods. Method 1: Surface labeling of hHSCs using CliniMACS (a dextran-coated IO agent) immobilized with anti-CD34 antibodies followed by the protocol from manufacturer (supplementary information). Method 2: Intracellular labeling cells with IO. IO NPs (100 μg/mL) were incubated with macrophages and HUVECs in culture at 37°C for 2 h.

Two flow protocols were used. Impact of flow conditions, ⁿMag seeding density, and number of infused cells on the efficiency of magnetic cell capture was investigated. To ascertain the efficacy of cell capture by the magnetized stent, we built a flow bioreactor set up that was operated in stasis or continuous modes as shown in [Figure 5a](#) under standard cell culture

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3 conditions (37 °C and 5% CO₂). Mg stents coated with a PLA layer containing 0, 5, 15, and 30%
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5 ¹²⁵I-Mag was deployed in transparent silicon tubing (4 mm inner diameter) connected via flexible
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7 tubes to a liquid reservoir and a peristaltic pump (Masterflex[®], Cole-Parmer Ins.). Cell media
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9 was circulated at 25 or 50 mL/min and 0.3 or 1 × 10⁶ of labeled cells were injected 3 cm away
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11 from the stent, while media continuously flew. For stasis flow control, flow was paused while
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13 cells were injected and resumed the flow. The cells were injected three times by dividing them
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15 into three parts (each injection: 0.1 and 0.33 × 10⁶ cells were injected for 1 and 3 min,
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17 respectively). To quantify number of captured cells, stents were imaged by a Bruker molecular
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19 imaging instrument (Carestream Health, Inc.) at the desired time points after washing them with
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21 a cell-free media. A calibration standard of fluorescently labeled cells was used to calculate
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23 actual number of cells captured. Degradation study of ¹²⁵I-Mag stent was performed in Dulbecco's
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25 Modified Eagle Medium (DMEM) in a flow chamber at the flow rate of 25 mL/min.
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