

1 Performance of intrinsic heteroatoms in CoP@ginkgo leave-based
2 carbon material on promoting the electrocatalytic activity during
3 HER and OER

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14

15 **Abstract:** Nitrogen (N) and sulfur (S) in ginkgo leaves were converted to carbon lattice
16 heteroatoms, making them a matrix. Herein, cobalt phosphide (CoP) was loaded on the
17 N/S/phosphorus (P) co-doped carbon matrix (CoP@NSPC) via the carbothermic
18 reduction method using the ginkgo leave-based carbon as the precursor. A N/P co-doped
19 carbon matrix (CoP@NPC) was also prepared using glucose-based carbon as the
20 support. Effects of intrinsic heteroatoms from ginkgo leaves were revealed through
21 XPS and UPS compared with CoP@NPC. The N/S heteroatoms accelerated the
22 electron transfer and adjusted the d-band center of CoP@NSPC, thus causing a faster

23 electrocatalytic process. The as-obtained CoP@NSPC exhibited excellent activity
24 toward HER (160 mV @ 10 mA cm⁻²) and OER (198 mV @ 10 mA cm⁻²). The assembly
25 feasibility and catalytic performance were further verified in overall water splitting and
26 exhibited high efficiency and long durability of CoP@NSPC.

27 **Key words:** ginkgo leaves, N/S/P co-doped carbon, CoP, d-band, water splitting

28

29 **1 Introduction**

30 Hydrogen gas (H₂) is an environment-friendly and carbon-neutral energy carrier that
31 may be a prospective alternative to traditional fossil fuels [1-4]. The current industrial
32 H₂ production was based on steam reforming, which consumes renewable fossil fuels
33 and causes the emission of greenhouse gases [5, 6]. By comparison, electrocatalytic
34 overall water splitting assembled with renewable energy has been regarded as a
35 sustainable route for H₂ generation [7, 8]. However, limited to the high cost, rare natural
36 abundance and inferior stability of precious metal-based catalysts, electrocatalytic H₂
37 production through water electrolysis still can not meet the requirements of large-scale
38 application. Hence, many research efforts are devoted to searching the affordable and
39 highly active alternatives of noble-metal catalysts [1, 9-12]. The transition metal (TM)-
40 based catalysts such as carbides, sulfides, phosphides, etc., have been widely explored
41 [1, 13-15]. In particular, the transition metal phosphides (TMPs) have drawn much
42 attention, where cobalt phosphide (CoP) was regarded as one of the appealing
43 candidates of noble metal-based catalysts for both hydrogen evolution reaction (HER)
44 and oxygen evolution reaction (OER). The phosphorus (P) atoms with high

45 electronegatively in CoP were proved to be the proton acceptor and draw electrons from
46 metal, thus accelerating the electron transfer and enhancing the electrocatalytic activity
47 [16-18].

48

49 However, the easy agglomeration and unsuitable adsorption/desorption energy with
50 intermediates (eg. HO^* , H^* , etc.) of CoP during the HER and OER process seriously
51 restricted its utilisation [19, 20]. Loading CoP on the carbon matrix is a feasible method
52 to protect the nanoparticles from agglomeration and optimise the binding energy with
53 intermediates [20, 21]. Furthermore, the introduction of foreign atoms into the carbon
54 lattice will further adjust the electronic structure and energy level of CoP supported
55 carbon catalysts, which will lead to moderate binding energy (BE) with various
56 intermediates in both HER and OER processes [22, 23]. Biomass is an attractive carbon
57 precursor. Its nitrogen (N) and sulfur (S) containing organic components could be
58 directly introduced into the carbon skeleton with a simple pyrolysis process [24-28],
59 thus endowing biomass-based carbon with a possibility to be the support for CoP. It is
60 noted that larger amounts of biomass waste were produced every year, while most of
61 them were burned or buried, even causing secondary pollution. In order to take full use
62 of the advantages of biomass, the intrinsic dopants of biomass-based carbon promoting
63 the electrocatalytic activity of CoP are interesting but lack in-depth study. Particularly,
64 the role of the dopants in the biomass-based carbon on the band structure and electronic
65 properties of CoP deserves detailed analysis.

66

67 The position of d-band centers is related to the binding energy of intermediates in the
68 electrochemistry process, which is critical for electrocatalytic activities. While the
69 dopants in carbon lattice reconstruct the d-band structure of CoP, the effect of dopants
70 from biomass on the d-band center of CoP is not clear. Therefore, in this work, ginkgo
71 leaves were used as the carbon sources to prepare the N/S doped carbon, which was
72 then loaded with CoP. The non-dopant carbon was obtained from glucose without any
73 N/S element widely existing in the ginkgo leaves [29]. The d-band center of such two
74 samples was investigated to analyse the role of N/S atoms from ginkgo leaves.
75 Moreover, the relationship between N/S atoms with high electronegativity in the carbon
76 lattice with the electron transfer was revealed.

77

78 **2 Experiment section**

79 **2.1 Materials**

80 Cobalt (II) nitrate hexahydrate ($\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, 99.99%), Zinc chloride ($\text{ZnCl}_2 \cdot 6\text{H}_2\text{O}$,
81 99.99%), Magnesium chloride ($\text{MgCl}_2 \cdot 6\text{H}_2\text{O}$, 99.99%), 2-methylimidazole (2-MeMI),
82 glucose, phytic acid (PA, 55~70 wt.%) were purchased from Sinopharm Chemical
83 Reagent Co., Ltd. and used without purification. The ginkgo leaves were collected in
84 Nanjing, Jiangsu Province, China, were washed with ultrapure water several times to
85 remove the dirt. The clean ginkgo leaves were dried for a whole night in an oven at 80
86 °C and then crushed to obtain the ginkgo leaf powders.

87

88 **2.2 Preparation of NSC and CoP@NSPC**

89 The $\text{ZnCl}_2 \cdot 6\text{H}_2\text{O}$ (1 g) and $\text{MgCl}_2 \cdot 6\text{H}_2\text{O}$ (2 g) was used as the activation agent and
90 template to grind with the ginkgo leave powders (1 g) for 20 mins and pyrolysis at 750
91 °C for 2 h with the protection of N_2 ; the as-obtained carbon was named nitrogen and
92 sulfur co-doped carbon (NSC).

93

94 The CoP@NSPC was prepared via the carbothermal reduction method using the NSC
95 as the carbon source and support. Firstly, 120 mg NSC was dispersed in 10 mL ultrapure
96 water to form homogenous suspension A. The 30 mg $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ and 42 mg 2-
97 MeMI were dissolved in 5 mL ultrapure water, respectively. Then, the two solutions
98 were mixed and stirred for 30 mins to obtain blue suspension B. The 250 μL PA was
99 injected into the suspension B and subsequently, suspension A was added to suspension
100 B. The mixed suspension was vigorously stirred for 6 h and evaporated moisture via
101 water bath at 60 °C, then dried at 80 °C in a vacuum. The dry powder was transferred
102 into the tube furnace and calcined at 900 °C for 2 h in an N_2 atmosphere. The as-
103 prepared black powders were named CoP@NSPC .

104

105 **2.3 Preparation of non-dopants carbon and CoP@NPC**

106 The non-dopants carbon and CoP@NPC were prepared via the same procedure as NSC
107 and CoP@NSPC , except the ginkgo leaves powders was replaced by glucose.

108

109 **2.4 Material characterisations**

110 The morphology of the as-obtained catalysts was revealed via scanning electron

111 microscopy (SEM, FEI quanta 400FEG), transmission electron microscopy (TEM, FEI
112 Tecnai 20 electron microscope) and the high-angle annular dark field-scanning
113 transmission electron microscope (HAADF-STEM, FEI TALOSF200S). The X-ray
114 powder diffraction (XRD, Bruker D8) was employed to study the structural
115 compositions and valence state of the catalysts. The specific surface area was analysed
116 via the N₂ adsorption experiment on the Quanta 250F and FEI by Brunauer-Emmett
117 Teller (BET). To further investigate the element dispersion and chemical states on the
118 materials surface, the K-Alpha X-ray photoelectron spectroscopy (XPS) was taken on
119 Thermo ESCALAB 250XI, and calibrated with the binding energy using C1s at 284.6
120 eV. The Raman spectra were conducted using iHR550 Raman microscope (HOR-IBA
121 scientific) at a range of 400~4000 cm⁻¹. The ultraviolet electron spectroscopy (UPS,
122 Phi5000 VersaProbeIII) was taken to obtain the valence band spectrum of materials.
123 Furthermore, the work function spectrum was measured via UPS with a bias voltage of
124 -10 V.

125

126 **2.5 Electrochemical measurement**

127 Electrocatalytic measurements for HER and OER were conducted in 0.5 M H₂SO₄ and
128 1 M KOH using a conventional three-electrode system on CHI 760E electrochemical
129 workstation at room temperature with N₂ and O₂ saturation, respectively. The graphite
130 rod was selected as the counter electrode. The mercury/mercurous sulfate electrode
131 (Hg/Hg₂SO₄) and mercury/mercuric oxide electrode (Hg/HgO) was used as the
132 reference electrode in 0.5 M H₂SO₄ and 1 M KOH, respectively. The glass carbon

133 electrode (GCE, 0.196 cm²) was served as the working electrode with the modification
134 of the catalysts ink. Such catalysts ink was prepared by mixing the 10 mg of the as-
135 obtained catalysts with 1000 μL of ethanol/deionised water ($V_{C_2H_5OH} : V_{H_2O} = 1 : 3$) and
136 sonicated for 30 minutes. The well-mixed catalysts ink of 10 μL was dropped onto the
137 GC and dried at 25 °C. Subsequently, the 10 μL Nafion (5 wt.%) was dropped onto the
138 GC and dried naturally. The linear sweep voltammetry (LSV) plots were recorded with
139 a scan rate of 2 mV s⁻¹. Cyclic voltammetry (CV) tests for HER and OER were
140 performed at a range of -0.5~-0.3 V and -0.2~0 eV, respectively, with a scan rate of 10,
141 30, 50, 70, 90 mV s⁻¹, respectively. The overall water splitting was performed with a H-
142 cell. Moreover, all the HER and OER tests were taken with IR-correct and the measured
143 potentials were converted to a reversible hydrogen electrode (RHE) using Eq.(a) and
144 Eq.(b), respectively.

$$145 \quad E_{vs.RHE} = E_{vsHg/HgSO_4} + E^{\theta}_{Hg/HgSO_4} + 0.059 pH \quad (a)$$

$$146 \quad E_{vs.RHE} = E_{vsHg/HgO} + E^{\theta}_{Hg/HgO} + 0.059 pH \quad (b)$$

147 Where, $E^{\theta}_{Hg/HgSO_4}$ was equal to 0.697 V and $E^{\theta}_{Hg/HgO}$ was equal to 0.098 V.

148

149 **3 Results and discussion**

150 **3.1 Physical-chemistry properties**

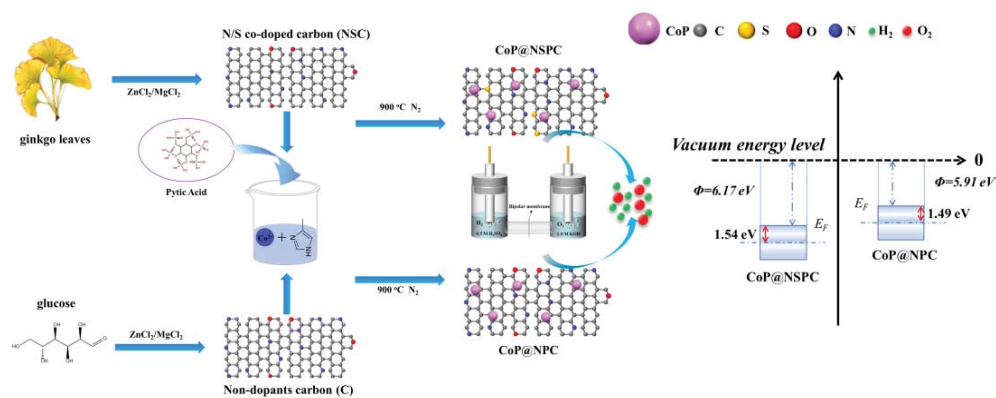
151 The mass ratio of Co(NO₃)₂·6H₂O and NSC was first explored for optimal preparation.

152 As shown in Figure S1, the HER and OER tests were conducted to prove the best mass

153 ratio of Co(NO₃)₂·6H₂O and NSC was 1:4. Then, compared with samples prepared via

154 non-dopants carbon skeleton derived from glucose, the role of inherent N/S atoms in

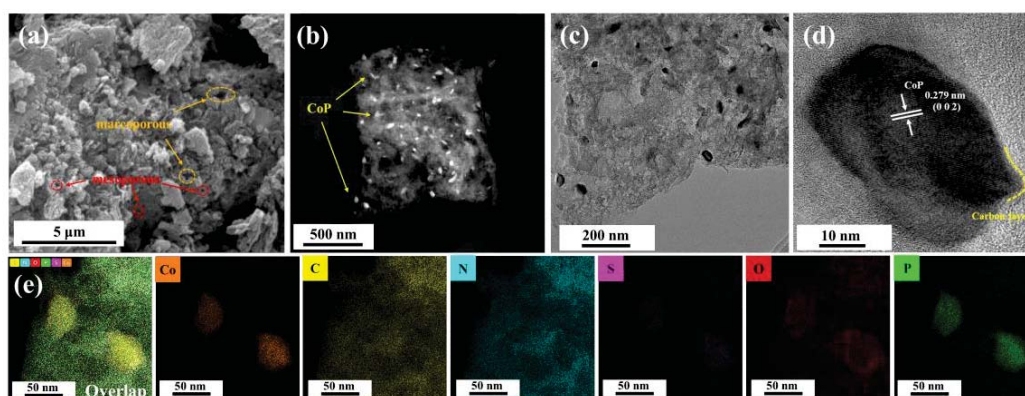
155 ginkgo leaves was further revealed. CoP@NSPC and CoP@NPC were prepared via the
 156 same steps as shown in scheme 1; thus, the effects of nitrogen atoms introduced by 2-
 157 MeMI is considered to be identical here.



158
 159 Scheme 1. Synthesis process CoP@NSPC and CoP@NPC

160
 161 The morphology and structures of CoP@NSPC and CoP@NPC were investigated
 162 through SEM. Benefiting from the activation of MgCl₂ and ZnCl₂, there are abundant
 163 pores on the surface of CoP@NSPC (Figure 1a) and CoP@NPC (Figure S2). The
 164 porous structure of CoP@NSPC and CoP@NPC mainly consists of microporous and
 165 mesoporous, which may meet the demands of fast mass transfer [1] and promote the
 166 exposure of active sites inside of the materials [30]. The HAADF-STEM was employed
 167 to explore further the dispersion of CoP nanoparticles (NPs) in CoP@NSPC. As shown
 168 in Figure 1b, the highly dispersed light flake particles in the dark field image are the
 169 CoP NPs with an average size of 25.26 nm (Figure S3). The fringe spacing with 2.79 Å
 170 corresponds to the (002) lattice planes of CoP.
 171 Moreover, the EDS elemental mapping shown in Figure 1e confirms that the elementals
 172 including C, N, O, S, P are distributed over the entire as-obtained materials, while the

173 Co is mainly dispersed as the nanoparticles in the form of CoP. Meanwhile, due to the
174 unavoidable surface oxidation upon exposure to air, the appearance of slightly oxidised
175 P and Co states can be usually observed in CoP@NSPC. The S element exists mainly
176 in the form of doping. The detailed analysis is discussed in the XRD and XPS sections.
177



178
179 Figure 1. (a) SEM, (b) HAADF-STEM, (c, d) HRTEM images and (e) EDS elemental
180 mapping images of CoP@NSPC.

181

182 As shown in Figure 2a, the XRD patterns further confirm the successful preparation of
183 CoP NPs loaded on the N/P/S co-doped carbon support, which was consistent with the
184 HAADF-STEM results. It is evident that the diffraction peak positions of CoP@NSPC
185 and CoP@NPC are the same. The broad peak at the 26° represents the (0 0 2) lattice
186 plane of graphite carbon [21]. The diffraction peaks at 32°, 36°, 46°, 48°, 52°, and 57
187 ° are indexed to the (011), (002), (200), (111), (102), (210), (112), (211), (103), (020)
188 and (212) planes of CoP according to the JCPDS-29-0497. In addition, the pore
189 structure of CoP@NSPC and CoP@NPC was studied by N₂ adsorption and desorption
190 experiments. The adsorption isotherm (Figure 2b) corresponds to the type IV isotherms,

191 which means the hole of CoP@NSPC and CoP@NPC mainly consists of mesoporous
192 [31]. The pore size distribution shown in Figure S4 has further confirmed the existence
193 of mesoporous and macroporous. Such hierarchical porous structure endows the
194 CoP@NSPC and CoP@NPC with a high specific surface area of 733.99 m²/g and 1069
195 m²/g, respectively. Since the CoP@NSPC and CoP@NPC owned similar crystal
196 structure and pore structure, the key role to result in the different performance on HER
197 and OER for CoP@NSPC and CoP@NPC needs in-depth discussion through Raman
198 and XPS spectrum.

199

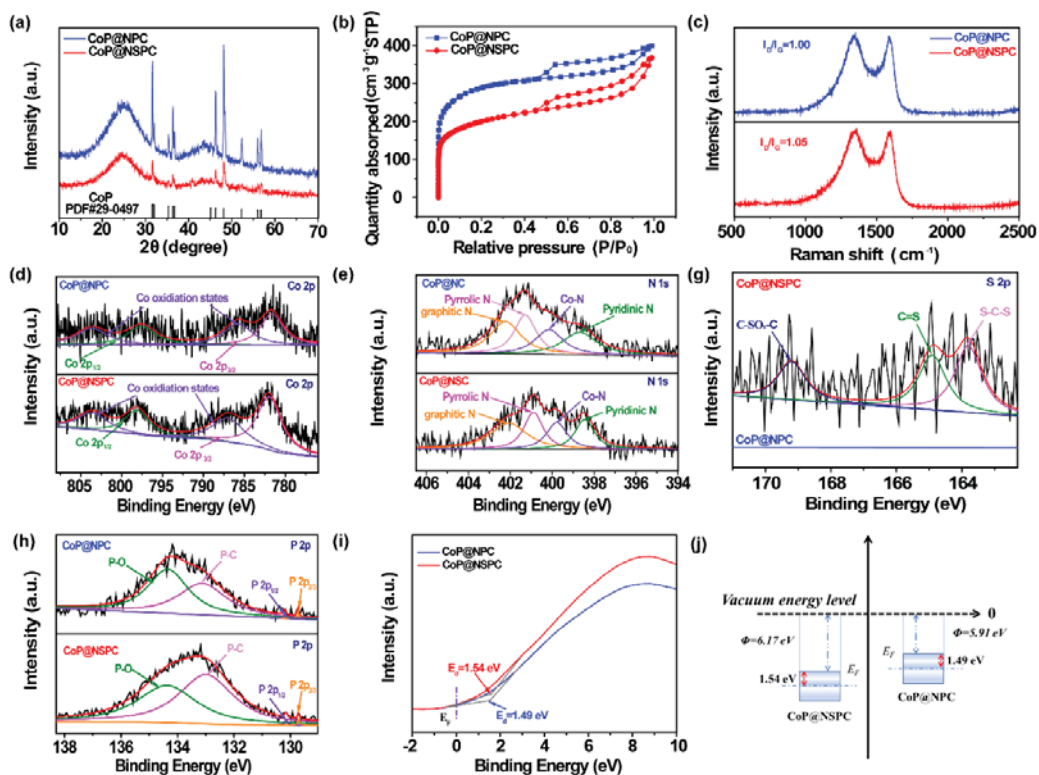
200 There are two distinct peaks in the Raman spectrum (Figure 2c), the peak at 1350 cm⁻¹
201 and 1580 cm⁻¹ named D and G band, representing the disordered carbon and graphite
202 carbon, respectively. The ratio of the intensity of the D band and G band indicated the
203 number of defects in the carbon support. At the same time, the I_D/I_G for CoP@NSPC is
204 1.05, greater than that of CoP@NPC (1.00). Hence, such results implied the
205 CoP@NSPC with more defects, which can be further confirmed by the XPS spectrum.
206 As shown in Table S1, there are more N/S contents in the CoP@NSPC compared with
207 CoP@NPC. Considering the N atoms introduced by 2-MeMI is roughly equal, the
208 difference of the N atoms is mainly caused by the pristine heteroatoms derived from
209 ginkgo leaves. The N, S, O, P was proved to be doped in the carbon skeleton of
210 CoP@NSPC, which is consistent with the element mapping image. The high-resolution
211 N 1s spectrum of CoP@NSPC can be subdivided into four peaks at 400.8, 398.45,
212 402.08 eV, and 399.78 eV (Figure 2d) represents the pyrrolic N, pyridinic N, graphitic

213 N and Co-N, respectively. By taking advantage of the rich amount of N containing
214 organic contents in ginkgo leaves, the atoms ratio of pyrrolic N, pyridinic N, graphitic
215 N and Co-N were both higher than CoP@NPC as shown in Table S2, which will offer
216 more active sites and accelerate the diffusion-limited current density [30]. The high-
217 resolution S 2p spectrum of CoP@NSPC can be deconvoluted into three peaks, while
218 there are no S elements in the CoP@NPC as shown in Figure 2g (Table S3). The three
219 peaks located on the 163.79 eV, 164.89 eV and 169.19 eV correspond to the presence
220 of S-C-S, C=S and C-SO_x-C, respectively [32]. There are four distinct peaks at 798.09
221 eV, 781.93 eV, 803.58 eV and 786.99 eV in the high-resolution Co 2p spectrum, namely
222 Co 2p_{1/2}, Co 2p_{3/2} and Co oxidations states [20-22, 33]. Compared with CoP@NPC, the
223 content of Co is increased, which may result from the promotion of Co-anchoring on
224 the carbon skeleton induced by the presence of more electronegative atoms doping [34].
225 Notably, compared with CoP@NPC, the BE of Co-N for N 1s of CoP@NSPC was a
226 negative shift, while the BE of Co-N for Co 2p was a positive shift (Table S4). The
227 plentiful N atoms significantly enhance the electron transfer for CoP@NSPC, which
228 may adjust the electronic structure and accelerate the electrocatalysis process [22, 35].
229 The high-resolution O 1s spectrum of CoP@NSPC and CoP@NPC can be
230 deconvoluted into three peaks at 530.86, 532.13 and 533.18, as shown in Figure S6,
231 corresponding to the O-Co, O-C and O-P. As shown in Figure 2h, the chemical states
232 of P were basically the same, which are four distinct peaks located at 134.36, 132.98,
233 130.15 and 129.71 in the high-resolution P 2p spectrum, corresponding to the P-O, P-
234 C, P 2p_{1/2} and P 2p_{2/3}.

235

236 Furthermore, the dopants could reconstruct the electronic structure of CoP, which may
237 affect the adsorption and desorption energy of various intermediates [36]. The d-band
238 center is the descriptor of the free energy of hydrogen (oxygen) adsorption. To uncover
239 the d-band center of CoP@NSPC and CoP@NPC, the UPS valence band spectra was
240 employed to calculate the work function and the location of the d-band center. Shown
241 in Figure S7 is the work function spectrum derived from the UPS valence band
242 spectrum with a -10 V bias. Since the secondary electron cut-off edges for CoP@NSPC
243 and CoP@NPC are 15.05 eV and 15.31 eV, the values of work function (Φ) for
244 CoP@NSPC and CoP@NPC are 6.17 eV and 5.91 eV, respectively. The higher work
245 function value indicates the Fermi level for CoP@NSPC shifts away from vacuum level
246 compared with CoP@NPC and its d-band shifts away from the vacuum level. Through
247 the raw UPS valence band spectrum (Figure S8), the d-band center can be obtained
248 from the linear intersection near the Fermi level, as shown in Figure 2e [37]. The d-
249 band centers for CoP@NSPC and CoP@NPC are 1.54 eV and 1.49 eV respectively.
250 The downshift of the d-band center optimises the adsorption/desorption energy with
251 intermediates of HER and OER, which is beneficial to the electrocatalytic activity of
252 CoP@NSPC.

253



254

255 Figure 2. (a) XRD patterns, (b) N₂ sorption isotherms, and (c) Raman spectrum of
 256 CoP@NSPC and CoP@NPC, High resolution XPS spectrum of (d) N 1s, (e) S 2p, (f)
 257 Co 2p and (g) P 2p of CoP@NSPC and CoP@NPC, (h) UPS valence band spectra of
 258 CoP@NSPC and CoP@NPC, (i) schematic of energy band structure.

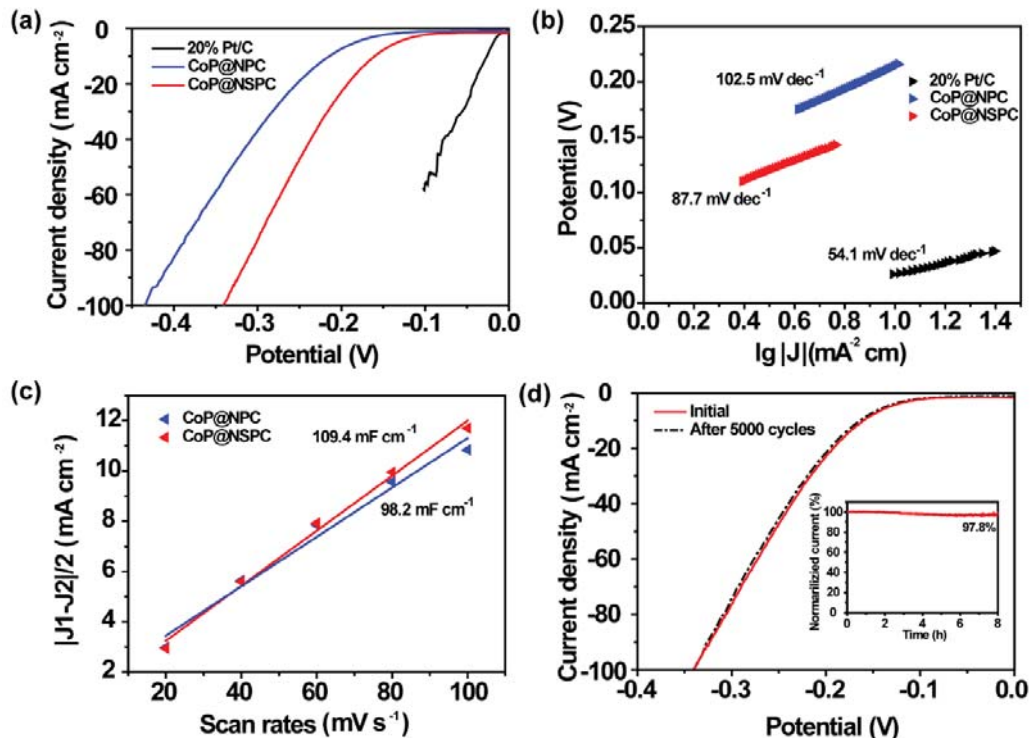
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260 3.2 Electrocatalytic activity and stability towards HER

261 The HER activity for CoP@NSPC, CoP@NPC and 20% Pt/C were assessed through
 262 the linear scan voltammogram (LSV) polarisation curves recorded in 0.5 M H₂SO₄.
 263 Shown in Figure 3a was the IR-compensated LSV curves. The CoP@NSPC manifests
 264 outstanding HER performances while the CoP@NPC has poor HER activity. To obtain
 265 10 and 50 mA cm⁻² current density, the designed CoP@NSPC delivers a low
 266 overpotential of 160 and 256 mV, as shown in Figure 3a, much smaller than CoP@NPC

267 (214 and 331 mV). The HER reaction kinetics of CoP@NSPC and CoP@NPC was then
268 determined via Tafel diagrams. As shown in Figure 3b, CoP@NSPC demonstrates a
269 lower Tafel slope of 87.3 mV dec⁻¹ than CoP@NPC, indicating better HER catalytic
270 activity. The Tafel slopes for CoP@NSPC do not match the ideal Tafel slopes of 29, 38
271 and 116 mV dec⁻¹ corresponding to different rate-determining steps of HER, suggesting
272 the HER proceeds CoP@NSPC via a Volmer–Heyrovsky mechanism [21]. The
273 electrochemical impedance spectroscopy (EIS) test was employed to evaluate the
274 charge transfer resistance (R_{ct}) of the as-prepared catalyst [38]. The Nyquist plots in
275 Figure S9a for CoP@NSPC have shown a smaller semicircle which indicated the lower
276 value of R_{ct} , revealing the higher conductivity and faster electron transfer of
277 CoP@NSPC compared with CoP@NPC, which is consistent with XPS results. The
278 efficient catalytic surface area (ECSA) was calculated to further understand the superb
279 catalytic activity of CoP@NSPC [17]. The double-layer capacitor (C_{dl}) associated with
280 ECSA arose from cyclic voltammetry (CV) with different scan rates, as shown in Figure
281 S10. As shown in Figure 3c, the fitted C_{dl} of CoP@NSPC is 109.4 mF cm⁻², higher than
282 CoP@NPC (98.2 mF cm⁻²), suggesting there are more accessible active sites in
283 CoP@NSPC. Besides, the stability of catalytic activity is also critical for its practical
284 application. The chronoamperometric response (i-t) was performed under the current
285 density of 10 mA cm⁻² to access the stability of CoP@NSPC. After 8 h operation, it has
286 shown a 97.8% current density retention (Figure 3d), which means the robust catalytic
287 activity of CoP@NSPC. The internal diagram is the LSV curves before and after 5000
288 CV cycles. The overpotential to drive the current density to 10 mA cm⁻² was with a

289 negligible increase of ~ 4 mV.



290

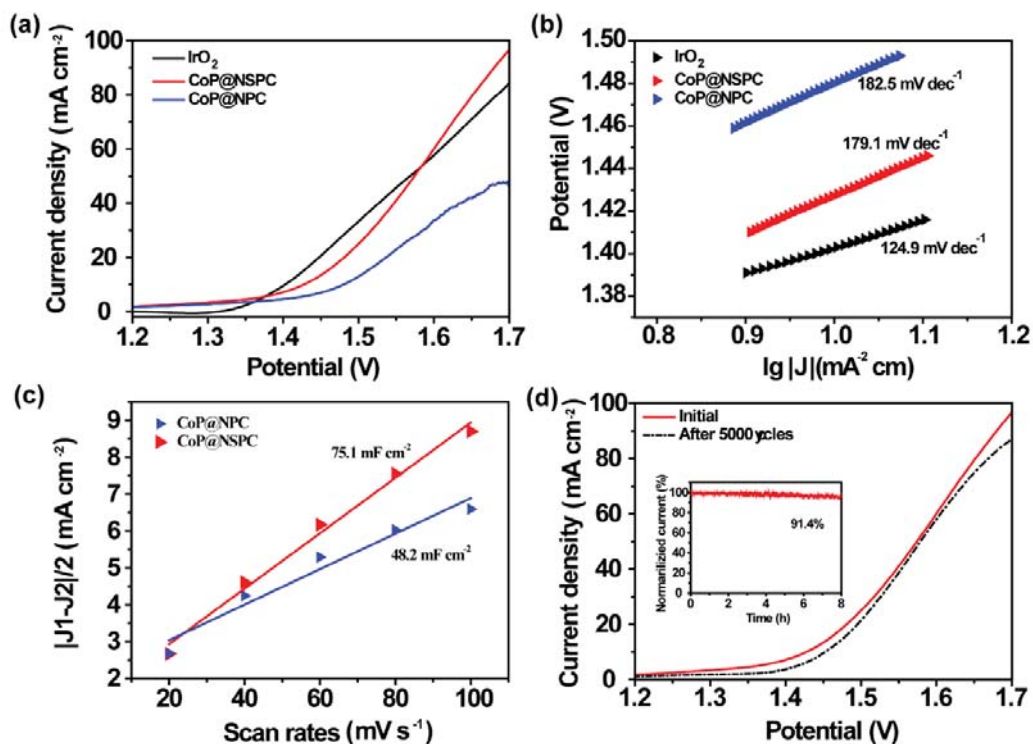
291 Figure 3. Electrocatalytic HER performance. (a) LSV curves and (b) Corresponding
292 Tafel plots in 0.5 M H₂SO₄, (c) C_{dl} stands for ECSA for CoP@NSPC and CoP@NPC,
293 (d) Durability test of CoP@NSPC.

294

295 3.3 Electrocatalytic activity and stability towards OER

296 The OER performance of CoP@NSPC, CoP@NPC and commercial IrO₂ was also
297 investigated via a standard three-electrolyte system in 1.0 M KOH. The LSV curves in
298 Figure 4a demonstrate that CoP@NSPC exhibit superb OER activity, in which only 198
299 and 345 mV overpotential is required to deliver 10 mA cm⁻² current density, comparable
300 to the commercial IrO₂ (173 and 339 mV). To reveal the reaction kinetics of OER
301 activity, the Tafel plots are obtained corresponded to the LSV plots, as shown in Figure

302 4b. The smaller Tafel plots of CoP@NSPC ($179.1 \text{ mV dec}^{-1}$) suggested better scaling
303 kinetics with voltage [39-41], which means faster kinetics and optimised OER activity.
304 Such a faster OER process was also proved by EIS testing. The semicircle of
305 CoP@NSPC with a smaller radius demonstrates the stronger charge transfer among the
306 OER process (Figure S9b). The C_{dl} for OER was also investigated based on CV curves
307 on different scan rates (Figure S11). As shown in Figure 4c, it is no doubt that the value
308 of C_{dl} for CoP@NSPC (54.9 mF cm^{-2}) is higher, which indicate it is favorable to expose
309 more active sites compared with CoP@NPC (48.2 mF cm^{-2}). Similar to the HER
310 process, the long-term stability was conducted via i-t testing at 10 mA cm^{-2} . The
311 CoP@NSPC performed robust durability with a mild density attenuation ($\sim 9.6\%$).
312 Furthermore, the stability evaluation is also performed by comparing the LSV curves
313 before and after 5000 CV cycles. As shown in the internal diagram of Figure 4d, the
314 overpotential to deliver 10 mA cm^{-2} current density exhibits a slight increase of 25 mV,
315 confirming the outstanding stability of CoP@NSPC for OER.
316



317

318 Figure 4. Electrocatalytic OER performance. (a) LSV curves and (b) Corresponding

319 Tafel plots in 1.0 M KOH, (c) C_{dl} stands for ECSA for CoP@NSPC and CoP@NPC,

320 (d) Durability test of CoP@NSPC.

321

322 The observation of the superior HER and OER activity of the CoP@NSPC compared

323 with CoP@NPC was mainly attributed to the following three reasons. First, the

324 adsorption energy of CoP with H₂ and O₂ was too strong, which will limit the process

325 of H₂/O₂ desorption. The carbon matrix with rich N/S dopants resulted in the downshift

326 of the d-band center, which optimised the adsorption/desorption with H₂/O₂, thus

327 causing a faster HER and OER process for CoP@NSPC. The dopants of N/S atoms

328 also can serve as the additional active sites, which was in accordance with the higher

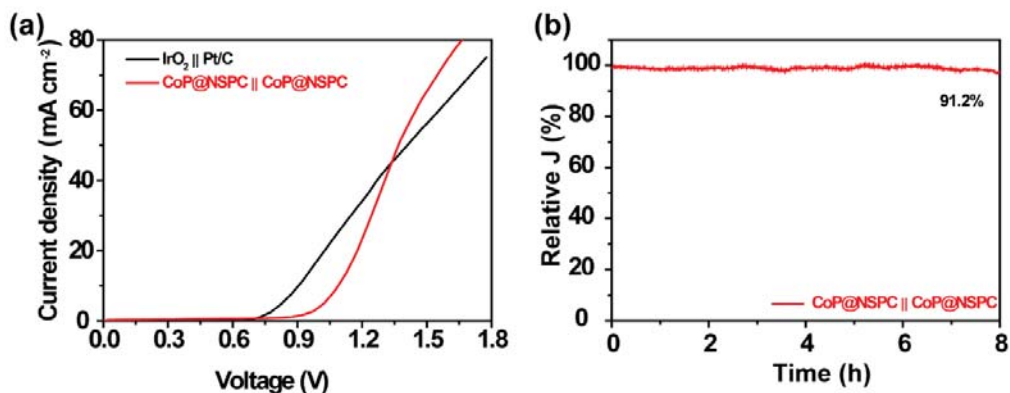
329 value of C_{dl} of CoP@NSPC. More than that, the dopants accelerated the electron

330 transfer between Co sites to the carbon matrix. The fine XPS spectrum of N species of
331 CoP@NSPC have shown a negative shift. In contrast, the peaks of Co species was
332 positively shifted compared with CoP@NPC, indicating there was favorable charge
333 transfer during the electrocatalytic process. Such results were further proved via EIS
334 tests for that the CoP@NSPC had a smaller impedance.

335

336 **3.4 Electrocatalytic activity and stability towards overall water splitting**

337 The overall water splitting was performed with a two-electrode device in H-type cell,
338 while the anode and cathode were assembled with the bifunctional electrocatalysts of
339 CoP@NSPC. The HER and OER process was conducted in 0.5 M H₂SO₄ and 1.0 M
340 KOH, respectively. The ion exchange membrane was used to exchange the ions of these
341 two electrolytes. As shown in Figure 5a, the CoP@NSPC || CoP@NSPC only needed a
342 cell voltage of 1.13 and 1.38 V to deliver the current density to 10 and 50 mA cm⁻²,
343 which was comparable to commercial IrO₂ || Pt/C. With the increase of the cell voltage,
344 the current density of CoP@NSPC || CoP@NSPC even exceeds the commercial IrO₂ ||
345 Pt/C. Figure 5b shows the durability test of CoP@NSPC || CoP@NSPC for water
346 splitting. After 10 h operation at the current density of 10 mA cm⁻², there was only a
347 slight current density attenuation of 8.8%, suggesting the excellent robustness for
348 overall water splitting.



349

350 Figure 5. Overall water splitting performance. (a) Polarization curves of IrO₂ || Pt/C and
 351 CoP@NSPC || CoP@NSPC, (b) Stability of CoP@NSPC || CoP@NSPC.

352

353 4 Conclusion

354 The as-prepared CoP@NSPC was fabricated using ginkgo leaves derived carbon via a
 355 facile carbothermic reduction strategy. The hybrid catalysts exhibit bifunctional
 356 catalytic activity toward the HER and OER processes. Owing to the intrinsic N/S atoms
 357 doping in the ginkgo leaves derived carbon, CoP@NSPC shows better catalytic activity
 358 for HER and OER compared with CoP@NPC. The heteroatoms in the carbon matrix
 359 of CoP@NSPC have been shown to induce faster electron transfer between Co NPs and
 360 carbon matrix. The d-band center of CoP@NSPC moves downwards, resulting in
 361 moderate adsorption and desorption of intermediates in the HER and OER processes.
 362 CoP@NSPC acted as both anode and cathode in the H-type cell, achieving a current
 363 density of 10 and 50 mA cm⁻² at 1.13 and 1.38 V, together with excellent stability.

364

365 Declaration of competing interest

366 The authors declare that they have no known competing financial interests or personal

367 relationships that could have appeared to influence the work reported in this paper.

368

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373

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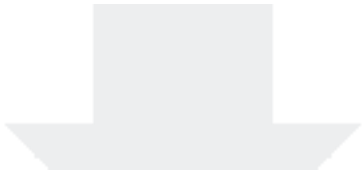
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