## Iron-intercalated Zirconium Diselenide Thin Films from the Low Pressure Chemical Vapor Deposition of $[Fe(\eta^5-C_5H_4Se)_2Zr(\eta^5-C_5H_5)_2]_2$

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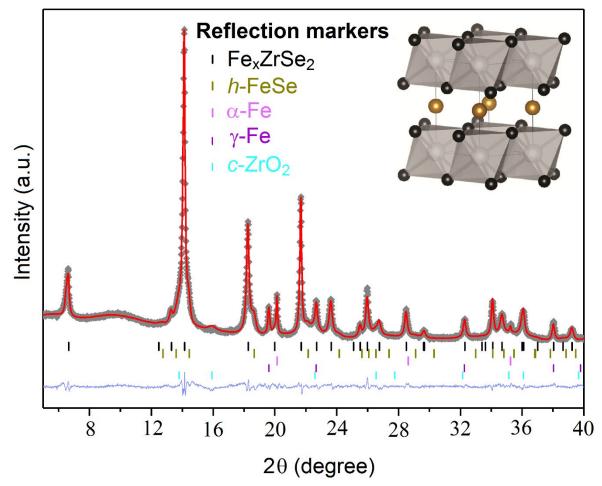
Rietveld refinements were performed using GSAS and EXPGUI.<sup>1,2</sup> All phases were modelled using a Pseudo-Voigt peak shape, GSAS profile function type 2. The large background contribution from Fefluorescence and an amorphous content from partially decomposed air-sensitive material was modelled using a shifted Chebyschev function with 20 terms. Preferred orientation of the highly 2dimensional target compounds,  $Fe_xZrSe_2$ , was observed and modelled using a 4<sup>th</sup> order spherical harmonic model resulting in a texture index 1.0104. Correlations between the preferred orientation parameters and the refined Fe occupancy within the  $Fe_xZrSe_2$  structures suggests the estimated errors associated with the refined occupancies are underestimated. The data was not of suitable quality for refinement of Se occupancies relative to the Zr, a key feature that has previously been shown to alter both lattice parameters and electronic properties.<sup>3,4</sup> The multiphase nature of the sample makes bulk elemental analysis to determine the Fe intercalant composition impossible.

Due to partial oxidation of the Fe<sub>x</sub>ZrSe<sub>2</sub> samples a significant contribution from ZrO<sub>2</sub> was observed. No Se was observed in our diffraction patterns likely due to a lack of crystallinity, adding to the observed amorphous background. Whilst zirconia, ZrO<sub>2</sub>, typically shows a monoclinic structure at room temperature, the incorporation of Fe has previously been shown to stabilise the cubic form, *c*-ZrO<sub>2</sub>, at room temperature.<sup>5</sup> The low crystallinity of these decomposition products results in extreme broadening of their associated diffraction peaks and so ZrO<sub>2</sub>, where only broad and ill-defined features are observed, the contributions were modelled using their cubic structures to minimise the number of free refinement parameters. Besides lattice parameters, no structural details were refined for the impurity phases and thermal parameters were fixed at a reasonable value.

The observation of  $\gamma$ (FCC)-Fe at room-temperature in bulk form is not expected and worth commenting on. It has previously been studied as a stabilised phase in epitaxially grown films on Cu(001) and diamond surfaces.<sup>6,7,8</sup> Lattice matching to the substrate (e.g. FCC-Fe  $a \sim 3.59$  Å, Cu(001) a = 3.61 Å, diamond a = 3.57 Å) stabilises the FCC structure and so it's appearance on Fe<sub>x</sub>ZrSe<sub>2</sub> (a = 3.77 Å) is not entirely unexpected here.

Phase	Space- group	a (Å)	<i>b</i> (Å)	<i>c</i> (Å)	β (°)	Phase fraction (wt. %)
	Atom	<i>x</i> / <i>a</i>	y/b	<i>z/c</i>	Occ	$U_{\rm iso}({\rm \AA}^2)$
$Fe_x ZrSe_2$	P-3m1	3.77239(6)	3.77239(6)	6.1336(3)	120	61.9(10)
	Zr	0	0	0	1.0	0.0233(13)
	Se	1/3	2/3	0.2616(6)	1.0	0.008(4)
	Fe	0	0	0.5	0.135(5)	0.025
FeSe	P6 <sub>3</sub> /mmc	3.6962(12)	3.6962(12)	5.999(3)	120	9.11(12)
	Fe	0	0	0	1	0.025
	Se	1/3	2/3	1/4	1	0.025
α-Fe	Im-3m	2.87086(17)	2.87086(17)	2.87086(17)	90	3.55(6)
	Fe	0	0	0	1	0.025
γ-Fe	Fm-3m	3.6116(3)	3.6116(3)	3.6116(3)	90	5.18(9)
	Fe	0	0	0	1	0.025
ZrO <sub>2</sub>	Fm-3m	5.1234(13)	5.1234(13)	5.1234(13)	90	20.25(19)
	Zr	0	0	0	1	0.025
	Ο	1/4	1/4	1/4	1	0.025

**Table S1**. Refined parameters for Fe<sub>x</sub>ZrSe<sub>2</sub>. The refinement quality of fit parameters  $\chi^2$ , R<sub>wp</sub> and R<sub>p</sub> are 8.443, 1.47 % and 1.10 %, respectively.



**Figure S1**. Rietveld refined powder diffraction data (crosses), fit (red line), difference curve (shifted blue line) and reflection markers for all phases (ticks).<sup>9-13</sup> Structural analysis was performed using the GSAS package.<sup>1</sup> Embedded in the right corner is an image of the Fe<sub>x</sub>ZrSe<sub>2</sub> structure drawn from structurally refined PXRD data using VESTA,<sup>14</sup> showing Zr and Se atoms (grey and black spheres) making up triangular layers of ZrSe<sub>2</sub> polyhedra in the ab plane, separated by Fe atoms (gold spheres) randomly occupying the octahedral vacancies within the van der Waals gap with a refined occupancy of ~14%.

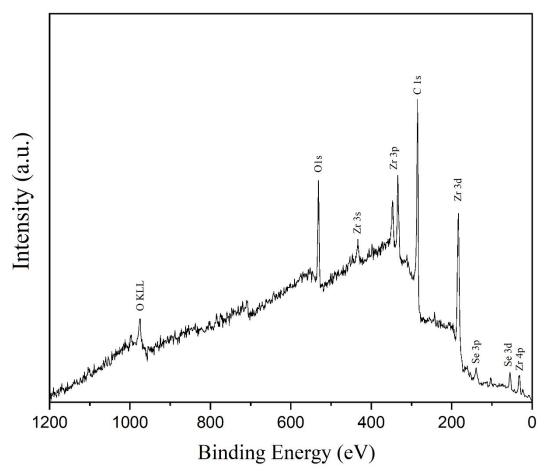
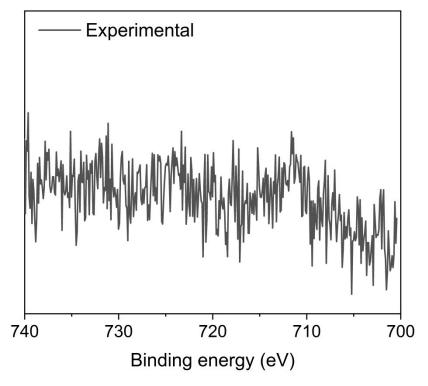


Figure S2. XPS surface survey scan of Fe<sub>0.14</sub>ZrSe<sub>2</sub> thin film sample (0 - 1200 eV).



**Figure S3.** XPS data for the surface of the  $Fe_xZrSe_2$  film showing the Fe 2p region with a low signal to noise ratio.

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