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Refining Energy Levels in ReS₂ Nanosheets by Low-Valent Transition-Metal Doping for Dual-Boosted Electrochemical Ammonia/Hydrogen Production

Feili Lai, Nan Chen, Xiaobin Ye, Guanjie He, Wei Zong, Katherine B. Holt, Bicao Pan, Ivan P. Parkin, Tianxi Liu, Renjie Chen**

Dr. F. Lai, W. Zong, Prof. T. Liu

Key Laboratory of Synthetic and Biological Colloids, Ministry of Education, School of Chemical and Material Engineering, Jiangnan University, Wuxi 214122, P. R. China

Email: txliu@dhu.edu.cn

Dr. N. Chen, Prof. R. Chen

School of Materials Science & Engineering, Beijing Institute of Technology, Beijing, 100081, China

Email: chenrj@bit.edu.cn

Dr. G. He, Prof. K. Holt, Prof. I. Parkin,

Christopher Ingold Laboratory, Department of Chemistry, University College London, 20 Gordon Street, London WC1H 0AJ, UK

X. Ye, Prof. B. Pan

Hefei National Laboratory for Physical Science at the Microscale, University of Science and Technology of China, Anhui 230026, China

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Abstract:

Electrocatalytic nitrogen reduction reaction (NRR) and hydrogen evolution reaction (HER) may provide intriguing approaches to nitrogen fixation and hydrogen production under ambient conditions, which need discover efficient and stable catalysts to light up the “green chemistry” future. However, it often forms bottlenecks on N₂/H₂O activation, the very first step of NRR/HER, by energetic electron injection from the surface of electrocatalysts. Herein, we report that the bottlenecks for both NRR and HER can be well tackled by engineering the energy level *via* low-valent transition-metal doping,

simultaneously, where ReS₂ is employed as a model platform to prove the concept. The doped low-valent transition-metal domains (e.g. Fe, Co, Ni, Cu, Zn) in ReS₂ provide more active sites for N₂/H₂O chemisorption and electron transfer, not only weakening the N≡N/O-H bonds for easier dissociation through proton coupling, but also elevating *d*-band center toward the Fermi level with more electron energy for N₂/H₂O reduction. As a result, we found that iron-doped ReS₂ nanosheets wrapped nitrogen-doped carbon nanofiber (Fe-ReS₂@N-CNF) catalyst exhibits superior electrochemical activity with 8-fold higher ammonia production yield of 80.4 μg h⁻¹ mg⁻¹_{cat.}, and lower onset overpotential of 146 mV and Tafel slope of 63 mV dec⁻¹, when comparing with the pristine ReS₂.

Ammonia (NH₃) is an indispensable carbon-free chemical to agriculture and other products,^[1-3] as well as potential carrier for hydrogen storage due to its high hydrogen density and liquefiable feature.^[4,5] However, the chemical inertness of dinitrogen (large bond energy of 940.95 kJ mol⁻¹),^[6,7] makes it a “Cadmean victory” to convert N₂ into NH₃. For instance, the famous reaction for industrial ammonia production is known as Haber-Bosch process, relying on iron-based catalysts and rigorous operating conditions (673-873 K, 20-40 MPa),^[8,9] which almost accounts for 1% of the worldwide energy consumption and 1.6% of the total global CO₂ emission.^[10]

In order to promote the progress of “green chemistry”, a clean, safe, and sustainable NH₃ synthesis route is in an urgent need. Electrochemical nitrogen reduction reaction (NRR) has gradually attracted much attention, and been regarded as an alternative approach.^[11-14] As an inevitably overwhelming rival, hydrogen evolution reaction (HER) is always wanted to be impeded, making it a key issue to find a reasonable way to enhance N species adsorption on the surface of catalyst instead of H species. For instance, Ling et al. put forward a reticular chemistry approach by exploiting MOF’s water-repelling and molecular-concentrated effects to overcome HER-imposed bottlenecks, achieving both excellent NRR selectivity (~90%) and boosted Faradaic efficiency by 10 percentage points.^[15] Recently, Yan et al. reported a strategy to simultaneously promote NRR selectivity and activity by

using bismuth nanocrystals and potassium cations, the Faradaic efficiency of which was achieved as 66%.^[16] Yan et al. also proposed it can suppress the HER process and improve the Faradaic efficiency of NRR to 56.55% by positively shifting the reaction potential and using single-atom iron as electrocatalyst.^[17] To our knowledge, however, traditional Haber-Bosch process will still be the main route for NH₃ production in the foreseeable future with a large amount usage of H₂ by steam reforming of fossil fuels, where HER would be a more attractive choice for renewable H₂ resource. In fact, the majority used hydrogen gas for Haber-Bosch process would go through a series of chemical treatment processes under high temperature, while less than 4% of the total H₂ gas comes from sustainable and mild electrocatalysis of water. Therefore, it would be also another possible perspective to explore dual-enhanced electrocatalysts for both ammonia and hydrogen production, as it is still difficult to suppress the HER process well for the majority electrocatalysts.

Bearing this in mind, rhenium disulfide (ReS₂) is recently discovered as next-generation transition metal dichalcogenide (TMD) materials for high-efficient electrocatalysis.^[18,19] It is mainly attributed to their layered structure with weak van der Waals interactions between individual sandwiched layers,^[20,21] and anisotropic structure with two-dimensional morphology to provide not only large specific surface area but also two-dimensional permeable channels for reactant adsorption/transport.^[22] In the past few years, abundant effort has been made to promote the development of TMD-based electrocatalysts for both nitrogen reduction reaction and hydrogen evolution reaction.^[23-25] In contrast with other TMD materials (such as MoS₂, MoSe₂, WS₂, and WSe₂), ReS₂ possesses a very weak interlayer coupling,^[25-27] thus leading to an enhanced reactant diffusion into its deep interlayer space for better NRR/HER processes. Besides that, traces of transition-metal doping, particularly for low-valent domains (such as Fe³⁺, Co²⁺, Ni²⁺, Cu²⁺, Zn²⁺),^[28-30] in atomically ordered ReS₂ nanosheets can disturb its crystalline lattice to expose more active sites for electrocatalysis. However, several significant issues in low-valent transition-metal doped ReS₂ nanosheets are still rarely reported and

urgently needed to be addressed and solved, namely, how does transition-metal incorporate into ReS₂, and where are the real electrocatalytic sites for NRR/HER?

Therefore, here we employ low-valent Fe³⁺/Co²⁺/Ni²⁺/Cu²⁺/Zn²⁺ salts deliberately for coprecipitation with ReO₄⁻ salt. By this, M⁴⁺-positioned Re could be replaced by low-valent transition-metal species (e.g. Fe, Co, Ni, Cu, Zn). The density functional theory (DFT) calculation firstly reveals that the energy level of ReS₂ can be well tuned by doping engineering with enhanced charge density around the low-valent cations. Taking iron-doped ReS₂ nanosheets wrapped nitrogen-doped carbon nanofiber (Fe-ReS₂@N-CNF) catalyst as an example, it exhibits superior electrochemical activity with an 8-fold higher ammonia production yield rate of 80.4 μg h⁻¹ mg⁻¹_{cat.} for NRR, and lower onset overpotential of 146 mV and Tafel slope of 63 mV dec⁻¹ for HER, when comparing with the pristine ReS₂. The deep mechanisms for NRR/HER processes are further discovered by in-situ spectroscopic characterization together with DFT studies that low-valent Fe domain is beneficial to providing more active sites for N₂/H₂O chemisorption and weakening the N≡N/O-H bonds for easier dissociation through proton coupling. These results clearly demonstrate the important role of low-valent transition-metal doping engineering on optimizing ReS₂-based NRR/HER electrocatalyst, which provides a novel but general route for green synthesis of NH₃ production.

Foreseen by computational study firstly, the pristine ReS₂ semiconductor shows a wide energy gap (E_g) of 1.31 eV (**Figure 1a**) as direct-band gap. This high E_g will certainly hinder the catalyst activation with inefficient electron transition. Interestingly, the other five kinds of optimized ReS₂ models were constructed by doping 5% low-valent transition-metals (e.g. Fe, Co, Ni, Cu and Zn), denoted as Fe-ReS₂ (**Figure 1b**), Co-ReS₂ (**Figure 1c**), Ni-ReS₂ (**Figure 1d**), Cu-ReS₂ (**Figure 1e**), and Zn-ReS₂ (**Figure 1f**), respectively. As a result, we found the E_g values of these doped ReS₂ samples significantly decreased to around 1.03 eV as indirect-band gaps, demonstrating it a feasible strategy for superior electron excitation of transition-metal doped ReS₂. The well reduced energy gap is attributed to the impurity band after doping of transition-metals, which can be either in the gap or in the

valence/conduction band. Here, the first-principle calculations show that impurity band will live in the gap of transition-metal doped ReS_2 , therefore the gap is decreased. Meanwhile, this may also be originated from the activate ReS_2 atomic surface with charge redistribution, especially centered around low-valance transition-metal doped domains (**Figure 1&S1**),^[31] which can be regarded as highly active electrocatalysts.

Inspired by the above theoretical results, iron-doped ReS_2 nanosheets wrapped nitrogen-doped carbon nanofiber ($\text{Fe-ReS}_2@N\text{-CNF}$) is firstly chosen as the key model system to investigate the benefits of low-valent transition-metal doping engineering. The samples are synthesized *via* a N-CNF templated hydrothermal method,^[32] in which Fe atoms are doped into ReS_2 lattice using $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ as a precursor as schematically illustrated in **Figure 2a**. Firstly, the typical scanning electron microscopy (SEM) image of N-CNF is displayed in **Figure 2b** with average diameter of 30-60 nm, proving that the heteroatom doping process for nitrogen atoms would not affect its morphology (**Figure S5**) and non-crystalline property (**Figure S6**). The nitrogen-rich carbon surface is also beneficial to the adsorption of $\text{ReO}_4^-/\text{Fe}^{3+}$ precursors for epitaxial growth of Fe-ReS_2 nanosheets. **Figure 2c-e** reveal the morphology of $\text{Fe-ReS}_2@N\text{-CNF}$ by SEM and transmission electron microscopy (TEM), highlighting a feature of uniform Fe-ReS_2 nanosheets covered hybrid carbon nanofiber with ultrafine diameter of ~ 85 nm and length extending to several micrometers. This hybrid strategy is beneficial to solving the difficult problem of Fe-ReS_2 aggregations (**Figure S7**), leading to significantly increased specific surface area from 12.1 to 24.6 $\text{cm}^2 \text{g}^{-1}$ (**Figure S8**). As displayed in high-resolution TEM (HRTEM) image (**Figure 2f**), the iron-doped ReS_2 nanosheets with a spacing of 0.62 nm can be assigned to (001) planes, indicating that the Fe-ReS_2 mainly grow along [001] direction. Energy-dispersive X-ray spectroscopy (EDS) mapping is further employed to ascertain the doping of Fe and N elements on $\text{Fe-ReS}_2@N\text{-CNF-5}$, revealing the uniform Fe and N doping (**Figure 2g**). X-ray diffraction (XRD) exhibits the same Bragg reflections of $\text{Fe-ReS}_2@N\text{-CNF-5}$ and non-doped

ReS₂@N-CNF (**Figure 2h**), implying no structural deformation caused by low-valent Fe substitution of Re⁴⁺.

X-ray photoelectron spectroscopy (XPS) is further applied to understand the chemical states and elemental compositions in the Fe-ReS₂@N-CNF-5. The binding energies of the Re⁴⁺ 4f_{7/2} and 4f_{5/2} electron peaks in Fe-ReS₂@N-CNF-5 are 41.0 eV and 44.4 eV, respectively (**Figure 3a**).^[33] As shown in **Figure 3b**, two characteristic peaks being located at 162.4 eV and 163.4 eV in Fe-ReS₂@N-CNF-5 are detected for the bonding configurations of S, corresponding to the core 2p_{3/2} and 2p_{1/2} level peaks of S²⁻, respectively.^[34] All these above results are consistent with values for ReS₂@N-CNF, indicating the immeasurable changes for chemical states of Re and S elements after doping iron atoms in Fe-ReS₂@N-CNF-5. Three types of pyridinic N (398.4 eV), pyrrolic N (400.3 eV), and graphitic N (401.1 eV) are identified from Fe-ReS₂@N-CNF-5 (**Figure 3c**), manifesting the successful incorporation of N atoms in the carbon nanofiber template. Meanwhile, the high pyridinic N content in Fe-ReS₂@N-CNF-5 can also promote the following ammonia production.^[35] The binding energies of the Fe 2p_{3/2} and Fe 2p_{1/2} electron peaks in Fe-ReS₂@N-CNF-5 are 708.1 eV and 721.2 eV (**Figure 3d**), respectively, confirming the successful incorporation of Fe atoms in ReS₂ structure. In this instance, an obviously negative shift of binding energy for the Fe 2p_{3/2} electron is observed when compared with other iron compounds, such as binding energies of Fe 2p_{3/2} in Fe₂O₃ (710.1 eV), FeCl₂ (710.7 eV), and FeCl₃ (711.6 eV).^[36] It can be ascribed to the bonding hybridization of Fe atom with its neighbor atoms, leading to the decreased oxidation state of Fe. After incorporating low-valent Fe element in ReS₂ nanosheets, the density distribution in Fe-ReS₂ is significantly destabilized to Fe-doped domains (*inset* in **Figure 3d**), due to the structural heterogeneity-led charge rearrangement. Furthermore, the quantitative analysis of XPS spectra (**Figure S9**) demonstrates the chemical formula of 5 mol % Fe-doped ReS₂ nanosheets can be fixed to Fe_{0.073}Re_{0.991}S₂ (**Table S1**).

Upon confirming the electronic structure of low-valent Fe doped ReS₂ compound, we are now in a position to evaluate the efficacy of as-obtained 5 mol % transition-metal doped ReS₂ nanosheets

wrapped nitrogen-doped carbon nanofiber (Fe/Co/Ni/Cu/Zn-ReS₂@N-CNF-5) for electrochemical HER/NRR performance, which were tested under 0.1 M Na₂SO₄ solution with continuous N₂ bubbling. As shown in the polarization curves (**Figure 4a**) and Tafel plots (**Figure 4b**), the low-valent transition-metal doping could moderately improve the HER activity of non-doped ReS₂@N-CNF (**Table S2**). Taking Fe-ReS₂@N-CNF as an example, it shows lower onset overpotential ($\eta = 146$ mV, the onset overpotential is defined as the difference between the reversible potential of the couple H₂/H⁺ and the potential where electrocatalysis occurs) and Tafel slope (63 mV dec⁻¹) than those in ReS₂@N-CNF ($\eta = 200$ mV; 87 mV dec⁻¹), implying its superiority to accelerate the hydrogen evolution reaction by low-valent transition-metal doping engineering. As an efficient conductive-frame, nitrogen-doped carbon nanofiber is also beneficial for the electron transfer along the interface of N-CNF and ReS₂ outer layer, leading to the enhanced HER performance (**Figure S12**). Fe-ReS₂@N-CNF-5 also shows remarkable durability (**Figure S13**), without dramatic change of the LSV curve after 5000 cycles. Apart from positive effect of low-valent transition-metal doping engineering in HER activity, we also found it is useful in the improvement of nitrogen reduction reaction (NRR). As shown in **Figure 4c**, the maximum NH₃ yield rate (80.4 $\mu\text{g h}^{-1} \text{mg}^{-1}_{\text{cat}}$) and Faradaic efficiency (FE = 12.3 %) of Fe-ReS₂@N-CNF-5 are much higher than those of non-doped ReS₂@N-CNF (10.3 $\mu\text{g h}^{-1} \text{mg}^{-1}_{\text{cat}}$, FE = 9.9%) at -0.2 V versus reversible hydrogen electrode (RHE). In order to make the quantitative method for NH₃ yield rate more convincing, nuclear magnetic resonance (NMR) method was further applied (**Figure S20a**). Based on the corresponding fitted curve (**Figure S20b**), the NH₃ yield rate can be calculated to be 79.8 $\mu\text{g h}^{-1} \text{mg}^{-1}_{\text{cat}}$, which approached to the NH₃ yield rate from indophenol blue method within the allowed error range, demonstrating the reliability of detection methods (NMR and indophenol blue methods). This enhanced NRR performance can also be observed in other low-valent transition-metal doped electrodes of Co/Ni/Cu/Zn-ReS₂@N-CNF-5 (**Figure S21 & Table S3**). To verify the source of ammonia production, NMR method was conducted as ¹⁵N isotopic labeling experiments. As displayed in **Figure S22**, only ¹⁴N triplets and ¹⁵N doublets were observed in the

$^{14}\text{NH}_4^+$ and $^{15}\text{NH}_4^+$ productions, respectively, by using $^{14}\text{N}_2$ and $^{15}\text{N}_2$ as feeding gases, which demonstrate not only the purity of as-used $^{14}\text{N}_2$ gas but also the stable structure of Fe-ReS₂@N-CNF-5 electrocatalyst without escaped N atoms during NRR process. Meanwhile, it was revealed that the Fe-doping concentrations would be closely related with the corresponding NRR performance (especially for NH₃ yield), 5% Fe-doping amount would be a more proper concentration to release the best electrocatalytic activity for Fe-ReS₂ samples (**Figure S24**). Both the NH₃ yield rates and FE_{NH₃} values for Fe-ReS₂@N-CNF-5 are also well maintained with a stability retaining of ~99% after 10 continuous electrochemical NRR tests (**Figure 4d**). From the corresponding XPS spectra (**Figure S25**), the Fe-ReS₂@N-CNF-5 still maintains the same chemical states and elemental compositions after NRR stability test as the ones before electrocatalysis (**Figure 3**), demonstrating its excellent electrocatalytic stability for long-term usage. As shown in **Figure S26**, the sum of the amounts for generated NH₃ production (S2) and H₂ production (S1) equals the total supplied charge amount (S1 + S3), which indicates all the supplied charge was used to reduce neither N₂ gas nor H₂O molecules with a Faradaic efficiency of 87.7% for HER process. The rough agreement of the charge amount (S2 ≈ S3) between experimentally quantified gases (NH₃ and H₂) and calculated one suggests the total FE is 100% for the sum of NRR and HER without hydrazine generation (**Figure S28**).

These enhanced NRR and HER performances in Fe-ReS₂@N-CNF-5 can be attributed to the refined energy levels of ReS₂ nanosheets after low-valent transition-metal doping, as predicted by DFT calculations in **Figure 1**. Experimentally, ultraviolet photoelectron spectroscopy (UPS) was used to determine the values of energy level (difference between the vacuum and Fermi level) of Fe-ReS₂ and pure ReS₂ as 6.11 and 7.27 eV (**Figure 4e&Figure S29**), respectively, by subtracting the width of the He I UPS spectra at excitation energy of 21.22 eV. Compared with the energy level of non-doped ReS₂ nanosheets, the one of Fe-ReS₂ is properly shifted to approach the reduction potentials of H₂O and N₂ (**Figure 4f**), making the electron exchange process (**Figure S30**) easier between electrocatalyst and the reacting species.^[30] Due to the increased energy levels of Fe-ReS₂ nanosheets, the energy required

for electron remove from the catalyst surface decreases, leading to the thermodynamically activated NRR and HER. In another perspective, the feasibility of electron transfer in low-valent transition-metal doped ReS₂ electrocatalysts would deliver lower onset overpotentials, demonstrating the Fe-ReS₂@N-CNF-5 a promised electrocatalyst for NRR and HER processes. Additionally, the incorporation of highly conductive N-CNF template is beneficial to increasing the electron transfer between two components in the Fe-ReS₂@N-CNF-5 for efficient reduction reaction as compared with that in Fe-ReS₂ bulk (**Figure S31**).

Apart from the perspective of thermodynamics, kinetics also plays a key role in these reactions. Attributing to N-CNF template and low-valance Fe doping, Fe-ReS₂@N-CNF-5 displays an increased number of electroactive sites (evidenced by the N₂ adsorption-desorption isotherms in **Figure S8** and optimized electronic structure in **Figure 3d**), leading to the optimal activation effect of H₃O⁺ and N₂ reagents on catalyst surface. For the HER process (**Figure 5a**), the H₃O⁺ molecule interacts between Re^{δ+}/Fe^{δ+}...O and S^{δ-}...H synergistically, as the first volmer step (H₂O + e⁻ → H_{ads} + OH⁻). After the adsorption of another H₂O molecule, the H₂ gas generates by following one kind of desorption steps (H_{ads} + H₂O + e⁻ → H₂ + OH⁻ or H_{ads} + H_{ads} → H₂). Revealed by DFT calculation further (**Figure S32**), the free energy value for H* (where * denotes as an adsorption site) is -0.36 eV for the pure ReS₂, but the water adsorption energy is up to 0.81 eV, which lead to the sluggish HER kinetics. Although the free energy value for H* in Fe-ReS₂ is relatively higher to be -0.49 eV, its water adsorption energy can be drastically reduced to 0.29 eV, indicating it thermodynamically favorable for water reduction on Fe-ReS₂ in neutral solution. To further investigate the nitrogen reduction reaction process (**Figure 5a**), in-situ surface-enhanced infrared absorption spectroscopy (SEIRAS) measurements were carried out for Fe-ReS₂@N-CNF-5 sample (**Figure 5b-d**). The IR bands at about 3420 and 1646 cm⁻¹ (**Figure 5b**) match well with the O-H stretching and H-O-H bending of water molecules, respectively. Their positive shift with the decrease of applied potential from 0 to -0.3 V is attributed to the change of the adsorption configuration of water.^[37] The peaks at 1554/1519, 1330 and 1102 cm⁻² are attributed to

the H-N-H bending,^[38,39] -NH₂ wagging,^[13,40] and N-N stretching^[41,42] of adsorbed N₂H_y species, respectively. These results demonstrate that the nitrogen reduction reaction on Fe-ReS₂ follows an associative mechanism, instead of dissociative mechanism in the Haber-Bosch process,^[43] with the simultaneous addition of hydrogen on N₂ to break the N≡N bond. The time-dependent SEIRAS spectra (**Figure 5d**) of Fe-ReS₂@N-CNF-5 catalyst at -0.2 V show similar absorption results with time goes, demonstrating its ultra-stable electrocatalysis. As reported in previous literature, enzymatic mechanism of N₂ fixation is more suitable for single-atom-embedded systems with lower energy barrier.^[44,45] Therefore, the possible NRR mechanism is depicted in **Figure S33** with DFT calculation results in **Figure 5e-g**. All the reaction intermediates were bounded simultaneously to the surface Re atoms of both Fe-ReS₂ and ReS₂, while the Re atoms neighboring Fe atom showed higher activity for the reactions of N-related species (**Figure 5e&4f**). The calculation results further indicate the active sites in Fe-ReS₂ are located in the region near doped Fe atoms (especially for the Re atoms), which is corresponded with the ELF mappings (**Figure 1**) and destabilized density distribution (*inset* in **Figure 3d**). Furtherly revealed by the free energy diagram (**Figure 4g**), the adsorption energy of nitrogen on Fe-ReS₂ (-0.28 eV) is much smaller than that on ReS₂ (-0.04 eV), which demonstrates that nitrogen is prefer to be adsorbed and activated on the Fe-doped surface of Fe-ReS₂. Subsequently, six consecutive protonation and reduction processes were canvassed as N₂ (g)* → NNH* → HNNH* → HNNH₂* → H₂NNH₂* → NH₂* + NH₃ → NH₃ (**Table S4**). For the reaction pathway on ReS₂, the NH₃ desorption step is the rate-determining step during NRR process due to the highest energy barrier demand of 2.15 eV; While the transformation of HNNH₂* → H₂NNH₂* becomes the rate-determining step for the reaction pathway on Fe-ReS₂ with a relatively lower energy barrier demand of 0.68 eV. The results indicate that the Fe-doped surface of Fe-ReS₂ is beneficial to providing more active sites for highly kinetic NRR process.

Conclusions

In summary, we demonstrate that modulating the energy level of electrocatalysts by low-valent transition-metal doping engineering is an effective strategy to improve the NRR and HER activity of ReS₂ based catalysts. By means of electrochemical analysis, in situ FTIR, and DFT studies, we show the introduction of low-valent Fe motifs into ReS₂ structures results in the formation of abundant anchor sites for direct adsorption and subsequent activation of N₂ molecules, which are responsible for the significantly improved NRR and HER intrinsic activities. Our work unravels the important role of low-valent transition-metal on energy level refining for electrocatalysis optimization, which may also be used in many other catalytic systems.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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References

- [1] R. Raja, G. Sankar, J. M. Thomas, *J. Am. Chem. Soc.* **2001**, *123*, 8153.
- [2] J. W. Erisman, M. A. Sutton, J. Galloway, Z. Klimont, W. Winiwarter, *Nat. Geosci.* **2008**, *1*, 636.
- [3] M. A. Legare, G. Belanger-Chabot, R. D. Dewhurst, E. Welz, I. Krummenacher, B. Engels, H. Braunschweig, *Science* **2018**, *359*, 896.
- [4] R. Lan, J. Irvine, S. W. Tao, *Int. J. Hydrogen Energy* **2012**, *37*, 1482.

- [5] A. Klerke, C. H. Christensen, J. K. Norskov, T. Vegge, *J. Mater. Chem.* **2008**, *18*, 2304.
- [6] D. Bao, Q. Zhang, F. L. Meng, H. X. Zhong, M. M. Shi, Y. Zhang, J. M. Yan, Q. Jiang, X. B. Zhang, *Adv. Mater.* **2017**, *29*, 16047993.
- [7] N. Zhang, A. Jalil, D. X. Wu, S. M. Chen, Y. F. Liu, C. Gao, W. Ye, Z. M. Qi, H. X. Ju, C. M. Wang, X. J. Wu, L. Song, J. F. Zhu, Y. J. Xiong, *J. Am. Chem. Soc.* **2018**, *140*, 9434.
- [8] A. Banerjee, B. D. Yuhas, E. A. Margulies, Y. B. Zhang, Y. Shim, M. R. Wasielewski, M. G. Kanatzidis, *J. Am. Chem. Soc.* **2015**, *137*, 2030.
- [9] S. Licht, B. C. Cui, B. H. Wang, F. F. Li, J. Lau, S. Z. Liu, *Science* **2014**, *345*, 637.
- [10] C. van der Ham, M. Koper, D. Hettterscheid, *Chem. Soc. Rev.* **2014**, *43*, 5183.
- [11] C. X. Guo, J. R. Ran, A. Vasileff, S. Z. Qiao, *Energy Environ. Sci.* **2018**, *11*, 45.
- [12] V. Kyriakou, I. Garagounis, E. Vasileiou, A. Vourros, M. Stoukides, *Catal. Today* **2017**, *286*, 2.
- [13] Y. Yao, S. Q. Zhu, H. J. Wang, H. Li, M. H. Shao, *J. Am. Chem. Soc.* **2018**, *140*, 1496.
- [14] C. Lv, Y. M. Qian, C. S. Yan, Y. Ding, Y. Y. Liu, G. Chen, G. H. Yu, *Angew. Chem. Int. Ed.* **2018**, *57*, 10246; *Angew. Chem.* **2018**, *130*, 10403.
- [15] H. K. Lee, C. Koh, Y. H. Lee, C. Liu, I. Y. Phang, X. M. Han, C. K. Tsung, X. Y. Ling, *Sci. Adv.* **2018**, *4*, eaar3208.
- [16] Y. C. Hao, Y. Guo, L. W. Chen, M. Shu, X. Y. Wang, T. A. Bu, W. Y. Gao, N. Zhang, X. Su, X. Feng, J. W. Zhou, C. W. Hu, A. X. Yin, R. Si, Y. W. Zhang, C. H. Yan, *Nat. Catal.* **2019**, *2*, 448.
- [17] M. F. Wang, S. S. Liu, T. Qian, J. Liu, J. Q. Zhou, H. Q. Ji, J. Xiong, J. Zhong, C. L. Yan, *Nat. Commun.* **2019**, *10*, 341.
- [18] F. Qi, X. Q. Wang, B. J. Zheng, Y. F. Chen, B. Yu, J. H. Zhou, J. R. He, P. J. Li, W. L. Zhang, Y. R. Li, *Electrochim. Acta* **2017**, *224*, 593.
- [19] J. Gao, L. Li, J. W. Tan, H. Sun, B. C. Li, J. C. Idrobo, C. V. Singh, T. M. Lu, N. Koratkar, *Nano Lett.* **2016**, *16*, 3780.
- [20] X. Huang, Z. Zeng, H. Zhang, *Chem. Soc. Rev.* **2013**, *42*, 1934.

- [21] J. F. Xie, J. J. Zhang, S. Li, F. Grote, X. D. Zhang, H. Zhang, R. X. Wang, Y. Lei, B. C. Pan, Y. Xie, *J. Am. Chem. Soc.* **2013**, *135*, 17881.
- [22] X. D. Zhang, Y. Xie, *Chem. Soc. Rev.* **2013**, *42*, 8187.
- [23] L. Zhang, X. Q. Ji, X. Ren, Y. J. Ma, X. F. Shi, Z. Q. Tian, A. M. Asiri, L. Chen, B. Tang, X. P. Sun, *Adv. Mater.* **2018**, *30*, 1800191.
- [24] X. D. Li, T. S. Li, Y. J. Ma, Q. Wei, W. B. Qiu, H. R. Guo, X. F. Shi, P. Zhang, A. M. Asiri, L. Chen, B. Tang, X. P. Sun, *Adv. Energy Mater.* **2018**, *8*, 1801357.
- [25] Q. Zhang, S. J. Tan, R. G. Mendes, Z. T. Sun, Y. T. Chen, X. Kong, Y. H. Xue, M. H. Rummeli, X. J. Wu, S. L. Chen, L. Fu, *Adv. Mater.* **2016**, *28*, 2616.
- [26] S. Tongay, H. Sahin, C. Ko, A. Luce, W. Fan, K. Liu, J. Zhou, Y. S. Huang, C. H. Ho, J. Y. Yan, D. F. Ogletree, S. Aloni, J. Ji, S. S. Li, J. B. Li, F. M. Peeters, J. Q. Wu, *Nat. Commun.* **2014**, *5*, 3252.
- [27] S. Horzum, D. Cakir, J. Suh, S. Tongay, Y. S. Huang, C. H. Ho, J. Wu, H. Sahin, F. M. Peeters, *Phys. Rev. B* **2014**, *89*, 15543315.
- [28] Z. Cai, D. J. Zhou, M. Y. Wang, S. M. Bak, Y. S. Wu, Z. S. Wu, Y. Tian, X. Y. Xiong, Y. P. Li, W. Liu, S. Siahrostami, Y. Kuang, X. Q. Yang, H. H. Duan, Z. X. Feng, H. L. Wang, X. M. Sun, *Angew. Chem. Int. Ed.* **2018**, *57*, 9392; *Angew. Chem.* **2018**, *130*, 9536.
- [29] S. H. Ye, Z. X. Shi, J. X. Feng, Y. X. Tong, G. R. Li, *Angew. Chem. Int. Ed.* **2018**, *57*, 2672; *Angew. Chem.* **2018**, *130*, 2702.
- [30] Y. Shi, Y. Zhou, D. R. Yang, W. X. Xu, C. Wang, F. B. Wang, J. J. Xu, X. H. Xia, H. Y. Chen, *J. Am. Chem. Soc.* **2017**, *139*, 15479.
- [31] F. L. Lai, J. R. Feng, X. B. Ye, W. Zong, G. J. He, Y. E. Miao, X. M. Han, X. Y. Ling, I. P. Parkin, B. C. Pan, Y. F. Sun, T. X. Liu, *J. Mater. Chem. A* **2019**, *7*, 827.
- [32] F. L. Lai, D. Y. Yong, X. L. Ning, B. C. Pan, Y. E. Miao, T. X. Liu, *Small* **2017**, *13*, 1602866.
- [33] C. M. Corbett, C. McClellan, A. Rai, S. S. Sonde, E. Tutuc, S. K. Banerjee, *ACS Nano* **2015**, *9*, 363.

- [34] J. Gao, L. Li, J. W. Tan, H. Sun, B. C. Li, J. C. Idrobo, C. V. Singh, T. M. Lu, N. Koratkar, *Nano Lett.* **2016**, *16*, 3780.
- [35] Y. Liu, Y. Su, X. Quan, X. F. Fan, S. Chen, H. T. Yu, H. M. Zhao, Y. B. Zhang, J. J. Zhao, *ACS Catal.* **2018**, *8*, 1186.
- [36] A. P. Grosvenor, B. A. Kobe, M. C. Biesinger, N. S. McIntyre, *Surf. Interface Anal.* **2004**, *36*, 1564.
- [37] K. Ataka, T. Yotsuyanagi, M. Osawa, *J. Phys. Chem.* **1996**, *100*, 10664.
- [38] S. D. Lin, A. C. Gluhoi, B. E. Nieuwenhuys, *Catal. Today* **2004**, *90*, 3.
- [39] L. K. Xu, Y. L. Xin, J. T. Wang, *Electrochim. Acta* **2009**, *54*, 1820.
- [40] G. Ramis, L. Yi, G. Busca, M. Turco, E. Kotur, R. J. Willey, *J. Catal.* **1995**, *157*, 523.
- [41] Y. Ikezawa, T. Sawatari, T. Kitazume, H. Goto, K. Toriba, *Electrochim. Acta* **1998**, *43*, 3297.
- [42] W. Gulaczyk, M. Kreglewski, A. Valentin, *J. Mol. Spectrosc.* **2003**, *220*, 132.
- [43] J. H. Montoya, C. Tsai, A. Vojvodic, J. K. Norskov, *ChemSusChem* **2015**, *8*, 2180.
- [44] X. Ma, J. Liu, H. Xiao, H. Li, M. H. Shao, *J. Am. Chem. Soc.* **2018**, *140*, 46.
- [45] J. X. Zhao, Z. F. Chen, *J. Am. Chem. Soc.* **2017**, *139*, 12480.

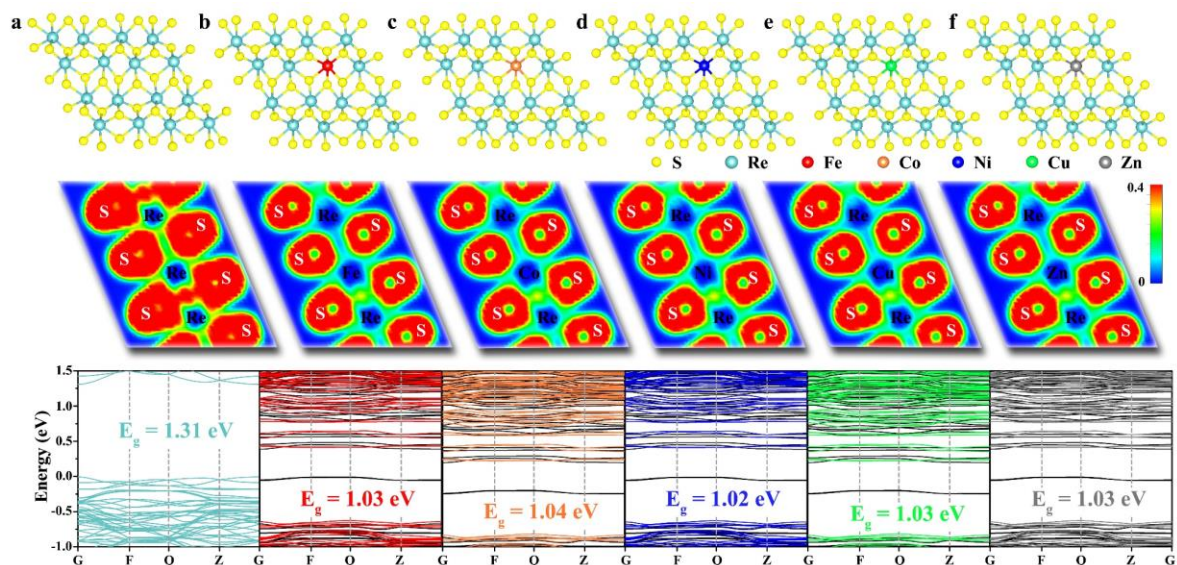


Figure 1. Atomistic models of crystal structure, two dimensional electron localization function (ELF) mappings, and calculated band structures for (a) ReS₂, (b) Fe-ReS₂, (c) Co-ReS₂, (d) Ni-ReS₂, (e) Cu-ReS₂, and (f) Zn-ReS₂, respectively.

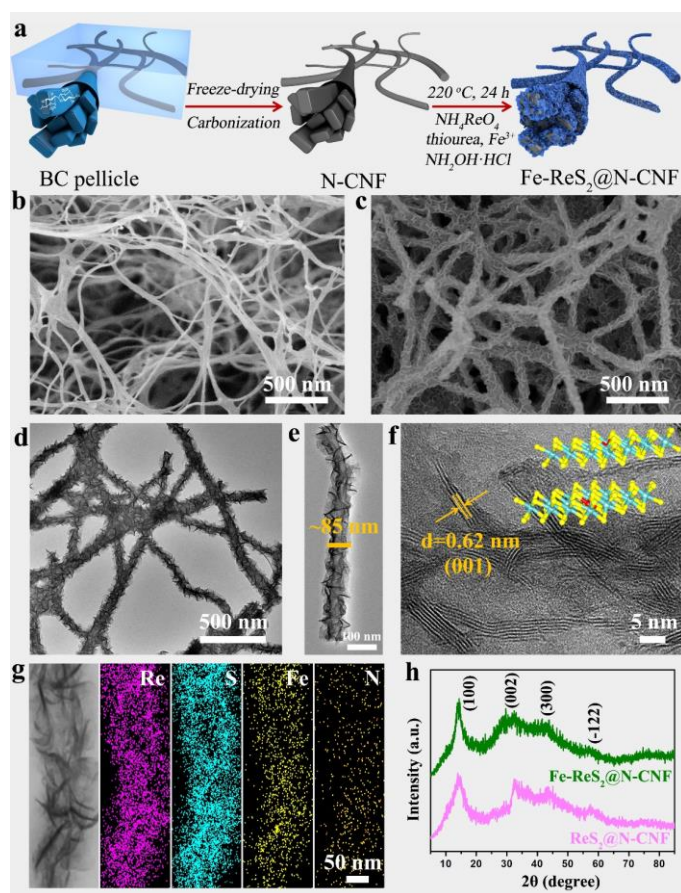


Figure 2. (a) Schematic illustration of preparation process of Fe-ReS₂@N-CNF. SEM images of (b) N-CNF and (c) 5 mol % Fe-doped ReS₂@N-CNF (Fe-ReS₂@N-CNF-5). (d-e) TEM images, (f) HRTEM image, and (g) corresponding element mapping of Fe-ReS₂@N-CNF-5. (h) XRD patterns of Fe-ReS₂@N-CNF-5 and ReS₂@N-CNF.

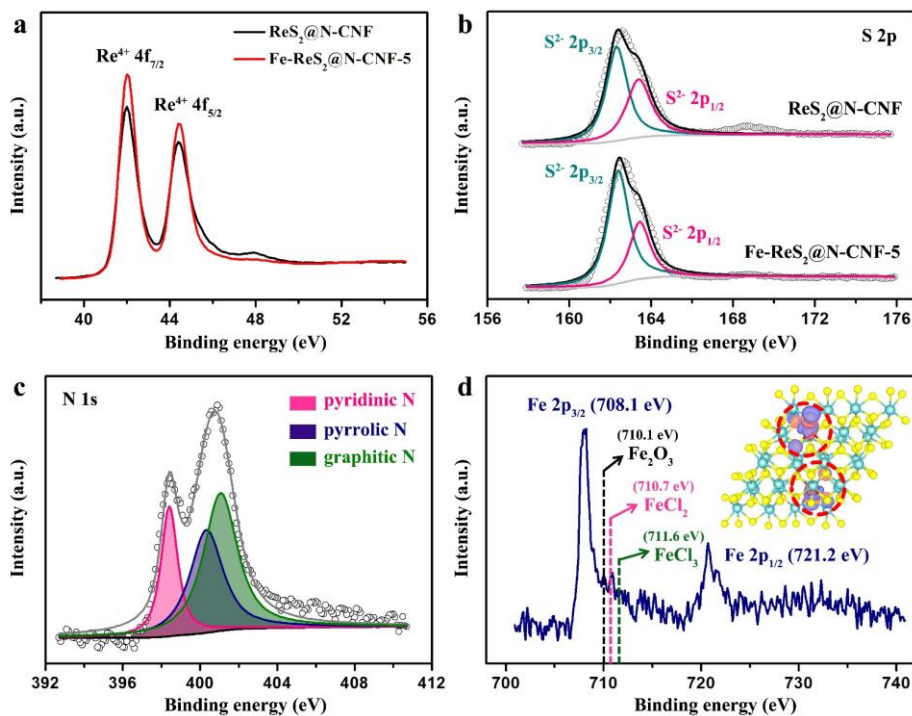


Figure 3. High-resolution XPS spectra of (a) Re 4*f* and (b) S 2*p* peaks for Fe-ReS₂@N-CNF-5 and ReS₂@N-CNF, respectively. High-resolution XPS spectra of (c) N 1*s* and (d) Fe 2*p* for Fe-ReS₂@N-CNF-5.

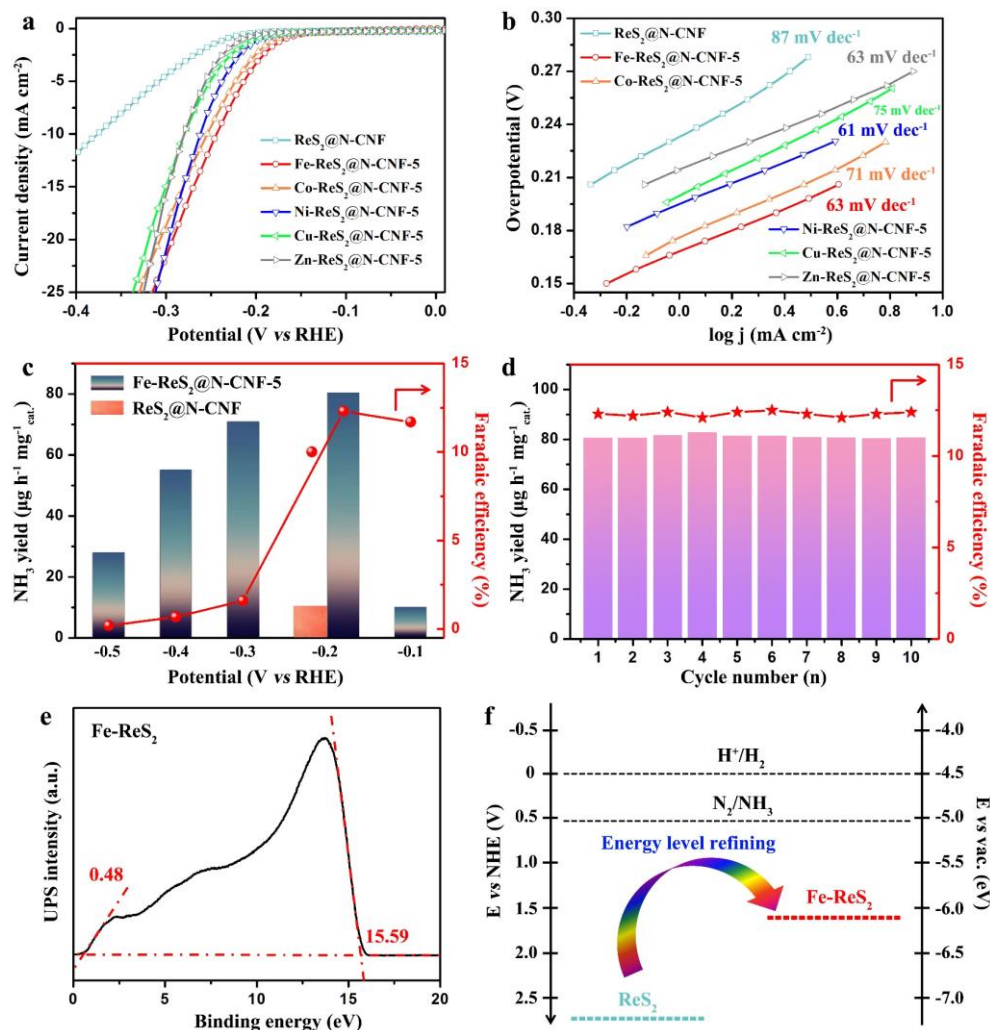


Figure 4. Electrocatalytic HER/NRR of 5 mol % transition-metal doped ReS₂ nanosheets wrapped nitrogen-doped carbon nanofiber (Fe/Co/Ni/Cu/Zn-ReS₂@N-CNF-5) and non-doped ReS₂@N-CNF catalysts in aqueous solution of 0.1 M Na₂SO₄ under ambient conditions. (a) Polarization curves. (b) Tafel curves. (c) The NH₃ production rates (bar graph) and FE_{NH3} (red line) of Fe-ReS₂@N-CNF-5 and ReS₂@N-CNF at different potentials. (d) NRR stability test of Fe-ReS₂@N-CNF-5 at -0.2 V versus RHE. (e) Ultraviolet photoelectron spectroscopy (UPS) spectra of Fe-ReS₂ nanosheets. (f) Band structure diagram for Fe-ReS₂ and pure ReS₂.

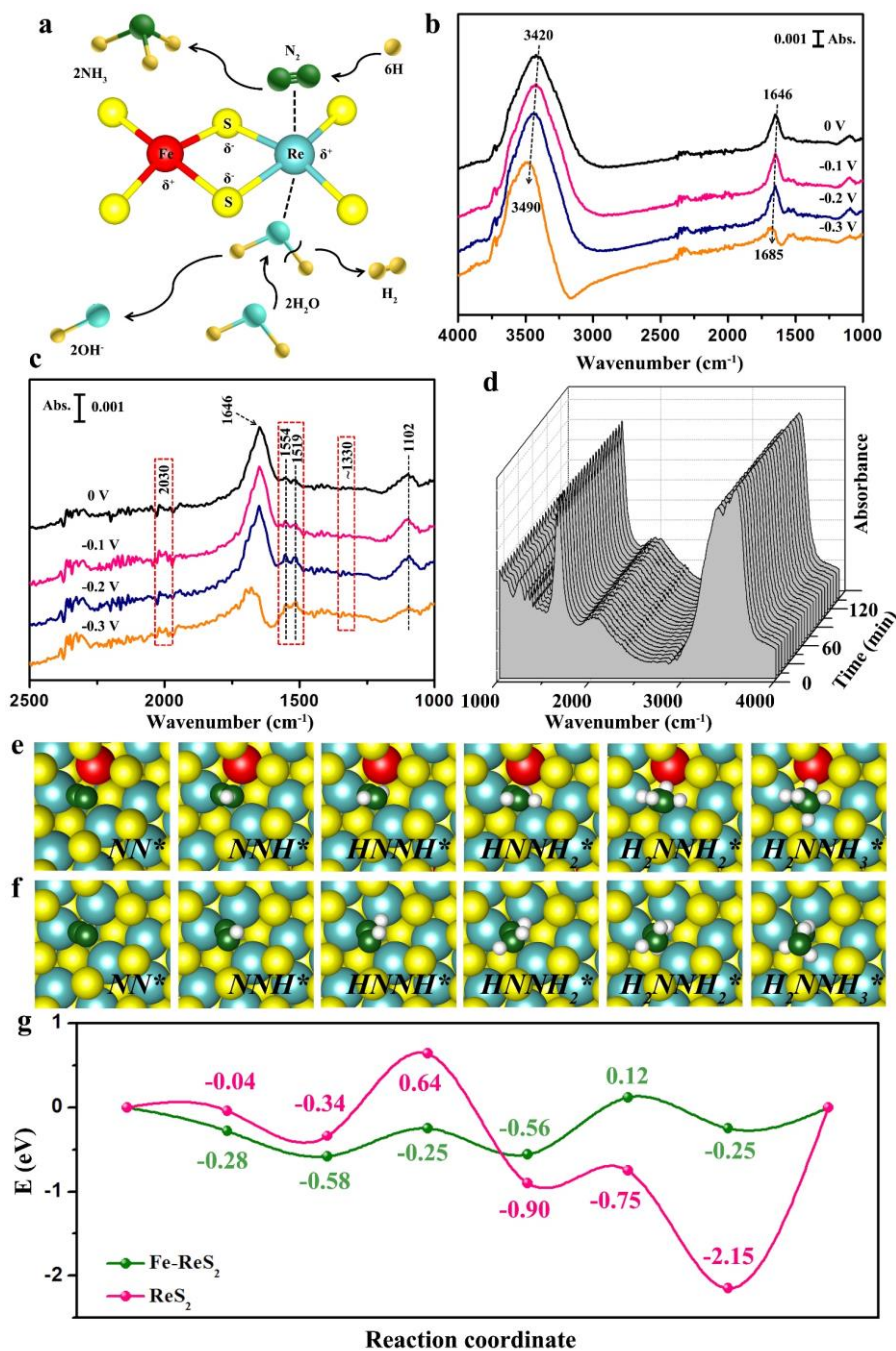


Figure 5. (a) The proposed mechanism on uncoordinated Fe (δ^+) and Re (δ^+) centers during NRR and HER processes. (b) In-situ SEIRAS spectra from 0 to -0.3 V on the Fe-ReS₂@N-CNF-5 catalyst in a N_2 -saturated 0.1 M Na_2SO_4 solution. (c) Amplification of the selected area in (b). (d) Time-dependent SEIRAS spectra of Fe-ReS₂@N-CNF-5 catalyst at a voltage of -0.2 V. Optimized structures of NH^* , NNH^* , HNNH^* , HNNH_2^* , H_2NNH_2^* , and H_2NNH_3^* intermediates on: (e) Fe-ReS₂ and (f) ReS₂. (g) Free energy diagram of NRR to NH_3 on Fe-ReS₂ and ReS₂.

Energy level refining engineering: The energy level of ReS_2 can be finely tuned *via* low-valent transition-metal (*e.g.* Fe/Co/Ni/Cu/Zn) doping to match both of nitrogen reduction reaction and hydrogen evolution reaction electrocatalysis.

Keywords: electrocatalysis; low-valence; transition-metal; nitrogen reduction reaction; rhenium disulfide

Feili Lai, Nan Chen, Xiaobin Ye, Guanjie He, Wei Zong, Katherine B. Holt, Bicao Pan, Ivan P. Parkin, Tianxi Liu,* Renjie Chen*

Refining Energy Levels in ReS_2 Nanosheets by Low-Valent Transition-Metal Doping for Dual-Boosted Electrochemical Ammonia/Hydrogen Production

ToC figure

